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Ultralow overpotential nitrate reduction to ammonia via a three step relay mechanism

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Abstract

Ammonia plays a significant role in agriculture and the next-generation carbon-free energy supply. Electrocatalytic nitrate reduction to NH₃ is attractive for nitrate removal and NH₃ production under ambient conditions. However, the energy efficiency is limited by the high reaction overpotential. Here, we propose a three-step relay mechanism composed of a spontaneous redox reaction, electrochemical reduction, and electrocatalytic reduction to overcome this issue. Ru_xCo_y alloys are designed and adopted as model catalysts. Ru₁₅Co₈₅ exhibits an onset potential of +0.4 V versus a reversible hydrogen electrode and an energy efficiency of 41.54 ± 1.72 %, which are both the best records. The high performance results in a low production cost of 0.49 ± 0.02 /kg_{ammonia} (0.58 - 1.02/kg_{ammonia} sold in the USA). Electrochemical *in situ* spectroscopy and theoretical simulations indicate that the three-step relay mechanism leads to excellent catalytic performance on Ru₁₅Co₈₅ and can be extended to Ru_xFe_y and Ru_xNi_y alloys.

Introduction

Ammonia (NH₃), as one of the most common industrial chemicals, is essential for the production of nitrogenous fertilizers and shows great promise as a next-generation hydrogen-rich fuel.¹⁻³ NH₃ with an annual global market of 150 million metric tons, is presently produced through the reaction of fossil-fuelderived hydrogen and nitrogen (N₂) under high-temperature high-pressure conditions (Haber-Bosch method). Thus, new routes for sustainable ammonia synthesis are required. Excessive nitrate from overfertilization and industrial sewage is discharged into surface water and underground aquifers, threatening human health.⁴ The electrocatalytic nitrate reduction reaction (NO₃⁻RR) represents a viable approach with advantages in environmental restoration and energy economics under ambient conditions (Fig. 1a). Moreover, NO3⁻RR can be coupled with plasma/photocatalysis-driven nitrogen oxidation techniques to achieve conversion from air to ammonia.⁵⁻⁸ Great progress has been made in the NO₃⁻RR area from the points of fundamental research and practical application.9-12 Very recently, an industrialrelevant current density (1 A cm⁻²) for ammonia synthesis was reported with an onset potential of + 0.22 V versus a reversible hydrogen electrode (RHE).⁹ Notably, the theoretical potential of the NO_3 RR is + 0.69 V vs. RHE under alkaline conditions (pH=14),¹³ much higher than the reported optimal onset potential.¹⁴ Overlarge overpotentials will not only lose energy but also promote the competitive hydrogen evolution reaction (HER) and reduce the Faradaic efficiency (FE) of NH₃, resulting in dissatisfactory energy efficiency (EE) for NO3⁻RR. The high overpotential is caused by the sluggish rate-determining step of nitrate electroreduction to nitrite.¹⁵ The rational design and construction of advanced electrocatalysts to overcome this issue based on the existing reaction mechanism is still challenging.

Theoretically, Co can undergo spontaneous redox with NO_3^- to produce $Co(OH)_2$ and NO_2^- with a Gibbs free energy change of -303.01 kJ/mol (Step 1 in Fig. 1b).¹⁶ In this way, the rate-determining step from

nitrate to nitrite can be easily completed. Subsequently, $Co(OH)_2$ and NO_2^- can be reduced into Co and NH_3 through electrochemical and electrocatalytic processes, respectively, with the participation of active hydrogen (Steps 2 and 3 in Fig. 1b). Thus, the production of active hydrogen species is crucial for the designed three-step relay mechanism (spontaneous redox between Co and NO_3^- , $Co(OH)_2$ electroreduction to Co, and electrocatalytic NO_2^- to NH_3). According to the classic scaling relation, the adsorption energy of a hydrogen atom can be regarded as the descriptor for active hydrogen formation.¹⁷ Ru, Rh, Pd, Ir, and Pt possess moderate adsorption energies for hydrogen atoms (Fig. 1c).¹⁸⁻²¹ Among them, the cost of Ru is the lowest. Moreover, the hollow structure has been suggested to be conducive to mass transfer and atomic utilization in the electrocatalytic process.²²

Herein, a series of Ru_xCo_y hollow nanododecahedrons (HNDs) are synthesized to carry out the designed three-step relay mechanism for NO₃⁻ electroreduction to ammonia. $Ru_{15}Co_{85}$ HNDs exhibit the highest performance with an onset potential of +0.4 V vs. RHE. Under optimal conditions, ammonia is generated at a rate as high as 3.21 ± 0.17 mol g_{cat}^{-1} h⁻¹ with a corresponding FE of 96.78 ± 5.1 %. The energy consumption and production cost per kilo ammonia are calculated as 16.33 ± 0.68 kWh and \$ 0.49 ± 0.02 at +0.3 V vs. RHE, which are both the lowest among all the reports. The high performance can be maintained under an industrial-relevant current density (1 A cm⁻²) and a wide concentration range for nitrate (10-2000 mM), demonstrating the promising application potential. Furthermore, the combined results of *in situ* electrochemical isotope-labeled attenuated total reflection Fourier transformed infrared spectroscopy (ATR-FTIR), X-ray structure analysis (XAFS), Raman, X-ray diffraction (XRD), and theoretical simulation reveal the function of the three-step relay mechanism in decreasing the overpotential. The asproposed three-step relay mechanism can also be extended to Ru_xFe_y and Ru_xNi_y for efficient NO₃⁻RR to ammonia.

