

# Synthesis and Investigation of Structural, Microstructural and Optical Properties of Iron-doped SnO<sub>2</sub> Nanoparticles : Application in Photocatalysis

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## Research Article

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# Synthesis and investigation of structural, microstructural and optical properties of iron-doped SnO<sub>2</sub> nanoparticles : Application in photocatalysis

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## Abstract

Iron-doped tin dioxide nanoparticles Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> (0% ≤ x ≤ 10% ), were prepared by coprecipitation method. X-ray diffraction (XRD) patterns indicate that Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> nanoparticles (NPs) crystallize in the tetragonal rutile-like structure. Scanning Electron Microscope (SEM) observations did not show more modification of the SnO<sub>2</sub> morphology with Fe addition. All the results are consistent with the fact that Fe is strongly soluble in SnO<sub>2</sub> host. The optical band gap energy decreased from 3.65 to 3.30 eV with increasing the iron doping concentration in the solution. Finally, the photocatalytic efficiency of Fe<sub>0.05</sub>Sn<sub>0.95</sub>O<sub>2</sub> nanoparticles (NPs) was examined for the degradation of MB in aqueous solution under UV irradiation. It was found that smaller bandgap of Fe doped SnO<sub>2</sub> photocatalyst resulted in a prominent increase in photocatalytic activity of nanoparticles against Methylene Blue (MB) under UV irradiation.

**Key words:** Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> nanoparticles, structural, microstructural and optical properties, photocatalysis.

## 1 Introduction

The development of nanoscale particles is a promising research theme, supported by different branches of the industry ranging from biomedical to cosmetic through catalysis. Indeed, such a particle size gives the material unique and new physical and chemical properties, not encountered with larger materials. [1-3]

Metallic oxides and in particular tin oxides are excellent examples for accessing new or improved properties. Tin oxide nanoparticles offer attractive magnetic, electrical and optical properties [4] as well as high chemical reactivity [5-7], giving them a wide range of fields of application such as biomedical, treatment of water, catalysis, gas sensors, lithium-ion accumulators [8, 9].

Tin oxide is a transparent conductive oxide of type n with a energy band gap of 3.6 eV, The structure of tin dioxide called cassiterite crystallizes in the tetragonal rutile type system represented with the parameters:  $a = b = 4.737 \text{ \AA}$  and  $c = 3.186 \text{ \AA}$  [10]. The elementary cell contains six atoms, four oxygen atoms and two tin atoms. Its space group is  $P4 / mnm$  [11]. Each tin ion  $\text{Sn}^{4+}$  is at the center of an octahedron formed by six oxygen ions  $\text{O}^{2-}$ , as long as each oxygen ion  $\text{O}^{2-}$  is at the center of an isosceles triangle formed by three ions of tin  $\text{Sn}^{4+}$ . The cation  $\text{Sn}^{4+}$  and the anion  $\text{O}^{2-}$  have the value 0.071 and 0.14 nm [12]. The oxygen vacancies formed by the transfer of an oxygen atom from a site to a normal site in the gaseous state, make it possible to obtain a semiconductor of this type. Indeed, the oxygen gap thus created has two electrons, it is then said to be neutral which can be yielded under the effect of temperature. There is then simple or double ionization thereof. The released electrons can attach to  $\text{Sn}^{4+}$  tin atoms. They then become  $\text{Sn}^{2+}$  and behave like electron donors [13,14]. It is arguably the most widely used compared to other oxides in the area of air pollution monitoring and the detection of toxic gases. [15] In fact, it has electrical properties linked to the remarkable surface adsorption. Tin oxide is also known for its catalytic properties, in particular it facilitates the decomposition of many hydrocarbons.

The doping of  $\text{SnO}_2$  by suitable elements such as Mn, Fe, Co, Al, Mg ... etc, can improve its optical, electrical and magnetic characteristics and accelerate the race for its practical applications[16].

The addition of transition metals in the photocatalyst influences the trapping of electrons, slows down the speed of recombination of electron / hole pairs, increases the efficiency of charge transfer. It also influences the optical gap of the photocatalyser by creating the energy levels inside the forbidden band which helps us to extend the spectrum of adsorption towards visible light and to take advantage of solar radiation.

Recently Fe-doped  $\text{SnO}_2$  has attracted attention as a candidate for diluted magnetic semiconductors (DMS) [17,18]. It has the advantage of a lower surface potential barrier than

metals and a high catalytic activity, which can be very vital in the field of electron emission [19].

Many methods of synthesis of SnO<sub>2</sub>nanopowders such as hydrothermal [20], chemical vapor deposition (CVD) [21], sol-gel [22], chemical coprecipitation [23] and spray pyrolysis have been developed [24].

