High Optical Kerr Nonlinearity of PdSe₂ Dichalcogenide 2D Films for Nonlinear Photonic Chips

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Abstract

As a novel layered noble metal dichalcogenide material, palladium diselenide (PdSe_2) has attracted wide interest due to its excellent optical and electronic properties. In this work, a strong third-order nonlinear optical response of 2D PdSe_2 films is reported. We conduct both open-aperture (OA) and closed-aperture (CA) Z-scan measurements with a femtosecond pulsed laser at 800 nm to investigate the nonlinear absorption and nonlinear refraction, respectively. In the OA experiment, we observe optical limiting behaviour originating from large two photo absorption (TPA) in the PdSe_2 film of $\beta = 3.26 \times 10^{-8} \text{ m/W}$. In the CA experiment, we measure a peak-valley response corresponding to a large and negative Kerr nonlinearity of $n_2 = -1.33 \times 10^{-15} \text{ m}^2/\text{W}$ – two orders of magnitude larger than bulk silicon. In addition, the variation of n_2 as a function of laser intensity is also characterized, with n_2 decreasing in magnitude when increasing incident laser intensity, becoming saturated at $n_2 = -9.96 \times 10^{-16} \text{ m}^2/\text{W}$ at high intensities. Our results show that the extraordinary third-order nonlinear optical properties of PdSe_2 have strong potential for high-performance nonlinear photonic devices.

1. Introduction

Two-dimensional (2D) layered materials such as graphene, [1–3] graphene oxide (GO), [4–9] transition metal dichalcogenides (TMDCs), [10–12] and black phosphorus (BP) [13, 14] have attracted a great deal of interest, enabling diverse nonlinear photonic devices with vastly superior performance compared to bulk materials. Amongst them, TMDCs (MX_2 , M = transition metal and X = chalcogen), with bandgaps in the near infrared to the visible region, have opened up promising new avenues for photonic and optoelectronic devices. [2, 15, 16] For instance, a few mono-layers of MoS_2 and WS_2 have been used as broadband, fast-recovery saturable absorbers for mode locking in pulsed fiber lasers. [2, 15] Nonlinear optical modulators and polarization dependent all-optical switching devices have been realized based on ReSe_2 [16] and SnSe . [17] As a new 2D noble metal dichalcogenide in the TMDC family, PdSe_2 has recently attracted significant interest. [18–21] Similar to the puckered structure of BP, it has a puckered pentagonal atomic structure – with one Pd atom bonding to four Se atoms and two adjacent Se atoms covalently bonding with each other. Due to this low-symmetry structure, PdSe_2 possesses unique in-plane anisotropic optical and electronic properties, [18, 19] featuring an in-plane noncentrosymmetric structure, in contrast to its cousin PtSe_2 . Further, PdSe_2 has a layer-dependent bandgap, varying from 0 eV (bulk) to 1.3 eV (monolayer) - a property well suited for photonic and optoelectronic applications – in particular, for wavelength tuneable devices. Moreover, different to BP which degrades rapidly under ambient conditions, PdSe_2 is highly air-stable, indicating its robustness and potential for practical applications. The high carrier mobility and anisotropic Raman spectroscopy of 2D PdSe_2 layers have been investigated [18, 20] as well as highly-sensitive photodetectors from the visible to mid-infrared wavelengths. [22, 23] Recently, the optical nonlinear absorption of PdSe_2 nanosheets has also been reported in the context of mode-locked laser applications. [24, 25] To date, however, its optical Kerr nonlinearity has not been investigated.

