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# 1 Room-Temperature Long-Range Ferromagnetic Order in a Confined

## 2 Molecular Monolayer

3  
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17

18 **Abstract:** Pursuing new ferromagnetic systems could bring about research breakthroughs for magnetism.  
19 Currently, studies of magnetic ordering are expanding from interatomic to intermolecular exchange  
20 interactions to produce ferromagnets in inorganic/organic crystals<sup>1-3</sup>. To date, the crystalline framework  
21 used for stabilizing parallel spin alignment via an ordered lattice seems to be indispensable to  
22 ferromagnetism<sup>4</sup>. In our case, we demonstrated room-temperature long-range ferromagnetic order in  
23 two-dimensional confined cobaltocene (Co(Cp)<sub>2</sub>) molecular monolayers. In a confined van der Waals  
24 (vdW) interlayer space, spontaneous uniform spin orientation of Co(Cp)<sub>2</sub> can be settled. Ferromagnetic  
25 coupling is established by an intermolecular vibronic superexchange interaction<sup>4,5</sup>, a new long-distance  
26 exchange interaction in a cooperative dynamic Jahn-Teller (J-T)<sup>5,6</sup> Co(Cp)<sub>2</sub> monolayer. As expected, the  
27 confined Co(Cp)<sub>2</sub> monolayers exhibit a high ferromagnetic transition temperature (above room  
28 temperature) with a saturation magnetization up to 4 emu.g<sup>-1</sup>.  
29

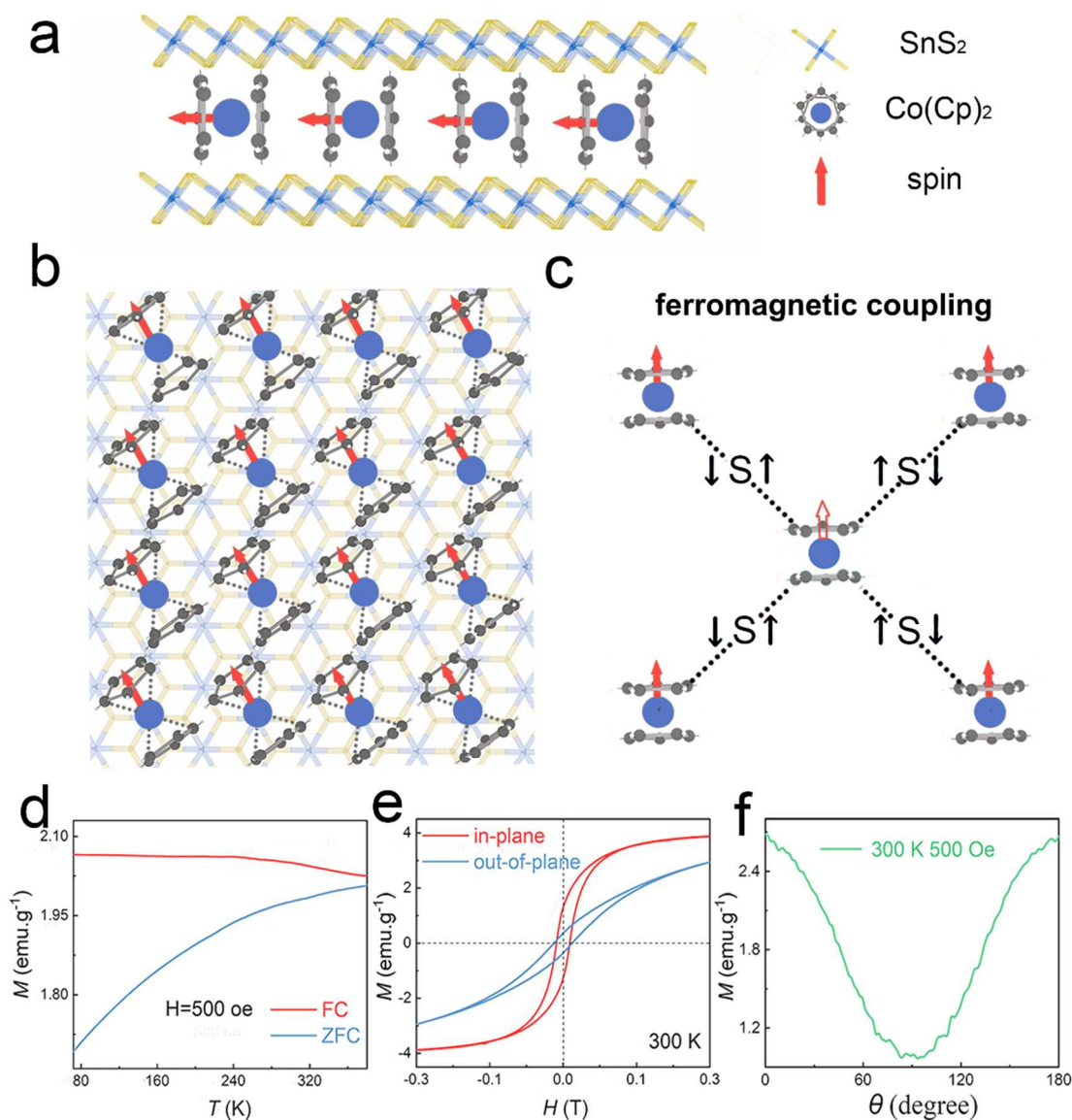
30 The origin of ferromagnetism, *i.e.*, spontaneous parallel spin ordering<sup>7,8</sup>, has been a long-standing pursuit  
31 of scientists. The spin-spin exchange interaction is the essential factor for ferromagnetic ordering<sup>9-12</sup>,  
32 which determines the ferromagnetic properties of a material, such as the transition temperature ( $T_c$ ) and  
33 saturation magnetization ( $M_s$ ). As the exchange interaction is extremely sensitive to the distance of  
34 adjacent spins<sup>13</sup>, ferromagnetism can restrictedly be found in a crystalline framework. Through  
35 translational symmetry, a crystalline framework with a uniform distance between adjacent lattice sites  
36 provides ideal conditions for spin ordering. Therefore, conventional ferromagnets with high  $T_c$  and large  
37  $M_s$  are found in highly crystalline inorganic systems with strongly bonded atoms, ranging from transition  
38 metals to lanthanide metals<sup>14,15</sup> and their oxides with perovskite and spinel structures<sup>16-18</sup>.  
39

