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Xianqi Song

Jilin University <https://orcid.org/0000-0002-4758-5865>

Chang Liu

Jilin University <https://orcid.org/0000-0003-0824-1098>

Quan Li (✉ liquan777@jlu.edu.cn)

Jilin University <https://orcid.org/0000-0002-7724-1289>

Yanming Ma

Jilin University <https://orcid.org/0000-0003-3711-0011>

Changfeng Chen

Department of Physics and Astronomy, University of Nevada

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Intrinsic dense twinning via release of native strain

Xianqi Song,¹ Chang Liu,^{1,2} Quan Li,^{1,2,*} Yanming Ma,^{1,2,†} and Changfeng Chen^{3,‡}

¹State Key Lab of Superhard Materials and International Center for Computational Method and Software, College of Physics, Jilin University, Changchun 130012, China

²International Center of Future Science, Jilin University, Changchun 130012, China

³Department of Physics and Astronomy, University of Nevada, Las Vegas, Nevada 89154, USA

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Icosahedral boron-rich solids exhibit outstanding mechanical properties, but the crystal structure of this prominent class of materials has long remained enigmatic. Here, we report on a surprising discovery of an unprecedented twinning induced structural stabilization that creates highly twinned crystal structures that are stabler than the prevailing single crystal structure comprising complex multi-atom structural units. This phenomenon is showcased via a symmetry guided optimization process that produces a series of increasingly twinned B_4C structures with progressively lower energy below that of the single crystal, and this behavior also occurs in multiple other boron-rich solids. These findings unveil a distinct paradigm of defect (twinning) induced structural stabilizing mechanism that reduces energy via release of native strains built in the complex structural units of the single crystal, creating an exceptional category of materials that comprise multiple domains of intrinsic dense twinning in the crystal structures.

A fundamental problem in condensed matter physics is to determine stable crystal structure and elucidate the influence on structural stability by defects that are ubiquitously present, by nature or by design, in all materials. It is generally accepted that the stable structure of a crystal hosts a periodic bonding pattern satisfying one of 230 space group symmetries and that defects such as vacancies, dislocations or grain boundaries break bonding symmetries and raise crystal energy¹⁻³. This long established paradigm is rooted in considerations of chemical bonding and crystal symmetry, and is broadly adopted as the basis for constructing structural prototypes and evaluating structure-property relations⁴⁻⁷. Exceptions, however, may arise in materials comprising multi-atom structural units with complex intra-unit and inter-unit bonding connection patterns that create intrinsic strains in the single crystal, thereby opening a path for defect mediated bonding adjustment to release the native strains and stabilize the structure. To demonstrate this scenario, we showcase a distinct group of boron-rich compounds for a systematic exploration and an in-depth elucidation.

The light-element boron-rich compounds B_nX ($X=C, N, O, \text{etc.}$) possess low density and high strength that are favorable for wide applications⁸⁻¹¹. The constituent atoms in these compounds have similar electronegativity values that facilitate formation of B-X covalent bonds, and multi-center boron bonds promote diverse bonding patterns¹², generating structural units containing a large number of atoms, which greatly increases the difficulty for crystal structure determination. Meanwhile, close atomic radii of the X and B atoms make it hard for experimental distinction of atomic occupation sites, hindering accurate structural characterization. Early studies introduced the rhombohedral prototype structure containing icosahedral cages¹³, which has been adopted for a large family of compounds, such as B_4C , $B_{13}C_2$, $\alpha\text{-B}$, and B_6X ($X=N, O, Si, P, S$)¹⁴. Later experimental and computational advances further identified that the sta-

ble structure of B_4C comprises $B_{11}C_p$ (CBC) icosahedral units with carbon atom substituting for boron on the polar site (C_p) instead of the equatorial site (C_e) of the B_{12} unit while the displaced boron atom joins the original three-atom carbon segment to form a CBC chain in the crystal structure¹⁵⁻¹⁸. Employing various crystal structure search methods¹⁹⁻²², several recent studies predicted nonrhombohedral structures of $\alpha\text{-B}$, B_4C , B_6N , and B_6O ²³⁻²⁶.

