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## Article

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# Chirality Affecting Reaction Dynamics of HgS Nanostructures Simultaneously Visualized in Real and Reciprocal Space

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**ABSTRACT:**

Chirality involved reactions enable to probe features in the fields of asymmetric synthesis and catalysis, which allow to gain insight into the fundamental mechanisms of topochemically controlled reactions. However, in situ observation of the chirality-associated reaction dynamics with simultaneous structural determination of new features has been lacking. Here, we report the direct visualization of the electron-beam-stimulated reaction dynamics of HgS nanostructures with chiral and achiral morphologies simultaneously in both real and reciprocal space. Under the electron-beam excitation of HgS nanostructures, the formation and evaporation dynamics of Hg nanodroplets were vividly pictured while the reciprocal space imaging revealed the structural transformation from monocrystalline to polycrystalline. Such induced changes were size-dependent, which were slowed down when involving the chirality in the nanostructures. The finding offers a fundamental understanding of topochemically controlled reaction mechanisms and holds promise of tuning asymmetric synthesis for catalysis related applications.

**KEYWORDS:** redox reactions; chirality; structural dynamics; inorganic chiral nanostructures; in-situ observation

Chirality is usually used to describe the geometric property of handedness that is not superimposable on its mirror image by any combination of translations and rotations. Chiral molecules exist in two stereoisomers (enantiomers) that are mirror images of each other, showing either left-handed or right-handed form by their configuration. Generally, both the left- and right-handed forms present similar chemical and physical properties, except when they are utilized in particular cases including asymmetric catalysis<sup>1,2</sup>, optical activities<sup>3</sup> and biological processes<sup>4-6</sup>. Chirality occurs commonly in organic and biological molecules/compounds<sup>7-10</sup>. Nevertheless, inorganic chiral entities possess distinctive characteristics such as the facile control of crystal structures and the tolerance to high-temperature stresses, which exhibit great promise for applications in chiral sensing<sup>11-13</sup>, advanced optical devices<sup>14-21</sup>, and asymmetric catalysis<sup>22-25</sup>.

Among the 230 types of space groups inorganic chiral nanocrystals belong to the 65 Sohncke space groups<sup>26,27</sup>, where the group is non-superimposable with its mirror image enantiomorph. Chirality in these systems exists at all length scales, from the atomic/molecular level (lattice chirality), up to the macroscopic scale (morphology chirality). Inspired by the chirality in nature and life, most studies of inorganic chiral nanocrystals have been devoted to the synthesis strategies in the past years<sup>28-36</sup>. Typical applications of chiral crystals, promising in optical devices and photocatalysis, require the involvement of the interaction with photons/electrons, such as irradiation. However, the reports of chiral effect on the irradiation dynamics are rare.

HgS, chiefly used as a pigment in paints throughout history, occurs as a red (cinnabar,  $\alpha$ -HgS, hexagonal) or black (metacinnabar,  $\beta$ -HgS, cubic) crystalline solid.  $\alpha$ -HgS belongs to the chiral space group  $P3_121$  or  $P3_221$ , showing both the lattice and morphology chirality<sup>34</sup> that would be ideal for studying the irradiation-induced chemical reaction dynamics. Under pulsed laser ablation with the

wavelengths from ultraviolet to near infrared, the phase transformation from  $\alpha$ - to  $\beta$ -HgS was observed by the micro-Raman spectroscopy and X-ray photoelectron spectroscopy methods<sup>37</sup>. Using the pump-probe microscopy, both the  $\beta$ -HgS and metallic liquid Hg were created by the ultraviolet light exposure of  $\alpha$ -HgS<sup>38</sup>. However, in situ observation of such irradiation-induced chemical reaction process as well as its subsequent structural evolution in space and time has been lacking. Involving the chiral effect on the e-beam irradiation stability have been reported in multiwalled carbon<sup>39</sup> and boron nitride<sup>40</sup> nanotubes. Revealing simultaneously the chemical reaction dynamics in both real and reciprocal space, especially when involving chiral effect, is essential to provide an unambiguous understanding of chirality mediated reactions and phase transitions, which paves the way for controlling the chirality-associated plasmonics and catalytic reactions.

Here, we report the direct observation of chirality-related chemical reaction dynamics of inorganic HgS nanostructures by using *in situ* transmission electron microscopy (TEM) simultaneously in real and reciprocal space. The technique enables the visualization of the formation and evaporation dynamics of liquid Hg nanodroplets as a result of the e-beam-stimulated chemical reduction reactions of  $\text{Hg}^{2+}$  to  $\text{Hg}^0$ , and allows the simultaneous determination of the structural transition from monocrystalline to polycrystalline. Interestingly and surprisingly, the HgS nanostructures, once involving chirality, present distinct reaction behavior under the e-beam excitation. The chiral nanostructures show dense surface steps, which are more tolerant against the e-beam irradiation than that of the achiral ones. Such e-beam-stimulated reaction behavior is size-dependent, namely, the tolerance reaches the highest as the size of the nanostructure is around 200 nm and then becomes weakened as the size increases. The chiral effect on the above phenomena is discussed.

## Results

**Synthesis and characterization of HgS nanostructures.** The inorganic HgS nanostructures were synthesized using the seed-mediated epitaxial method with the growth rate controlled by a syringe pump (Fig. 1, top; see Methods, the left corner is an example of the as-synthesized product). The HgS seeds had a size of ~8-15 nm (Supplementary Fig. 2), which were used for the subsequent epitaxial growth of various nanostructures. The nanostructures can crystallize in the chiral space group  $P3_121$  or its enantiomorph  $P3_221$ . Shown in Fig. 1 (middle left) is the atomic model of HgS with the  $P3_121$  space group<sup>41</sup>. The helical arrangement of Hg and S atoms along the crystallographic  $c$  axis represents an atomic scale primary chiral unit (Fig. 1, middle center). The chiral unit acts as a template for the epitaxial growth of chiral HgS nanostructures in the presence of chiral molecules (Fig. 1, middle right). Fig. 1 (bottom left) shows a typical TEM image of the as-synthesized HgS nanostructures. The HgS displays twisted bipyramid morphology with the geometric left-handedness<sup>34</sup>, which has a width and length of ~150, 400 nm in size, respectively. Detailed growth conditions of HgS with different sizes are listed in Supplementary Table 2 and the corresponding morphologies are presented in Supplementary Fig. 3. The energy dispersive X-ray (EDX) spectrum and mapping results indicate the existence of the Hg and S elements in the nanostructure, and show no obvious aggregation of clusters before long-term electron beam (e-beam) illumination (Fig. 1, bottom center and right).

**Experimental implementation of reaction dynamics studies.** We investigated the chemical reaction dynamics of the HgS nanostructures with chiral and achiral morphologies in both real and reciprocal space using the in situ TEM (Fig. 2, left). An e-beam is employed to excite the specimen so that chemical reactions are triggered at targeted locations. The subsequent reaction dynamics are recorded simultaneously in real (structural imaging) and reciprocal (phase determination) space by a digital

camera. Fig. 2 (bottom right) shows the morphologies of the chiral HgS before (0 s) and after 300 s e-beam irradiation, respectively. In comparison with the initial one, the nanostructure after the irradiation presents significant changes of contrast and the occurrence of many dispersed nanoclusters with dark contrast (marked by the red arrow).

High resolution TEM image reveals that the nanostructure after the e-beam irradiation was still  $\alpha$ -HgS phase, without the transformation to  $\beta$  phase (Supplementary Fig. 4), which is different from that observed in the samples after the laser exposure<sup>37</sup>. It indicates that the behavior in this work might be associated with the chemical reactions. Because chemical species exhibit quite different mass loss rate, the overall chemical composition of the sample may change during the e-beam irradiation. Upon comparison of the EDX mapping results between the two states (0 and 300 s), the losses of both Hg and S were clearly observed after the irradiation (Supplementary Fig. 5). The EDX spectra show that the intensities from both Hg and S elements decreased, where the ratio of Hg:S quantitatively changed from  $\sim 1:1$  of the initial stoichiometric sample to  $\sim 1:2$  of the e-beam-irradiated one (Fig. 2, upper right). To understand the above phenomena, the reaction dynamical processes were in situ revealed in the following sections.

**Reaction dynamics of chiral HgS nanobipyramids.** The structural evolution dynamics of a single chiral HgS nanobipyramid were investigated during the e-beam exposure, as shown in Fig. 3 (Supplementary Movie 1). The nanobipyramid is  $\sim 60, 120$  nm in width and length, respectively. For convenience, we neglected the time of the TEM alignment, and denoted the initial image as time zero (0 s). The image and its live fast Fourier transform (FFT) pattern indicate that the as-synthesized HgS is monocrystalline, corresponding to the trigonal structure ( $a= 4.145 \text{ \AA}$ ,  $b= 4.145 \text{ \AA}$  and  $c=9.496 \text{ \AA}$ )<sup>42</sup>. The direction along the length of the nanobipyramid is parallel to the [001] orientation. After the e-

beam excitation for 50 s, several Hg nanodroplets ( $\sim 8$  nm in size) were seen (marked by the dotted circles). The occurrence of the Hg nanodroplets might be due to the e-beam-induced chemical reaction of HgS. The nanodroplets gradually decreased their size during the e-beam irradiation, and finally vanished in about 31 s (Supplementary Fig. 6). With the prolonged irradiation (120 and 180 s), the nanovoids gradually grew and coalesced into the big ones. In comparison with the initial state, the diffraction spots became less and the intensity was weaker, indicating a lower symmetry of the HgS nanostructure. As the e-beam irradiation continued (240 s), the HgS nanostructure was severely destroyed and only several diffraction spots were still visible. Eventually, a hollow nanostructure with the twisting shell layer remained on the supporting film after the 300 s e-beam irradiation.