Results

Preparation and characterization of Ru_xCo_y **HNDs.** The catalysts of Ru_xCo_y HNDs are prepared via a twostep chemical conversion method, as displayed in Fig. 2a. First, dodecahedral ZIF-67 (Supplementary Fig. 1) react with different concentrations of Ru³⁺ and Co²⁺ under hydrothermal conditions, in which the hydrolysis process and the Kirkendall effect result in the formation of Ru_xCo_yO_z HNDs with uniformly dispersed Ru and Co (Supplementary Figs. 2 – 3).^{23–25} After annealing under a H₂ atmosphere, Ru_xCo_yO_z HNDs are reduced into Ru_xCo_y while maintaining a hollow dodecahedral structure (Fig. 2b,c and Supplementary Fig. 4). The interplanar spacing of Ru₁₅Co₈₅ HNDs measured by high-resolution transmission electron microscopy (HRTEM) is 0.19 nm (Fig. 2d), corresponding to the (101) plane of the Ru₁₅Co₈₅ alloy.²⁶ XRD patterns show the shift of diffraction peaks to lower angles with increasing Ru content (Fig. 2e and Supplementary Fig. 5) due to the larger atomic radius of Ru than Co.²⁷ The calculated contents of Ru and Co based on the Vegard law and Bragg equation are close to the results of inductively coupled plasma optical emission spectrometry (ICP-OES) and energy dispersive spectroscopy (EDS, Supplementary Table 1). Moreover, the surface content analysis of Ru and Co based on X – ray photoelectron spectroscopy (XPS, Supplementary Figs. 6,7 and Supplementary Tables 1,2) is also similar. The uniform distribution of high Ru content in Co originates from the confinement effect of the ZIF-67 porous structure to evenly store Ru³⁺ and Co²⁺ ions.²⁸ Locally enlarged XPS (Fig. 2f,g) shows that the binding energy of Ru⁰ and Co⁰ in Ru_xCo_y HNDs exhibit positive ($\Delta_{max,Ru} = 0.46 \text{ eV}$) and negative ($\Delta_{max,Co} = -0.32 \text{ eV}$) shifts compared to pure Ru and Co, respectively, indicating electron transfer from Ru to Co.²⁹ This phenomenon is also confirmed by X-ray absorption fine structure (XAFS). The three-dimensional wavelet transforms of *K* – edge extended XAFS (Fig. 2h,i) exhibit the contoured feature with the maximum intensity as metallic Ru and Co in Ru₁₅Co₈₅ HNDs. The slight reverse shift of the absorption threshold of Ru and Co in Ru₁₅Co₈₅ HNDs (Supplementary Fig. 8) reveals the poor-electron and rich-electron states of Ru and Co, respectively.

Electrocatalytic NO₃⁻ RR performance over Ru_xCo_y. The hydrogen reversible reaction is first used to calibrate the reference electrode (Fig. 3a and Supplementary Fig. 9).³⁰ As illustrated in the linear sweep voltammetry (LSV) curves (Fig. 3b and Supplementary Fig. 10), the *j* of all catalysts shows significant increases after adding nitrate, implying the proceeding of NO₃ ⁻RR over Ru_xCo_v HNDs. With increasing Ru content, the *j* values first increase and then decrease. Ru₁₅Co₈₅ HNDs exhibit the highest *j* and the lowest Tafel slope (Fig. 3c), implying the fastest electron transfer frequency for the NO₃ ⁻RR. After the chronoamperometry measurements for NO₃ -RR (Supplementary Fig. 11), the products of nitrite and ammonia were analyzed and quantified by coloration and nuclear magnetic resonance (NMR) methods (Supplementary Figs. 12–14).³¹ The plotted heatmaps (Fig. 3d, e and Supplementary Fig. 15a, b) intuitively exhibit the FE and yield_{NH3} rate over Ru_xCo_v HNDs at given potentials. Among them, Ru₁₅Co₈₅ HNDs exhibit optimal performance. After normalization by the electrochemical surface area (ECSA) (Supplementary Figs. 15c,16,17 and Supplementary Note 1), Ru₁₅Co₈₅ HNDs still possess the highest performance for the NO₃ ⁻RR under low overpotential, implying their high intrinsic activity. The onset potential over Ru₁₅Co₈₅ HNDs is + 0.4 V vs. RHE, close to the theoretical potential (+ 0.69 V vs. RHE). ¹³ Under + 0 V vs. RHE, ammonia is generated at a high FE of 96.78 ± 5.1% with a corresponding yield rate of 3.21 ± 0.17 mol g_{cat}⁻¹ h⁻¹.¹⁴ The energy efficiency (EE) for ammonia synthesis reveals an optimal value $(41.54 \pm 1.72\%)$ at + 0.3 V vs. RHE for Ru₁₅Co₈₅ HNDs (Fig. 3f and Supplementary Fig. 18). For comparison, Co and Ru₁₅Co₈₅ nanoparticles (NPs) are prepared via the coprecipitation method. The results show that the HND structure can increase the number of active sites and promote the intrinsic activity of active sites for NO₃ ⁻RR (Supplementary Figs. 18, 19 and Supplementary Note 2). Considering the upcoming industrialization of NO₃ $^{-}$ RR,^{11,12} the cost of as-produced ammonia is significant. On the basis of the price of renewable electricity (\$0.03 / kWh),³² the energy consumption and production cost per kilo ammonia over Ru₁₅Co₈₅ HNDs are calculated as 16.33 ± 0.68 kWh and \$ 0.49 ± 0.02, close to the commercial price (\$0.58 ~ 1.02, sold in the USA).³³ Notably, this is a simple cost accounting based on electricity price, without considering capital costs and ohmic losses. Taking the environmental benefit of