Here, we characterize the third-order nonlinear optical properties of PdSe₂ multilayer films via Z-scan technique with femtosecond optical pulses at 800 nm. Both OA and CA measurements are performed to investigate the nonlinear absorption and nonlinear refraction of PdSe₂. Experimental results show that PdSe₂ films exhibit a large and negative (self-defocusing) Kerr nonlinearity (n_2) of $\sim -1.33 \times 10^{-15} \text{ m}^2/\text{W}$, two orders of magnitude larger than bulk silicon. In the OA measurement, we observe a large nonlinear absorption β of $\sim 3.26 \times 10^{-8} \text{ m/W}$, which originates from TPA in the PdSe₂ films. In addition, we investigate the intensity dependence of the nonlinear response of PdSe₂, finding that the absolute magnitude of the Kerr nonlinearity n_2 initially decreases slightly with incident laser intensity, becoming saturated at higher intensities. These results verify the large third-order nonlinear optical response of PdSe₂ as well as its strong potential for high-performance nonlinear photonic devices.

2. Material Preparation And Characterization

The atomic structure of PdSe₂ crystals is shown in Fig. 1(a). PdSe₂ exhibits a unique puckered pentagonal structure, different to other TMDCs like MoS₂ and WS₂. The Se-Pd-Se layers stack with weak van der Waals interactions to form a layered structure. [18, 19] In each monolayer, the pentagonal rings are formed with one Pd atom bonding to four Se atoms and two adjacent Se atoms covalently bonding with each other, which is similar to the puckered structure of BP, and yields both anisotropic and non-centrosymmetric properties of PdSe₂. More importantly, unlike the rapid degradation of BP under ambient conditions, PdSe₂ has significantly better air-stability. [22, 23] Together, these properties of PdSe₂ make it promising for high performance photonic and optoelectronic applications.

Here, we investigate large-area multilayer PdSe₂ films deposited on transparent sapphire substrates. The PdSe₂ films were synthesized via Chemical vapor deposition (CVD). [26] The films were polycrystalline, as is typical for CVD synthesized films, with crystal sizes varying from 10's of nanometres up to 100 nm. Because of the polycrystalline nature of the films, the inversion symmetry breaking properties (i.e., non-centrosymmetric) of the single PdSe₂ crystals could not be observed on optical wavelength scales in the macroscopic PdSe₂ continuous films studied in this work. Figure 1(b) shows the photography of the prepared PdSe₂ film. The morphology image and height profile of the prepared PdSe₂ films were characterized by atomic force microscopy (AFM), as illustrated in Fig. 1(c). The film thickness was measured to be $\sim 8 \text{ nm}$, which corresponds to ~ 20 layers of PdSe₂. [19, 26]

Raman spectrum of the prepared PdSe₂ film excited with a laser at 514 nm is shown in Fig. 2(a). Three representative phonon modes can be observed, including the A_g^1 ($\sim 145.5 \text{ cm}^{-1}$) and B_{1g}^2 ($\sim 222.5 \text{ cm}^{-1}$) vibrational modes that correspond to the movement of Se atoms and the A_g^3 ($\sim 258.8 \text{ cm}^{-1}$) mode that relates to the relative movements between Pd and Se atoms. [20, 26] To further characterize the film quality, X-ray photoelectron spectroscopy (XPS) was employed to measure the binding energy of PdSe₂. Figure 2(b) shows the XPS results of Pd 3d and Se 3d core levels for the PdSe₂. The peaks of the fit at $\sim 342.2 \text{ eV}$ and $\sim 336.9 \text{ eV}$ are attributed to the Pd 3d_{3/2} and Pd 3d_{5/2}, respectively, whereas the peaks at $\sim$

55.7 eV and ~ 54.9 eV correspond to Se $3d_{3/2}$ and $3d_{5/2}$, respectively. [20, 26] To characterize the linear absorption and optical bandgap, the optical absorption spectrum (from 400 nm to 2500 nm) of the PdSe₂ film was measured with ultraviolet-visible (UV-vis) spectrometry, as shown in Fig. 2(c). The inset of Fig. 2(c) shows the Tauc plot extracted from the linear absorption spectrum, where the optical bandgap of the PdSe₂ film is estimated to be ~ 0.7 eV. We also characterize the in-plane (TE-polarized) refractive index (n) and extinction coefficient (k) of the PdSe₂ film via spectral ellipsometry, as depicted in Fig. 2(d). The refractive index first increases dramatically with wavelength to reach a peak at ~ 700 nm and then decreases more gradually at longer wavelengths. The extinction coefficient exhibits a significant decrease from 600 nm to 1200 nm, and then the rate of decrease slows down at longer wavelengths. This shows an agreement with the trend of the UV-vis absorption spectrum in Fig. 2(c).