40 Magnetic molecules are a new class of magnetic moment units of great potential interest, and currently,  
41 pursuing magnetic order in molecular systems, especially two-dimensional (2D) system beyond room

42 temperature is a very important research area. Different from conventional ferromagnetic systems relying  
43 on the exchange interaction of unpaired electrons of d or f shells, for molecular systems, the cooperative  
44 spin-spin exchange interaction to achieve ferromagnetic coupling is usually based on their p shells or  $\pi$ -  
45  $\pi$  stacking<sup>19,20</sup>, which enlarges the distance of the exchange interaction between the local magnetic  
46 moments. The long-distance exchange interaction in molecular systems brings a new system of  
47 ferromagnetic ordering beyond inorganic crystals, such as the Prussian blue framework<sup>21</sup> and metal-  
48 organic frameworks (MOFs)<sup>22</sup>. However, for molecular systems, a stable crystalline lattice is still  
49 indispensable to achieve ferromagnets with long-range ferromagnetic order. In fact, high-temperature  
50 ferromagnetic order is strongly dependent on stable lattice constraints for resistance to thermal agitation,  
51 which is more vulnerable in atomic limit<sup>23,24</sup>. Therefore, challenges still remain in robust ferromagnetic  
52 ordering of independent molecules and their aggregates in 2D limit, which have no stable crystalline  
53 lattice. And a new exchange interaction is required for long-range ferromagnetic order in molecular  
54 aggregate systems, which is vital for not only fundamentally understanding the nature of magnetism but  
55 also opening new avenues towards a new magnetic material platform.

56  
57 Herein, we achieved room-temperature long-range ferromagnetic order in a confined molecular  
58 monolayer as a new class of ferromagnetic configurations under atomic thickness. A molecular  
59 monolayer of  $\text{Co}(\text{Cp})_2$  can be achieved in a confined van der Waals (vdW) interlayer space. An  
60 intermolecular vibronic superexchange interaction, as a new exchange interaction induced by the  
61 cooperative vibronic Jahn-Teller (J-T) effect, produces strong ferromagnetic coupling between  $\text{Co}(\text{Cp})_2$   
62 molecules and realizes high  $T_c$  ( $> 300$  K) and large  $M_s$  ( $4 \text{ emu}\cdot\text{g}^{-1}$ ). The construction of robust  
63 intermolecular ferromagnetic coupling among 2D confined molecular aggregates will promote the  
64 understanding of the ferromagnetic mechanism and the search for low-dimensional ferromagnets.

65



66

**Figure 1: Ferromagnetism of the confined Co(Cp)<sub>2</sub> monolayers.** **a.** Scheme for confined Co(Cp)<sub>2</sub> between SnS<sub>2</sub> interlayers **b.** Top-view of the Co(Cp)<sub>2</sub>/SnS<sub>2</sub> structure. **c.** Scheme for the intermolecular vibronic superexchange interaction of adjacent Co(Cp)<sub>2</sub> molecules through S atoms. **d.** Magnetization versus temperature under a magnetic field of 500 Oe for ZFC and FC processes. **e.** In-plane and out-of-plane magnetization versus field hysteresis loops at 300 K. **f.** Magnetization versus angle under 500 Oe at 300 K.

