In-situ formed and low-temperature deposited Nb: TiO2 compact-mesoporous layer for hysteresis-less perovskite solar cells with high performance

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Abstract

Recently, reported perovskite solar cells (PSCs) with high power conversion efficiency (PCE) are mostly based on mesoporous structures containing mesoporous titanium oxide (TiO2) which is the main factor to reduce the overall hysteresis. However, existing fabrication approaches for mesoporous TiO2 generally require a high temperature annealing process. Moreover, there is still a long way to go for improvement in terms of increasing the electron conductivity and reducing the carrier recombination. Herein, a facile one-step, in situ and low-temperature method was developed to prepare an Nb: TiO2 compact-mesoporous layer to serve as both scaffold and electron transport layer (ETL) in PSCs. The Nb: TiO2 compact-mesoporous ETL based PSCs exhibit suppressed hysteresis, which is attributed to the synergistic effect of the large interface surface area caused by nano-pin morphology and the improved carrier transportation caused by Nb doping. Such a high-quality compact-mesoporous layer allows the PSC to achieve a remarkable PCE of 19.74%. This work promises an effective approach for creating hysteresis-less and high-efficiency PSCs based on compact-mesoporous structures with lower energy consumption and cost.

Introduction

Organic-inorganic hybrid perovskites have been attracting great interest as promising light-absorbing materials owing to their large absorption coefficients, high carrier mobility, and ease of fabrication [1-5]. Perovskite-based solar cells, photodetectors, light-emitting diodes (LEDs), and even memory devices have been widely investigated and established [6-8]. Since the year 2009, the power conversion efficiencies of perovskite solar cells (PSCs) have maintained rapid growth from 3.8% to over 25% under standard AM 1.5 illumination [9-12]. PSCs are generally fabricated with a mesoporous or planar structure [13-15]. To this day, the reported PSCs with high power conversion efficiency (PCE) are typically based on mesoporous structures containing an indispensable mesoporous metal oxide [16]. Titanium oxide (TiO_2) has been commonly used as an electron transport layer and the typical mesoporous-type PSC presented by Seok has a structure of FTO/compact TiO_2/mesoporous TiO_2 and perovskite composite layer/ perovskite upper layer/PTAA/Au [17]. As is often reported in the literature, mesoporous TiO_2 contributes most to mesoporous-type PSC for reducing the overall hysteresis [18]. However, the fabrication of a mesoporous TiO_2 layer often requires a high-temperature (>450°C) annealing treatment, leading to large energy consumption and limiting its application in flexible devices [19-21]. Compared with the mesoporous-type PSCs, planar-type PSCs can be fabricated using a low temperature and low-cost processes[22]. However, planar-type PSCs usually suffer from poor electron conductivity, severe charge recombination, and relatively lower crystallinity, resulting in low PCE with severe hysteresis behavior [23-24].

Extensive efforts have been made to develop high-quality TiO_2 electron transport layers (ETLs) with high electron mobility, such as through morphology optimization, surface modifications, and doping. In particular, a wide range of elements have been chosen to prepare TiO_2 doping layers in PSCs, including

Indium (Li) [25-26], Niobium (Nb) [27-28], Platinum (Pt) [29], Sodium (Na)[30], Neodymium (Nd) [31], and Aluminum (Al) [32]. For instance, it is reported that the Li-doped TiO_2 is beneficial to the performance of the mesoporous-structure PSCs, especially the hysteresis effect [26]. Liao et al. reported a Pt-doped TiO_2 ETL which was demonstrated to improve the charge carrier extraction and injection efficiency in n-i-p PSCs [29]. Other ions such as Na, Nb and transition metal ions [30-31, 33-35] were used to modify surface or passivate defect of TiO_2, contributing to less non-radiative recombination. Among these elements, Niobium metal (Nb) is a good candidate as a doping material for titanium oxide electron transport materials due to its similar radius with that of titanium. The results shown by Yin et al. demonstrated that Nb doping can make an improvement in both conductivity and mobility, simultaneously decrease the trap-state density of TiO_2 ETLs for PSCs [27]. Despite this progress, a relatively high temperature (150 °C) treatment is mandatory and large hysteresis is still observed in PSCs based on Nb-doped TiO_2. As is well known, current density-voltage(*J-V*) hysteresis is a critical issue that occurs frequently, especially in planar-structure PSC devices. Severe hysteresis can lead to instability of PSCs and degradation of PCE. For this reason, it richly rewards the effort to develop a hysteresis-less PSC utilizing a simple and low-temperature method.

