

Effects of Nanocellulose on the Structure of Collagen: Insights from Molecular Dynamics Simulation and Umbrella Sampling

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28 **1. Introduction**

29 As a valuable new type of green biological nanomaterial, nanocellulose has the advantages of a fine
30 nanostructure, good mechanical strength, and low thermal expansion coefficient, as well as recoverability
31 and sustainability (Fernandes and Madhuranthakam 2020; Zhang et al. 2019). Moreover, nanocellulose
32 in the form of nanoparticles, tablets, gas-liquid gel system, and fiber membrane system is widely used in
33 the field of energy storage (Chen et al. 2018), wastewater treatment (Tang et al. 2019), 3D bio-
34 printing(Dorishetty et al. 2020), biosensor(Golmohammadi et al. 2017), drug delivery(Salimi et al. 2019)
35 and other biological fields. Compared with graphene (Luan, Huynh and Zhou 2016), graphene oxide (He
36 et al. 2019), molybdenum disulfide (Gu et al. 2016; Gu et al. 2017), carbon nanotube(El-Sayed et al.
37 2016), and other 2D nanomaterials, nanocellulose as a carrier material has a lower nanotoxicity, which
38 makes it more widely available. Although nanocellulose and its derivatives have been widely concerned
39 in medicine and biological tissue, the long-term retention of nanocellulose in the human body makes its
40 toxicity study very important due to the lack of cellulose-degrading enzymes in the human body.

41 Collagen can also be a candidate for biomaterials such as tissue-engineered scaffolds and wound
42 dressings (Lee et al. 2019; Ge et al. 2018; Sorushanova et al. 2019). However, the application of pure
43 collagen materials is limited due to their low water resistance, fast biodegradation perishability, and poor
44 thermal stability (Ge et al. 2018). While cellulose and collagen nanocomposite materials overcome the
45 weaknesses of pure collagen materials. Cellulose and its derivatives can be widely used to strengthen
46 various polymer matrix materials due to their high specific surface area, high crystallinity, low density,
47 and high elastic modulus (Manhas et al. 2015; Salimi et al. 2019; Liu et al. 2018; Li et al. 2017).
48 Researchers have proved that collagen/nanocellulose composite has good properties and stability better
49 than pure collagen. Animal experimental studies (Liu et al. 2020b; Liu et al. 2020a) (Collagen/cellulose
50 nanofiber hydrogel scaffold: physical, mechanical and cell biocompatibility properties; A 3D porous
51 microsphere with multistage structure and component based on bacterial cellulose and collagen for bone
52 tissue engineering (Zhang et al. 2020); demonstrated that collagen and nanocellulose composite is a
53 promising material for wound dressings and tissue engineering scaffolds.

54 The advantages and disadvantages of nanocellulose and collagen can effectively complement each
55 other to form a more potential nanocomposite material (Cudjoe et al. 2017), which makes their

56 composites have a broader application prospect. However, the interactions between nanocellulose and
57 collagen, which are significantly related to the strength of composites, are still obscure. Furthermore, in
58 vivo and in vitro experiments have shown that nanocellulose and its derivatives have adverse effects on
59 intestinal microorganisms (DeLoid et al. 2019), liver cells (Otuochere et al. 2020), and lung cells(Sai and
60 Fujita 2020). In addition, nanocellulose biological dressings and tissue-engineered materials, as the main
61 components of human tissues, will directly interact with collagen when they contact the human body.
62 Therefore, it is necessary to study the toxicity of nanocellulose.