67

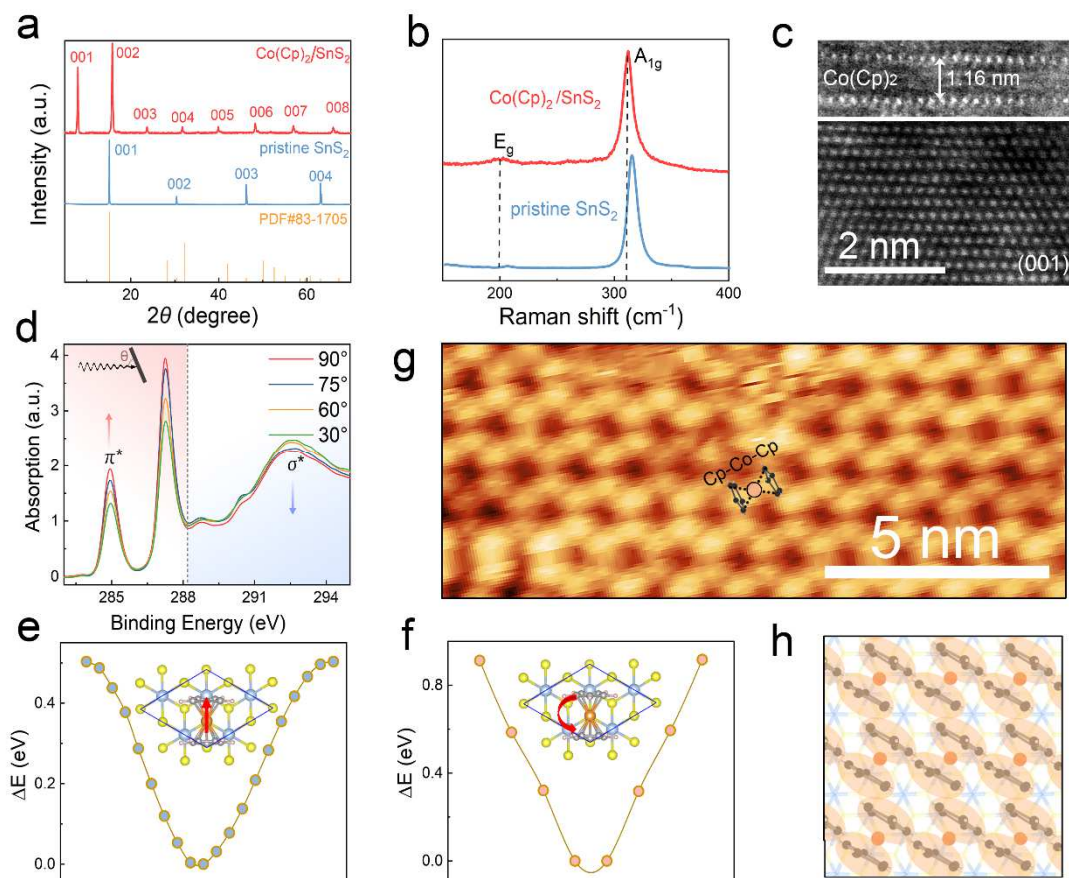
68 **Room-temperature ferromagnetic property:**

69 Ferromagnetic molecular monolayers confined in van der Waals (vdW) interlayer space have been  
 70 synthesized via a solvothermal method. SnS<sub>2</sub> is a typical transition metal dichalcogenide (TMD)  
 71 possessing a layered structure similar to MoS<sub>2</sub><sup>25</sup>; thus, the 2D interlayers in SnS<sub>2</sub> provide a confined  
 72 space for limiting the orientation of the structural and spin orders of Co(Cp)<sub>2</sub>, as shown in the scheme of  
 73 Fig. 1a. When a heterointerface between organic-2D materials is established, charge transfer plays a role  
 74 in the formation of organic acceptors on solid surfaces, resulting in strong bonding between the organic  
 75 molecules and the substrate<sup>26</sup> and leading to anisotropy<sup>27,28</sup>. Thus, a 2D ordered Co(Cp)<sub>2</sub> monolayer can  
 76 be formed in the SnS<sub>2</sub> interlayer space, as shown in Fig. 1b. The dynamic charge transfer from Co(Cp)<sub>2</sub>  
 77 molecules to SnS<sub>2</sub> not only introduces an interaction between adsorbates and the substrate but also

78 establishes an intermolecular vibronic superexchange effect<sup>29</sup> between two adjacent J-T molecules  
79  $\text{Co}(\text{Cp})_2$ <sup>30,31</sup> through S atoms (Fig. 1c), forming long-range ferromagnetic order.