nitrate contaminant removal into account, the reported Ru15Co85 HNDs for the NO3 -RR show great attraction. Meanwhile, the pivotal performance parameters are summarized and compared with recently reported works (Fig. 3g and Supplementary Table 3). 10,13,14,34-36 The onset overpotential, EE, and ammonia price over Ru₁₅Co₈₅ HNDs are all the best as far as we know. Moreover, the high performance is maintained well in the continuous 30 cycles (Supplementary Fig. 21). After the chronoamperometry test, the three-dimensional skeleton structure, lattice spacing and Ru content are almost the same as those of the sample before the electrochemical test (Supplementary Fig. 22). These results indicate the high durability of Ru₁₅Co₈₅ HNDs for the NO₃ ⁻RR. To further highlight the industrial production potential, a series of experiments are performed. First, the electrolyte with different NO_3 ⁻ concentrations (0.01-2 M) were studied, and the high performance was maintained well over a wide range (Fig. 3h). Then, a larger electrolytic cell equipped with a peristaltic pump to circulate the electrolyte is used for the NO3 TRR test (see details in the Methods section and Supplementary Fig. 23a). The chronopotentiometry measurements can be operated over a wide range of current densities (50-1000 mA cm⁻², Supplementary Fig. 23b and Fig. 3i). The highest ammonia yield rate can achieve 3.83 ± 0.08 mmol cm⁻² h⁻¹ at 1000 mA cm⁻², and over 100 hours of long-term stability is obtained at 200 mA cm⁻² (Fig. 3j). These results demonstrate the promising application potential of Ru₁₅Co₈₅ HNDs for the NO₃ ⁻RR.

Mechanistic studies. After soaking Ru₁₅Co₈₅ HNDs in the electrolyte solution (0.1 M KOH and 0.1 M KNO₃), partial nitrate is converted to nitrite (Fig. 4a, Supplementary Fig. 24, and Supplementary Note 3). Similar phenomena can also be found for Co HNDs but are not suitable for Ru and Ru₁₅Co₈₅O₇ HNDs, implying the proceeding of spontaneous redox between Co^0 and NO_3^- to produce NO_2^- . The appearance of characteristic Raman peaks of Co-O bonds and XRD peaks (Supplementary Fig. 25) suggests the spontaneous formation of Co(OH)₂ (Eq. 1). The absence of Raman and XRD signals for Ru HNDs excludes its participation in the spontaneous redox process. The electrochemical in situ Raman spectra of Ru₁₅Co₈₅ HNDs show the disappearance of the Co-O characteristic peak (688 cm⁻¹) below 0 V vs. RHE (Fig. 4b), which is more positive than that of Co HNDs (-0.5 V vs. RHE, Supplementary Fig. 26, 27).³⁰ These results demonstrate that redox-derived Co(OH)₂ species can be electroreduced in situ to metallic Co⁰ (Eq. 2), and the existence of Ru can promote the electrochemical reduction process. The electrochemical *in situ* XRD characterization confirms the electroreduction of $Co(OH)_2$ into Co^0 (Fig. 4d). The first LSV curve of $Ru_{15}Co_{85}$ HNDs after soaking in NO_3^- electrolyte reveals that the Co(OH)₂ initial reduction potential starts from + 0.4 V vs. RHE (Fig. 4e), which corresponds to the onset potential of the NO3⁻RR over Ru15Co85 HNDs (+0.4 V vs. RHE, Fig. 3e and Supplementary Fig. 15b). Interestingly, this potential is more positive than the theoretical equilibrium potential of Co⁰/Co(OH)₂ redox couple. One possible speculation is that the kinetic effect induced by the coupling of a chemical reaction to an electron transfer process causes the potential shift. Under constant electroreduction conditions, the characteristic Raman peak of Co(OH)₂ (688 cm⁻¹) intermittently appears and disappears with the addition of a small amount of nitrate and after the chronoamperometry reaction (Fig. 4f and

Supplementary Fig. 28), indicating the continuous proceeding of the redox reaction (Eq. 1) and the electroreduction reaction (Eq. 2). To further confirm this conclusion, electrochemical *in situ* XANES of Ru₁₅Co₈₅ HNDs (Fig. 4g) was performed. Co valence increases in the presence of nitrate. Taking the valence of Co in samples Co and CoO at 0.4 of normalized absorption as standards, we obtain the linear relation between valence state and *k*-edge energy of Co (black and yellow balls in Fig. 4h).³⁷ Co valences in Ru₁₅Co₈₅ HNDs under different conditions are determined (Fig. 4g,h, Supplementary Fig. 8b,29, and Supplementary Note 4). Interestingly, the nominal valence of Co in Ru₁₅Co₈₅ HNDs is + 0.87 and + 0.37 with nitrate under 0 V and - 0.5 V vs. RHE (-0.5 V is the most negative potential of our test range), respectively. However, the valence of Co in Ru₁₅Co₈₅ HNDs is - 0.16 without nitrate at 0 V vs. RHE. The indelible Co^{δ+} (0 < δ < 2) indicates the oxidation of Co to Co(OH)₂ by nitrate (Eq. 1) and the electroreduction of Co(OH)₂ to Co (Eq. 2) simultaneously proceed during NO₃⁻RR, and a dynamic Co-valence cycle is achieved.