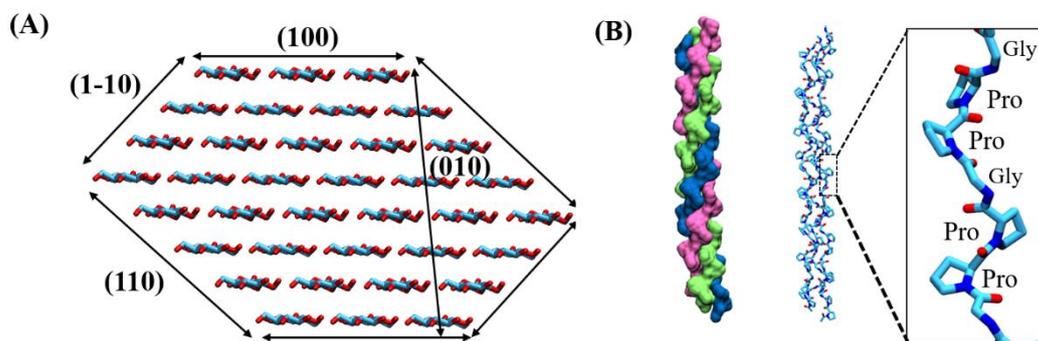
63 In this study, molecular dynamics simulations were carried out to study the interactions between
64 collagen and nanocellulose. Nanocellulose materials are characterized by a high degree of crystallinity
65 thus it is feasible to employ cellulose crystal to model the properties of nanocellulose. Native crystalline
66 cellulose contains a mixture of faces, which complicates the direct assignment of cellulose-collagen
67 interactions to a specific face by experimental methods such as NMR spectroscopy. Molecular dynamics
68 simulations provide a good solution to bypass this limitation. A single crystal face in nanoscale can be
69 constructed by molecular modeling. Molecular dynamics simulations have been carried out by Crowley
70 et al. to study the interactions between cellulose and lignin and gain insights into quantitative
71 relationships between different cellulose faces and specific lignin chemistries (Vermaas, Crowley and
72 Beckham 2019). Moreover, molecular dynamics simulations have also been used to probe the
73 interactions between water molecules and cellulose, which shed light on the wetting mechanisms of
74 cellulose(Malaspina and Faraudo 2019); Cellulose nanocrystals produced using recyclable sulfuric acid
75 as hydrolysis media and their wetting molecular dynamics simulation; Molecular simulation of surface
76 reorganization and wetting in crystalline cellulose I and II, Wetting the (110) and (100) Surfaces of I β
77 Cellulose Studied by Molecular Dynamics. Furthermore, many the interaction between two-dimensional
78 nanomaterials and biomacromolecules have also been investigated by molecular dynamics simulations.
79 Zhou et al., adopted molecular dynamics simulation (MD) to conduct thorough research on the interfacial
80 properties of two-dimensional nanomaterials and found that graphene (Luan et al. 2016), graphene
81 oxide(He et al. 2019; Mathesh et al. 2016), defective graphene(Gu et al. 2019) and other carbon
82 nanomaterials can induce protein and nucleic acid denaturation, which provided theoretical support for
83 the potential nanotoxicity of nanomaterials when used in biological systems. ,

84 In this work, I β -cellulose, which is one of the main components of higher plants, was selected to

85 model nanocellulose. Collagen type I was employed as the model collagen, which is the most abundant
86 and widely distributed natural structural protein in the human body (Lin and Liu 2006). The adsorption
87 of collagen on different crystal planes of cellulose was simulated by molecular dynamics. Then the
88 structural changes and specific interactions were characterized in detail. This study investigated the
89 interactions between nanocellulose and collagen at a molecular scale and evaluated the structural changes
90 of collagen, which revealed the possible biological effects of nanocellulose and provided theoretical
91 guidance for the design of nanocellulose-collagen complex at the same time.

92 2. Methods

93 The initial structure of collagen is obtained by extracting three chains from the crystal structure
94 (Berisio et al. 2009) (PDB code 1K6F) as shown in Fig. 1(B) All nanocellulose crystal faces of different
95 sizes (Fig. 1A) were constructed using Cellulose-Builder (Gomes and Skaf 2012). The thickness of (100),
96 (110), (1-10) and (010) varied to weaken the effects of surface tension. Collagen was placed above the
97 surface of the nanocellulose crystal at different angles (0° , 30° and 45°) in VMD software (Humphrey,
98 Dalke and Schulten 1996) to form the initial coordinates of the simulation system and the minimum
99 distances between cellulose faces and collagens were ranging from 0.8 nm to 1 nm (Fig. 2A). As shown
100 in Fig. 2(A), three parallel simulations of each system were carried out for 500 ns. The composite system
101 was solvated in a cubic box with a TIP3P water model (Mark and Nilsson 2001; Jorgensen et al. 1983)
102 and modeled by a CHARMM36 force field (Lee et al. 2014; Boonstra, Onck and van der Giessen 2016).



103
104 **Fig. 1.** (A) The crystalline plane of cellulose I β . (B) Collagen in initial configuration (PDB ID: 1K6F).