80  
81 The room-temperature ferromagnetic signature of  $\text{Co}(\text{Cp})_2$  layers was affirmed by examining the  
82 temperature dependence of magnetization measurements, which were obtained on a vibrating sample  
83 magnetometer (VSM). Under a constant magnetic field ( $H = 500$  Oe), field cooling (FC) and zero-field  
84 cooling (ZFC) processes were applied to measure the temperature dependence of magnetization (M-T)  
85 curves from 70 to 380 K (Fig. 1d). For the FC curve, the magnetization presents a downward trend during  
86 the heating process, whereas for the ZFC curve, the magnetization shows an increase with elevating  
87 temperature. Note that no intersection exists between the FC and ZFC lines, indicating that the transition  
88 temperature of  $\text{Co}(\text{Cp})_2/\text{SnS}_2$  exceeds the range limit of the instrument (380 K). In comparison, the  
89 magnetic property measurements of the individual  $\text{SnS}_2$  crystal as well as  $\text{Co}(\text{Cp})_2$  powder show a  
90 diamagnetic property and a paramagnetic property, respectively, as displayed in Fig. S1. This result  
91 indicates that the ferromagnetic property comes from the combination of  $\text{Co}(\text{Cp})_2/\text{SnS}_2$ , namely, confined  
92  $\text{Co}(\text{Cp})_2$  monolayers. The magnetism of  $\text{Co}(\text{Cp})_2/\text{SnS}_2$  was further studied by taking isothermal  
93 magnetization versus field (M-H) curves at 300, 350 and 380 K. The obvious S-shaped curves  
94 accompanied by a typical magnetic hysteresis loop (Fig. S2) manifest the room-temperature  
95 ferromagnetic property of  $\text{Co}(\text{Cp})_2$  monolayers.

96  
97 In addition, the magnetic anisotropy, which is typical characteristics of two-dimensional magnets, can be  
98 observed in  $\text{Co}(\text{Cp})_2/\text{SnS}_2$ . The in-plane and out-of-plane M-H curves of the sample at 300 K are depicted  
99 in Fig. 1e. When the magnetic field is perpendicular to the c-axis of the sample plane, susceptibility  
100 saturation can be reached faster than in the vertical case, indicating that the easy magnetization axis  
101 (EMA) is along the in-plane direction. To further probe the anisotropy, VSM measurements with a rotator  
102 were performed under constant temperature ( $T = 300$  K) and magnetic field ( $H = 500$  Oe). A cosine curve  
103 with maxima at  $\theta = 0^\circ$  and  $180^\circ$  and a minimum at  $\theta = 90^\circ$  is displayed in Fig. 1f, further confirming that  
104 the EMA is along the in-plane direction. Based on the above results, we propose a strong ferromagnetic  
105 property and magnetic anisotropy in  $\text{Co}(\text{Cp})_2/\text{SnS}_2$  originating from the confined  $\text{Co}(\text{Cp})_2$  monolayers,  
106 implying the successful preparation of a 2D room-temperature ferromagnet.



**Figure 2: Structure of the confined  $\text{Co}(\text{Cp})_2$  monolayer.** **a.** XRD patterns of the  $\text{Co}(\text{Cp})_2/\text{SnS}_2$  and pristine  $\text{SnS}_2$ . **b.** Raman spectrum of a  $\text{Co}(\text{Cp})_2/\text{SnS}_2$  nanoflake and a pristine  $\text{SnS}_2$  nanoflake. **c.** HAADF-STEM cross-sectional images of a  $\text{Co}(\text{Cp})_2/\text{SnS}_2$  nanoflake from the side view (top) and top view (bottom). **d.** C K-edge XAS spectra of  $\text{Co}(\text{Cp})_2/\text{SnS}_2$  from  $30^\circ$  to  $90^\circ$ . Potential surface scanning through **e.** translation and **f.** rotation along the high-symmetry direction. **g.** Constant current STM image of a  $\text{Co}(\text{Cp})_2$  monolayer on a  $\text{SnS}_2$  surface. **h.** Scheme of the  $\text{Co}(\text{Cp})_2$  monolayer on a  $\text{SnS}_2$  surface.