Here, we proposed a facile one-step, in situ and low-temperature (70°C) strategy to develop a hysteresis-less PSC, in which an Nb: TiO_2 compact-mesoporous layer served as both scaffold and ETL. The Nb: TiO_2 layer contained a compact TiO_2 layer with nano-pin morphology on the surface, which was utilized as a scaffold layer. The hysteresis index decreased significantly from 24.39% for the PSC based on bare TiO_2 to 3.19% for that based on 2% Nb: TiO_2 layer due to the collaborative effect of the large interface surface area caused by nano-pin morphology on the surface and the improved carrier transportation rate because of the presence of Nb. The high-quality mesoporous layer allowed the PSC to achieve a remarkable PCE of 19.7%. This work promises an effective approach for achieving hysteresis-less and high-efficiency PSCs through scalable and inexpensive methods at low temperature.

Methods

Sample Preparation

First, the FTO substrates were successively put into acetone, alcohol, and deionized water for ultrasonic cleaning for 30 minutes each. After that, the cleaned substrates were treated by a UV-ozone cleaner for 20 min and then placed in a petri dish. Second, 0.1 M $TiCl_4$ was dropped into deionized ice water mixture to prepare $TiCl_4$ aqueous solution. Third, 2% 0.1 M $NbCl_5$ was put into the ethanol near the temperature of 0 ℃ to obtain $NbCl_5$ ethanol solution. Then, the $NbCl_5$ ethanol solution and $TiCl_4$ aqueous solution were sequentially dropped into the petri dish containing FTO substrates. After hydrothermal reacting at 70°C for 60 min, the Nb: TiO_2 nano-pin features were formed on the FTO substrate.

The perovskite absorption layer was deposited with the dynamic two-step spin coating method[36]. First, the PbI_2 precursor solution was obtained by adding 0.462 g PbI_2 into 1 mL DMF. Meanwhile, the CH_3NH_3I

(MAI) precursor solution was obtained by adding 0.1 g MAI into 2 mL isopropanol (99.5%, Aladdin). Second, 55 μL PbI_2 precursor solution was spun onto the as-prepared Nb: TiO_2 ETL film at 3000 rpm for 10 s. At this moment, 55 μL MAI precursor solution was dropped onto the sample immediately, and spinning was continued for 20 s. Finally, the whole film was annealed at 150°C for 15 min.

The precursor was obtained by stirring 1 mL chlorobenzene solution, which contained 72.3 mg Spiro-OMeTAD, 28 μL 4-tert-butylpyridine, and 17 μL Li-TFSI solution (520 mg mL^{-1}). The precursor was spin-coated onto perovskite film at 2000 rpm for 30 s. Then, the Spiro-OMeTAD HTL was obtained.

Characterization Methods

A field-emission scanning electron microscope (FE-SEM, SU8010, Hitachi) was carried out to study the morphologies of the samples. The absorption spectra were recorded with a UV-vis spectrophotometer (Shimadzu, UV-3600). Electrochemical impedance spectroscopy (EIS) was employed to understand the carrier transportation process by an electrochemical workstation (Autolab, PGSTAT 302N). The current density–voltage (*J-V*) measurement was recorded using a digital source (Keithley 2400) with the assistance of the solar simulator (ABET Technologies, SUN 3000).

Results And Discussion

A schematic of the PSC structure and the Nb: TiO_2 synthesis procedure is shown in Fig. 1. First, the cleaned FTO substrates were placed in the bottom of the petri dish. Second, 1 mL $NbCl_5$ ethanol solution and 49 mL $TiCl_4$ aqueous solution were poured on the dish sequentially. Third, the dish was transferred into a oven for hydrothermal reacting at 70°C for 1 h. Finally, the Nb-TiO_2 layer with nano-pin morphology was formed on the FTO substrates. For the preparation of control TiO_2 layer, only $TiCl_4$ aqueous solution (without $NbCl_5$ ethanol solution) was dropped into the dish containing FTO substrates.