Grain boundaries are a common type of structural defects, among which coherent twin boundaries (TBs) are most stable because the favorable lattice matching at the boundary minimizes the energy cost of forming the defective TBs relative to the single crystal²⁷. Crystal twinning further enriches structural diversity of boron-rich complex covalent bonding solids. Synthesized boron-rich solids are often densely twinned²⁸⁻³³, and first-principles calculations show that twinning raises the system energy but may also strengthen the resulting crystal structure^{26,34-40}, which is similar to observations made in nanotwinned diamond and cubic boron nitride⁴¹⁻⁴⁴. Despite extensive past studies, there is still a lack of understanding of the nature of TBs in boron-rich solids, especially the rich variety and complex hierarchy of their structural forms and the atomistic mechanisms that determine their relative stability.

In this work, we report on an intriguing crystal twinning induced structural stabilization mechanism that optimizes the relation of local bonding symmetry and global stacking symmetry in a broad class of icosahedral boron-rich solids, which host complex multi-atom cages and chains as the basic structural building units. Remarkably, a symmetry guided optimization procedure produces twinning structures with energies that are below that of the established single crystal and become progressively lower with rising defect (twinning) density. This behavior runs counter to the common wisdom that structural defects would raise the crystal energy.

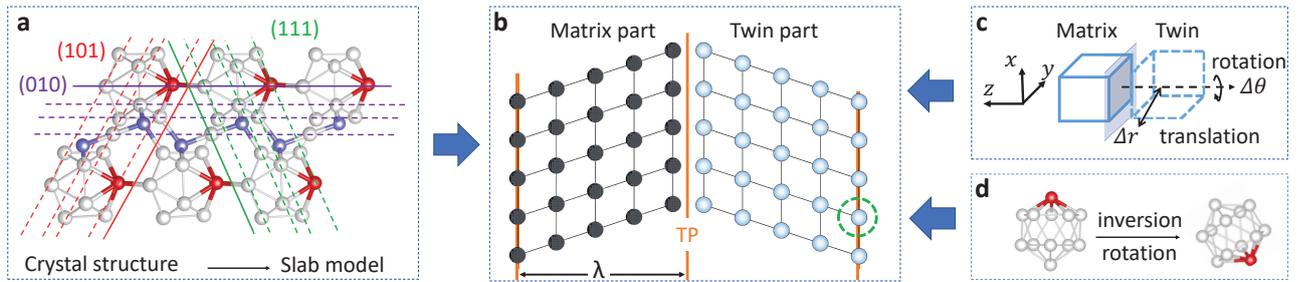


FIG. 1: **Construction of twinned crystal.** **a** A slab is cut from the single crystal to construct the matrix part of a TB structure. Here we use B_4C as an exemplary material template, where boron atoms, carbon atoms on CBC chains and carbon atoms on icosahedra are represented by gray, lavender and red spheres, respectively. The solid lines indicate suitable cutting planes in the (101), (111), and (010) orientations for building the matrix part, while the parallel dotted lines in each orientation show improper cutting planes (more discussion on TP construction is given in Supplementary Information). **b** The twinning structure is formed by joining the matrix part and twin part at the twin plane, and the spacing between adjacent TPs is set by the thickness (λ) of the slab. **c** Splicing the matrix and twin part via selective translation or rotation of the twin part to achieve favorable bonding configurations at the TB. **d** Adjustment of local structure units (e.g., icosahedral $B_{11}C$ cage) via selective inversion or rotation for further optimization of the bonding connection at the TB.

95 Such unusual behaviors stem from a twinning-mediated¹³²
 96 release of native strains inherent in the complex bonding¹³³
 97 network, and this phenomenon is widely present in a¹³⁴
 98 variety of icosahedral boron-rich solids, such as B_4C ,¹³⁵
 99 B_6O , B_6N , and $B_{13}CN$, that are conducive to hosting¹³⁶
 100 multiple energetically close densely twinned crystal¹³⁷
 101 domains. We further identified a broader variety of¹³⁸
 102 twinned boron-rich crystals that exhibit energies only¹³⁹
 103 slightly above that of the single crystal. These findings¹⁴⁰
 104 showcase a robust twinning-induced crystal stabilization¹⁴¹
 105 mechanism among a prominent class of complex covalent¹⁴²
 106 bonding crystals, creating a new paradigm that enriches¹⁴³
 107 fundamental structure-stability relations for defects in¹⁴⁴
 108 crystalline solids.