107

108

### 109 Structure of $\text{Co}(\text{Cp})_2$ monolayers in $\text{SnS}_2$ interlayers:

110 The structural of confined molecular was confirmed using a combination of various techniques, as shown  
 111 in Fig. 2. Owing to the form of  $\text{Co}(\text{Cp})_2$  layers, the hybrid structure underwent substantial expansion  
 112 along the  $[001]$  zone axis, resulting in a significant increase in the interlayer space, which can be reflected  
 113 by the powder X-ray diffraction (XRD) pattern in Fig. 2a. After intercalation, the corresponding  
 114 interlayer distance of  $\text{Co}(\text{Cp})_2/\text{SnS}_2$  increases from  $6 \text{ \AA}$  to  $11.6 \text{ \AA}$ , with an observable peak shift towards  
 115 lower angles along with a new set of  $(00l)$  peaks, showing that the  $\text{Co}(\text{Cp})_2$  monolayers are in intact  $\text{SnS}_2$   
 116 interlayers. As shown in Fig. 2b, a slightly redshift is observed in the  $\text{Co}(\text{Cp})_2/\text{SnS}_2$  Raman peaks  
 117 compared with those of pristine  $\text{SnS}_2$ . This results from the weakening of the vdW forces induced by the  
 118 expanded interlayer gap<sup>32</sup>, indicating the existence of  $\text{Co}(\text{Cp})_2$ . The interlayer distance of  $\text{Co}(\text{Cp})_2/\text{SnS}_2$   
 119 can be specifically measured via high-angle annular dark-field scanning transmission electron  
 120 microscopy (HAADF-STEM) from the cross section. Using the Z-contrast imaging principle, visually  
 121 revealing the complete atomic structure in the  $\text{Co}(\text{Cp})_2/\text{SnS}_2$  system, the Sn atoms are found to maintain  
 122 the same layered parallel arrangement (Fig. 2c). The interlayer distance increases from  $5.7 \text{ \AA}$  to  $11.6 \text{ \AA}$ ,  
 123 which well matches the XRD patterns. In addition, a hexagonal lattice pattern is clearly observed by top-

124 view HAADF-STEM (Fig. 2c bottom), which well agrees with the SnS<sub>2</sub> template, revealing the retained  
125 good crystallinity after the formation of Co(Cp)<sub>2</sub> monolayers. The elements, structure and thermal  
126 stability of Co(Cp)<sub>2</sub>/SnS<sub>2</sub> are further shown in Fig. S3-S7. Therefore, through the solvothermal method,  
127 Co(Cp)<sub>2</sub> monolayers were successfully obtained in the SnS<sub>2</sub> interlayer space.

128

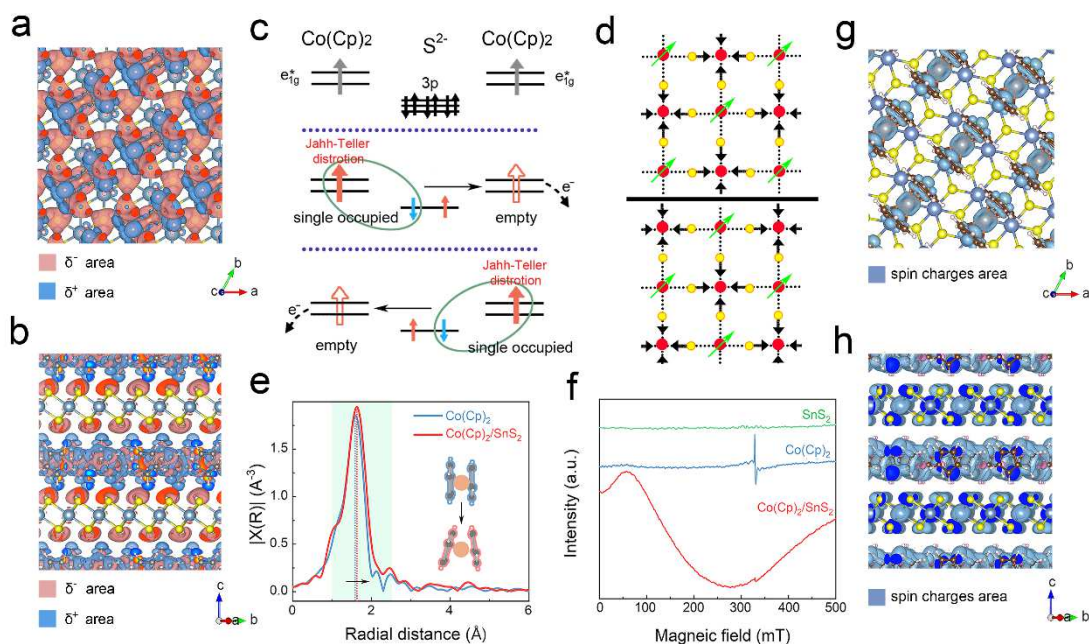
129 Due to the aromatic properties of the two characteristic cyclopentadienyl (Cp) rings, the orientation of  
130 confined Co(Cp)<sub>2</sub> molecules can be revealed using angle-resolved carbon K-edge X-ray absorption near-  
131 edge structure (XANES) spectra. The X-ray absorption structure (XAS) spectra for various angles ( $\theta$ ) of  
132 the incident light with respect to the Co(Cp)<sub>2</sub>/SnS<sub>2</sub> plane were collected, as shown in Fig. 2d. The 284.9  
133 and 287.3 eV peaks come from  $\pi^*$  resonances, while the others at 288.7, 290.4 and 292.6 eV originate  
134 from  $\sigma^*$  resonances. With increasing  $\theta$ , the intensity of the  $\pi^*$  resonances shows an uptrend, reaching a  
135 maximum at  $\theta = 90^\circ$ , while the trend of the  $\sigma^*$  resonances is the opposite. The increased intensity of the  
136  $\pi^*$  resonances demonstrates the process in which the incident light and 1s- $\pi^*$  become parallel, and the  
137 decreased intensity of the  $\sigma^*$  resonances reveals a perpendicular tendency, indicating that the Cp axes of  
138 Co(Cp)<sub>2</sub> molecules are parallel to the SnS<sub>2</sub> plane according to this linear dichroism.