**Size-dependent effect on reaction dynamics of chiral HgS nanobipyramids.** To understand the formation of the Hg nanodroplets, we further studied the HgS nanobipyramids with different sizes under the e-beam excitation. Fig. 4 shows the tendency of the stability of HgS nanobipyramids with respect to their sizes (length ranged from 80 to 400 nm; an example is displayed in the inset) under the same e-beam irradiation condition. The period from the initial state to the one that the morphology and corresponding FFT pattern of the nanostructure without change obviously during the irradiation was referred to the total reaction time in the figure. With increasing the size of the nanobipyramid, the total reaction time increased to  $\sim 1400$  s for the nanobipyramid with the size of  $\sim 200$  nm, and then decreased to  $\sim 400$  s for the  $\sim 400$  nm one (Fig. 4, top). Four typical HgS nanobipyramids with the different sizes ( $\sim 120$ , 200, 240 and 340 nm) are displayed in the bottom panel of the figure, corresponding to the dotted circles marked in the top panel. Under the identical e-beam irradiation for 240 s, no obvious changes of the morphology and the corresponding FFT pattern were observed in the  $\sim 200$  nm nanobipyramid. In contrast, significant changes were seen in the nanobipyramids with other sizes. For

example, the hollow nanostructure was already formed in the  $\sim 120$  nm nanobipyramid. For the large ones, a number of nanovoids appeared in the nanobipyramids with the sizes of  $\sim 240$  and  $340$  nm. It suggests that the  $\sim 200$  nm HgS nanobipyramid is more persistent to the e-beam irradiation than that with other sizes, resulting in the distinct chemical reaction rates observed above. It is noteworthy from the FFT pattern that the  $\sim 340$  nm nanobipyramid presents obvious transformation from monocrystalline to polycrystalline after the irradiation for the same time. For details, the reaction characteristics of HgS with other sizes were displayed in Supplementary Fig. 7-16.

**Reaction dynamics of achiral HgS nanobipyramid.** To further understand the chiral effect on the reaction dynamics, the e-beam irradiation of the achiral nanostructures was performed for the comparison. Fig. 5 shows the structural evolution of HgS nanobipyramid with chiral (top left) and achiral (top right) morphologies under the same e-beam excitation condition. The initial nanobipyramid is approximately  $280$  nm in size. When the irradiation time increased, the chiral HgS showed no obvious morphological changes and the diffraction spots with monocrystalline form were readily seen (0-20 s). However, significant changes were observed in the achiral HgS, i.e., several Hg nanodroplets with the range of  $3-8$  nm in diameter were seen (marked by the dotted circles) after 20 s e-beam irradiation. It indicates that the achiral HgS nanostructures were less tolerance to the e-beam irradiation, in opposite to that observed in the chiral cases. Note that the evaporation time of the Hg nanodroplets for the achiral nanobipyramid was about  $12$  s (Supplementary Fig. 17), which was shorter than that for the chiral one ( $\sim 31$  s). The atomic force microscopy (AFM) results show that the surface of the chiral nanobipyramid is rougher than that of the achiral one (Fig. 5, bottom). Moreover, the surface step heights of the chiral and achiral nanobipyramids are estimated to be  $\sim 10$  and  $20$  nm, respectively, indicating the denser surface steps on the chiral samples (Supplementary Fig. 18). It

suggests that the chirality-affected chemical reaction dynamics might be associated with the surface characteristics of the nanostructures.

## Discussion

Generally, sample damage caused by the e-beam can be through either elastic scattering such as knock-on atomic displacements or inelastic scattering including specimen heating and radiolysis. It has been reported that continuous ultraviolet light exposure can generate Hg droplets through local heating of HgS<sup>38</sup>. Besides, for pure  $\alpha$ -HgS sample, the phase transformation to  $\beta$ -HgS occurs at  $\sim 673$  K and is completed at  $\sim 698$  K<sup>43</sup>. Under the e-beam irradiation shown here, however, the e-beam-induced temperature rise is no more than 1 K<sup>44</sup> (see detailed discussion in Supplementary Information including Supplementary Fig. 1 and Table 1). This suggests that heating is impossible to cause phase transformation and has negligible effect on the formation of Hg nanodroplets.

It is seen from Figure 2 (top right) that the loss of Hg (higher atomic mass) is faster than that of S after the e-beam irradiation, suggesting that the knock-on mechanism is not the major factor for the sample damage. With the exposure to high-energy e-beam, the ionizing radiation (radiolysis) would cause the breakage of chemical bonds in materials. Based on these considerations and the experimental results, a pictorial understanding of the e-beam-induced chemical reactions in HgS system is illustrated in Fig. 6 (top). Under the e-beam stimulus, the cleavage of Hg-S chemical bonds and the creation of vacancies readily occur, which may induce the reactions of the reduction of Hg<sup>2+</sup> to Hg<sup>0</sup> and the oxidization of S<sup>2-</sup> to S<sup>0</sup>. The Hg<sup>0</sup> atoms diffuse and aggregate so that the Hg nanodroplets nucleate and grow. As seen from the FFT patterns (Supplementary Fig. 7-16.), a lower symmetry of the HgS nanostructure appears at this stage (Supplementary Movie 2). Owing to the subsequent evaporation of

many Hg nanodroplets, a number of nanovoids emerge, which divide the original single crystal into many pieces of small polycrystals. As the Hg nanodroplets continue to move and evaporate, the nano-sized polycrystals would gradually rotate and change their orientations, giving rise to the FFT ring patterns (structural transformation from monocrystalline to polycrystalline) observed in the later stage of the e-beam irradiation. Finally, the nanostructures are disordered during the long-term e-beam irradiation, resulting in the amorphization of the most parts of the specimen.

In general, the tolerance against the e-beam irradiation increase/decrease linearly with respect to the size of nanostructured materials<sup>45-47</sup>. The size-dependent chemical reaction dynamics observed in this work, different from general cases, may involve the interplay between the surface energy of steps and the free energy of irradiation-induced defects (e.g., vacancies). During the e-beam bombardment, such generated point defects would diffuse and be absorbed to the surface steps of HgS nanobipyramid, and finally annihilate there. The excess free energy  $\Delta G_d$  of these point defects is proportional to the square of nanobipyramid size  $L$ , i.e.,  $\Delta G_d \propto L^2$ , while the free energy change of surface steps  $\Delta G_s$  is described as  $\Delta G_s \propto 1/L$ <sup>48</sup>. Thus, the change in Gibbs free energy of the irradiated HgS nanobipyramid is the sum  $\Delta G_{sum}$  of the two, namely,  $\Delta G_s + \Delta G_d$ , as shown in Fig. 6 (bottom left). When the crystal size is in the range of  $L < L_1$  or  $L > L_2$ , the Gibbs free energy barrier of the transition satisfies  $\Delta G_t < \Delta G_{sum}$ , indicating that the transition (a change in crystal structure or amorphization) would be easily induced by the e-beam irradiation. Our experimental results of HgS nanobipyramids with the sizes of 80~160 nm or 240~400 nm follow the criteria in this region. On the other hand, it becomes harder for the transition when  $L_1 < L < L_2$  since  $\Delta G_t > \Delta G_{sum}$ . As observed in Fig. 4, the high irradiation tolerance occurred in the HgS nanobipyramids with the sizes of 180~220 nm provides the information of size-dependent stability in HgS nanostructured system during the irradiation.

Chirality has a significant role in the surface morphology of HgS nanobipyramids, resulting in the distinct dynamics observed during the e-beam stimulus (Fig. 5). Upon comparison of the phenomena in the chiral and achiral HgS, it suggests that the chiral morphology with high density of surface steps is more resistant to the e-beam irradiation than the achiral one. The great quantity of the surface steps would function as the sinks for the removal of irradiation-induced defects. Meanwhile, the wetting process on the nanobipyramid surface would determine the nucleation of Hg nanodroplets as a result of the chemical reactions induced by the e-beam excitation. The formation of Hg nanodroplets depends on the interplay of several contributions (Fig. 6, bottom center) to the Gibbs free energy that are expressed as<sup>49</sup>,

$$\Delta G = G_{vol} + G_{surface} \quad (1)$$

$$G_{vol} = g_{vol}V \quad (2)$$

where  $G_{vol}$  is the volume free condensation energy,  $G_{surface}$  the surface energy,  $g_{vol}$  the free energy per unit volume for Hg ( $-13.43 \text{ eV/nm}^3$ )<sup>50,51</sup> and  $V$  the volume of Hg nanodroplet. The surface energy  $G_{surface}$  involves the contributions from the surface tensions and the contact areas of solid-liquid and liquid-vacuum interfaces. According to the Young's equation, the interfacial tensions are expressed with the contact angle as,

$$\cos \theta = (\gamma_{sv} - \gamma_{sl})/\gamma_{lv} \quad (3)$$

where  $\gamma_{lv}$ ,  $\gamma_{sl}$  and  $\gamma_{sv}$  are the interfacial tensions of liquid–vacuum, solid-liquid and solid-vacuum, respectively. The exact value of the volume free condensation energy and surface energy depend on the contact angle  $\theta$ . Consequently, the size of the critical nucleus  $R^*$  could be determined by  $G_{vol}$  and  $G_{surface}$  energies, which is varied with the change of contact angle  $\theta$  (Fig. 6, bottom right). Assuming the classical cap shape for the nanodroplets, the expressions of the critical radius and the

critical total free energy change  $\Delta G^*$  for their heterogeneous nucleation are described as follows (see detailed calculations in Supplementary Information),

$$R^* = -\frac{2\gamma_{lv}}{g_{vol}} \sin\theta \quad (4)$$

$$\Delta G^* = \frac{4\pi\beta\gamma_{lv}^3}{3(g_{vol})^2} \quad (5)$$

$$\beta = (2 + \cos\theta)(1 - \cos\theta)^2 \quad (6)$$

$$J = J_0 \exp(-\Delta G^*/k_B T) \quad (7)$$

Substituting the interfacial tension  $\gamma_{lv}$  value (0.486 N/m)<sup>52</sup> into eq. (4), the critical radius and total free energy for the Hg nanodroplet nucleation on the HgS surface was determined with respect to the contact angle  $\theta$ , as shown in Fig. 6 (bottom right) and Supplementary Fig. 6. The critical total free energy  $\Delta G^*$  and nucleation rate  $J$  for heterogeneous nucleation of Hg nanodroplets depend on the intrinsic wettability of the surface via the contact angle. The contact angle could be obtained from the evaporation dynamics of the Hg nanodroplets. The evaporation time  $\tau$  is given by<sup>53</sup>

$$\tau = \frac{1}{4} \left(\frac{3}{\pi}\right)^{2/3} \frac{V_{L0}^{2/3}}{D} \frac{k_B T \beta^{1/3}}{P_0 V_m f} \frac{C_f}{e^{\lambda/R} - 1 + x} \quad (8)$$