109 Results

110 Principles and procedures in constructing

111 **twinned structure.** Here, we construct and evaluate¹⁴⁸
 112 TBs in a variety of boron-rich solids, focusing on¹⁴⁹
 113 the effect of bonding symmetry on the total energy.¹⁵⁰
 114 The procedure for constructing twinning structures is¹⁵¹
 115 illustrated in Fig. 1 for the exemplary case of B_4C . The¹⁵²
 116 first step is to identify a suitable twin plane (TP) among¹⁵³
 117 various crystal orientations and cut a slab from the¹⁵⁴
 118 single crystal (Fig. 1a). A TB is then formed by joining¹⁵⁵
 119 a pair of matrix and twin parts that are in mutual¹⁵⁶
 120 mirror symmetry at a specific TP with adjustable slab¹⁵⁷
 121 thickness (Fig. 1b). Icosahedral boron-rich compounds¹⁵⁸
 122 contain complex, non-atom structural units, such as¹⁵⁹
 123 cages and chains, and different TP choices even with¹⁶⁰
 124 the same Miller index may lead to distinct TBs. To¹⁶¹
 125 obtain optimal TP, it is imperative to find good bonding¹⁶²
 126 connections between the matrix and twin part via¹⁶³
 127 crystal matching operations, such as translational and¹⁶⁴
 128 rotational adjustments, as illustrated in Fig. 1c. More¹⁶⁵
 129 over, positional variations of the structural units with¹⁶⁶
 130 different bonding patterns (Fig. 1d) may also notably¹⁶⁷
 131

132 impact structural stability³⁹. Since the TB dictated
 133 mirror symmetry alters the atomic positions and crystal
 134 orientations of the structural units from those of the
 135 single crystal, compatibility of the local structural-unit
 136 bonding symmetry with global TB stacking symmetry
 137 plays a crucial role in minimizing the twinning energy.

Following the procedure outlined above, we have
 138 constructed and examined a series of twinned B_4C
 139 crystals based on the prevailing single-crystal structure
 140 $R-B_{11}C_p(CBC)$. We find that bonding alignment at
 141 TP has a major impact on the relative stability of the
 142 resulting twinning structure. The twinned crystals with
 143 (111) and (101) oriented TPs (Supplementary Fig. 1)
 144 require substantial bonding adjustments across the TPs
 145 as outlined in Fig. 1c, d, including new bonding connec-
 146 tions to avoid the energetically unfavorable intra-cage C_e
 147 position and intercage C_p-C_p bonds. Meanwhile, for the
 148 (010) oriented TP, TBs can be formed via orientational
 149 adjustments of the CBC chains and icosahedral cages to
 150 satisfy the mirror symmetry without the need to change
 151 any bonding connections inherited from the original
 152 single crystal. This preservation of bonding connectivity
 153 makes the (010)-twinned B_4C energetically favorable.

154 Structural stability of twinned B_4C crystals.

155 Twinned crystals possess the characteristic symmetry
 156 with the lattice sites on the two sides of TPs forming
 157 mutual mirror images; but for multi-component materials,
 158 occupation of the lattice sites by different atomic species
 159 can lead to bonding configurations that do not strictly
 160 obey the mirror symmetry. In the case of B_4C , a (010)
 161 oriented twin model, referred to as Twin A_n hereafter
 162 (Fig. 2a-c) was previously proposed. In this structure,
 163 the lattice sites obey the TP mirror symmetry before the
 164 structural relaxation caused by atomic (carbon) substi-
 165 tution in the otherwise all-boron B_{12} icosahedral cage;
 166 however, the energetically dictated incorporation of the