 $NO_3^- + Co + H_2O \rightarrow NO_2^- + Co(OH)_2$ (1)

 $Co(OH)_2 + 2e^- \rightarrow Co + 2OH^- (2)$

 $NO_2^- + 5H_2O + 6e^- \rightarrow NH_3 + 70H^-$ (3)

Afterward, the nitrite electroreduction reaction $(NO_2^{-}RR)$ to ammonia (Eq. 3) was studied by a mixedisotopic labeling experiment (Fig. 4i). When we use equimolar ¹⁴NO₃⁻ and ¹⁵NO₃⁻ as the electrolyte, the generation of equimolar ¹⁴NH₄⁺ and ¹⁵NH₄⁺ can exclude the isotopic effect in the NO₃⁻RR. Notably, only ¹⁴NH₄⁺ is generated using equimolar ¹⁴NO₂⁻ and ¹⁵NO₃⁻ as the reactants, implying that NO₂⁻RR occurs prior to NO₃⁻RR. Meanwhile, the ammonia FE and yield rate both increase, especially in low overpotential regions (0.5 V to 0 V vs. RHE), by using nitrite to replace nitrate (Supplementary Fig. 30). These results demonstrate the advantage of the aforementioned three-step relay mechanism in decreasing the overpotential for the NO₃⁻RR (Fig. 1). Moreover, this mechanism can also be extended to Ru₁₅Ni₈₅ and Ru₁₅Fe₈₅ HNDs, which both exhibit high performance for nitrate reduction to ammonia (Supplementary Figs. 31, 32).

The yield rate of ammonia in an alkaline environment (pH \ge 7) exhibits insensitive changes under different pH values, indicating the concerted proton-electron transfer (CPET) pathway for the hydrogenation process (Fig. 5a and Supplementary Fig. 33).³⁸ Ru₁₅Co₈₅ HNDs and Ru HNDs possess similar underpotential H deposition zones, while Co HNDs show no hydrogen zone in the test range (Supplementary Fig. 34). Moreover, electrochemical quasi-*in situ* electron paramagnetic resonance (EPR) reveals that the introduction of Ru is conducive to the formation of H radicals (Fig. 5b), implying the positive effect of Ru on active hydrogen formation. Subsequently, electrochemical isotope-labeling *in situ* ATR-FTIR spectra (Fig. 5c,d and Supplementary Figs. 35,36) and online DEMS (Fig. 5e) are adopted to capture possible intermediates. During the NO₂⁻RR process (step 3), the characteristic peaks of *¹⁴NH₂ (3190, 3037, and 1160 cm⁻¹) are detected and shifted to lower wavenumbers for *¹⁵NH₂ (3157, 2987, and 1110 cm⁻¹) owing to the isotope effect. *¹⁵NO₂ characteristic peak is centred at ~ 1250 cm⁻¹. Although the *¹⁴NO (~ 1610 cm⁻¹) characteristic peak overlaps with that of *H₂O (1620 cm⁻¹), the negative shift of *¹⁵NO (~ 1552 cm⁻¹) indicates the existence of the *NO intermediate during the NO₂⁻RR.^{39,40} Notably, Ru₁₅Co₈₅ HNDs exhibit the same *in situ* ATR-FTIR spectra as Co HNDs but are quite different from Ru HNDs (Supplementary Fig. 35), implying that the active sites for NO₂⁻RR lie in metal Co sites. The electrochemical *in situ* ATR-FTIR spectra of NO₃⁻RR are similar to those of NO₂⁻RR (Supplementary Fig. 36). Electrochemical online DEMS tests show the m/z signals of NO (30), NH₃ (17), N₂ (28), HNO (31), and NH₂OH (33) (Fig. 5e). On the basis of the aforementioned results, the possible reaction pathways for NO₂⁻RR over Ru₁₅Co₈₅ HNDs, including dissociative, distal-O associative, distal-N associative and alternating-N associative, are deduced and summarized in Supplementary Fig. 37 and Supplementary Note 5.

Theoretical Simulation. Assuming the conversion from NO₃⁻ to NH₃ totally goes through CPET pathways, the minimum applied potential (U) of -0.28 V vs. RHE is required to generate ammonia thermodynamically, as marked by the vertical dotted line. The phase diagram of the abovementioned adsorbates on Ru₁₅Co₈₅ is presented as an example in Fig. 6a (Supplementary Note 6–7, Supplementary Fig. 38, and Supplementary Table 4). The transformation of *NO to *N is the potential-determinant step. The relationship among the applied potential/CPET steps/Ru ratio is presented in Fig. 6b (surface models are displayed in Supplementary Fig. 40). For almost all Ru_xCo_v samples (Supplementary Tables 5-8 and Note 8), the transformation of *NO₃ to *N demands negative applied potentials, especially for the *NO/*N couple (where the area is dark red with U between - 0.37 V ~ - 0.21 V vs. RHE), which implies that nitrate reduction through the pure electrochemical pathway is most likely thermodynamically limited by the N-O bond-breaking steps that demand negative applied potentials as low as - 0.37 V vs. RHE. However, such results contradict the experimental findings in Fig. 3e, where ammonia is already generated at U > 0 V vs. RHE. These findings indicate the existence of pathways other than the electrochemical method, especially for the N-O bond-breaking process. Figure 6c presents the as-proposed three-step relay pathway for $NO_3^ _{(aq)}$ reduction to ammonia. The reduction of NO₃ $_{(aq)}$ to NO₂ $_{(aq)}$ occurs through a redox reaction with Co⁰, which is experimentally supported and matches well with the well-known findings that active metals reduce nitrate to nitrite in basic media.⁴¹ The chemical cleavage of the N - O bond in *NO₂ and *NO is thermodynamically highly favored, with reaction Gibbs energies of - 2.0 eV and - 2.6 eV, respectively. The corresponding activation energies on the typical Ru₁₅Co₈₅ are 0.67 eV and 0.19 eV, respectively. The images of the various adsorbates and the related transition states (TS) are inserted in Fig. 6c. Interestingly, volcano-shaped curves for the adsorption of *NH₂ and *NO on various surfaces are observed, as shown in Fig. 6d, and Ru₇Co₉₃ and Ru₁₅Co₈₅ are close to the top areas, which is coincident with the experimental findings (Fig. 3a-f) that these two alloys show better activity than the others.