In this paper, Fe doped SnO<sub>2</sub> nanoparticles are synthesized by coprecipitation method for photocatalytic application in the treatment of water purification. The objective that we set ourselves in this work is to study the influence of doping on the structural, microstructural, and morphological properties of the of synthesized nanostructures and their photo-catalytic performance was investigated with the degradation of Methylene Blue (MB) dye

## **2 Experimental Procedure**

### **2.1 Synthesis of SnO<sub>2</sub> and Fe doped SnO<sub>2</sub> nanoparticles**

The iron-doped tin oxide nanoparticles, Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> (x= 0.00, 0.02, 0.05and 0.10), were prepared by chemical co-precipitation method[25]. For the synthesis of Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> nanoparticles, tin chloride pentahydrate (SnCl<sub>4</sub>.5H<sub>2</sub>O), ferric chloride dihydrate (Fe Cl<sub>2</sub>.2H<sub>2</sub>O) and sodium hydroxide (NaOH) were used as precursors without further purification. All the chemicals used in the present preparation are highly pure (more than 99%) and AR grade purchased from Merck.

Initially, 0.1M SnCl<sub>4</sub>.5H<sub>2</sub>O was dissolved in 100 ml double distilled water and maintained under agitation for 10 min. 0.3M NaOH solution was prepared separately by dissolving appropriate amount of NaOH in 50 ml double distilled water. The prepared NaOH solution was added drop wise into the initial aqueous solution under constant stirring up to 2hrs at room temperature to reach the pH value of 12. The formed gelatinous white precipitates were filtered and then washed seven times using double distilled water and ethanol to eliminate the impurities and chlorine ions from precipitates. Dry for overnight at room temperature and dried in oven at 100 °C for six hours. The dried white precipitates then annealed at 600 °C for two hours followed by grinding to get fine particles. The same procedure is repeated for the remaining samples synthesized with nominal compositions of Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> (x= 0.00, 0.02, 0.05and 0.10). These were used for various characterization techniques [26-28]

## 2.2 Characterization Techniques

The crystal structure of Fe-doped SnO<sub>2</sub> nanoparticles was determined by X-ray diffractometer (MiniFlex600) with Cu-K $\alpha$  radiation, 0.15418 nm). The surface morphology of the prepared nanoparticles was studied using a scanning electron microscopy (TESCAN\_VEGA3) equipped with energy dispersive X-ray spectroscopy (EDS) for chemical analysis. The infrared absorption (FTIR) spectra were recorded to examine the structure by using Thermo-Nicolet equipment in the 4000–400 cm<sup>-1</sup> region. The absorbance property was studied using Parkin Elmer UV–VIS–NIR Lambda 19 spectrophotometer in the 190–1800 nm spectral range.

## 2.3 Photocatalytic experiments

The photocatalytic activity of Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> [x= 0.00,0.02,0.05 and 0.10] nanocomposites was evaluated by measuring the decolourization of MB under visible light irradiation at ambient temperature using a 100 W incandescent lamp as a light source. The solutions were pretreated by magnetic stirring in the dark for 30 min to achieve adsorption–desorption equilibrium of MB on the catalyst surface. Then, the solutions were irradiated each 30 min, the concentration of MB was determined by recording the variations of the absorption band maximum (664nm) of MB using Parkin Elmer UV–VIS–NIR Lambda 19 spectrophotometer

## 3 Results and discussion

### 3.1 X-ray diffraction – structural analysis

The XRD pattern of Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> [ x = 0, 2, 5 and 10 atm.%] nanoparticles are shown in Fig. 1. In the scan range from 10° to 90°, both the samples show the four major peaks (100), (101), (200) and (211), which correspond well with that of cassiterite phase (JCPDS file number 01-071-0652). No secondary peak characteristic of ferric oxide phase appear for all Fe-doped samples was observed within the detection limit of used XRD, indicating that Fe ions successfully occupy lattice sites rather than interstitial sites of SnO<sub>2</sub> which possess a high solubility [29], and the good atomic-level blending of all constituent elements through the co-precipitation method. This important solubility of iron in SnO<sub>2</sub> has also been recorded for many other trivalent ions, such as Ga and In [30,31].

However, the addition of Fe influences the SnO<sub>2</sub>, XRD pattern either in intensity or in broadening of peaks, suggesting that the crystallinity and the particle size are modified by iron doping.