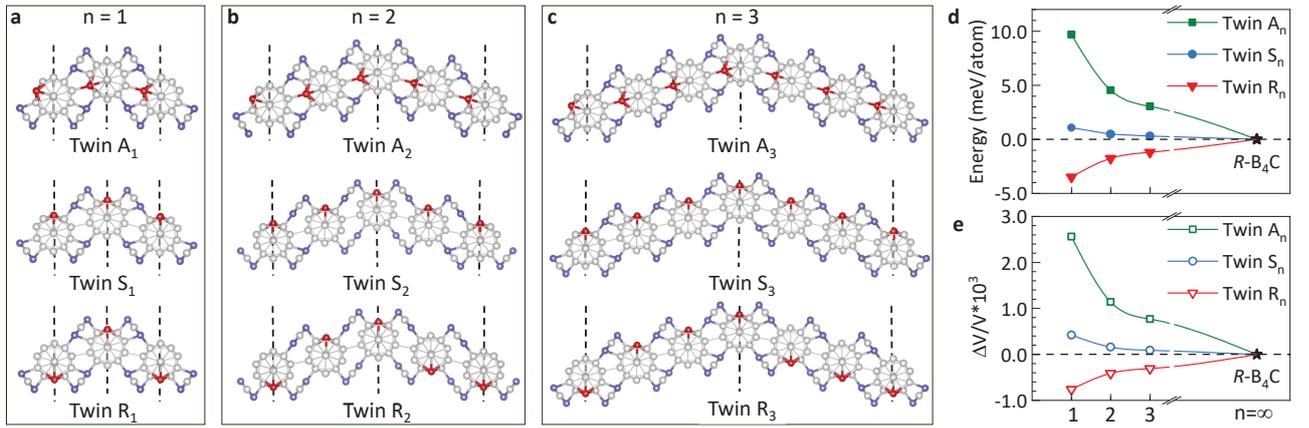


FIG. 2: **Structures and stability of twinned B₄C.** Three lowest-energy twinned crystals, Twin A_n, Twin S_n and Twin R_n, with distinct (010) slab thicknesses marked by structure-unit index **a** $n=1$, **b** $n=2$, and **c** $n=3$. The incorporation of a carbon atom into each B₁₁C cage causes a structural polarization as indicated by the directional alignment of the carbon atoms in the cages with respect to each other and to the TPs represented by vertical dashed lines (structural details are given in Supplementary Tables 1-3). **d,e** Energy and volume changes relative to the single crystal values, which are set to zero. In the $n \rightarrow \infty$ limit, all the twinned crystals tend to the prevailing single crystal rhombohedral R-B₄C structure.

169 carbon atom generates distortion in the resulting B₁₁C₂₀₅
 170 cage and breaks the structural symmetry at the atomic₂₀₆
 171 bonding level¹⁵⁻¹⁸. As a result, the bonding pattern in₂₀₇
 172 the twinned crystal becomes asymmetric about the TP₂₀₈
 173 (quantitative details are shown below); consequently, the₂₀₉
 174 resulting Twin A crystals no longer obey the mirror sym-₂₁₀
 175 metry, thus violating the basic requirement for TBs. To₂₁₁
 176 remedy this problem, we have constructed two different₂₁₂
 177 (010) oriented twinned crystals by rotating the B₁₁C cage₂₁₃
 178 to align the carbon atoms on the cages inside TPs to sat-₂₁₄
 179 isfy the full mirror symmetry of the structure units rela-₂₁₅
 180 tive to the TP at the atomic bonding level. The resulting₂₁₆
 181 twinned crystals, referred to as Twin S_n and Twin R_n,₂₁₇
 182 respectively, as also shown in Fig. 2a-c, exhibit detailed₂₁₈
 183 matching between the local (on-cage) bonding and TP₂₁₉
 184 mirror symmetry. 220

185 To assess the structure-stability relation of the low-₂₂₁
 186 energy twinned B₄C, we evaluate the energy variations₂₂₂
 187 with changing twinning density (Fig. 2d). Rising twin-₂₂₃
 188 ning density causes an obvious energy uptick in Twin₂₂₄
 189 A_n structures; meanwhile, Twin S_n series remain nearly₂₂₅
 190 degenerate with the single crystal even at the highest₂₂₆
 191 twinning density ($n=1$). Most remarkably, the energies₂₂₇
 192 of Twin R_n structures become progressively lower than₂₂₈
 193 the single crystal as twinning density increases. 229