139

140 Furthermore, the most stable configuration of confined Co(Cp)<sub>2</sub> between SnS<sub>2</sub> interlayers was confirmed  
141 with the aid of theoretical studies by scanning the potential energy surface, as shown in Fig. 2e and 2f.  
142 To validate the orientation of interlayer Co(Cp)<sub>2</sub>, we compare the energy of intercalated Co(Cp)<sub>2</sub> relative  
143 to translational and rotational motions confined in the in-plane direction. By scanning the angular and  
144 translational potential energy surface, the initial configuration of interlayer Co(Cp)<sub>2</sub> between SnS<sub>2</sub> layers  
145 is found to remain the lowest in energy. Moreover, the energy difference between the most stable  
146 configuration and the configurations after translational and rotational motions could be as large as 0.50  
147 eV and 0.92 eV, indicating a unified orientation for confined Co(Cp)<sub>2</sub> molecules. Based on the above  
148 analysis, the orientation of interlayer Co(Cp)<sub>2</sub> is confirmed and robust. Thus, ordered 2D arrays can be  
149 obtained. A scanning tunnelling microscope (STM) was operated to image the Co(Cp)<sub>2</sub> monolayer arrays  
150 on SnS<sub>2</sub>. As shown in Fig. 2g, all Co(Cp)<sub>2</sub> molecules lie flat in a particular direction with the Cp plane  
151 erect, which is consistent with the angle-resolved XAS analysis. Particularly, due to the interface  
152 rearrangement effect, 2D quasiperiodic arrays of Co(Cp)<sub>2</sub> can be observed. The two-lobe symmetrical  
153 patterns, representing the double Cp rings, partially overlap, which is consistent with the observed  
154 phenomena in previous experiments<sup>33</sup> and reflects the exchange interaction between Co(Cp)<sub>2</sub> molecules.  
155 The scheme under the STM image shows the arrangement of the confined Co(Cp)<sub>2</sub> monolayer more  
156 specifically. This molecular order provides favourable conditions for the formation of an intermolecular  
157 exchange interaction.

158





159

**Figure 3: Charge density distribution and ferromagnetic coupling of Co(Cp)<sub>2</sub> monolayer.** Charge density difference between the SnS<sub>2</sub> substrate and Co(Cp)<sub>2</sub> molecules: **a.** top view and **b.** side view. The isosurface values are  $4 \times 10^{-3}$  and  $1 \times 10^{-3}$  e/bohr<sup>3</sup>, respectively. The red area represents the electron-gain ( $\delta^-$ ) area, and the blue indicates the electron-loss ( $\delta^+$ ) area. **c.** Intermolecular vibronic superexchange of Co(Cp)<sub>2</sub> via S atoms. **d.** Long-range magnetic ordering in the Co(Cp)<sub>2</sub> lattice; the red circle is Co(Cp)<sub>2</sub>, and the yellow circle is S. **e.** Co FT-EXAFS spectra for isolated Co(Cp)<sub>2</sub> (blue) and confined Co(Cp)<sub>2</sub> (red). **f.** ESR spectra of pristine SnS<sub>2</sub> (green), Co(Cp)<sub>2</sub> (blue) and Co(Cp)<sub>2</sub>/SnS<sub>2</sub> (red). Spatial distribution of spin charges for Co(Cp)<sub>2</sub>/SnS<sub>2</sub> from the top view **g.** and side view **h.** The isosurface values are  $4 \times 10^{-3}$  and  $1 \times 10^{-3}$  e/bohr<sup>3</sup>, respectively.