In summary, we propose a three-step relay mechanism to decrease the reaction overpotential for the NO₃ ⁻RR. A series of Ru_xCo_y HNDs catalysts are designed and prepared. Ru₁₅Co₈₅ HNDs exhibit the optimal catalytic performance (onset potential: +0.4 V vs. RHE, EE: 41.54 ± 1.72%, ammonia cost: \$0.49 ± 0.02 /kg), outperforming all the reports. The electrochemical *in situ* ATR-FTIR, XAFS, Raman, XRD, and DFT calculations reveal that the high NO₃ ⁻RR performance originates from the enhanced three-step relay processes on Ru₁₅Co₈₅ HNDs (spontaneous redox between Co and NO₃ ⁻, Co(OH)₂ electroduction to Co and electrocatalytic NO₂ ⁻ to NH₃). The introduction of Ru can promote the electrocatalytic reduction of NO₂ ⁻ to ammonia and the electroreduction of Co(OH)₂ to Co benefiting from its excellent hydrogen supply capacity. Moreover, the efficient NO₃ ⁻RR performance on Ru₁₅Fe₈₅ and Ru₁₅Ni₈₅ confirm the universality of the relay mechanism. This work provides a novel reaction pathway to improve the energy efficiency of the NO₃ ⁻RR and inspires the design of efficient catalysts to extend to other electrocatalytic processes.

Methods

Material characterization. Scanning electron microscopy (SEM) images were taken with a Hitachi S-4800 scanning electron microscope. Transmission electron microscopy (TEM) images were obtained using a JEOL JEM-200 microscope. The X-ray diffraction (XRD) pattern was recorded with a Bruker D8 Focus Diffraction System by using a Cu Ka source (λ = 0.154178 nm). X-ray photoelectron spectroscopy (XPS) spectra were collected on a Thermo Fisher Scientific K-Alpha + instrument using monochromatic Al Ka (*hv* = 1486.6 eV) radiation. All spectra were collected at a vacuum pressure of $< 2 \ge 10^{-7}$ Pa, and the cumulative scanning number of each sample was 20. The sample was prepared and transferred into an Ar environment to avoid oxidation. All binding energies were referenced to the C 1 s peak at 284.6 eV.⁴² An inductively coupled plasma optical emission spectrometer (ICP-OES) was taken with an Agilent 7700x. For sample preparation, we dissolved 1 mg catalyst (0.5 mL of 2 mg mL⁻¹ catalyst suspension) in 50 mL of aqua regia. After stirring overnight, KOH solution was used to adjust the pH to above 4, and then deionized water was used to dilute the concentration of ions. The ultraviolet-visible (UV-Vis) absorbance spectra were measured on a Beijing Purkinje General T6 new century spectrophotometer. Fourier transform infrared spectroscopy (FTIR) spectroscopy was carried out with a MAGNA1IR 750 (Nicolet Instrument) FTIR spectrometer. The EPR spectra were recorded on a Bruker EMX-8 spectrometer operated at 9.5 GHz 100 K. The NMR spectra were recorded on Varian Mercury Plus instruments at 600 MHz (1H NMR). The pH values of the electrolytes were determined using a pH meter (LE438 pH electrode, Mettler Toledo, USA). The Raman spectra were obtained on a Renishaw inVia reflex Raman microscope under an excitation of 532 nm laser light with a power of 20 mW. The XAFS was performed at the 1W1B beamline of the Beijing Synchrotron Radiation Facility (BSRF). The XAFS spectra were analyzed with the ATHENA software package. The k-weighting was set to 1 for the Fourier transforms. Fourier transforms of $\chi(k)$ were performed in the k-range of 2-8 Å⁻¹ with the Hanning window function. All the EXAFS spectra were obtained without phase correction. The signal acquisition time was 15-20 min for each scan. The XAS signal of powders was collected in transmission mode under ambient conditions. The powders were

uniformly coated on magic tape (2\$30 cm², produced by the Scotch company) and folded into a 2\$2 cm² sample.

XRD analysis. The lattice spacing can be calculated by the characteristic peaks of XRD based on the Bragg equation.

$$2d{
m sin} heta=n\lambda$$

Where *d* represents the lattice spacing and θ represents the angle between the incident ray and the crystal plane, corresponding to the characteristic peak angle of XRD. Notably, the abscissa of XRD (Fig. 2e) is 2 θ . *n* is a natural number; here, we take *n* = 1. λ is the wavelength of the X-ray, and we take λ = 0.1544178 nm.

We calculated the content of Ru in the alloy by the Vegard law.

$$d = x_{
m Ru} d_{
m Ru} + x_{
m Co} d_{
m Co}$$

Here, $x_{
m Ru}+x_{
m Co}=1$

Where d, d_{Ru} , and d_{Co} represent the lattice spacings of the alloy, Ru, and Co, respectively. Among them, d_{Ru} and d_{Co} can be obtained by the Bragg equation of our catalysts (pure Ru and Co samples). x_{Ru} and x_{Co} are the molar contents of Ru and Co in the alloy, respectively.