194 The sensitive microstructure dependent energy of₂₃₀
 195 twinned B₄C showcases two major channels of strain₂₃₁
 196 release for structural optimization. The first channel is₂₃₂
 197 associated with the alignment of the carbon atom on₂₃₃
 198 the icosahedral cage that maximizes the local bonding₂₃₄
 199 symmetry with respect to the TP, thus minimizing the₂₃₅
 200 strains inside the cages. Specifically, the rotation of₂₃₆
 201 the B₁₁C_p cage to align the carbon atoms on the TP₂₃₇
 202 optimizes the local bonding symmetry relative to the₂₃₈
 203 TP and lowers the energy via strain reduction inside the₂₃₉
 204 icosahedral cages, which is responsible for the signifi-₂₄₀

cantly reduced energy of Twin S_n versus Twin A_n. The second channel is related to the more subtle long-range strains built in the single crystal, which is reduced by breaking the translational symmetry via alternating structural (cage) polarization marked by the position of the carbon atom on the B₁₁C_p cage, which further lowers the crystal energy of Twin R_n and, surprisingly, brings the energy of the twinned B₄C crystal below that of the single crystal.

Discussion

Bonding configurations of single and twinned B₄C crystals. To elucidate the microstructure-energy relation, we make a comparative study of the bonding structures of Twin A₁, Twin S₁ and Twin R₁ in comparison with the single crystal R-B₄C. The results (Fig. 3) show that the single crystal hosts an inhomogeneous distribution of bond lengths throughout the structural units. The average bond lengths on the two halves of the icosahedral B₁₁C cage are notably different and so are the B-C bond lengths on the CBC chain, all of which indicate the presence of considerable native strains that cause bonding distortions in the crystal. Meanwhile, Twin A₁ possesses nearly identical bond-length disparities, reflecting similar amount and extent of native strains in the crystal, but the average bond lengths on the cage is slightly longer, leading to the higher energy compared to the single crystal. With the rotational adjustment aligning the on-cage carbon atoms inside the TPs, Twin S₁ hosts a more symmetric bonding arrangement relative to the TP, with identical average bond lengths on the two halves of each cage, but the results on cages in adjacent layers are different and the C-B bonds on the CBC chain also have different lengths, indicating reduced but residual strains in the crystal structure. Finally, after further adjusting the adjacent cages by alternating the alignments of the

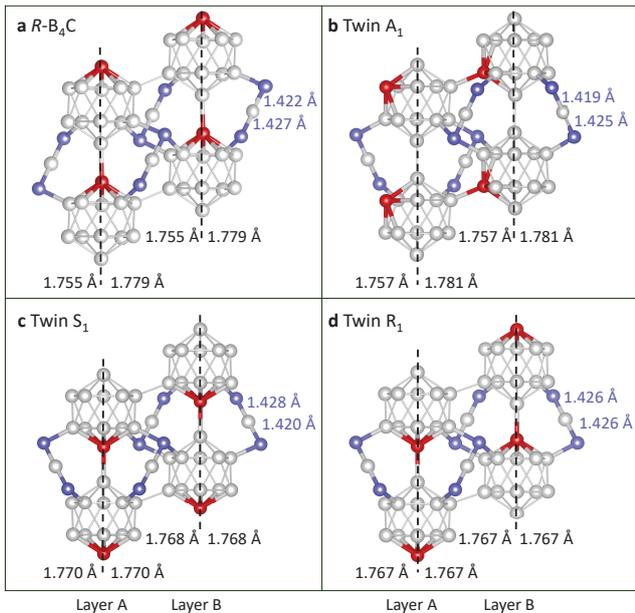


FIG. 3: **Bonding configurations of single and twinned B_4C crystals.** **a** Single crystal $R-B_4C$. **b** Twin A_1 . **c** Twin S_1 . **d** Twin R_1 . Listed are the average bond lengths on the left and right halves of the $B_{11}C$ cages in the two distinct primary stacking layers and those on the adjacent CBC chain in each case (details on bond-length data are given in Supplementary Table 4). The vertical dashed lines represent the (010) plane in the single crystal and TPs in the twinned crystal structures.

carbon-atom dictated structural polarization of the cages in neighboring slabs, Twin R_1 exhibits fully symmetric bonding arrangements with identical average bond lengths on the two halves of every cage and the same C-B bond lengths on the CBC chain throughout the crystal, indicating the effective removal of native strains originally built in the single crystal. This symmetry guided strain release creates a fully relaxed crystal with energy lower than the single crystal. The intrinsic strain states of various crystal structures are reflected in their volume change (Fig. 2e), which correlates perfectly with the energy (Fig. 2d), showing the strain effect on structural relaxation. This analysis highlights the mechanisms for energetic optimization by TB bonding adjustments.