$$P_V = (1 - x)P_0 \quad (9)$$

$$f = 0.5(0.00008957 + 0.6333\theta + 0.116\theta^2 - 0.08878\theta^3 + 0.01033\theta^4) \quad (10)$$

where  $J_0$  is the kinetic constant,  $V_{L0}$  the initial volume of the nanodroplet with a contact diameter  $R$ ,  $V_m$  the Hg atom volume in the liquid phase ( $2.48 \times 10^{-2}$  nm<sup>3</sup>),  $k_B$  the Boltzmann's constant,  $T$  the temperature (25 °C),  $D$  the diffusion constant ( $14.2 \times 10^{-6}$  m<sup>2</sup>/s)<sup>54</sup>,  $P_0$  the saturated vapor pressure ( $1.6 \times 10^{-4}$  Pa)<sup>55</sup>,  $\theta$  the contact angle in radians,  $\lambda$  the characteristic length scale for Hg liquid drops (5.81 nm),  $C_f$  the factor describing the nanodroplet evaporation in the nanoscale<sup>56,57</sup>,  $P_V$  the external vapor pressure. From eq. (9) the value of  $x$  was calculated to be 0.99 by knowing that the external vapor pressure was  $\sim 10^{-7}$  Pa inside the TEM chamber. From the total evaporation time of the Hg

nanodroplets on the chiral ( $\sim 31$  s, Fig. 3 and Supplementary Fig. 6) and achiral ( $\sim 12$  s, Supplementary Fig. 17) HgS surfaces, the contact angles were estimated from eq. (8-10) to be  $\sim 126^\circ$  and  $\sim 66^\circ$ , respectively. At these contact angles, the total free energy of the Hg nanodroplet was 2.31 eV for the chiral morphology while it decreased to 0.55 eV for the achiral one. An estimate of the nucleation rate on a surface with contact angle of  $\sim 66^\circ$  about  $10^{29}$  times higher than that of  $\sim 126^\circ$  surface from eq. (7). This indicates that it became easier for the Hg nanodroplet nucleation on the achiral HgS surface than that on the chiral one. Our experimental observations agree with the tendency reported in the water droplet system that the calculated nucleation rate on its surface with a contact angle of  $\sim 25^\circ$  was considerably higher than that of  $\sim 110^\circ$  surface<sup>58</sup>. In the extreme case, when the contact angle is zero (condition for liquid spreading), the total free energy change  $\Delta G$  equals zero, suggesting no energy barrier for the heterogeneous nucleation. Thus, it becomes relatively hard to trigger the chemical reactions in the chiral HgS nanostructures (large contact angles) under the e-beam stimulus, as observed experimentally in Fig. 5.

In conclusion, the real-time observations of the e-beam-stimulated reaction dynamics involved in the chiral and achiral HgS nanobipyramids, together with the simultaneous identification of their structural transformation, reveal the nature of size-dependent chiral effect on the nucleation and evaporation dynamics of the reaction-induced liquid nanodroplets. With such in situ real and reciprocal space visualization, we obtain the evaporation rate of the Hg nanodroplets for both the chiral and achiral system, facilitating the quantitative understanding of the chirality associated reaction dynamics of inorganic enantiomers in terms of the free energies accounting for the nucleation and evaporation processes. This approach has the potential for unique applications in chemistry and materials science, providing the key parameters accounting for the transient behaviour of topochemically controlled

reactions at the nanoscale.

## **Methods**

### **Synthesis of $\alpha$ -HgS seeds**

In a solution-based synthesis, 70.0 mg of  $\text{Hg}(\text{NO}_3)_2 \cdot \text{H}_2\text{O}$  and 10.0 mL of deionized water were mixed in a 25 mL glass vial, followed by injection of 2.0 mL of aqueous solution containing 26.9 mg of D-penicillamine (or L-penicillamine or equal mixture of D- and L- penicillamine). Then, a 0.6 mL of aqueous solution containing 24.0 mg of NaOH was added into the vial under magnetic stirring. With the dropwise addition of 2.0 mL of thioacetamide solution (0.1 M), the color of the mixture solution turned from colorless to dark brown. The reaction was proceeded in a 38 °C water bath for 15 h under stirring. The product was collected by centrifugation at 6,000 rpm for 10 min, washed three times using isopropanol, and finally dispersed in 5.0 mL of deionized water.

### **Seed mediated epitaxial growth of $\alpha$ -HgS with chiral and achiral morphology**

The epitaxial growth of  $\alpha$ -HgS with chiral morphology was performed using a syringe pump at the specific growth temperature (Figure 1, top). In a typical synthesis, 50  $\mu\text{L}$  of  $\alpha$ -HgS seeds colloidal solution and 5 mL water were added into a three-neck round-bottom flask submerged in the 38 °C water bath. 1.0 mL of aqueous solution containing 13.5 mg of D-penicillamine (or L-penicillamine or equal mixture of D- and L-penicillamine) and 0.6 mL of aqueous solution containing 24.0 mg of NaOH were successively added into the flask under magnetic stirring. Then, 6.0 mL of aqueous solution containing 74.4 mg of  $\text{Hg}(\text{NO}_3)_2 \cdot \text{H}_2\text{O}$  (Hg precursor solution) and 10 mL of aqueous solution containing 19 mg of thioacetamide (S precursor solution) were co-injected into the flask through the syringe pump with an injection rate of 1.0 mL/h. To obtain the chiral  $\alpha$ -HgS with different sizes, the

injection time was controlled from 2 to 6 h. After reaction, the obtained colloidal solution (orange color) was centrifuged at 6,000 rpm. The precipitates were washed three times using isopropanol, and finally dispersed in 5.0 mL of deionized water.

### **TEM sample preparation and characterization**

The TEM samples were prepared by drying the drops of the  $\alpha$ -HgS colloidal suspension on the carbon-coated copper grids under ambient conditions. To avoid the aggregation of the nanocrystals, the colloidal suspension was sonicated for about 10 minutes before dropping on the copper grids. The structural characterization of the samples were carried out using a ThermoFisher Talos F200X TEM with an accelerating voltage of 200 kV. The energy-dispersive X-ray detector (EDX) was used to conduct elemental analysis of the samples. As-synthesized HgS nanostructures were thoroughly characterized by atomic force microscope (AFM, FastScan Bio, Bruker, USA). AFM topography imaging and line profile analysis were used to reveal and measure the step height and thickness.

### ***In situ* visualization of structural dynamics**

*In situ* observations of the e-beam-induced structural dynamics of  $\alpha$ -HgS nanocrystals were performed in the TEM with a constant e-beam dose rate of  $700 \text{ e}/\text{\AA}^2\text{s}$ . We recorded simultaneously the structural evolution of  $\alpha$ -HgS nanocrystals and their corresponding fast Fourier transform (FFT) patterns by a digital camera. All the movies for each kind of specimen were recorded at the same condition. The EDX and high-angle annular dark-field (HAADF) imaging were employed to analyze the composition changes of the  $\alpha$ -HgS nanocrystals at different e-beam irradiation time.

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## Figure captions

**Figure 1. Synthesis and structural characterization of  $\alpha$ -HgS nanobipyramids.** Upper: Schematic of the seed-mediated epitaxial growth of chiral  $\alpha$ -HgS nanostructures. The inset shows the orange  $\alpha$ -HgS colloidal suspensions. Middle left: Schematic representation of the atomic structure of  $\alpha$ -HgS (Space group  $P3_121$ ). Middle center: Helical chains of Hg and S atoms along the crystallographic  $c$  axis. Middle right: Schematic illustration of the epitaxial evolution from the chiral lattice to the chiral morphology by involving the chiral molecules. Bottom left: Typical TEM image of  $\alpha$ -HgS with chiral morphology. The dotted lines are the twisting orientations in the nanobipyramids. Bottom center: Elemental mapping of Hg and S, respectively. Bottom right: EDX spectrum of  $\alpha$ -HgS. The inset is the corresponding high-angle annular dark field (HAADF) image for EDX.

**Figure 2. Experimental setup for in-situ observations of chemical reaction dynamics.** Left: Schematic of e-beam-induced chemical reaction dynamics visualized in both real and reciprocal space. The sample dynamics are triggered by an e-beam, while their structural evolution and corresponding diffraction patterns are simultaneously recorded by a charged-coupled device (CCD) camera. Upper

right: EDX spectra of  $\alpha$ -HgS before and after the 300 s e-beam irradiation. Bottom right: TEM images of  $\alpha$ -HgS before and after the 300 s e-beam irradiation. The arrow indicates the occurrence of nanoclusters due to the irradiation.

**Figure 3. Structural evolution in the chiral  $\alpha$ -HgS nanobipyramid.** Time-resolved TEM images and live FFT patterns show the formation and evaporation dynamics of Hg nanodroplets as well as the creation and coalescence of nanovoids. The initial HgS was monocrystalline (0 s). After the 50 s e-beam irradiation, the Hg nanodroplets with the diameter of  $\sim 6$  nm were observed (marked by the dotted circles). Such nanodroplets gradually shrank and finally vanished as the e-beam irradiation continued. Following the creation, growth and coalescence of nanovoids, the HgS nanobipyramid was gradually destroyed with the prolonged irradiation (120, 180 and 240 s). Eventually (300 s), a hollow nanostructure with the twisting shell layer was seen.

**Figure 4. Size-dependent effect on the reaction dynamics of chiral HgS.** Upper: Total reaction times of HgS nanobipyramids as a function of their sizes (length ranged from  $\sim 80$  to 400 nm) under the same e-beam excitation condition. The total reaction time firstly increased with the size, then reached the maximum at the size of  $\sim 200$  nm and after that decreased with increasing the crystal size. The inset illustrates the length of a nanobipyramid used for this work. Bottom: TEM images and corresponding FFT patterns of four typical HgS nanobipyramids with the different sizes ( $\sim 120, 200, 240$  and  $340$  nm), corresponding to the dotted circles marked in the top panel. These nanobipyramids were irradiated under the identical condition for 240 s. The comparison of such structural evolution indicates the distinct irradiation stability of HgS with different sizes.

**Figure 5. Chiral effect on the reaction dynamics of HgS.** Upper: TEM images and corresponding FFT patterns of HgS nanobipyramids (~280 nm) with chiral (left) and achiral (right) morphology under the 20 s e-beam irradiation. Several Hg nanodroplets with the diameter of 3-6 nm occurred in the achiral HgS (marked by the dotted circles), whereas no droplets were observed in the chiral one. Bottom: AFM topography images of HgS nanobipyramids with chiral (left) and achiral (right) morphology. The chiral surface was rougher than the achiral one.