The particle size can be calculated by the full width at half maximum (FWHM) of XRD based on the Debye-Scherrer equation.

$$D = \frac{K\lambda}{B\cos\theta}$$

Where *D* represents the particle size. *K* is the Scherrer constant; here, we take K = 0.89. B represents the FWHM of the XRD characteristic peak. θ represents the Bragg diffraction angle, λ is the wavelength of the X-ray, and we take $\lambda = 0.1544178$ nm. For the calculation of FWHM and particle size, we used the relevant functions in "*Jade 6*" software for integration and calculation.

Electrochemical NO₃ RR measurements. Electrochemical NO₃ RR measurements were performed with an lvium – n–Stat electrochemical workstation (lvium Technologies B.V.). A typical three-electrode H-cell was used, including a working electrode, an Ag/AgCl electrode (saturated KCl solution) as the reference electrode, and a carbon rod counter electrode in 0.1 M KOH or 0.1 M KNO₃ + 0.1 M KOH electrolyte, which was separated into a cathode cell (25 mL) and anode cell (30 mL) by an anion membrane (Alkymer AE-115). For catalytic potential, we do not use *iR* correction, except for special instructions. For the chronoamperometry test, carbon paper ($0.5 \diamond 0.5 \text{ cm}^2$) decorated with 0.15 mg catalysts was used as the working electrode. In a typical procedure, an evenly distributed catalyst suspension was prepared by ultrasonically mixing 20 mg catalyst into 8 mL H₂O, 2 mL isopropyl alcohol, and 50 µL Nafion. The 75 µL suspension was covered on the carbon paper surface (0.25 cm^2). The current density was normalized by the catalyst mass, the geometric area of the electrode, and the electrochemical surface area (ECSA). All electrochemical data (except stability testing) were repeated more than 3 times, and the error bar represents the standard deviation of the data. All potentials were calibrated to the reversible hydrogen electrode (RHE) by the following equation:

$$E_{
m RHE} = E_{
m Ag/AgCl} + 0.0591 imes
m pH + \phi_{
m reference}$$

Where $E_{Ag/AgCl}$ represents the experimental applied potential. Notably, the correction term (0.0591pH + $\varphi_{reference}$) is calibrated by a hydrogen reversible reaction (Fig. 3**a**, hydrogen evolution reaction and hydrogen oxidation reaction).

For the ECSA, we used the double-layer capacitance method in 0.1 M KOH solution during the non-Faradic potential range with different scan rates from 10 to 100 mV s⁻¹. The ECSA of the working electrodes was calculated according to the following equation:

$$egin{aligned} I_{ ext{c}} = v C_{ ext{dl}} \ ECSA = rac{C_{ ext{dl}}}{C_{ ext{s}}} \end{aligned}$$

Where I_c represents the charging current with different scan rates. v is the scan rate. C_{dl} is the double-layer capacitance. C_s represents the specific capacitance for a flat metallic surface, which is generally in the range of 20–60 µF cm⁻². According to the reports, we assume it is 40 µF cm⁻².

The current density conversion formula from mass-normalized to geometric area-normalized is:

$${j_{ ext{geo.}}} = rac{{j_{ ext{mass}}} imes m}{S}$$

Where j_{geo} . represents the current density normalized by geometric area, and j_{mass} represents the current density normalized by mass. *m* is the mass of the supported catalyst (0.15 mg). *S* is the carbon paper geometric area of the supported catalyst (0.25 cm⁻²).

The linear voltammetry profile measurements were conducted under a flow of N₂ using the rotating disk electrode (RDE) deposited with the catalysts (40 μ g) as the working electrode at a rotation rate of 100 ~ 1,600 rpm and a sweep rate of 2 mV·s⁻¹. All polarization curves were subjected to 80% *iR* correction.

Ammonia Faradaic efficiency was calculated according to the following equation:

$$FE_{NH3} = \frac{Q_{NH3}}{Q} = \frac{n_{NH3}Vc_{NH3}F}{Q}$$

Where *Q* represents the applied overall coulomb quantity (C). Q_{NH3} represents the coulomb required to produce ammonia. *n* is the electron transfer number; for 1 mol ammonia, it is 8. *V* is the volume of the

catholyte of the cathode chamber, which is 25 mL. C_{NH3} is the concentration of ammonia produced. *F* is the Faraday constant (96,485 C•mol⁻¹).

The energy efficiency (EE) was defined as the ratio of fuel energy to applied electrical power, which was calculated with the following equation:

$$ext{EE}_{ ext{NH3}} = rac{\left(E_{ ext{OER}}^{ heta}-E_{ ext{NH3}}^{ heta}
ight) imes ext{FE}_{ ext{NH3}}}{E_{ ext{OER}}-E_{ ext{NH3}}}$$

Where E_{NH3}^{θ} represents the equilibrium potential of nitrate electroreduction to ammonia, which is 0.69 V vs. RHE under alkaline conditions. E_{OER}^{θ} represents the equilibrium potential of the oxygen evolution reaction, which is 1.23 V vs. RHE. FE_{NH3} is the Faradaic efficiency for ammonia. E_{OER} and E_{NH3} are the applied potentials (the overpotential of OER refers to the recently reported literature⁴⁶).

For the amplified ammonia product process, we used chronopotentiometry to demonstrate the industrial application potential. Herein, the composition and type of electrolytic cell are the same as those of the previous chronoamperometry test, except that the volume of the electrolytic cell is changed to 80 mL. For the preparation of the electrode, carbon felt $(0.5 \diamond 0.5 \text{ cm}^2)$ decorated with 0.6 mg catalysts was used as the working electrode, and titanium mesh $(1.5 \diamond 2 \text{ cm}^2)$ was used as the counter electrode. A Ag/AgCl electrode (saturated KCl solution) was used as the reference electrode. Moreover, we employ a peristaltic pump to promote mass transfer, and the liquid circulation speed is 200 mL min⁻¹.