160

### 161 Intermolecular vibronic superexchange interaction in the Co(Cp)<sub>2</sub> monolayer:

162 Recently, exchange interactions have been found between two Ni(Cp)<sub>2</sub><sup>34,35</sup>, which confirms the existence  
 163 of intermolecular exchange interactions. To better understand the ferromagnetic coupling of confined  
 164 Co(Cp)<sub>2</sub> layer, first-principles calculations were performed to investigate the geometric, electronic and  
 165 magnetic properties of the intercalated system. As shown in the differential charge density between  
 166 Co(Cp)<sub>2</sub> and the SnS<sub>2</sub> substrate (Fig. 3a, 3b), the electron accumulation and depletion regions are mainly  
 167 located at the surface S atoms of SnS<sub>2</sub> and the Cp rings, respectively, indicating significant charge transfer  
 168 between confined Co(Cp)<sub>2</sub> and the SnS<sub>2</sub> substrate. Bader charge analysis shows that one Cp molecule  
 169 could transfer as many as  $0.47 e^-$  electrons to the SnS<sub>2</sub> substrate, and the binding energy of Co(Cp)<sub>2</sub> is  
 170 calculated to be 1.54 eV per Co(Cp)<sub>2</sub> molecule, inducing a strong interaction between Co(Cp)<sub>2</sub> molecules  
 171 and the substrate (Table S1 and Fig. S8), thus facilitating unified orientation of confined Co(Cp)<sub>2</sub>  
 172 molecules to form Co(Cp)<sub>2</sub> monolayers.

173

174 The fact that approximately  $0.5 e^-$  effective average charge transfer, means that half of the Co(Cp)<sub>2</sub> will  
 175 transfer an electron to S at a certain instant (correspondingly become [Co(Cp)<sub>2</sub>]<sup>+</sup>). We proposed that  
 176 strong ferromagnetic coupling can be achieved in the configuration of Co(Cp)<sub>2</sub> -S<sup>2-</sup>- [Co(Cp)<sub>2</sub>]<sup>+</sup> in the  
 177 Co(Cp)<sub>2</sub> lattice via intermolecular vibronic superexchange. As the ground state of Co(Cp)<sub>2</sub> is  $e_{1g}^4 a_{1g}^2$   
 178  $e_{1g}^{*1}$  (Fig. S9), with half-filled  $e_{1g}^*$  orbital, the J-T effect<sup>36</sup> must be taken into consideration, which leads  
 179 to structural distortion; while there will be no J-T distortion happened on [Co(Cp)<sub>2</sub>]<sup>+</sup> with an empty  $e_{1g}^*$

180 orbital. Owing to the repel and attraction from  $\text{Co}(\text{Cp})_2$  and  $[\text{Co}(\text{Cp})_2]^+$ , respectively, the middle  $S^2$ -shift  
181 to  $[\text{Co}(\text{Cp})_2]^+$ . Based on the Pauli exclusion principle, exchange coupling can be established between the  
182  $\text{Co}(\text{Cp})_2$  and  $[\text{Co}(\text{Cp})_2]^+$  by virtual electron transfer under the framework of superexchange, as shown in  
183 Fig. 3c. According to Goodenough–Kanamori rules, there would be ferromagnetic coupling when the  
184 virtual electron transfer is from a half-filled to an empty orbital, such as the case of  $\text{Co}(\text{Cp})_2$  and  
185  $[\text{Co}(\text{Cp})_2]^+$ . Because the J-T distortion is dynamic, the transform between  $\text{Co}(\text{Cp})_2$  and  $[\text{Co}(\text{Cp})_2]^+$  is  
186 cooperative and in a dynamic equilibrium, which couples the spins of neighboring  $\text{Co}(\text{Cp})_2$  to form long-  
187 range ferromagnetic planes. The mechanism of long-range ferromagnetic order in the  $\text{Co}(\text{Cp})_2$  monolayer  
188 is displayed in Fig. 3d. In addition, the existence of a cooperative dynamic J-T effect in the 2D  $\text{Co}(\text{Cp})_2$   
189 monolayer generates intrinsic magnetic anisotropy via spin-lattice coupling, achieving 2D  
190 ferromagnetism.