**Figure 6. Mechanism of the e-beam-induced chemical reactions of HgS.** Upper: Schematic of the structural evolution of order-disorder transition upon the e-beam excitation. With the incoming high-energy electrons, the Hg-S chemical bonds are cleaved, and the reduction of  $\text{Hg}^{2+}$  to  $\text{Hg}^0$  as well as the oxidation of  $\text{S}^{2-}$  to  $\text{S}^0$  readily occur. The  $\text{Hg}^0$  atoms aggregate to nucleate the nanodroplets. The Hg nanodroplets grow and evaporate, and subsequently a number of nanovoids arise, leading to the formation of many nano-sized crystals. Such small crystals gradually rotate and change their orientations, giving rise to the ring-patterned polycrystalline character. With the continuous irradiation, the nanostructures are severely disordered so that most parts of the sample finally become amorphous. Bottom left: The free energy of surface steps ( $\Delta G_s$ ), defects creation ( $\Delta G_d$ ) and launch of structural transition ( $\Delta G_t$ ) in the irradiated HgS nanobipyramid as a function of the crystal size  $L$ . Bottom middle: Free energy diagram for nucleation with respect to the radius of the nucleus. Bottom right: Evaporation time and the total free energy of the Hg nanodroplets as a function of the contact angle  $\theta$ . The inset shows the schematic of the Hg nanodroplet on a solid surface.

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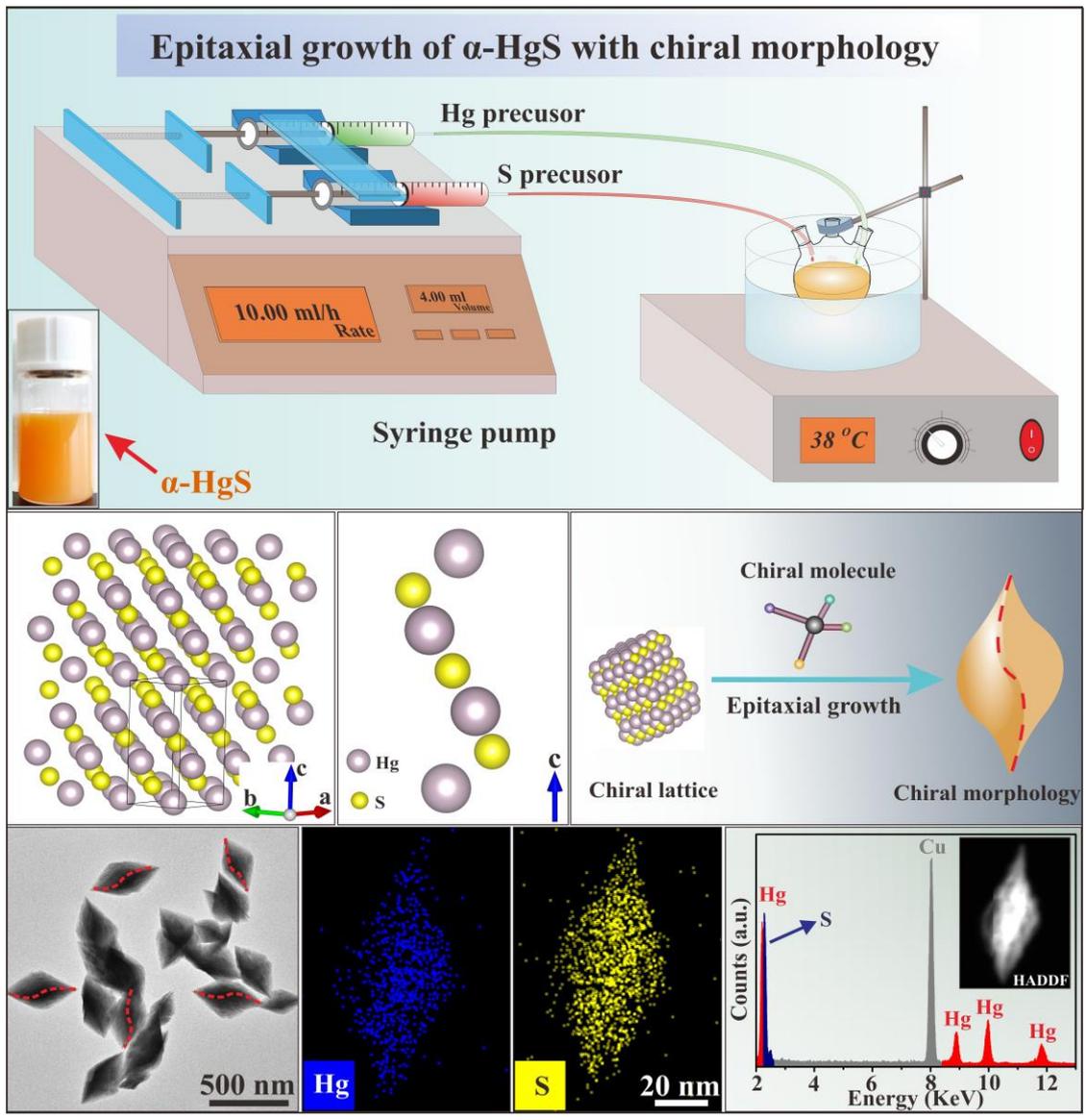
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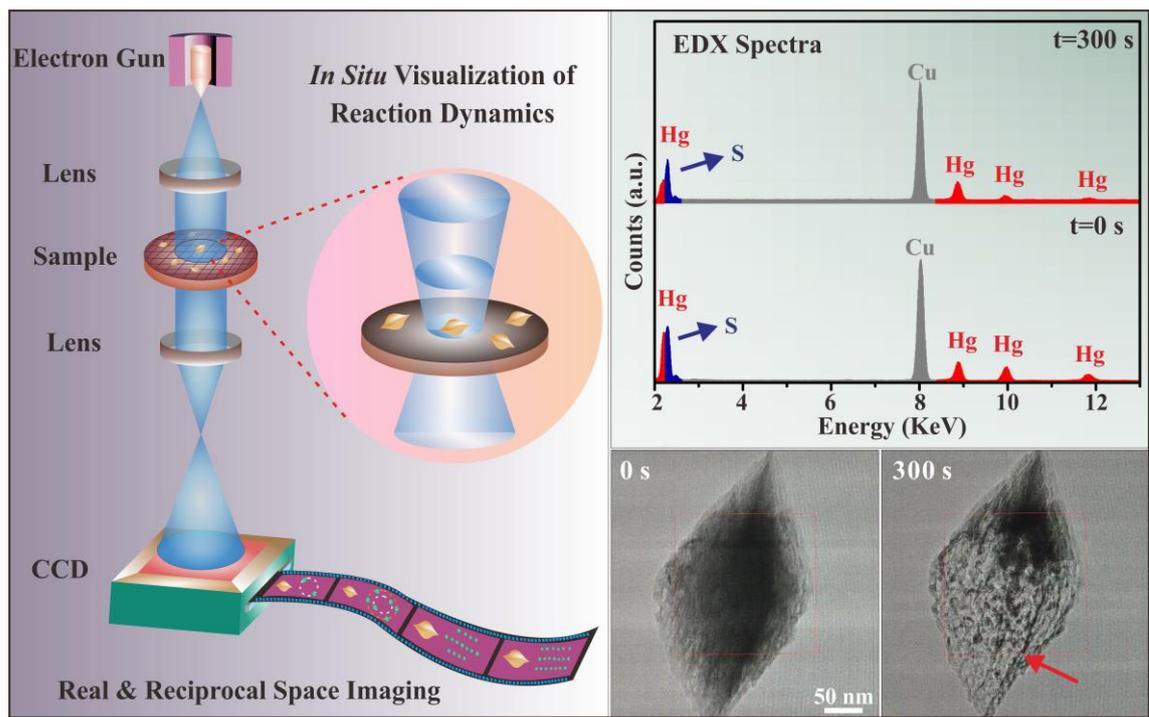
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**Figure 1**