105 All MD simulations were performed in GROMACS-5.1 (Berendsen, Spoel and Drunen 1995; Van
106 Der Spoel et al. 2005) package All the systems were equilibrated carefully in the beginning of simulation.
107 The energy minimization process was carried out with 1000 cycles of steepest descent and 1,000 cycles

108 of conjugate gradient minimization. Then, equilibration runs were performed for 5 ns in the NVT
109 ensemble and 5 ns in the NPT ensemble with the heavy atoms of protein and cellulose fixed. Finally, 500
110 ns production runs were simulated in the NPT ensemble with the restriction of the protein released. The
111 long-range electrostatic interactions were treated by the particle mesh Ewald (PME) method (Petersen
112 1995), while the short-range van der Waals interactions were calculated with a cutoff distance of 1.0 nm.
113 All covalent bonds containing hydrogen atoms were constrained by the LINCS algorithm (Hess et al.
114 2008). The V-rescale thermostat (Berendsen et al. 1984) was used to heat the system to 300 K and the
115 Parrinello-Rahman Pressure coupling (Parrinello and Rahman 1981; Nosé and Klein 2006) kept the
116 system pressure at 1 bar. The integration step size of the simulation process is 2 fs. Periodic boundary
117 conditions were applied in all directions with glycosidic bonds formed between mirror images.

118 The potential of mean force (Roux 1995) (PMF) obtained by pulling simulation and umbrella
119 sampling (Hub 2015) was used to calculate the binding free energy of the system. The cellulose surface
120 was used as a reference point and a harmonic potential was applied to the collagen as a pulling point.
121 The last frame of the MD simulations was selected as the initial conformation, 300 ps umbrella traction
122 was provided for collagen along the z-axis to increase the center of mass (COM) distance between
123 collagen and cellulose. The spring constant used was $2000 \text{ kJ mol}^{-1} \text{ nm}^2$ and the pull rate was 0.01 nm/ps.
124 More than 13 umbrella sampling windows were selected according to the interval size of COM values.
125 1ns of simulations in NPT was performed on each sample, then 10 ns of MD process was carried out.
126 Finally, weighted histogram analysis (Hub 2015) (WHAM) was used to calculate PMF.

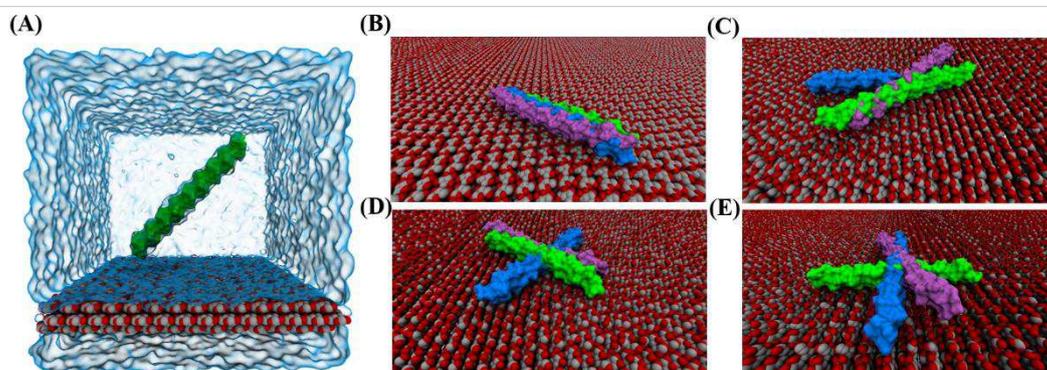
127 The relevant modules in GROMACS were used to calculate the backbone root mean square
128 deviation (RMSD) and backbone root mean square fluctuation (RMSF) of proteins during the whole
129 simulation process. G_hbond was used to calculate the number of hydrogen bonds the distance of 0.30
130 nm and angle 30° as criterial.