Electrochemical in situ **Raman tests.** The *in situ* Raman measurement was carried out by the aforementioned Raman microscope and electrochemical workstation. The cell was made up of a Teflon with a quartz window between the sample and the objective. The working electrode was immersed into the electrolyte through the wall of the cell, and the electrode plane was kept perpendicular to the laser. A platinum wire and Ag/AgCl served as the counter and reference electrodes, respectively. LSV curves were conducted from 0.5 to – 0.6 V vs. RHE with a scan rate of 2 mV s⁻¹. Electrochemical intermittent *in situ* Raman measurement was carried out in 0.1 M KOH solution under + 0.3 V vs. RHE. After collecting the first Raman spectrum, we added 1 ml 0.1 M KNO₃ solution to the electrolyte. After 1 min, we collected the second Raman spectrum. After 20 min, nitrate was totally reduced, and then the Raman spectrum was collected again. We repeated this cycle test 4 times.

Electrochemical in situ **ATR-FTIR tests.** Electrochemical *in situ* ATR-FTIR measurements were performed on a Linglu Instruments ECIR-II cell mounted on a Pike Veemax III ATR with a single bounce silicon crystal covered with an Au membrane in internal reflection mode. Spectra were recorded on a Thermo Nicolet Nexus 670 spectrometer. The electrolyte was degassed by bubbling N₂ for 30 min before the measurement. The single-bounce silicon crystal covered with a Au membrane was prepared through the following procedure. (1) 0.12 g NaOH, 0.23 g NaAuCl₄·2H₂O, 0.13 g NH₄Cl, 0.95 g Na₂SO₃, and 0.62 g Na₂S₂O₃·5H₂O were dissolved in 100 mL H₂O (denoted as Solution A). (2) Monocrystal silicon was immersed in aqua regia ($V_{\text{concentrated HCI}}$: V_{HNO3} = 1:1) for 20 min and then polished using Al powder for 10 min. After washing three times with water and acetone, clean monocrystal silicon was obtained. (3) The above monocrystal silicon was immersed in a mixture of H₂SO₄ and H₂O₂ ($V_{\text{concentrated H2SO4}}$: V_{H2O2} = 1:1) for 20 min. (4) After washing three times with water, the above monocrystal silicon was then immersed in 40% NH₄F aqueous solution and washed three times with water. (5) Monocrystal silicon was immersed in a mixture of 15 mL solution A and 3.4 mL 2% NH₄F aqueous solution. (6) After 5 min, Aucoated monocrystal silicon was obtained.

Electrochemical online DEMS test. 0.1 M KOH + 0.1 M KNO₃ electrolyte kept flowing into a homemade electrochemical cell through a peristaltic pump. Glassy carbon electrodes coated with $Ru_{15}Co_{85}$ HNDs catalyst, Pt wire, and Ag/AgCl electrodes were used as the working electrode, counter electrode, and reference electrode, respectively. Then, the applied voltage (- 0.2 V vs. RHE) was employed alternately, and the interval was 2 minutes. After the electrochemical test was over and the mass signal returned to baseline, the next cycle started using the same conditions to avoid accidental error. After seven cycles, the experiment ended.

Electrochemical in situ **XAS test.** Electrochemical *in situ* XAS at the Co K-edge was carried out at the 1W1B beamline of the Beijing Synchrotron Radiation Facility (BSRF). The electrolytic cell was made inhouse with Teflon containing 0.1 M KOH or 0.1 M KOH + 0.1 M KNO₃ electrolyte, in which a graphite rod and Ag/AgCl electrode were used as the counter electrode and reference electrode, respectively. Carbon paper ($2\diamond 2 \text{ cm}^2$) loaded with an electrocatalyst (3 mg) was used as the working electrode, and the catalyst was concentrated in the center of the carbon paper with an effective area of more than 1 cm². The *in situ* XAS signal was collected in fluorescence mode at chronoamperometry measurement with 0 V or – 0.5 V vs. RHE. No pretreatment was required before the electrochemical chronoamperometry test.

Electrochemical in situ **XRD test.** Electrochemical *in situ* XRD patterns were measured on a Rigaku Smartlab9KW Diffraction System using a Cu Ka source ($\lambda = 0.15406$ nm). The electrolytic cell was made up of Teflon with Pt wire as the counter electrode and a Hg/HgO electrode as the reference electrode. Carbon paper ($0.5 \diamond 0.5 \text{ cm}^2$) loaded with catalyst (0.6 mg) was used as the working electrode, and the patterns were collected under chronoamperometry measurement under different potentials. Before collecting data, we ran the chronoamperometry test under the applied potential for 5 min. The pattern was collected in the 2 θ ranging from 20° to 70° under the applied potential from – 0.4 V to 0.5 V vs. RHE. Each diffraction pattern was collected for 5 min for statistical analysis.

Other experimental details have been added to the **Supplementary Methods**, including Materials, Preparation of catalysts, Materials characterizations, determination and quantitation of ammonia/nitrite, and computational methods.

Online content

Any methods, additional references, Nature Research reporting summaries, source data, extended data, supplementary information, acknowledgments, peer review information; details of author contributions and competing interests; and statements of data and code availability are available at https://doi.org/xxxxxx.

Declarations

Data availability

The data that support the findings of this study are available from the corresponding authors.

Correspondence and requests for materials should be addressed to Bin Zhang.