**Figure 2**

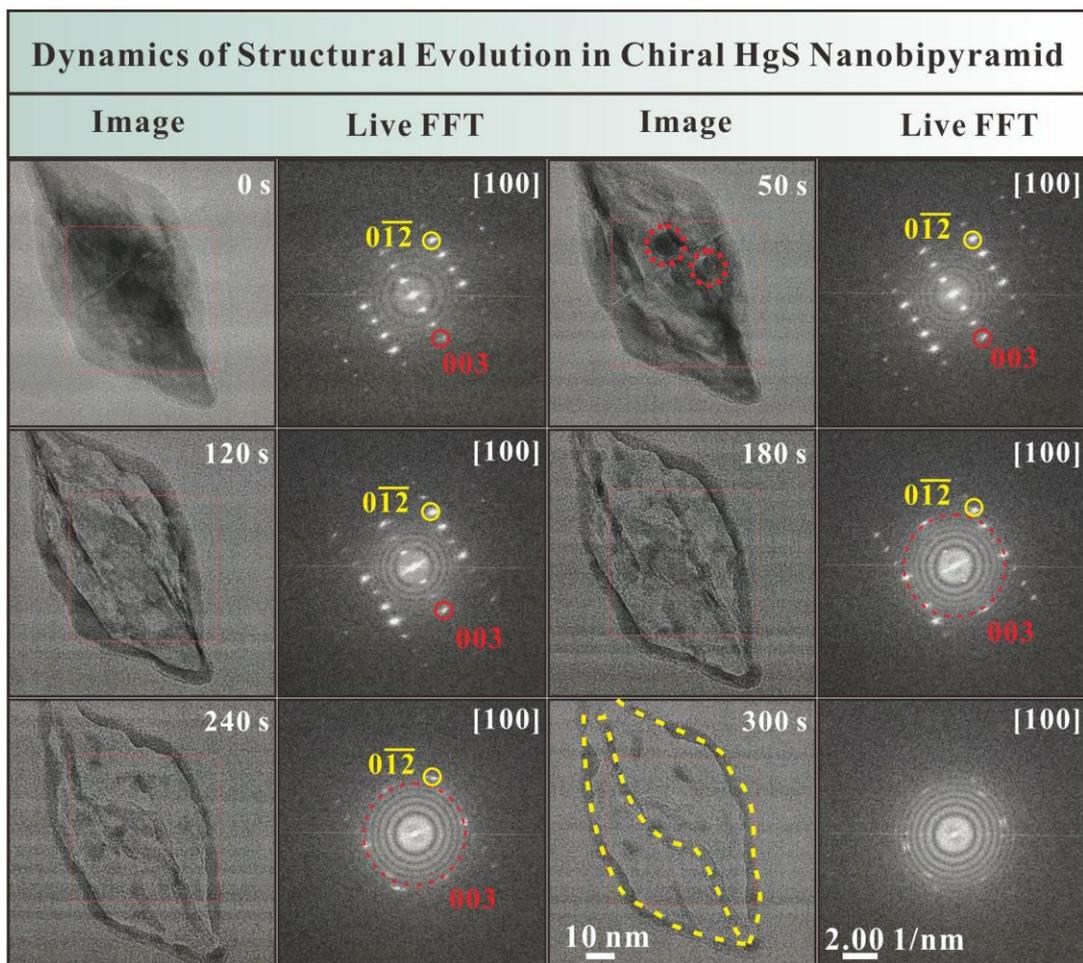
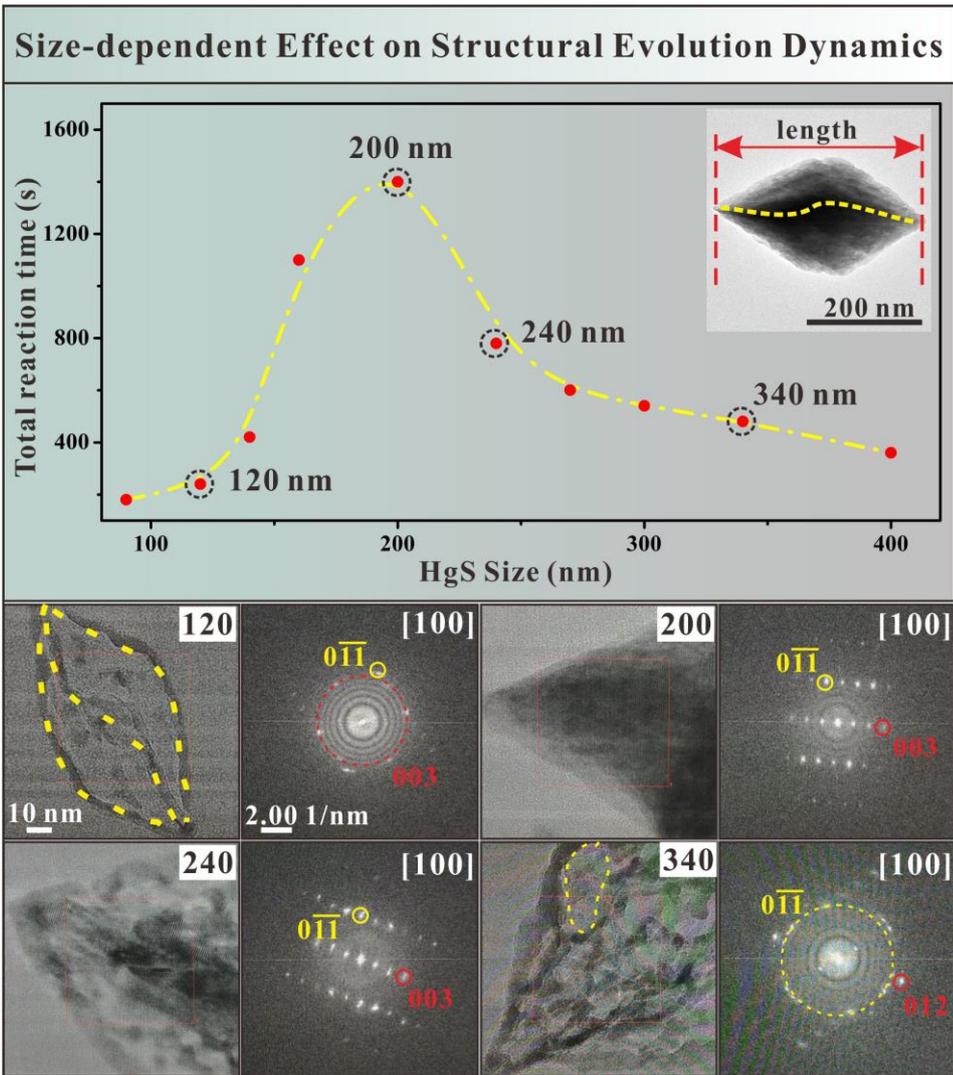


Figure 3



**Figure 4**

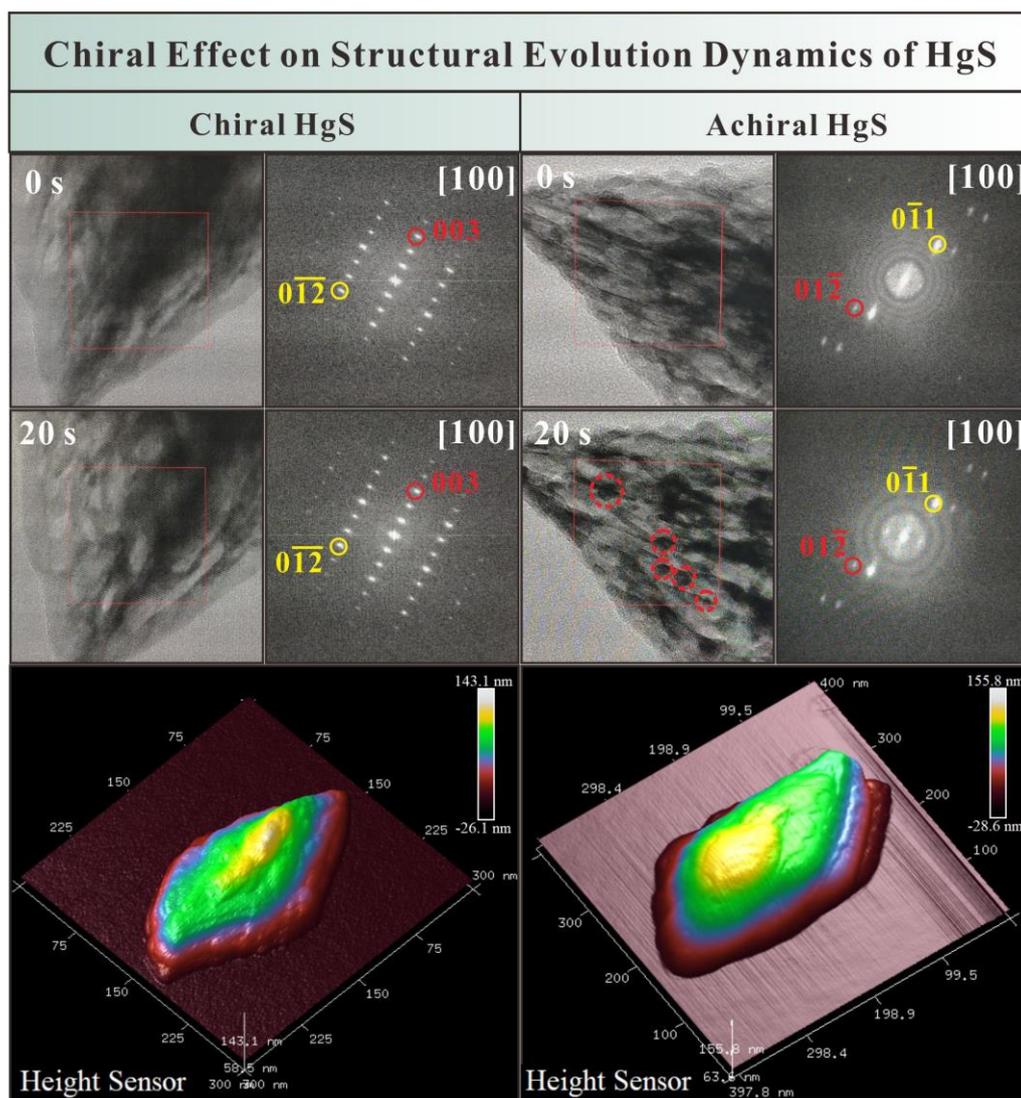
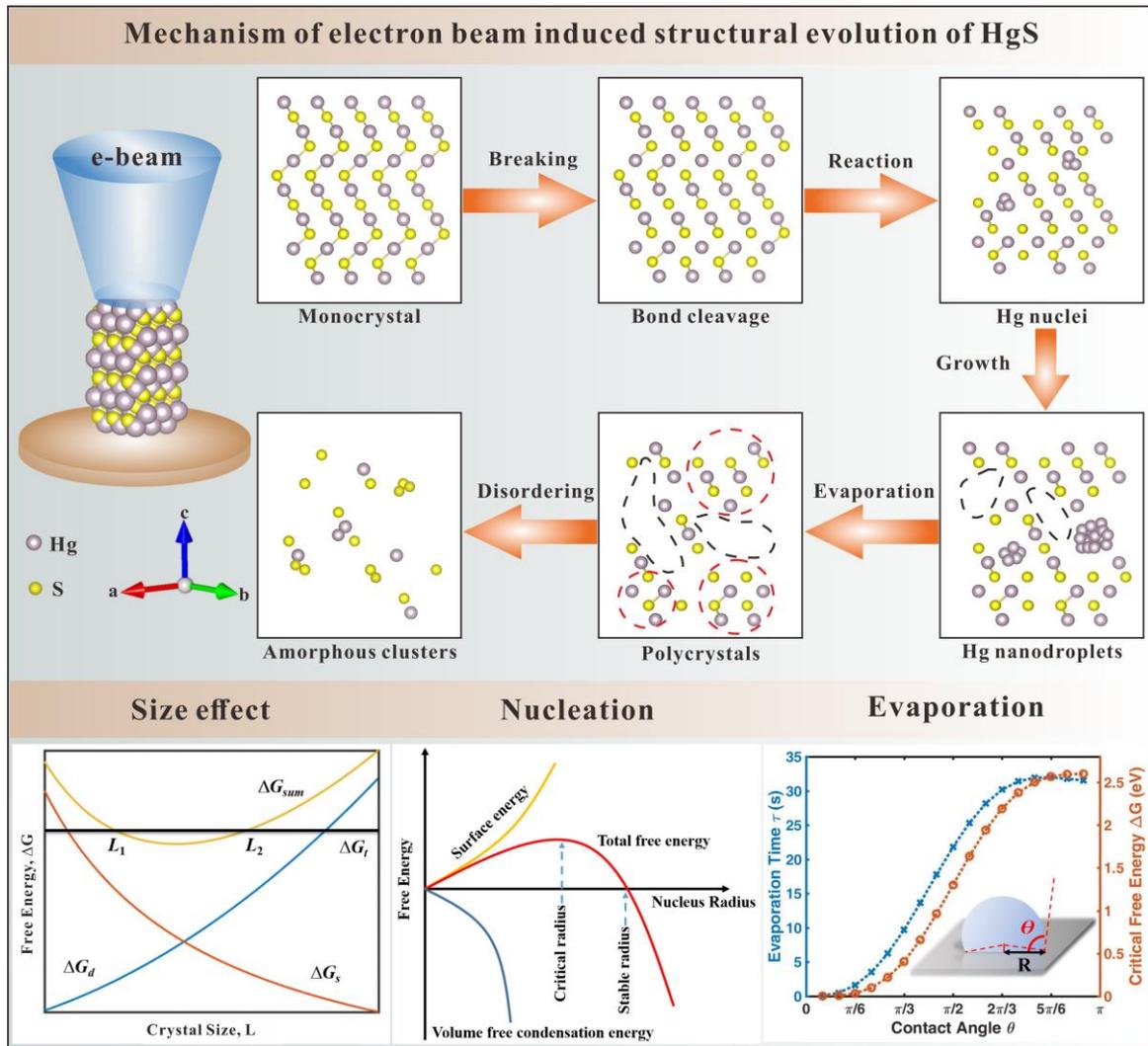