3. Z-scan Measurements

To investigate the third-order nonlinear optical properties of PdSe₂, we characterized the nonlinear absorption and refraction of the prepared PdSe₂ films via the Z-scan technique, [27-29] where a femtosecond pulsed laser with a centre wavelength at ~ 800 nm and pulse duration of ~ 140 fs was used to excite the samples. A half-wave plate combined with a linear polarizer was employed to control the power of the incident light. A beam expansion system consisting of a 25-mm concave lens and 150-mm convex lens was used to expand the light beam, which was then focused by an objective lens (10 $\times$, 0.25 NA) to achieve a low beam waist with a focal spot size of ~ 1.6 μm . The prepared samples were oriented perpendicular to the beam axis and translated along the Z-axis with a linear motorized stage. A high-definition charge-coupled-device (CCD) imaging system was used to align the light beam to the target sample. Two photodetectors (PDs) were employed to detect the transmitted light power for the signal and reference arms.

Figure 3(a) shows the OA Z-scan results for the PdSe₂ film at three representative intensities. A typical optical limiting behaviour was observed in the OA curves, with the transmission decreasing as the PdSe₂ sample was moved through the focal point. We measured the response of pure sapphire substrate and did not observe any significant nonlinear absorption, indicating that the observed optical limiting response was induced by the PdSe₂ film. We also note that the transmittance dip of the OA curve decreased when the incident laser intensity was increased. In principle, the optical limiting behaviour can be induced by several mechanisms such as nonlinear light scattering (NLS), reverse saturable absorption (RSA), two-photon absorption (TPA) and multi-photon absorption (MPA). [30,31] However, apart from the basic condition that the total energy of the photons involved in each process (eg., two photons, for TPA, one photon for SA etc.) must be larger than the bandgap, there is no a-priori reason for any particular process to dominate. For thin PdSe₂ film in our case, though, we first exclude the NLS effect since it usually dominates for dispersion or solution samples with laser-induced microbubbles. [30,31] According to the UV-vis spectra, the bandgap of the few-layer PdSe₂ film is estimated to be 0.7 eV, which is lower than a single photon energy of the incident laser at 800 nm. Therefore, all the above processes can occur. While SA at low laser intensities and RSA at high laser intensities might be expected for the Z-scan

measurements, we did not observe this. This could possibly be because the single photon transition is inefficient under 800-nm femtosecond laser excitation due to the indirect band structure of the few-layer 8-nm-thick PdSe₂ films, or possibly parallel band absorption effects. [31] Considering this, RSA is unlikely to dominate the nonlinear absorption in PdSe₂ films due to its one photon process. Given the high peak power of the incident femtosecond pulses, TPA is likely to account for the optical limiting behaviour observed in our Z-scan measurements.

To extract the TPA coefficient β of PdSe₂, we fit the measured OA results with the well-established theory. [27,28] The TPA coefficient β for the PdSe₂ film is shown in Figure 3(b) at different laser intensities. A large $\beta = 3.26 \times 10^{-8}$ m/W is observed, which is comparable to the reported values of graphene, and higher than that of WS₂, highlighting the strong optical limiting effect in PdSe₂ film. In addition, the TPA coefficient β is relatively constant with incident laser intensity, reflecting the fact that we are working in an intensity regime where the material properties of the PdSe₂ films are not changing much. The slight fluctuation in β with laser intensity may arise from light scattering in the PdSe₂ film surface.