131 **3. Results and discussions**

132 **3.1 Structural of collagen remained intact on nanocellulose**

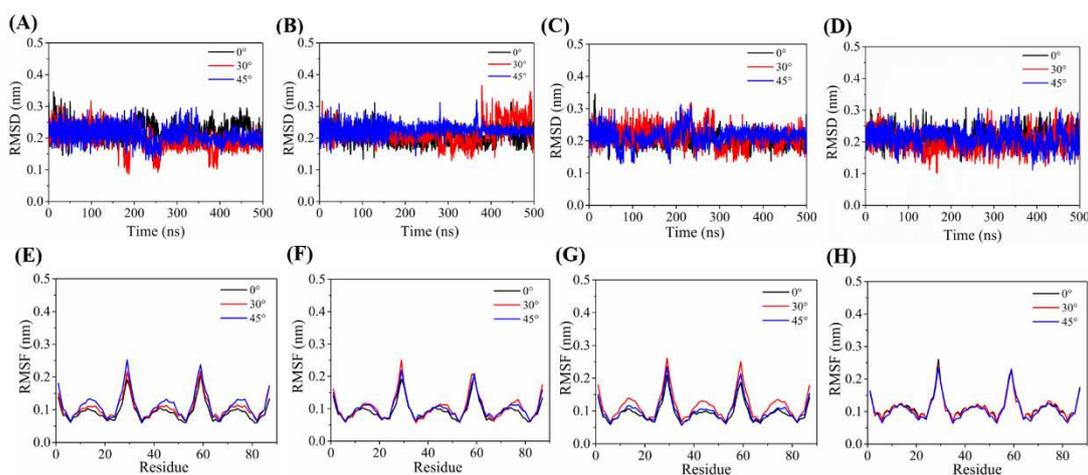
133 In the initial simulation system, the distance between protein and cellulose crystal faces was
134 controlled between 0.8-0.1nm as shown in Fig. 2A and three groups of different model analyses were
135 conducted for each crystal face. Although the hydrophilicity and hydrophobicity of the selected cellulose

136 crystal faces were different, the adsorption behaviors of collagen on different crystal faces were similar.
137 The protein slowly contacted the cellulose surface. As shown in Fig. 2(B-E), no significant structural
138 changes were observed except that the overall structure of collagen was slightly bent at the end of the
139 simulation.



140
141 **Fig. 2.** (A) The initial system configuration of (100) planes. The water box is rendered with a cyan surface.
142 (B-E) Snapshots of four crystal planes at 500ns. The protein shows green (0°), purple (30°), and blue
143 (45°) to indicate three tracks, and the cellulose crystal plane is shown in sphere (carbon, gray; oxygen,
144 red), hydrogen is not shown for clarity.

145 The calculation of the root-mean-square deviation of the protein was carried out to quantitatively
146 measure the change of collagen structure during the simulation. RMSD values of collagen on different
147 nanocellulose crystal faces are kept between 0.2 nm-0.3 nm as shown in Fig. 3(A-D). The profiles of
148 RMSD exhibit very small fluctuation during the whole process. RMSD values in this range indicated
149 that the protein structure did not change significantly during the adsorption process. Furthermore, RMSF
150 was calculated to evaluate the free movement degree of each residue in collagen molecules. As shown in
151 Fig. 3(E-H), the profiles of RMSF has good accordance with each other regardless of the properties of
152 different cellulose faces. Each chain of the collagen model is composed of 29 residues and the peak of
153 the line represents the end of each chain, which indicates that the two ends of the polypeptide chain are
154 more flexible and the structure of other residues located in the middle of collagen is stable. In conclusion,
155 the overall structure of collagen was not damaged during the process of adsorption on different crystal
156 faces of cellulose. It is widely recognized that the change of protein structure induces the loss of its
157 biological function (Feng et al. 2017), thus it can be inferred that nanocellulose has good biocompatibility.
158