To understand the effect of Nb doping on the evolution of the TiO_2 layer, the morphologies of the control TiO_2 and Nb-doped TiO_2 were investigated using scanning electron microscopy (SEM) which is shown in Fig. 2. The bare TiO_2 exhibits a much smoother surface, which is a typical morphology of compact TiO_2 layers in planar PSCs. However, 2% of Nb-doped TiO_2 shows a nano-pin texture distributed on the compact layer. The length of the nano-pin was determined to be 50 ± 20 nm. This indicates that the Nb: TiO_2 layer contains a compact TiO_2 layer with a nano-pin morphology on the surface, which is regarded as a mesoporous layer. Therefore, this in-situ-formed Nb: TiO_2 compact-mesoporous layer, which was obtained by a one-step process, actually serves as both a scaffold and an ETL in the PSC. The formation of nano-pin morphology resulted from the hydrothermal reacting with the assistance of $NbCl_5$ ethanol solution.

The XPS spectra of 2% Nb: TiO_2 film is shown in Fig. 3. Fig. 3a shows the intensity at the entire binding energy range of the 2% Nb: TiO_2 layer. It is found that the Nb/Ti ratio of 1.3% is closed to the element ratio of 2% in the precursor mixture. The peaks located at 458 eV and 464 eV correspond to the binding energy of Ti $2p_{3/2}$ and Ti $2p_{1/2}$ as shown in Fig.3b. The Gaussian fitted lines of Nb^{5+} can be deconvolved into two individual peaks which are associated with Nb $3d_{5/2}$ and Nb $3d_{3/2}$ at the binding energy of 207 eV and 209 eV (Fig. 3c). The XPS spectra demonstrates the successful doping of Nb in the TiO_2 film.

Fig. 4a shows the absorption spectra of FTO, bare TiO_2/FTO, and Nb-doped TiO_2/FTO. Both bare TiO_2 and Nb-doped TiO_2 exhibit the main absorption edge in the wavelength range of 300-350 nm. The absorption curve of Nb-doped TiO_2 almost overlaps that of bare TiO_2. The energy band gap (Eg) can be calculated based on the absorption spectra using the Tauc equation, which is shown in Fig. 4b. The Eg is 4.55 eV for FTO and 3.5 eV for both bare TiO_2 and Nb doped TiO_2. Therefore, it can be concluded that Nb doping has little influence on the absorption of TiO_2. The light permeation is also not changed during the Nb doping as shown in Fig. S1.

Fig. S2 presents the SEM images of $CH_3NH_3PbI_3$ perovskite films spin-coated on the bare TiO_2 and Nb-doped TiO_2 films. Thanks to our previously developed non-substrate-selective dynamic two-step spin coating strategy [36], the perovskite films exhibit fewer pinholes and full surface coverage, suggesting that the dynamic spin-coating procedure allows for better control of the film uniformity and coverage. Besides, the average crystalline grain sizes of the perovskite films are very similar. Fig. S3 presents the absorption spectra of the perovskite films deposited on the bare TiO_2 and Nb-doped TiO_2 films. No obvious difference in absorption peak is observed between the perovskite films. These results suggest that the nano-pin morphology formation on the Nb-doped TiO_2 compact-mesoporous layer could have little effect on the perovskite crystallization by dynamic two-step spin coating strategy.

To understand the carrier transportation crossing the ETL/perovskite interfaces, the electrical impedance spectroscopy (EIS) was employed. PSCs were fabricated with the structure of FTO/TiO_2/perovskite film/Spiro-OMeTAD/Au. Fig. 5 shows the Nyquist plots of PSCs based on bare TiO_2 and 2% Nb: TiO_2 layers, and the corresponding equivalent circuit model is shown in the inset. The parameters of EIS were listed in Supplementary Table S1. It is known that the EIS contains two circular arcs [37]. The high-frequency component is attributed to the charge transport resistance (R_{ct}), and the low-frequency component is mainly related to the recombination resistance (R_{rec}) [38]. In the present work, the perovskite/HTL interface is identical for both PSCs, and the only variable affecting R_{ct} and R_{rec} is the perovskite/ETL interface. Compared to the bare TiO_2 device, the Nb: TiO_2 device exhibits smaller R_{ct} and larger R_{rec}. The small R_{ct} contributes to more efficient electron extraction, and the large R_{rec} proves lower

charge recombination. These results confirm that the Nb: TiO_2 compact-mesoporous layer is an effective ETL for simultaneously improving charge transportability and reducing the carrier recombination rate.