To further investigate the nonlinear absorption of the PdSe₂ film, we measured the minimum transmittance with the sample at the focal point of the Z scan setup, for different incident laser intensities. Figure 3(c) shows the transmittance of PdSe₂ at the focal point as a function of laser intensity, where the transmittance fluctuates around a relatively constant value at low intensities and then decreases significantly as the laser intensity increased. The experimental data fits the theory well, [31] verifying our assumption of TPA being the dominant process for nonlinear absorption in the PdSe₂ film. The order of the observed nonlinear absorption can also be confirmed by examining the relation between $\ln(1 - T_{OA})$ versus $\ln(I_0)$: [32]

$$\ln(1 - T_{OA}) = k \ln(I_0) + C, \quad (1)$$

where k is the slope showing the order of the nonlinear absorption and C is a constant. For pure TPA, the slope is equal to 1. [32] We obtain a slope of 1.18 (Figure 3(d)), suggesting the observed nonlinear absorption is mainly attributed to TPA in the PdSe₂ film.

We also performed CA Z-scan measurements to investigate the Kerr nonlinearity (n_2) of the PdSe₂ films. The values of n_2 for the PdSe₂ film at different laser intensities were extracted by fitting the measured CA results. Figure 4(a) shows a representative CA result for PdSe₂ at a laser intensity of 17.15 GW/cm². The transmittance of the sample exhibited a transition from peak to valley when the sample passed through the focal plane. Such a peak-valley CA behaviour corresponds to a negative Kerr coefficient n_2 and indicates an optical self-defocusing effect in the PdSe₂ film. The noise in the CA data is mainly induced by the light scattering in the PdSe₂ film surface. By improving the film uniformity, such noise can be further reduced. As discussed above, TPA results in the transfer of electrons from valence band to

conduction band, increasing the free carrier density in the film. Therefore, the observed negative Kerr nonlinearity potentially originates from the TPA-induced free carrier nonlinear refraction and interband blocking. [33,34]

Figure 4(b) shows the measured Kerr coefficient n_2 of PdSe₂ versus laser intensity, showing a large n_2 of $-1.33 \times 10^{-15} \text{ m}^2/\text{W}$. Table 1 compares the β and n_2 of PdSe₂ with other 2D layered materials. As can be seen, the value of n_2 for PdSe₂ is lower than those of graphene and GO, but still more than two orders of magnitude higher than bulk silicon. [35, 36] Such a high n_2 suggests that PdSe₂ is an extremely promising material for self-defocusing based nonlinear photonic applications. For example, a negative Kerr nonlinearity can be used to self-compress ultrashort pulses in the presence of positive group-velocity dispersion. [37] Another application of a negative Kerr nonlinearity is mode locking of lasers using the Kerr mode-locking technique [35, 38] as well as the possibility of achieving net parametric modulational instability gain under normal dispersion conditions. [35, 39]

In addition, as shown in Figure 4(b), the absolute value of n_2 initially decreases with laser intensity and then saturates at higher intensities. In theory, the optical nonlinear refraction originates mainly from the free-carrier and bound-electron nonlinearities. [33, 40-44] We assume that the two mechanisms co-exist in the PdSe₂ film. It has been shown that, near the half-bandgap the two-photon resonance typically yields a positive n_2 . However, at higher photon energies, the bound-electron contribution to the n_2 nonlinearity becomes negative, while the free-carrier nonlinearity is usually also negative. [33,44] We therefore infer that either, or both, processes contribute to the nonlinearity since we observed a negative Kerr nonlinearity for the PdSe₂ film. This is further complicated by the fact that PdSe₂ is an indirect bandgap material. The Kerr nonlinearity is dominated by direct transitions at all energies, whereas the nonlinear absorption is dominated by indirect transitions in energy regions where the direct transitions are not allowed (eg., below half of the direct bandgap for TPA). [45]

The refractive index change in the PdSe₂ film can be expressed by $\Delta n = n_2^* I_0 + \sigma_r N$, where n_2^* is the nonlinear refraction related to bonding electrons, σ_r is the free carrier refractive coefficient and N is the charge carrier density. [33] Therefore, the effective $n_2 = \Delta n / I_0 = n_2^* + \sigma_r N / I_0$, is an intensity dependent parameter, which can explain the n_2 variation as a function of laser intensity observed in our measurements.