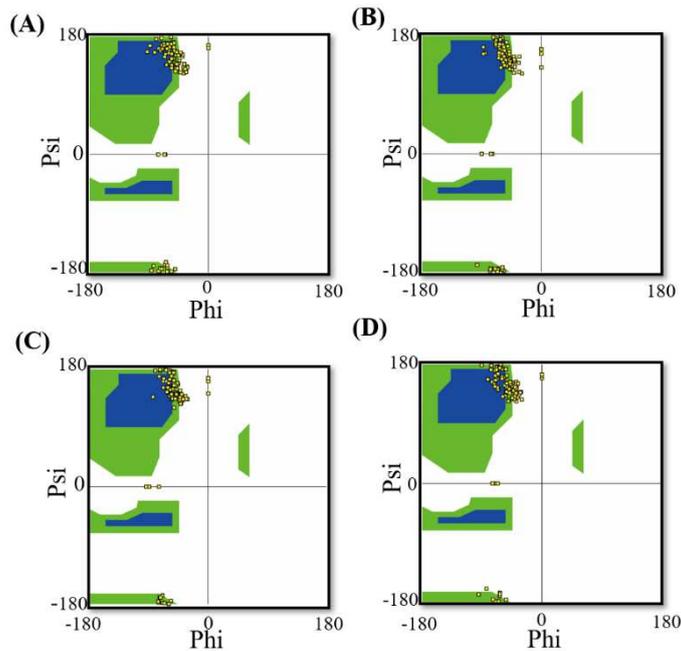


159

160 **Fig. 3.** (A-D) Time evolution of root mean squared deviation (RMSD) values of collagen on (100), (110),
 161 (1-10) and (010) faces over time, respectively. (E-H) root mean square fluctuation (RMSF) variation of
 162 each residue that constitutes collagen on different crystal planes.

163 3.2 Evolution of the secondary structure of collagen

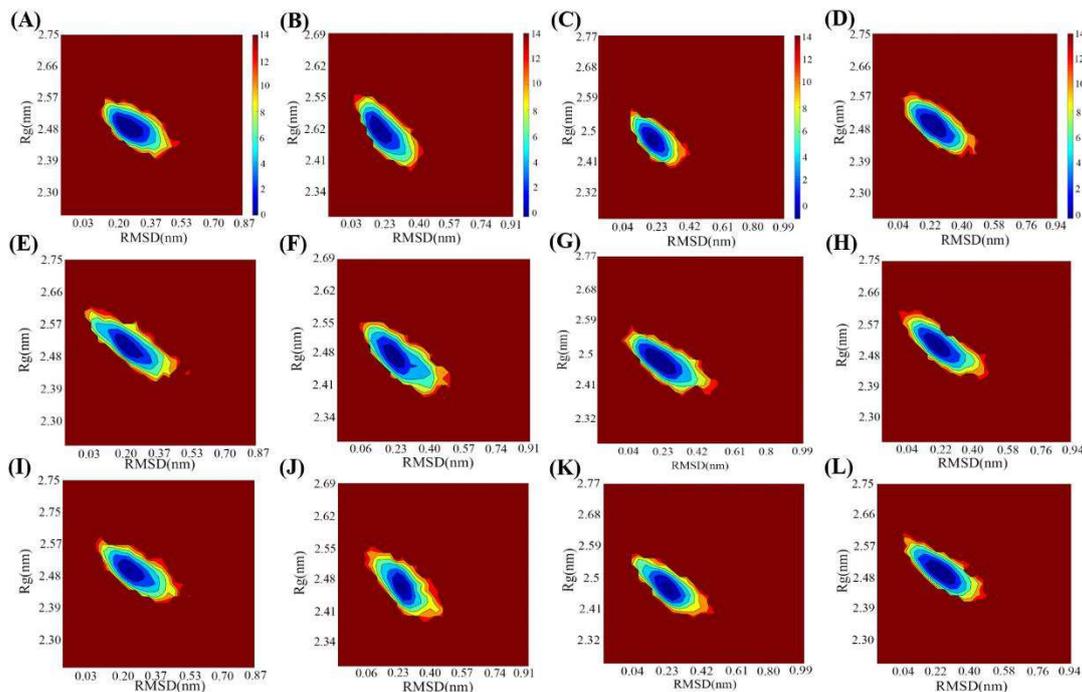
164 The structure of proteins plays an important role in their activity and biological effects. To analyze
 165 the structural changes of collagen in the process of adsorption in more detail, the Ramachandran plot
 166 (Hoof, Sander and Vriend 1997) was used to characterize the changes in the secondary structure of
 167 proteins. Collagen is a coil, but one with distinct tertiary and quaternary structures: three separate
 168 polypeptides, called α chains are super-twisted about each other. The superhelical twisting is right-handed
 169 in collagen, opposite in sense to the left-handed helix of the α chains. Thus, it is ambiguous to evaluate
 170 the secondary structure of collagen by designating the structure helix, sheet, or coil. Therefore,
 171 Ramachandran plots were employed. The collagen helix is a unique secondary structure with $\Phi = -$
 172 51° and $\Psi = +153^\circ$, which is quite distinct from the α helix (Köppen, Ohler and Langel 2007). As shown
 173 in Fig. 4, Ramachandran plots of the last frames extracted from MD simulations display similar
 174 characteristics and most of Φ and Ψ values are confined within the region corresponding to the
 175 structure of collagen. Random coil structures have also been observed with Φ and Ψ values appearing
 176 around $(-90, -180)$. It was found by comparison that the Φ and Ψ angles of collagen not significantly
 177 deviated (Fig. 4A-D) from the specified collagen conformation values during the adsorption of collagen
 178 in different crystal planes of cellulose. Therefore, it is indicated that all models exhibit partial structural
 179 interruption but are not damaged during adsorption.