As shown in Fig. 6, the dependence of the PCE of PSCs on the Nb doping contents was investigated. The detail parameters for PSCs with different Nb doping concentrations varying from 0% to 8% was shown in Table 1. It is found that the doping ratio affects the open circuit voltage (V_{oc}) and fill factor (FF), which were first increased and then decreased with increasing Nb doping. The device with a 2% Nb-doped TiO_2 layer exhibits the highest V_{oc} of 1.19 eV, J_{sc} of 23.52 mA/cm^2, and FF of 70.74%, leading to a PCE as high as 19.74% for the champion devices. Thanks to better carrier transportation, all parameters show notable improvement. However, superfluous doping would strengthen the carrier scattering and lead to poor mobility. The incremental recombination will weaken the carrier transport improvement and eventually harm the PCE.

Table 1 Performance parameters of PSCs based on different Nb doping concentrations

Solar cells	V_{oc} (V)	J_{sc} (mA cm^{-2})	FF (%)	PCE (%)
0% Nb	1.04	23.80	63.56	15.70
2% Nb	1.19	23.52	70.74	19.74
4% Nb	1.17	23.58	70.28	19.31
6% Nb	1.12	23.36	68.88	18.17
8% Nb	1.11	22.93	59.94	15.21

The measured *J-V* curves of the control and champion device are shown in Fig. 7. It is well known that *J-V* hysteresis behavior often occurs, especially in planar structure PSC devices. In this work, the hysteresis of *J-V* curves of bare compact TiO_2-based PSC and 2% Nb:TiO_2 compact-mesoporous layer-based PSC were examined. The hysteresis index, (PCE of reverse scan - PCE of forward scan)/PCE of reverse scan [30], reduced markedly from 24.39% for the PSC based on bare compact TiO_2 to 3.19% for the PSC based on 2% Nb-doped TiO_2 layer. It is well known that PSCs based on a mesoporous TiO_2 layer can collect electrons and effectively achieve balance between the hole flux and electron flux due to its larger surface area, thereby exhibiting less hysteresis [17]. The suppression of hysteresis of the *J-V* curve for the Nb-doped TiO_2 device resulted from the existence of the conductance increasing nano-pin morphology on the surface that is likely to be a mesoporous TiO_2 layer. Charge accumulation caused by interfacial capacitance at the ETL/perovskite interface would be reduced and result in hysteresis-less character.

Conclusion

We have developed a facile one-step, in situ and low-temperature approach to achieve an Nb: TiO_2 compact-mesoporous layer that serves as both scaffold and ETL in PSCs. As a result, a PSC device based on 2% Nb-doped TiO_2 exhibits a maximum PCE of 19.74%, which is dramatically higher than that of the control device based on the bare TiO_2. The Nb:TiO_2 layer contains a compact TiO_2 layer with nano-pin morphology on the surface, which is utilized as a mesoporous layer. Due to the collaborative effect of a large interface surface area and improved carrier transportation rate, the hysteresis of the *J-V* curve is markedly reduced, with the hysteresis index decreasing significantly from 24.39% to 3.19%. This work promises an effective approach for achieving hysteresis-less and high-efficiency PSCs through scalable and inexpensive methods at low temperature.

Abbreviations

PSCs: Perovskite solar cells; PCE: Power conversion efficiency; TiO_2: Titanium oxide; ETL: Electron transport layer; SEM: Scanning electron microscope; EIS: Electrochemical impedance spectroscopy; B_g: Band gap; E_g: Energy band gap; V_{oc}: Open circuit voltage; FF: Fill factor; J_{sc}: Short circuit current density.

Declarations

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Authors' Contributions

YM prepared the materials and fabricated the devices. SH and HX carried out XPS, EIS, SEM measurements. YY and ZW wrote the manuscript. All authors read and approved the final manuscript.

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Availability of Data and Materials

The authors declare that the materials and data are available to the readers, and all conclusions made in this manuscript are based on the data which are all presented and shown in this paper.

Competing Interests

The authors declare that they have no competing interests.