Table 1. Comparison of β and n_2 for various 2D layered materials

Material	Laser parameter	Thickness	β (m/W)	n_2 (m ² /W)	n_2 ($\times n_2$ of Si ¹)	Ref.
Graphene	1150 nm, 100 fs	5-7 layers	3.8×10^{-8}	-5.5×10^{-14}	-1.22×10^4	[42]
GO	800 nm, 100 fs	$\sim 2 \mu\text{m}$	4×10^{-7}	1.25×10^{-13}	2.75×10^4	[29]
MoS ₂	1064 nm, 25 ps	$\sim 25 \mu\text{m}$	$(-3.8 \pm 0.59) \times 10^{-11}$	$(1.88 \pm 0.48) \times 10^{-16}$	41.32	[11]
WS ₂	1040 nm, 340 fs	$\sim 57.9 \text{ nm}$	$(1.81 \pm 0.08) \times 10^{-8}$	$(-3.36 \pm 0.27) \times 10^{-16}$	-73.85	[12]
BP	1030 nm, 140 fs	$\sim 15 \text{ nm}$	5.845×10^{-6}	-1.635×10^{-12}	-3.59×10^5	[14]
PtSe ₂	800 nm, 150 fs	$\sim 4.6 \text{ nm}$	-8.80×10^{-8}	—	—	[43]
PtSe ₂	1030 nm, 340 fs	17 layers	—	$(-3.76 \pm 0.46) \times 10^{-15}$	-8.26×10^2	[32]
PdSe ₂	800 nm, 140 fs	$\sim 8 \text{ nm}$	$(3.26 \pm 0.19) \times 10^{-8}$	$(-1.33 \pm 0.23) \times 10^{-15}$	-2.92×10^2	This work

¹ n_2 for silicon = $4.55 \times 10^{-18} \text{ m}^2/\text{W}$ (ref. [36])

4. Conclusion

In summary, we report a large third-order nonlinear optical response of PdSe₂ films measured with the Z-scan technique. Experimental results show that PdSe₂ has a strong TPA response with a large β of $\sim 3.26 \times 10^{-8} \text{ m/W}$. The Kerr nonlinearity (n_2) of PdSe₂ is also investigated. We find that n_2 is negative, and with an absolute magnitude that is more than two orders of magnitude larger than bulk silicon. Furthermore, we characterize the variation in n_2 of PdSe₂ with laser intensity, finding that n_2 initially increases (decreasing in absolute magnitude) with incident laser intensity and then saturates at higher intensities. Our results verify PdSe₂ as a promising 2D material with prominent nonlinear optical properties similar to graphene oxide [46–49], and potentially even for mid IR applications on platforms such as Si-Ge [50–57].

Declarations

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Figures

Figure 1

(a) Crystal structure of PdSe₂. (b) Photograph of prepared multilayer PdSe₂ film. The unit for the numbers on the ruler is centimeter. (c) AFM height profile of the multilayer PdSe₂ film.

Figure 2

Characterization of the prepared PdSe₂ film. (a) Raman spectrum excited via a 514-nm laser. (b) X-ray photoelectron spectroscopy (XPS) spectra. (c) UV-vis absorption spectrum. Inset shows the extracted Tauc plot. (d) Measured refractive index (n) and extinction coefficient (k).

Figure 3

(a) OA Z-scan results of PdSe₂ film at different intensities. (b) TPA coefficient β of PdSe₂ film versus laser intensity. (c) Normalized transmittance of PdSe₂ film at the focal point as a function of laser

intensity. (d) The plot of $\ln(1-\text{TOA})$ versus $\ln(I_0)$ to determine the order of nonlinearity. The measured and fit results are shown by scatter data points and solid lines, respectively.

Figure 4

(a) Representative CA Z-scan result of PdSe2 film at intensity of 17.15 GW/cm². (b) Kerr coefficient n_2 of PdSe2 film versus laser intensity.