Figure 5



**Figure 6**

# Figures

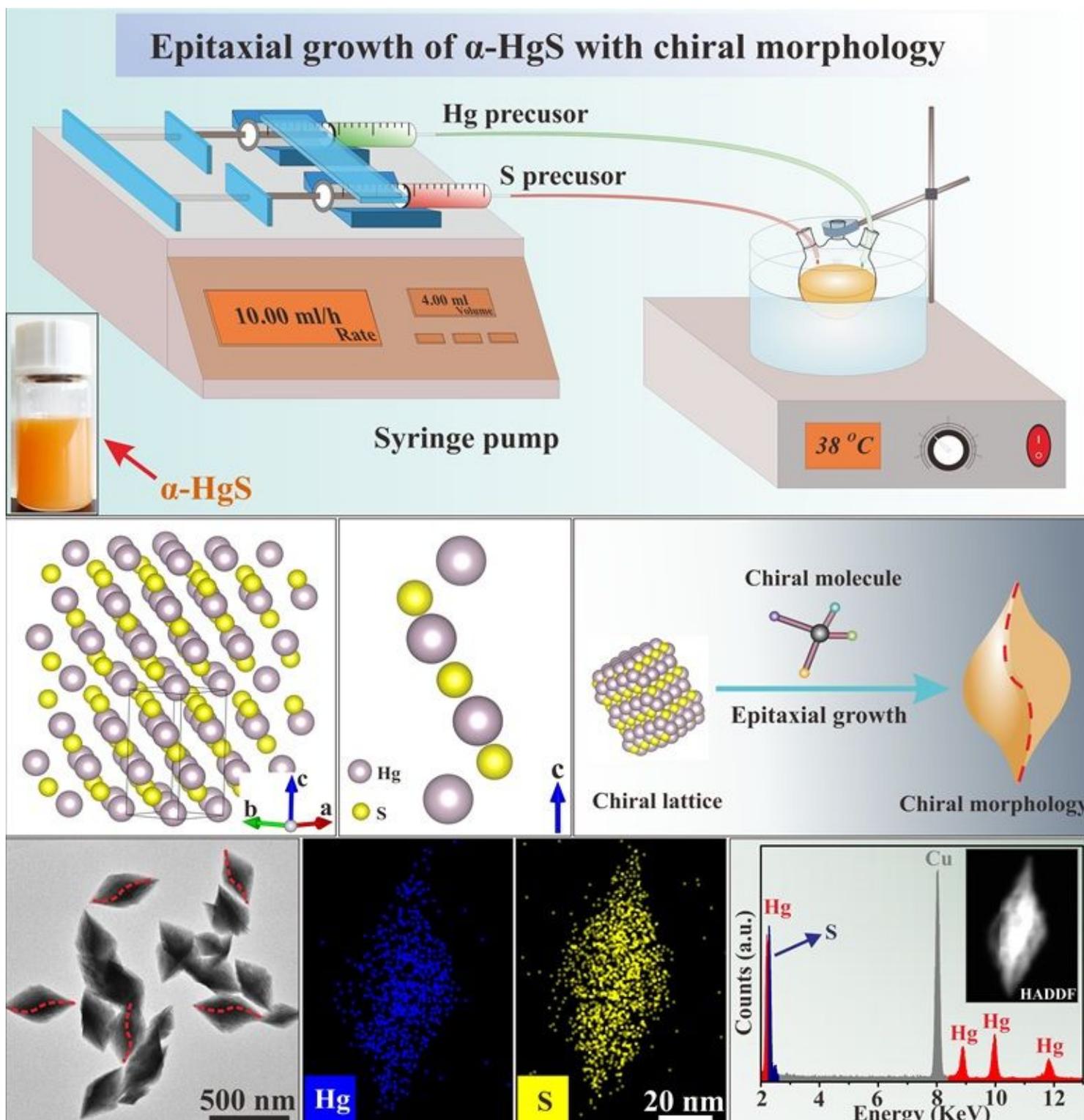
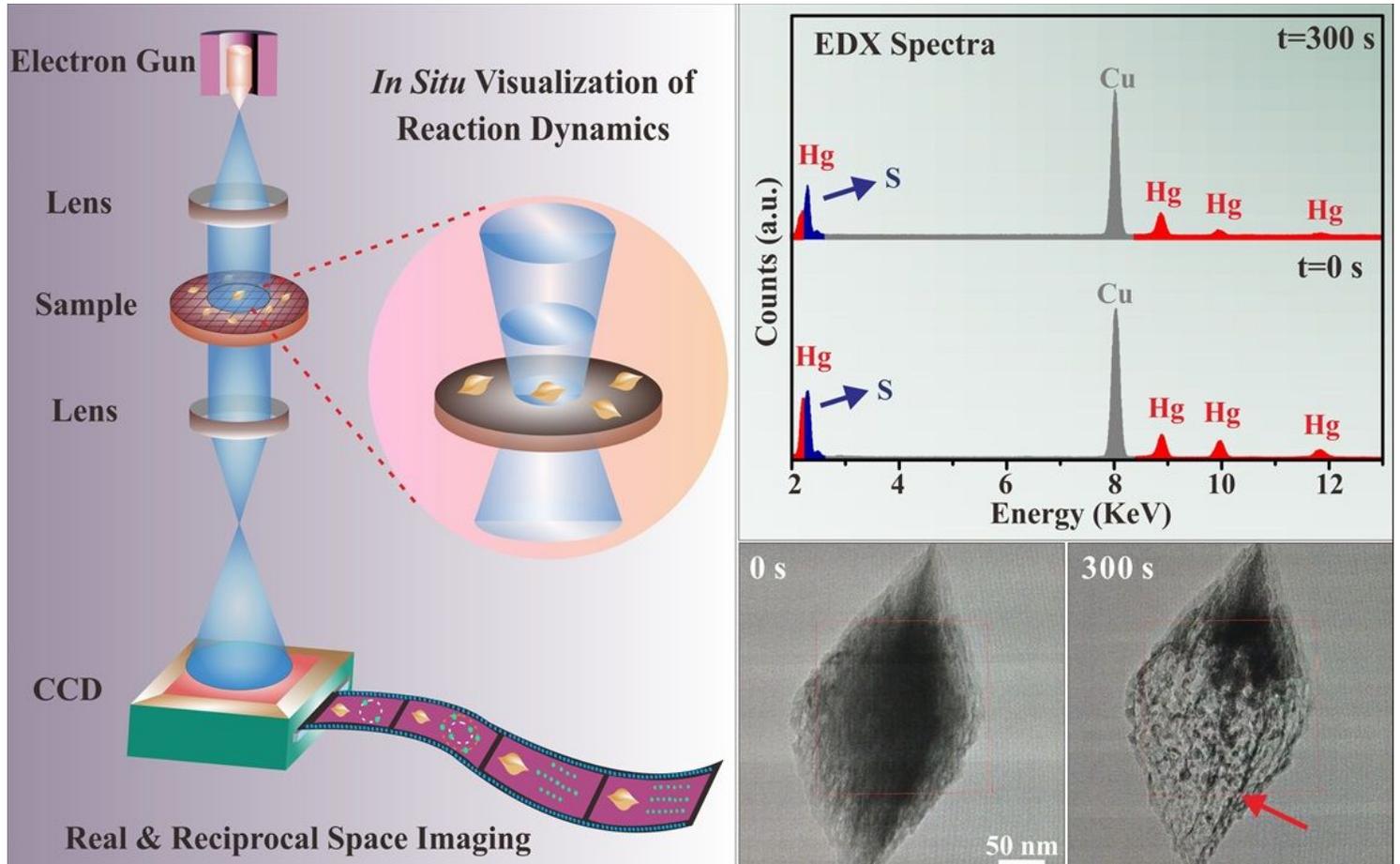


Figure 1

Synthesis and structural characterization of  $\alpha$ -HgS nanobipyramids. Upper: Schematic of the seed-mediated epitaxial growth of chiral  $\alpha$ -HgS nanostructures. The inset shows the orange  $\alpha$ -HgS colloidal suspensions. Middle left: Schematic representation of the atomic structure of  $\alpha$ -HgS (Space group