In Sn<sub>0.98</sub>Fe<sub>0.02</sub>O<sub>2</sub> sample, the maximum intensity peak characterizing the structure rutile corresponding to (110), (101), (200) and (211) plane which decreasing with doping level. The reduced diffraction intensity with doping may be due to impurities that oppose the growth of SnO<sub>2</sub>. It is also due to the movement of Sn<sup>4+</sup> ions in the interstitial sites and also to an increase in the amorphous phase and disorder. A decrease in the intensity of the diffracted peaks with the increase of dopant level was also reported in sol-gel synthesized Fe doped SnO<sub>2</sub> [32].

Diffraction analysis is a very important tool for studying the crystal growth of a nanoparticles. It allows us to calculate the size of crystallites which can play an important role in the physical properties (electrical, optical,...) of materials.

The grain sizes were evaluated from the width at mid-height (in 2θ) of the diffraction peak using Scherrer's formula [33]:

$$D = \frac{0.9\lambda}{\cos \theta} \quad (1)$$

Where:  $\lambda = 1.54056 \text{ \AA}$ ,  $\theta$  is the angle of incidence.

From relation (1) the Values a and c are calculated as follows:

$$d_{hkl} = \frac{a}{\sqrt{h^2 + k^2 + \left(l \frac{a}{c}\right)^2}} \quad (2)$$

Table .1 represents the values of the grain size(D), c/a ratio, micro-strain (ε), dislocation (δ), specific surface area (S) and the lattice parameters “a” and “c” of SnO<sub>2</sub>, Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> nanoparticles.

We valued the lattice constant a to lie between 4.7227 and 4.7269 Å, while lattice constant c lies between 3.1878 and 3.1806 Å, for undoped SnO<sub>2</sub> and Sn<sub>0.90</sub>Fe<sub>0.10</sub>O<sub>2</sub> respectively, which due to the substitution of iron in SnO<sub>2</sub>. The c/a fraction decreases with the doping rate which

shows a progression in the basal plane, note that the decrease in unit cell volume with the rate doping confirms the insertion of Fe<sup>3+</sup> in SnO<sub>2</sub> lattice which can be considered as a greater distortion of the cell.

Indeed, the size deduced by Scherrer's relation, It is obvious that calcination (600°C) can strengthen the crystallinity of the samples, but also increase the size of the crystallite grains, for the undoped sample with a value of 47 nm, while for doped samples the crystallite size was 27 nm for  $x = 0.02$ , 18 nm for  $x = 0.05$ , and 16 nm for  $x = 0.10$ , this decrease in crystallite size is clearly observed with the doping rate which confirms the substitution of  $\text{Sn}^{4+}$  by  $\text{Fe}^{3+}$  ions which is explained by their proximity to ion rays (0.72 Å for  $\text{Sn}^{4+}$  [34] and 0.64 Å for  $\text{Fe}^{3+}$  [35]), this substitution of  $\text{Sn}^{4+}$  by  $\text{Fe}^{3+}$  ions is very possible. It is probable that the relative 3+ oxidation state of the iron ions, which is already contained in the precursor used ( $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$ ), is mainly kept in the  $\text{SnO}_2$  matrix [36,37]. It is clear that the addition of Fe blocked the progression of the  $\text{SnO}_2$  crystal grains. Note that the decrease in unit cell volume with the rate doping confirms the insertion of  $\text{Fe}^{3+}$  in  $\text{SnO}_2$  lattice which can be considered as a greater distortion of the cell.

The lattice strain, it usually means the inhomogeneous local strain in the material which gives rise to x-ray peak broadening. It is only one factor causing peak broadening. The other factors are small crystallite size and presence of lattice defects.

The determination of the stresses produced in the nanoparticles due to lattice distortion and impurities was carried out using the Stoke Wilson formula [38]:

$$\varepsilon = \frac{\beta}{4 \tan \theta} \quad (3)$$

Pure  $\text{SnO}_2$  shows a strain of  $0.17 \times 10^{-3}$  while with a doping (Fe) content of 2%, it increases to  $0.3 \times 10^{-3}$ ,  $0.45 \times 10^{-3}$  with 5% doping, and  $0.50 \times 10^{-3}$  with 10% doping. The positive signs of the stress values correspond to tensile stress. These strain variants may occur due to the deformation generated in the samples at the time of thermal expansion in different directions. In addition, the crystallite size is reversed proportionally to the deformation, which implies that a change in the crystallite size will cause a variability of the deformation in the sample.