180

181 **Fig. 4.** The Ramachandran plots of collagens adsorbed on the planes (100) (A), (110) (B), (1-10) (C),
 182 and (010) (D).

183 To further investigate the effects of nanocellulose on the global structure of collagen, principal
 184 component analysis contour maps were constructed, which is also an effective method to analyze protein
 185 structure. Fig. 5 displays the free energy contour map of collagen after PCA in 12 simulated trajectories.
 186 As shown in Fig5, the free energy contour values of each model are located in a similar region with only
 187 one global minimum, which indicates that the effects of different crystal faces on collagen structures are
 188 too little to induce obvious changes. All the global minima of the 12 MD simulations are restricted within
 189 narrow ranges with RMSD about 0.23 nm and Rg about 0.42 nm, which further indicates the intactness
 190 of collagen on the surface of nanocellulose. The conformational space of collagen on the surface of
 191 nanocellulose provided information about the structural state of collagen in nanocellulose-collagen
 192 composites. Moreover, the study also partially evaluated the biological effects of nanocellulose when it
 193 is applied in the human body as a biomedical material. It exerted limited influence on the structure of
 194 collagen, which further validated the biocompatibility of nanocellulose.



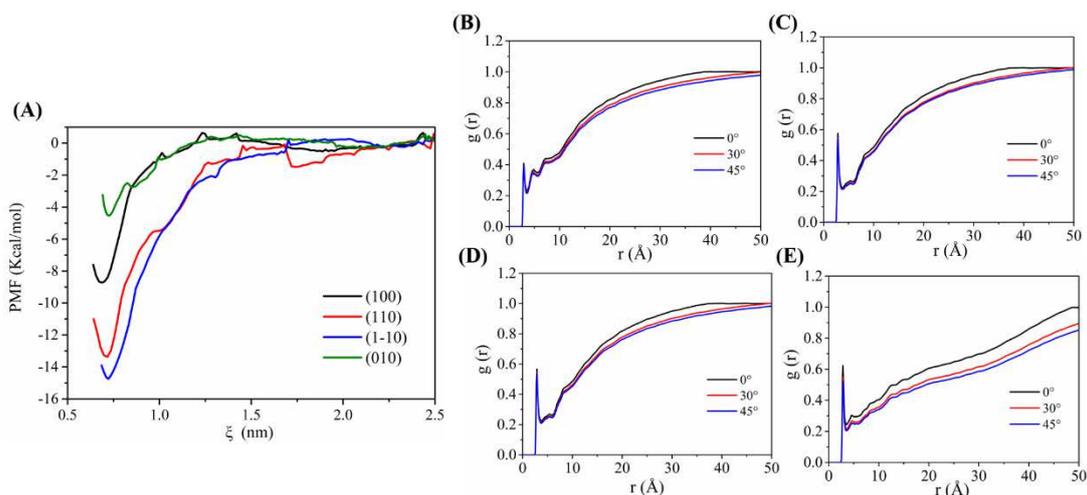
195

196 **Fig. 5.** Free energy landscape obtained from principal PCA of all models. From left to right, (100), (110),
 197 (1-10), and (010) sides. The horizontal rows from left to right are (100), (110), (1-10), and (010),
 198 respectively. The vertical rows from top to bottom represent 0° , 30° , and 45° .

199 3.3 Interaction between collagen and nanocelluloses during adsorption

200 To understand the adsorption of collagen onto nanocellulose, umbrella sampling simulations were
 201 performed for all the four faces. Prior to the umbrella sampling simulation, the centers of all the cellulose
 202 models move to original points. Reaction coordinates were corrected by deducting half of its thickness
 203 in order to eliminate the effects of different thicknesses of cellulose slab and better illuminate the
 204 properties of different surface morphologies. As shown in Fig. 6(A), the lowest PMF values are found
 205 near the layer surface for all the systems, indicating that collagen tends to bind to cellulose. PMF shows
 206 that free energies are close for faces (110) and (1-10) with values of -15.5 kcal/mol and -14.8 kcal/mol
 207 respectively, which imply that almost the same adsorption strength of collagen onto the (110) and (1-10)
 208 surface. Free energies of (100) (-9.5 kcal/mol) are a little higher than those of (110) and (1-10) faces,
 209 indicating that the affinity between (100) layer and collagen are slightly weaker. In contrast with PMF
 210 profiles of (110), (1-10), and (100), the binding energies of (010) are much lower (-5.2 kcal/mol). These
 211 free energies derived from umbrella samplings indicate that collagen tends to migrate toward all the faces