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Figures

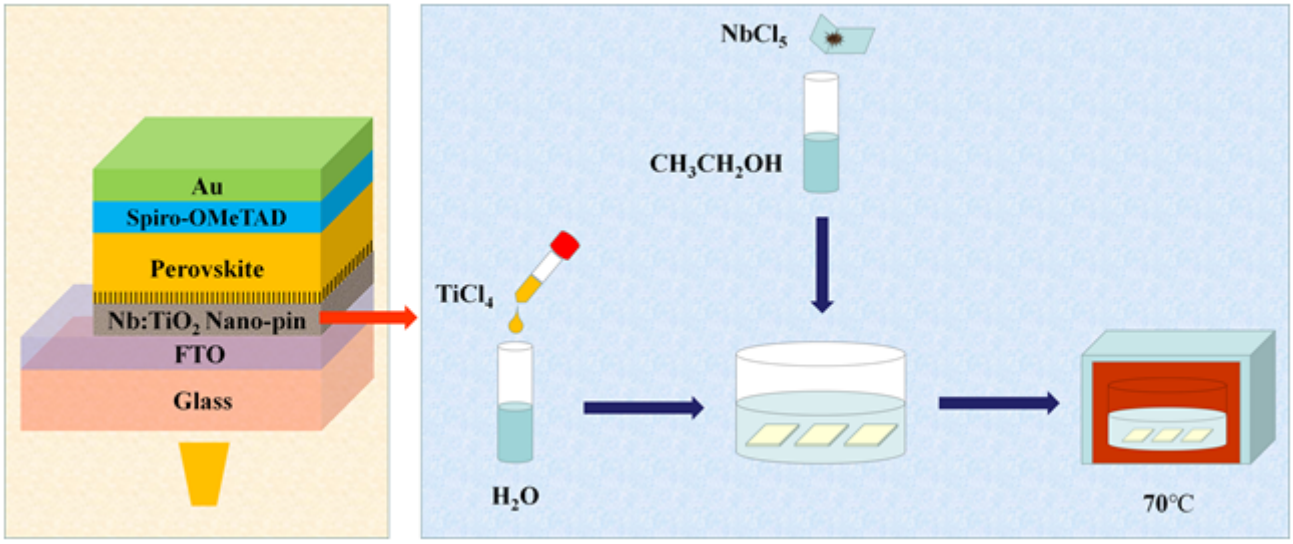


Figure 1

Schematic of PSC structure and Nb: TiO2 synthesis procedure.

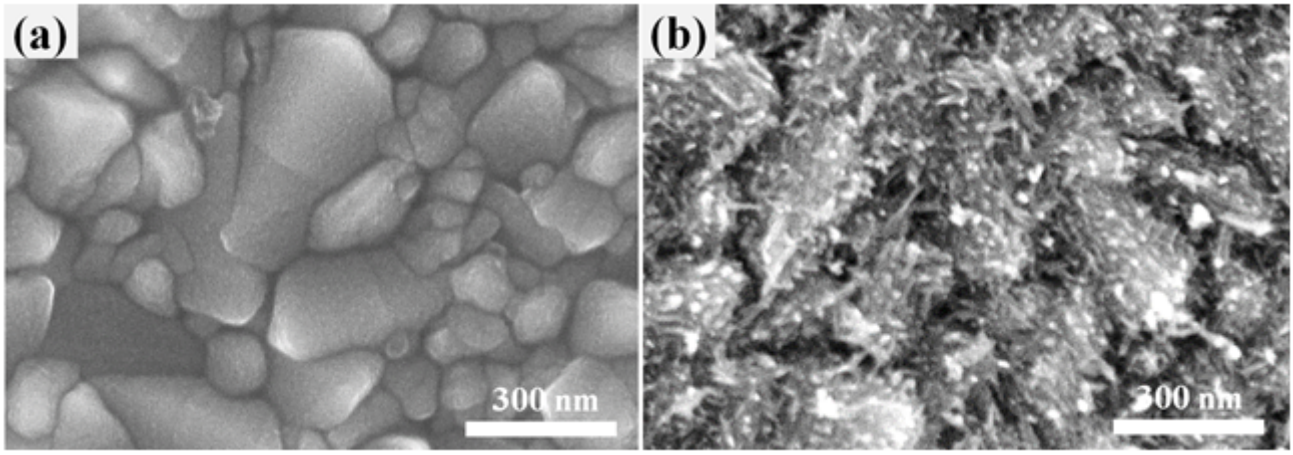


Figure 2

Top-view SEM images of (a) TiO2/FTO, and (b) 2% Nb: TiO2/FTO.