Releasing strain energy via paired structural defects with contrasting topological features is known to occur in diverse physical systems, such as paired dislocations with opposite chirality or handedness in solids and liquid crystals^{45–47} or vortex-antivortex pairs in magnets^{48–52} and superconductors^{48–52}. These phenomena share the overarching mechanism in that distortions caused by individual defects are effectively compensated by the properly paired defects, thereby reducing the overall disturbance to the system thus lowering the energy cost. The $R-B_4C$ crystal hosts an intrinsic microstructural polarization due to the carbon atom incorporation into the boron icosahedron, which produces bonding

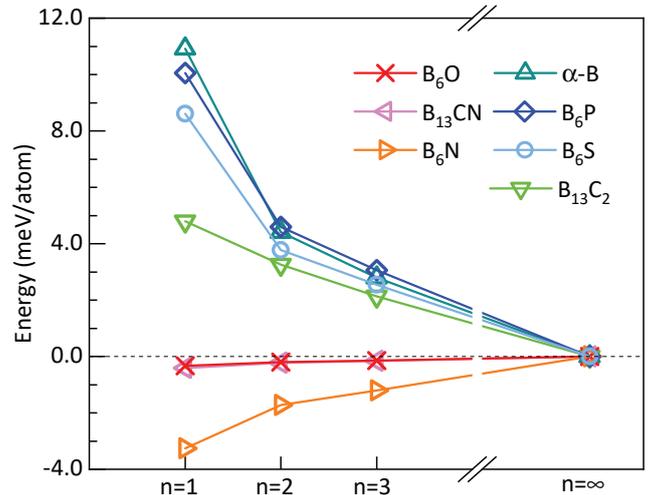


FIG. 4: **Stability of several representative twinned boron-rich solids.** Calculated energies of the twinning structures measured relative to the respective single-crystal results versus slab thickness index n for indicated select boron-rich solids.

distortions and the resulting native strains. The alternating stacking pattern with opposite polarization in neighboring matrix and twin parts of Twin R_n crystal alleviates the long-range strain, leading to steadily decreasing energy below the single-crystal value with rising twin density. This finding expands the long-established optimally paired-defect stabilization paradigm to include TB induced strain release, which produces remarkable energetic optimization that is better than the original defect-free system, greatly enriching this prominent class of physical phenomena.

Structure and stability of several twinned boron-rich solids. To evaluate twinning stabilization of crystal structure as a broader phenomenon, we further examined a range of boron-rich solids and identified cases of twinning structures that are stabler than their respective prevailing single crystals (Fig. 4). A twinned $B_{13}CN$ structure (see Supplementary Fig. 2 for a snapshot) is lower in energy than single crystal $R-B_{13}CN$ ⁵³ driven by a rotation-inversion chain configuration at the TB that is effective in releasing the intrinsic strain energy built in the single crystal. Additional cases involve $R-B_6N$ and $R-B_6O$ (see Supplementary Fig. 2 for snapshots). Here, both N and O atoms have smaller atomic radii than B atom and the strong covalent B-N and B-O bonds tighten the distance between the neighboring cages, generating native strains in the single crystal. Twinning modifies the orientation of the B-N and B-O bonds to release the intrinsic strains, thereby lowering the energy to below the single-crystal values.

We also constructed twinning structures of $B_{13}C_2$, $\alpha-B$, B_6P , and B_6S (see Supplementary Fig. 2 for structural snapshots), which exhibit slightly higher

but competitive energies compared to their respective single crystals (Fig. 4). Several factors prevent deeper twinning induced energy reduction here. For $B_{13}C_2$ the BCB chains do not support the rotation-inversion mechanism seen in $B_{13}CN$; meanwhile, B_6P and B_6S host large inter-cage distances due to the large radii of P and S atoms and α -B comprises multicenter bonding configurations¹², all of which do not promote twinning induced strain release. Nevertheless, the energetic relations indicate that these boron-rich solids also favor multi-domain twinning structures that coexist with the single crystal phase.