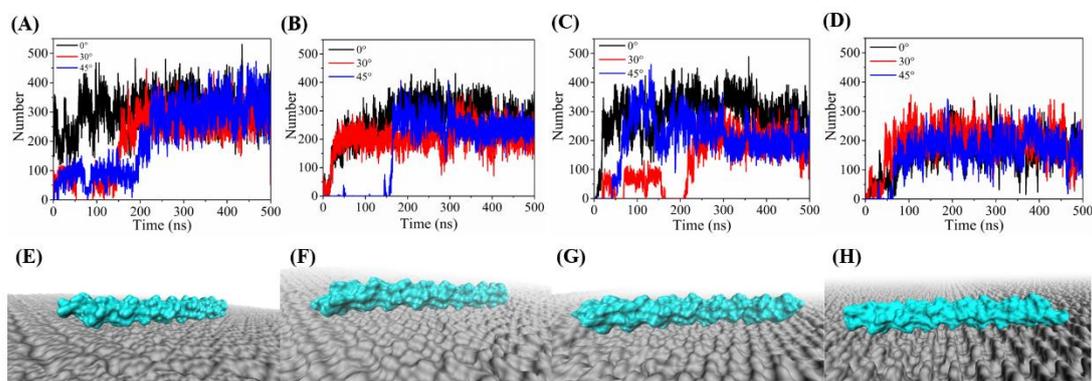
212 of cellulose. In particular, (110) and (1-10) faces displaying the strongest affinity to collagen. (100) face,
 213 which is more hydrophobic, displays weaker adsorptive capability with collagen. Due to the structural
 214 anisotropy of cellulose, $I\beta(110)$ and $I\beta(1-10)$ faces are hydrophilic while $I\beta(100)$ face is hydrophobic.
 215 The chain of collagen is composed of repeating tripeptide sequence Gly-Pro-Pro, which are all polar
 216 amino acids. Polar interactions between hydrophilic cellulose faces and collagen enhance their affinity.
 217 Though (010) face is hydrophilic, the interaction between these faces and collagen is the weakest among
 218 the four systems, which seems quite counterintuitive. The origin of this behavior is attributed to the
 219 topography of (010) faces, which hinders the binding of collagen with half of the hydroxyl grouping
 220 shielded. The steric hindrance weakens the interactions between collagen and (010) face, which is
 221 consistent with the results of the analysis of contact number. (110) and (1-10) are more hydrophilic than
 222 (100) but less rough than (010) face (Fig. 7E-H), which may result in a stronger affinity with collagen.



223
 224 **Fig. 6.** (A) Potential of mean force plots describing the binding of collagen and cellulose. The radial
 225 distribution function of oxygen in exposed hydroxyl groups on water and cellulose surfaces, with
 226 cellulose surfaces represented by (B) (100), (C) (110), (D) (1-10) and (E) (010) respectively.

227 To further validate the deduction, radial distributions of water molecules were calculated to evaluate
 228 the hydrophilicity of the four faces. As shown in Fig. 6(B-E), their first peak positions occur at the same
 229 position 3.8 Å, respectively. The height of the first $g(r)$ peak belonging to (100) face is about 0.4, which
 230 is significantly lower than that of (110), (1-10) and (010) faces with the heights of the first peak all about
 231 0.6. Thus, the heights of the first peaks describe a distinguishable difference in hydrophilicity among the
 232 four faces and the surface hydration of (100) face is weaker than the other three faces. To evaluate the
 233 effects of surface morphology on the interaction between collagen and cellulose, the contact number of