191

192 To investigate the electronic structure and valence state of confined  $\text{Co}(\text{Cp})_2$ , Co K-edge XANES spectra  
193 were obtained. An obvious pre-edge peak at 7730.25 eV can still be observed, suggesting a typical Co-  
194 Cp coordination structure, as shown in Fig. S10. The Fourier transform extended X-ray absorption fine  
195 structure (FT-EXAFS) spectra reveal a slight increase in the peak position from 1.60 to 1.63 Å, reflecting  
196 an increase in the bond length between the central Co atom and the Cp rings, as depicted by the inset in  
197 Fig. 3e. Theoretical calculations confirm the twisting of the two Cp rings (Fig. S11), causing the Co-C  
198 bond length to be dissimilar and elongate. The breaking of the Cp parallel structure intensifies the  
199 anisotropy of  $\text{Co}(\text{Cp})_2$  and influences the degree of J-T distortion, promoting vibronic superexchange  
200 between molecules. Electron spin resonance (ESR) spectra were obtained to detect the magnetic  
201 interaction in  $\text{Co}(\text{Cp})_2$  monolayers. No ESR signal was recorded for the pristine  $\text{SnS}_2$  crystal, as shown  
202 in Fig. 3f, due to its nonferromagnetic state. The sharp signal at approximately 320 mT in the pure  
203  $\text{Co}(\text{Cp})_2$  spectrum can be assigned to the signal of Cp rings<sup>37,38</sup>. Of note, this signal is retained in confined  
204  $\text{Co}(\text{Cp})_2$ , reflecting the undestroyed molecular structure for confined  $\text{Co}(\text{Cp})_2$ . However, the signal  
205 intensity is greatly reduced compared to pure  $\text{Co}(\text{Cp})_2$ , reflecting the electron transfer from  $\text{Co}(\text{Cp})_2$  to  
206 the  $\text{SnS}_2$  layer. In addition, a new broad signal is resolved at approximately 150 mT in the  $\text{Co}(\text{Cp})_2/\text{SnS}_2$   
207 spectrum, which indicates a ferromagnetic interaction in the system, reflecting the ferromagnetism in  
208  $\text{Co}(\text{Cp})_2$  monolayers.

209

210 To provide further insights into the magnetic properties of  $\text{Co}(\text{Cp})_2$  monolayers, the spatial distribution  
211 of spin charges was calculated, as shown in Fig. 3g and 3h. These results show that the spin charge  
212 density is mainly contributed by the confined  $\text{Co}(\text{Cp})_2$  molecule, consistent with the VSM measurement  
213 results. The calculated magnetocrystalline anisotropy energy (MAE) for  $\text{Co}(\text{Cp})_2/\text{SnS}_2$ , shown in Table  
214 S2, indicates that the EMA is parallel to the  $\text{SnS}_2$  substrate, along the axial direction of the  $\text{Co}(\text{Cp})_2$   
215 molecule, and the MAE ranges from 89 to 336  $\mu\text{eV}$ , which stabilizes the long-range magnetic order. To  
216 explore the interactions between in-plane molecules, we calculated the orbital-resolved density of states  
217 of Cp rings, as shown in Fig. S12. The density of states of the Cp ring is mostly contributed by the in-  
218 plane  $p_x$  and  $p_y$  orbitals, and the spin direction can be inferred to be along the in-plane direction. In  
219 addition, the two Cp rings share similar energy levels, reflecting a strong coupling between Cp rings, as  
220 shown in Fig. S13. Long-range ferromagnetic ordering in molecular monolayers has been observed in  
221 TCNQ arrangements<sup>27</sup>. Thus, a strong intermolecular ferromagnetic interaction between neighbouring  
222  $\text{Co}(\text{Cp})_2$  molecules, which is responsible for the long-range magnetic order, can be reasonably concluded.

223

224 **Conclusion**

225 In summary, we have demonstrated a new class of ferromagnetic configurations in confined Co(Cp)<sub>2</sub>  
226 monolayers that is different from crystalline ferromagnetic systems. Owing to an intermolecular vibronic  
227 superexchange interaction, spin order can be realized in Co(Cp)<sub>2</sub> monolayers, which is the first room-  
228 temperature ferromagnetic molecular monolayer. The vdW interlayer space not only stabilizes molecular  
229 monolayers via a confinement effect but also provides interface stress and charge transfer to promote the  
230 superexchange interaction between molecules. Our discovery of room-temperature molecular  
231 ferromagnetism provides a new way to search for novel ferromagnetic materials and explore exotic  
232 properties via intermolecular vibronic superexchange interactions.

233

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247

248 **Contributions**

249 C.W. conceived the idea, experimentally realized the study, cowrote the paper, supervised the entire project and is  
250 responsible for the infrastructure and project direction. Y.G. experimentally realized the study, cowrote the paper and  
251 supervised the entire project. Y.L., H.L. and B.Y. contributed equally to this work; they experimentally realized the  
252 study, analysed the data and cowrote the paper. These works were assisted by J.P., Y.S., Y.L.Y.Q., and J.Z. HAADF-  
253 STEM data collection was performed by Y.L. and X.L. Theoretical calculations were carried out by H.L. and X.W.  
254 Y.X. supervised the whole experimental procedure and cowrote the paper. All authors discussed the results and  
255 commented on and revised the manuscript.

256

257 **Competing interests**

258 The authors declare no competing financial interests.

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