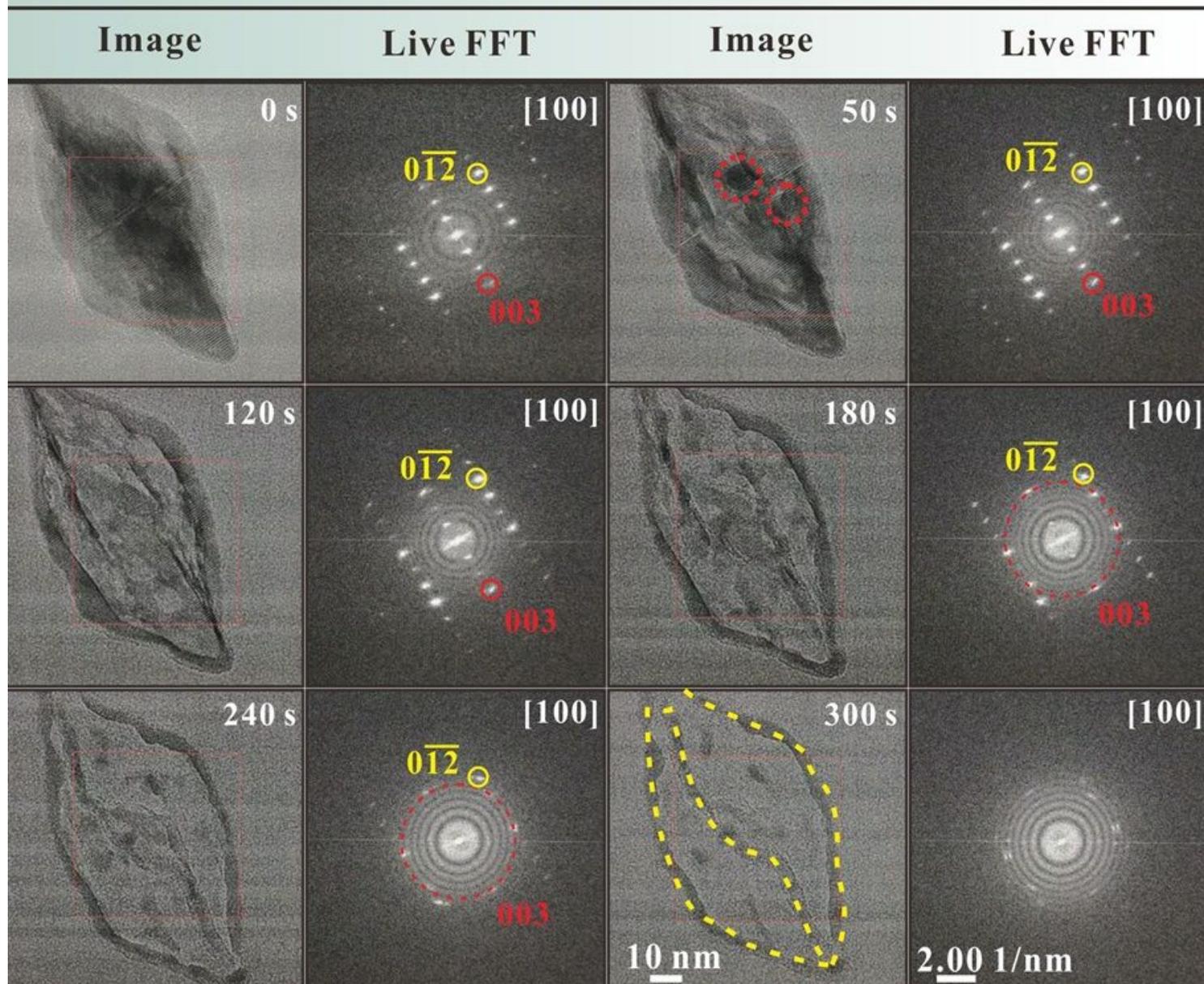
P3121). Middle center: Helical chains of Hg and S atoms along the crystallographic c axis. Middle right: Schematic illustration of the epitaxial evolution from the chiral lattice to the chiral morphology by involving the chiral molecules. Bottom left: Typical TEM image of  $\alpha$ -HgS with chiral morphology. The dotted lines are the twisting orientations in the nanobipyramids. Bottom center: Elemental mapping of Hg and S, respectively. Bottom right: EDX spectrum of  $\alpha$ -HgS. The inset is the corresponding high-angle annular dark field (HAADF) image for EDX.



**Figure 2**

Experimental setup for in-situ observations of chemical reaction dynamics. Left: Schematic of e-beam-induced chemical reaction dynamics visualized in both real and reciprocal space. The sample dynamics are triggered by an e-beam, while their structural evolution and corresponding diffraction patterns are simultaneously recorded by a charged-coupled device (CCD) camera. Upper right: EDX spectra of  $\alpha$ -HgS before and after the 300 s e-beam irradiation. Bottom right: TEM images of  $\alpha$ -HgS before and after the 300 s e-beam irradiation. The arrow indicates the occurrence of nanoclusters due to the irradiation.

## Dynamics of Structural Evolution in Chiral HgS Nanobipyramid



**Figure 3**

Structural evolution in the chiral  $\alpha$ -HgS nanobipyramid. Time-resolved TEM images and live FFT patterns show the formation and evaporation dynamics of Hg nanodroplets as well as the creation and coalescence of nanovoids. The initial HgS was monocrystalline (0 s). After the 50 s e-beam irradiation, the Hg nanodroplets with the diameter of  $\sim 6$  nm were observed (marked by the dotted circles). Such nanodroplets gradually shrank and finally vanished as the e-beam irradiation continued. Following the creation, growth and coalescence of nanovoids, the HgS nanobipyramid was gradually destroyed with the prolonged irradiation (120, 180 and 240 s). Eventually (300 s), a hollow nanostructure with the twisting shell layer was seen.

## Size-dependent Effect on Structural Evolution Dynamics

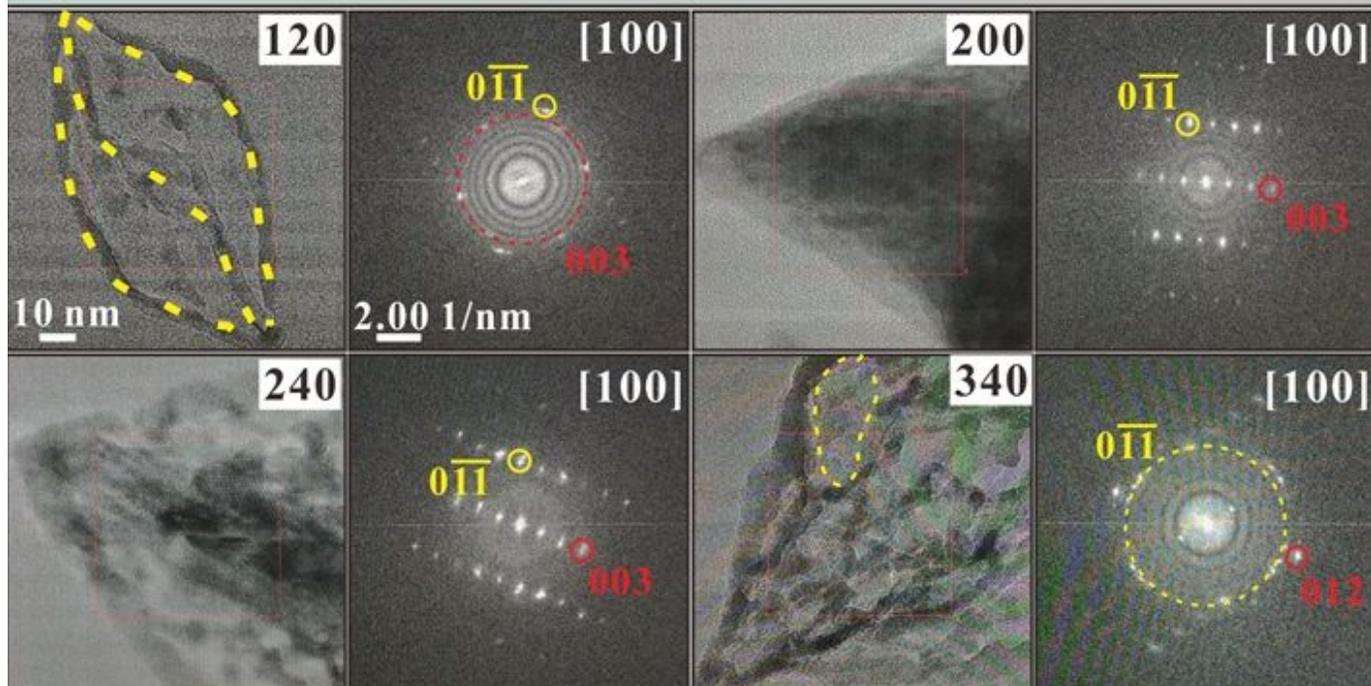
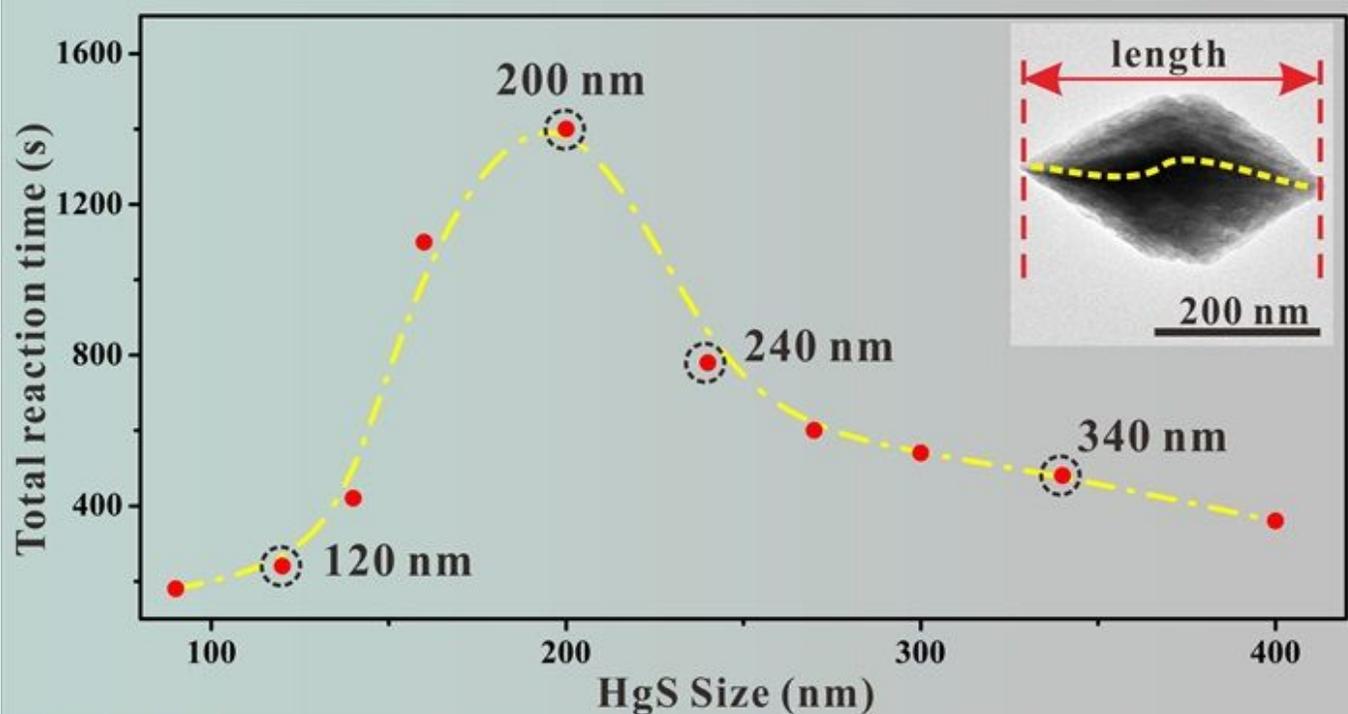