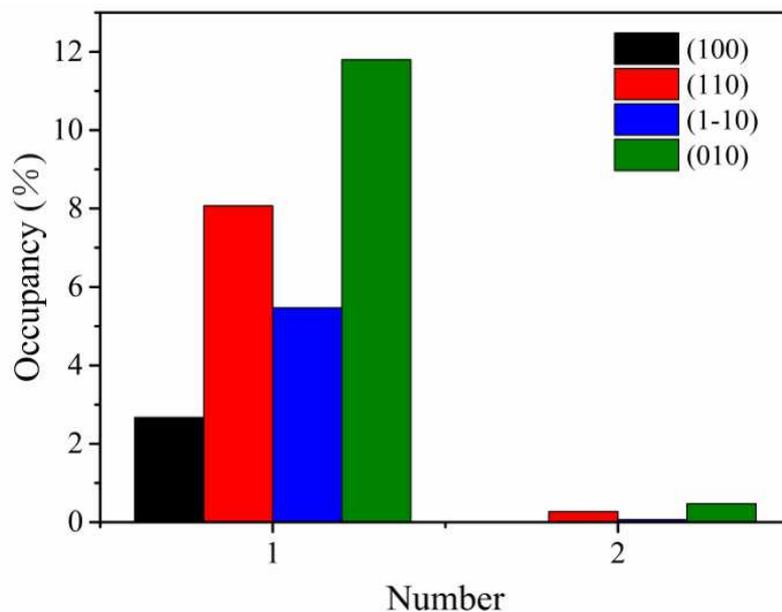
234 heavy atoms was calculated with 0.5 nm as the threshold. In general, the loading of collagen on the
 235 cellulose surface was fast with molecules of collagen adsorbed on cellulose within 200 ns. Based on the
 236 heavy atom contact numbers between collagen and cellulose, it has been deduced that there is an obvious
 237 correlation between surface roughness and contact numbers. As shown in Figure 7, (100) face displays
 238 the largest contact number with the smoothest surface, while, (010) face exhibits the least contact number
 239 with the greatest surface roughness. (110) and (1-10) faces are in between with the contact numbers larger
 240 than that of (100) face but less than that of (010) face. As the fundamental part of molecular interactions,
 241 the block of direct contact between collagen and (010) face impaired their affinity. The combinations of
 242 mediocre hydrophilicity and smoothness made (110) and (1-10) faces stand out from the other two crystal
 243 facets in the adsorption process.



244
 245 **Fig. 7.** Change of heavy atomic contact number of collagen and (100) (A), (110) (B), (1-10) (C), (010)
 246 (D) faces over time. (E-H) The last frame trajectory of the model when the collagen was 0° on the four
 247 crystal planes. Collagen is blue, cellulose overall structure is gray.

248 Cellulose molecules contain a large number of free hydroxyl groups, which might be involved in
 249 hydrogen bonding interactions. Therefore, the average occupancy of hydrogen bonds between the
 250 collagen and cellulose in the four systems was calculated respectively. As shown in Fig. 8, the values of
 251 hydrogen bonding occupancies are all lower than 12%, which indicates that hydrogen bonding interaction
 252 is not the predominant force driving the binding of collagen and cellulose. (010) face of cellulose exhibits
 253 the highest tendency to form hydrogen bonds with collagen, which is consistent with the outward
 254 orientation of its surface hydroxyl groups. The hydrophobic (100) facet is less inclined to be involved in
 255 hydrogen bonding interactions with hydroxyl groups mainly forming interchain hydrogen bonds. (110)
 256 and (1-10) facets have a median performance in hydrogen bonding interactions. Collagen adsorption may

257 be mainly driven by the dispersion interaction between the collagen and cellulose surface.



258

259 **Fig. 8.** Hydrogen bond occupancy between collagen and cellulose crystalline faces.

260 **4. Conclusion**

261 In this study, molecular dynamics simulations were carried out to investigate the adsorption
262 behavior of collagen on the ideal nanocelluloses surface. It has been observed that the structural integrity
263 of collagen has been maintained in the process of adsorption, which may shed light on the
264 biocompatibility of cellulose. (110) and (1-10) crystal faces exhibited the strongest affinity with collagen,
265 which was attributed to the combination of hydrophilicity and roughness. Hydrogen bonding interactions
266 are not frequent and are not the predominant force driving the binding of collagen and nanocellulose.
267 This study provides theoretical guidance for the design and fabrication of collagen-nanocellulose
268 composites. Furthermore, the intactness of collagen structure supported the viewpoint that nanocellulose
269 is quite biocompatible.

270 **Acknowledgments**

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