In summary, we have developed a method for atomic construction and evaluation of stable twinning structures in complex covalent crystals via selection of energetically favorable twin-boundary orientation and stacking pattern combined with optimization of bonding symmetry of multi-atom structural units relative to the twin plane. This method has led to the surprising discovery of intrinsically densely twinned structures among icosahedral boron-rich solids, including B_4C , $B_{13}CN$, B_6O , and B_6N , which are stabler than their respective prevailing single crystals. This discovery establishes a distinct paradigm of defect (twinning) induced stabilization of crystals comprising complex structural units, resulting in multiple domains with variable twinning densities that coexist with the single crystal phase.

It is interesting to note that experimental techniques commonly used to reduce defect density and purify single crystal structure, such as thermal annealing would have the opposite effect here by enhancing the density of twinning phases that have lower energy than the single crystal phase; consequently, pure phase single crystal structures are necessarily unattainable in this class of materials. Instead, specimens obtained under variable synthesis conditions may comprise different degrees of mixture of distinct twinning structures that are energetically close to each other. It is possible that other energetically competitive twinning patterns also appear in some cases. These extraordinary material behaviors expand the textbook description of the effect of twinning on crystal stability and define a distinct category of intrinsically twinned materials. These results enrich the knowledge and understanding of twinning in crystalline solids and provide a unique platform to explore the structure-property relations stemming from the newly identified native microstructural features, which may offer insights into distinct mechanisms governing deformation modes and stress responses that dominate structural evolution and dictate mechanical properties, opening unique paths for rational performance tuning and optimization.

Methods

First-principles calculations. The first-principles total-energy calculations are carried out using local density approximation exchange correlation potential^{54,55} as implemented in the Vienna *Ab initio* Simulation Package⁵⁶, adopting the projector augmented wave approach⁵⁷ to describe electron-ion interaction with $2s^22p^1$ and $2s^22p^2$ valence electron configurations for boron and carbon atoms, respectively. The plane wave basis set is constructed with an energy cutoff of 600 eV and the Brillouin zone is sampled under the Monkhorst-Pack scheme⁵⁸ with a k -point resolution of $2\pi \times 0.03 \text{ \AA}^{-1}$, achieving an energy convergence around 1 meV per atom.

Construction of twinned B_4C crystal. A twinned B_4C crystal is constructed by cutting a slab from its single crystal along a selected crystal plane, which serves as the twin plane (TP), then placing the mirror image of the slab across the TP to form the twinned structure. There are, however, constraints imposed by symmetry and energy that guide the choice of the most proper TPs. For example, the $(1\bar{1}0)$ plane is a mirror-symmetry plane of single-crystal B_4C and thus cannot serve as a TP since the construction simply reproduces the original single crystal. On the other hand, the (111) plane can serve as a TP, but relative displacements are needed to match the matrix and twin part, as illustrated in Supplementary Fig. 1a. Here, the TPs are the vertical bisectors of the CBC chain with the $B_{11}C$ cages located on both sides of the TPs. Further adjustments of carbon-atom positions are made to avoid the high-energy intra-cage C_e atoms and inter-cage C_p - C_p bonds. Meanwhile, the (101) plane that goes through the center B atom on the CBC chain does not bisect the atomic chain, making it an improper TP choice; but the (101) plane that separates adjacent icosahedra and CBC chains can serve as a proper TP, as illustrated in Supplementary Fig. 1b, where the equatorial-position planes or polar-position planes are not parallel to the TP. The structural adjustments relative to the single crystal introduced to achieve adequate bonding at the TP renders these twinned crystals to become less energetically favorable than the single crystal. The lowest energy twinned crystal is built with the TPs aligned in the (010) orientation as described in the main text of the paper. Here, the TPs traverse the B_{11} cages and divide them into equal halves without altering the CBC chains or breaking and rearranging bonds across the TPs, resulting in the most energetically favorable twinned B_4C crystal structure.

Data availability

The authors declare that the main data supporting the findings of this study are available within the article and its Supplementary Information file.

- * Electronic address: liquan777@jlu.edu.cn
- † Electronic address: mym@jlu.edu.cn
- ‡ Electronic address: changfeng.chen@unlv.edu
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Author contributions

Q.L., Y.M., and C.C. designed research; X.S. and C.L. performed research; X.S., C.L., Q.L., Y.M., and C.C. analyzed data and wrote the paper.

X.S. and C.L. contributed equally to this work.

Competing interests

The authors declare no competing interests.

Additional information

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Correspondence and requests for materials should be addressed to Quan Li, Yanming Ma or Changfeng Chen.

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