Figure 4

Size-dependent effect on the reaction dynamics of chiral HgS. Upper: Total reaction times of HgS nanobipyramids as a function of their sizes (length ranged from ~80 to 400 nm) under the same e-beam excitation condition. The total reaction time firstly increased with the size, then reached the maximum at the size of ~200 nm and after that decreased with increasing the crystal size. The inset illustrates the length of a nanobipyramid used for this work. Bottom: TEM images and corresponding FFT patterns of

four typical HgS nanobipyramids with the different sizes (~120, 200, 240 and 340 nm), corresponding to the dotted circles marked in the top panel. These nanobipyramids were irradiated under the identical condition for 240 s. The comparison of such structural evolution indicates the distinct irradiation stability of HgS with different sizes.

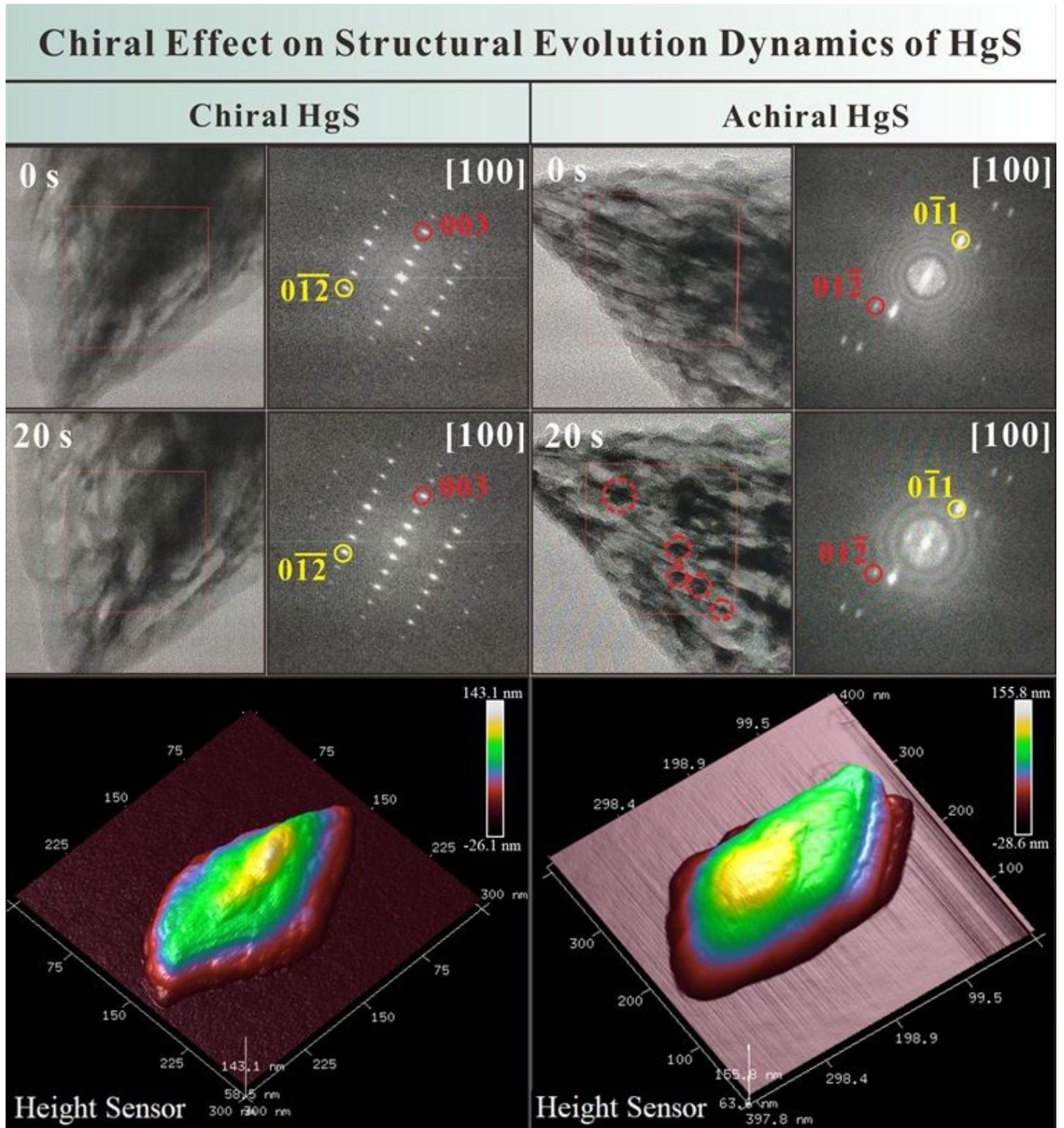


Figure 5

Chiral effect on the reaction dynamics of HgS. Upper: TEM images and corresponding FFT patterns of HgS nanobipyramids (~280 nm) with chiral (left) and achiral (right) morphology under the 20 s e-beam irradiation. Several Hg nanodroplets with the diameter of 3-6 nm occurred in the achiral HgS (marked by the dotted circles), whereas no droplets were observed in the chiral one. Bottom: AFM topography images of HgS nanobipyramids with chiral (left) and achiral (right) morphology. The chiral surface was rougher than the achiral one.

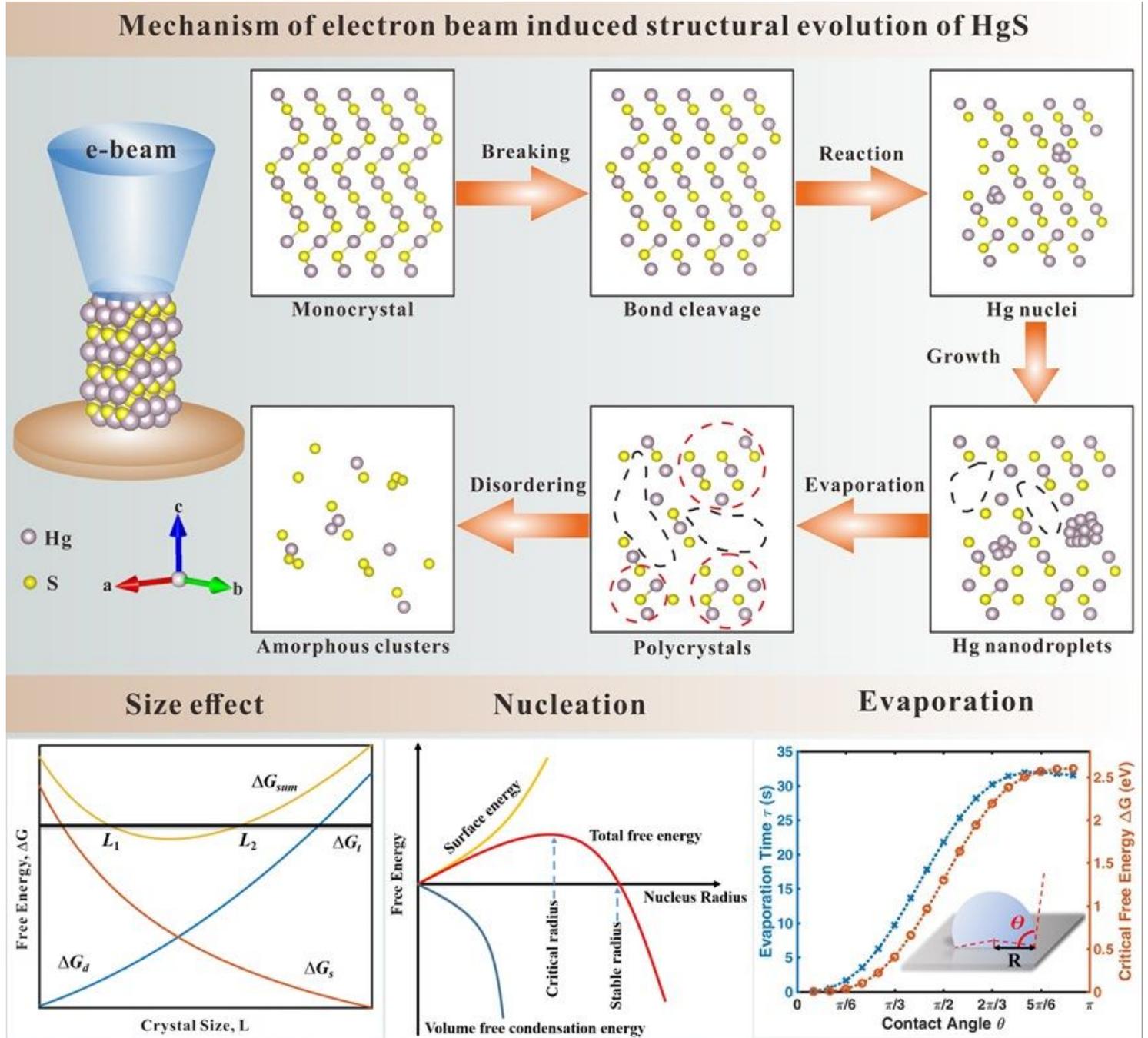


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Mechanism of the e-beam-induced chemical reactions of HgS. Upper: Schematic of the structural evolution of order-disorder transition upon the e-beam excitation. With the incoming high-energy electrons, the Hg-S chemical bonds are cleaved, and the reduction of Hg<sup>2+</sup> to Hg<sup>0</sup> as well as the

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