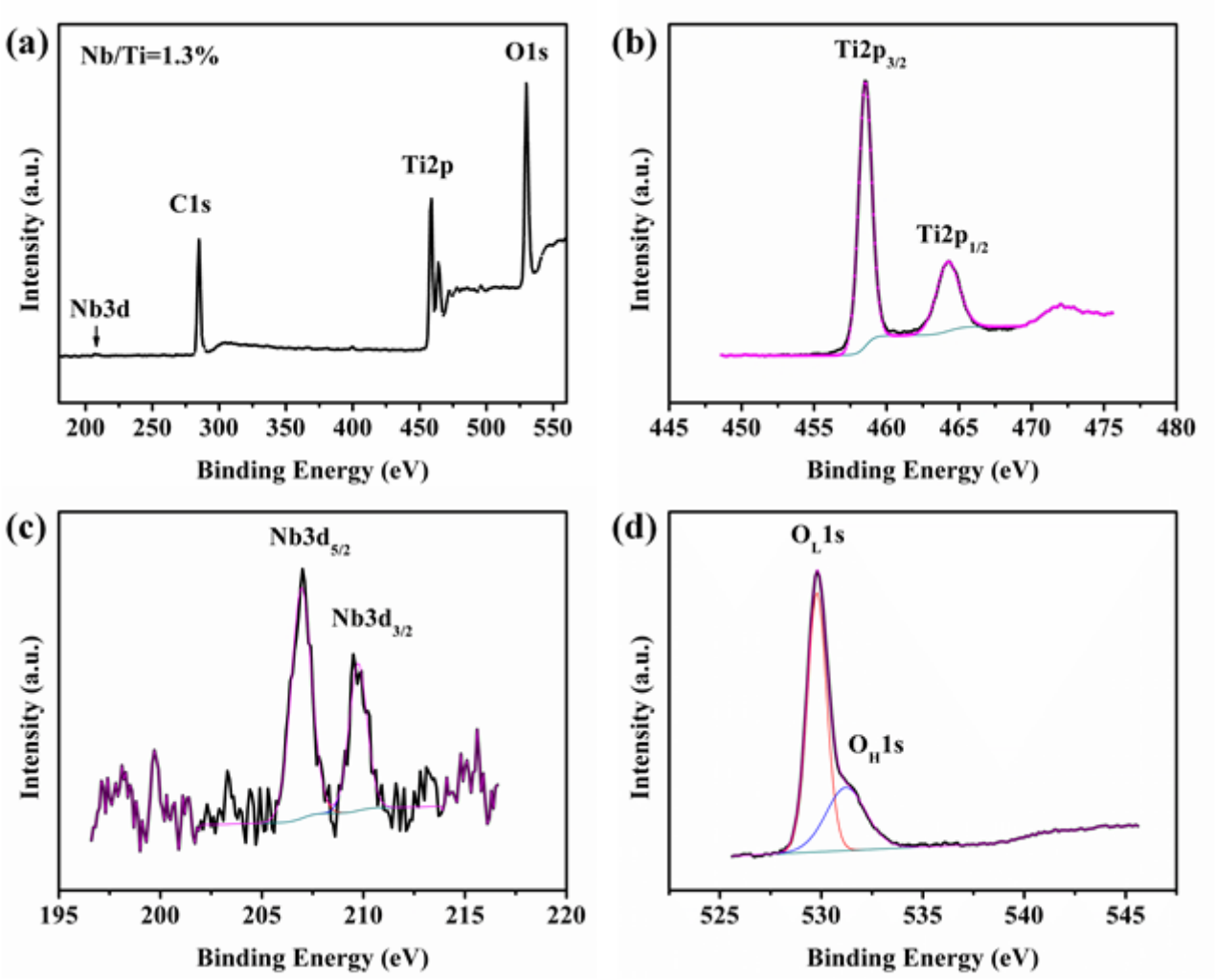


Figure 3

XPS spectra of 2% Nb:TiO2. (a) Survey, (b) Ti 2p, (c) Nb 3d, and (d) O 1s.