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Author Contributions

B. Z. and Y. Y. conceived the idea and supervised the project. B.Z., Y.Y., and S.H. designed the experiments. S.H. synthesized the materials and carried out electrochemical measurements. H.L. and R.Y. performed theoretical calculations. S.H. and T.L. carried out *in situ* experiments. S.H. and F.C. drew the schematic diagram. B.Z., Y.Y., and S.H. wrote the paper with comments from all authors.

Additional information

Supplementary information The online version contains supplementary material available at https://doi.org/xxxxxxx.

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Competing Financial Interests

The authors declare no competing interests.

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Figures



Figure 1

Catalyst design for the NO₃ RR. a, Schematic illustration of the electrochemical NO₃ RR to NH₃ under ambient conditions using renewable energy. **b**, The designed three-step relay mechanism for the NO₃ RR. **c**, Volcano plot of the exchange current density and Gibbs free energy of adsorbed atomic hydrogen for different metals.



Figure 2

Structural characterization of Ru_xCo_yO_z and Ru_xCo_y HNDs. a, Schematic illustration of Ru_xCo_y HNDs synthesis. b, c, d, SEM (b), TEM and corresponding EDS mapping (c), and HRTEM (d) images of $Ru_{15}Co_{85}$ HNDs. e,f,g, Locally enlarged XRD patterns (e) and XPS spectra (f and g) of Ru_xCo_y HNDs. h, i, Wavelet

transforms of Ru (**h**) and Co (**i**) for the k^3 -weighted XAFS signals in Ru₁₅Co₈₅ HNDs and contrast samples.



Figure 3

Electrocatalytic performance for the NO₃⁻**RR. a,** CV curves of Ag/AgCl electrode calibration in 0.1 M KOH solution. **b,** LSV curves of NO₃⁻**RR** over Ru_xCo_y HNDs under 1600 rpm with 80% *iR* correction. **c,** Tafel slopes of NO₃⁻**RR** over Ru_xCo_y HNDs. **d, e, f**, Heatmaps of FE (**d**), yield rate (**e**), and EE (**f**) for ammonia synthesis over Ru_xCo_y HNDs. **g,** Comparison of NO₃⁻**RR** performance over $Ru_{15}Co_{85}$ HNDs with recently

reported catalysts. **h**, The NO₃⁻RR performance over Ru₁₅Co₈₅ HNDs at different concentrations of KNO₃. **i**, Chronopotentiometry curves and ammonia yield rate of NO₃⁻RR (0.1 M KNO₃ + 0.1 M KOH) over Ru₁₅Co₈₅ HNDs at different current densities with 80% *iR* correction on potential. **j**, Long-term electrocatalytic stability test of NO₃⁻RR (0.1 M KNO₃ + 0.1 M KOH) over Ru₁₅Co₈₅ HNDs at 200 mA cm⁻² using a continuous flow system in an H-cell. The black arrow represents the renewal of fresh electrolytes.



Figure 4

Mechanistic studies for NO₃⁻RR over Ru₁₅Co₈₅ HNDs. a, The concentration variation of NO₃⁻, NO₂^{-,} and NH₃ in the electrolyte after the redox reaction with Ru₁₅Co₈₅ HNDs for different times without an applied potential. **b,c**, Electrochemical *in situ* Raman spectra of Ru₁₅Co₈₅ HNDs with (**b**) and without (**c**) NO₃⁻. **d**, Electrochemical *in situ* XRD patterns of NO₃⁻RR over Ru₁₅Co₈₅ HNDs. The * represents the characteristic peak of Co(OH)₂ (PDF # 45-0031), corresponding to (002), (102), (110), and (111) of hexagonal phase P-3m1. **e**, Continuous LSV curves of Ru₁₅Co₈₅ in 0.1 M KOH solution. The Ru₁₅Co₈₅ sample was first

soaked in NO₃⁻ electrolyte for 10 minutes and then washed with distilled water for 10 minutes. **f**, The intermittent *in situ* Raman peak intensity at 688cm^{-1} over Ru₁₅Co₈₅ HNDs at +0.3 V vs. RHE for the NO₃⁻ RR with alternate conditions of NO₃⁻ consumption and NO₃⁻ addition. **g**, Electrochemical *in situ* XANES of Ru₁₅Co₈₅ and *ex situ* XANES of standard samples (Co and CoO). **h**, Relation between the Co L₃⁻ edge absorption energy and valence states. **i**, ¹H NMR of NH₄⁺ in the mixed isotope labeling experiments.



Figure 5

 NO_3 -RR pathway over $Ru_{15}Co_{85}$ HNDs. **a**, Yield rate of NH_3 in different pH buffer solutions. **b**, Electrochemical quasi-*in situ* EPR tests at 0 V vs. RHE. **c**, **d**, Isotope-labeling electrochemical *in situ* ATR-FTIR spectra of $Ru_{15}Co_{85}$ HNDs using ${}^{14}NO_2^{-}$ (**c**) and ${}^{15}NO_2^{-}$ (**d**) as reactants. **e**, Electrochemical online DEMS of $NO_3^{-}RR$ over $Ru_{15}Co_{85}$.



Figure 6

Theoretical calculation results for NO3–RR. a, Phase diagram of various adsorbates on Ru15Co85. The vertical dotted line indicates the minimum applied potential required to reach thermodynamic feasibility. **b**, Mapping of the minimum applied potential required to reach the thermodynamic feasibility of various electrochemical transfer steps on RuxCoy surfaces. **c**, The activation and reaction energies of the N–O bond breaking elementary steps. The inserted images show the corresponding adsorption geometries and transition states on Ru15Co85 (101). **d**, The binding strength of *NO and *NH2 on RuxCoy surfaces. The ordinate is the absolute value of Binding strengthen.

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