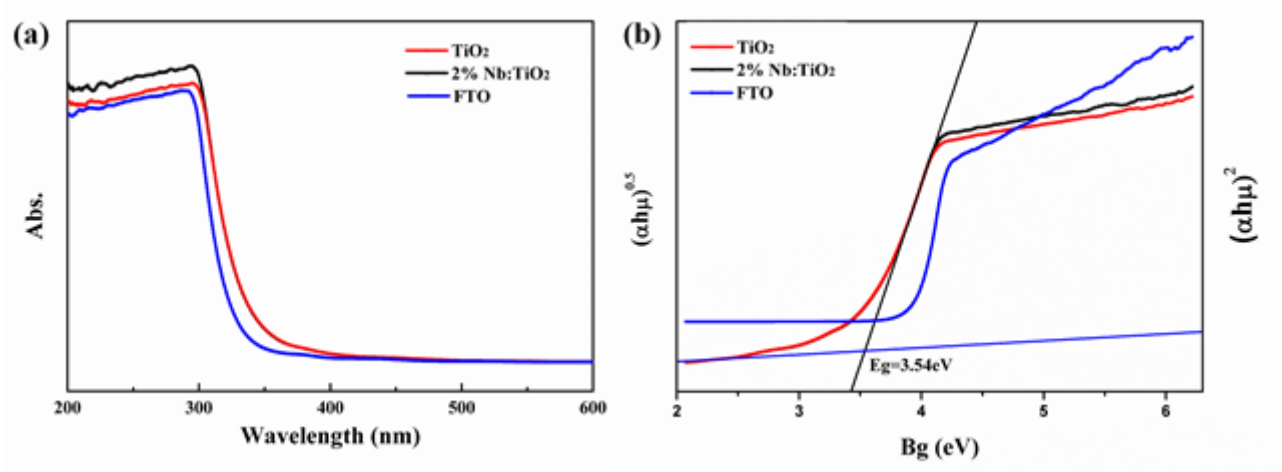


Figure 4

(a) The absorption spectra of the FTO substrate, TiO2/FTO and 2% Nb:TiO2 /FTO. (b) Tauc-plots of the FTO substrate, TiO2/FTO and 2% Nb:TiO2 /FTO.

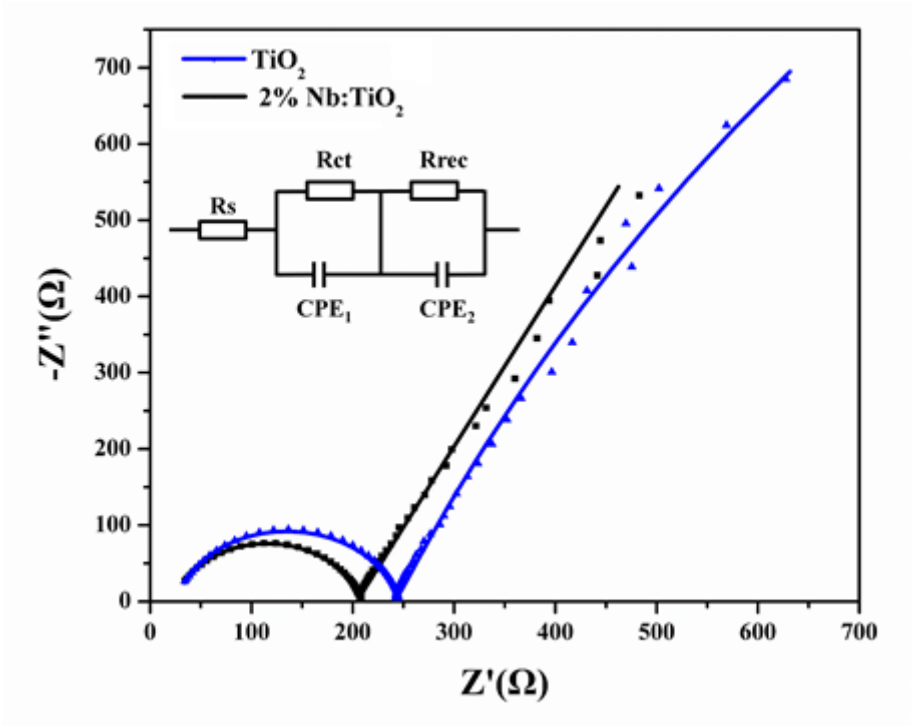


Figure 5

Nyquist plots of devices based on bare TiO2 and 2% Nb-doped TiO2 layers.

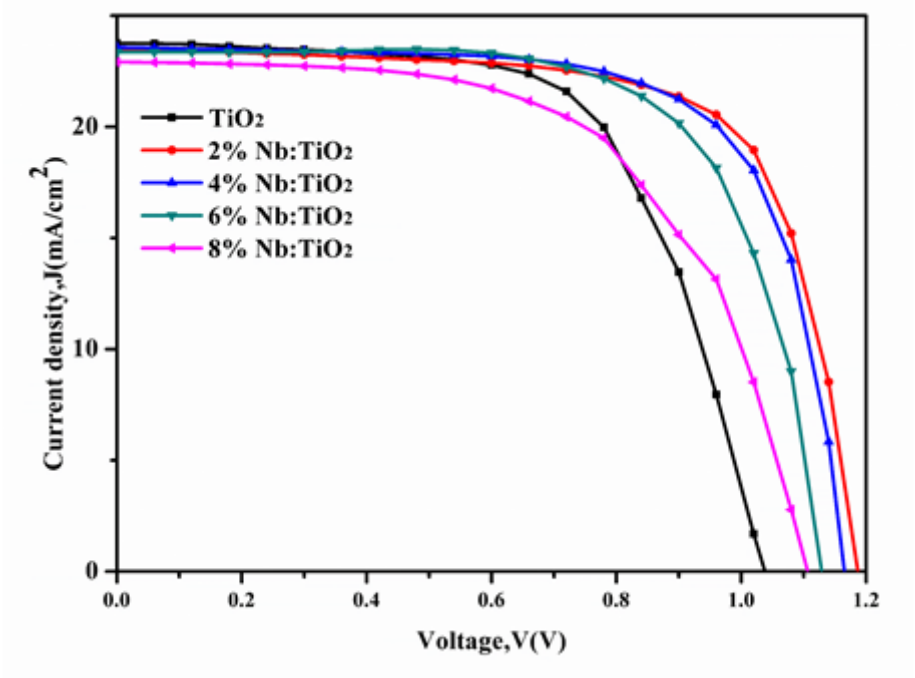
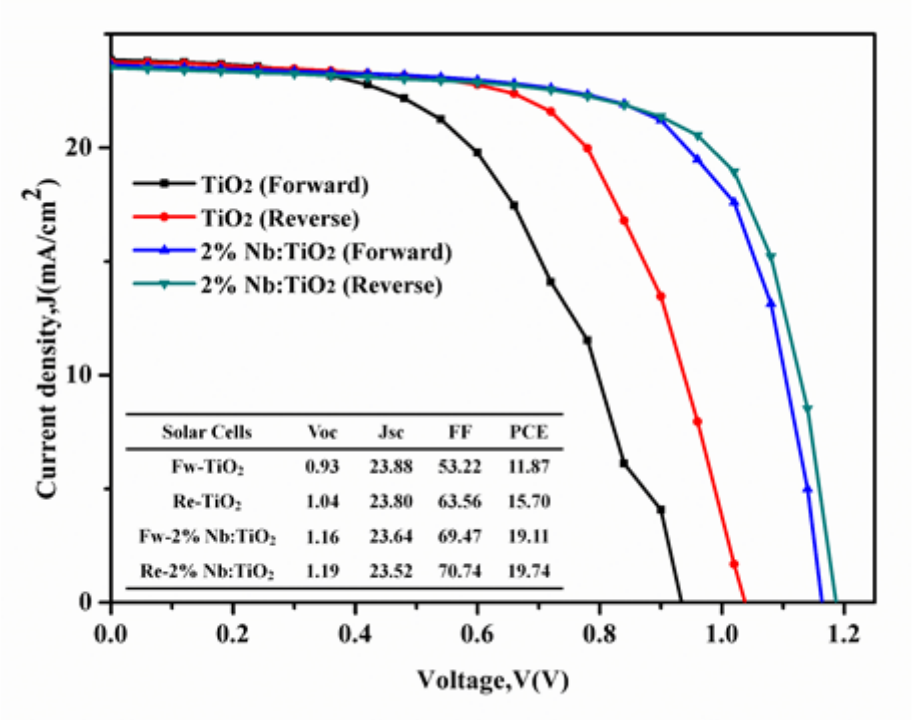


Figure 6

J-V curves of PSCs based on different Nb doping concentrations



Solar Cells	Voc	Jsc	FF	PCE
Fw-TiO_2	0.93	23.88	53.22	11.87
Re-TiO_2	1.04	23.80	63.56	15.70
Fw-2% Nb:TiO_2	1.16	23.64	69.47	19.11
Re-2% Nb:TiO_2	1.19	23.52	70.74	19.74

Figure 7

The J-V hysteresis behavior of the PSCs based on bare TiO2 and 2% Nb: TiO2 layer under AM 1.5 illumination.

Supplementary Files

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