The dislocation density value ( $\delta$ ) represents the number of dislocation lines per unit volume of the crystal, which is the size of the defects in a crystal. In other words, the dislocation density value will show the degree of crystallinity of the nanoparticle profile.

$$\delta = \frac{1}{D^2} \quad (4)$$

The result of the dislocation density ( $\delta_{np}$ ) of the  $\text{Sn}_{1-x}\text{Fe}_x\text{O}_2$  [ $x=0, 2, 5$  and  $10$  at%] nanoparticles is  $0.444 \times 10^{-3} \text{ nm}^{-2}$ ,  $1.306 \times 10^{-3} \text{ nm}^{-2}$ ,  $2.929 \times 10^{-3} \text{ nm}^{-2}$  and  $3.631 \times 10^{-3} \text{ nm}^{-2}$  respectively. Based on the results of this calculation is known that the dislocation density ( $\delta_{np}$ ) of the nanoparticles obtained in this study is increase with increasing of rate doping wich indicates that  $\text{SnO}_2\text{:Fe}$  nanoparticles have been produced had a low degree of crystallinity because of presence of impurty (Fe). which confirmed the presence of  $\text{Fe}^{3+}$  ions in the  $\text{SnO}_2$  lattice.

### 3.2 Morphological properties: SEM

Scanning Electron Microscope (SEM) observations have also been carried out. on all synthesized nanocatalysts in order to evaluate the evolution of crystallite morphology and size. SEM characterizations were performed at two magnifications for all synthesized pure  $\text{SnO}_2$  and Fe-doped powders. The images in Figure 2, show the presence of nanoparticle agglomerates consisting of crystallites of mainly quasi-spherical shape with a relatively small and homogeneous particle size distribution.

### 3.3 EDS characterization

The EDS spectra of the chemical composition of our Fe-doped  $\text{SnO}_2$  layers are as follows represented in figures 3. The analysis by energy dispersive X-ray spectroscopy (EDS) allowed us to verify the chemical composition of nanoparticles. All spectra show an apparent presence of tin and oxygen. The main objective of using this technique (EDS) is the confirmation of the existence of Fe. All the spectra show the presence of this element, coming from doping and we have noticed an increase in the intensity of the iron peak with the increase in the rate of doping.

The stoichiometric of the  $\text{Sn}_{1-x}\text{Fe}_x\text{O}_2$  nanostructure is measured exclusively from the Fe and Sn signals. According to the following relation (Eq. (5)) [39,40]:

$$x = \frac{r}{(1+r)} \quad (5)$$

Where  $x$  represents the Fe atom content and  $r$  the ratio of the Sn and Fe EDS signals (see Table 2).

Figure 4a, b show the variation of the ratio between the EDS signals of Fe and Sn as well as the calculated value atomic Fe content, both plotted against the "expected" Fe content of the deposit. The incorporation of Fe in the high network is very clear, for all doping concentrations

### 3.4 FTIR Spectroscopy

The study by FTIR spectroscopy performed on the different samples (in the wavelength range from 400 to 4000  $\text{cm}^{-1}$ ) is presented in figure 5, the absorption bands are recorded for wavelengths 3436, 1628, 1254, 1022 and 620  $\text{cm}^{-1}$ . The large vibrations existing at 3436  $\text{cm}^{-1}$  and 1628  $\text{cm}^{-1}$  indicate the presence of hydrogen bonds involved in O-H oscillators. The peaks at 1623  $\text{cm}^{-1}$  correspond to the O-H bending bonds associated with some residual water molecules, while small peaks around 1022  $\text{cm}^{-1}$  and 1254  $\text{cm}^{-1}$ , have also been recorded and can be attributed to hydrogen bonds involved in O-H oscillators. The vibration of the antisymmetric functional groups of the  $\text{SnO}_2$ , O-Sn-O bridge presents about 620  $\text{cm}^{-1}$  [41].

### 3.5 UV-Vis-NIR spectroscopy

The synthesized nanoparticles were analyzed by UV-Vis-NIR spectroscopy in the 300-700nm wavelength region. Fig. 6 shows the optical absorbance spectra of  $\text{SnO}_2$ ,  $\text{Sn}_{1-x}\text{Fe}_x\text{O}_2$  [x=0, 2, 5 and 10 at%] nanoparticles.

The absorption edge position is located between 364 nm and 375nm depending on the Fe content. The ultraviolet cut-off we notice can be due to the photo-excitation of electrons from the valence band to the conduction band, specifies the electronic transition from the valence band  $\text{O}^{2-}$  (2p) to the conduction band  $\text{Sn}^{4+}$  (5p), i.e. the direct recombination of holes and electrons in the valence and conduction bands of O-2p and Sn-5p respectively.

The absorption value decreases with doping and minimizes at Fe = 10%. During Fe substitution, the absorption edge is shifted to the higher wavelength side. The nanoparticle absorption is expected to be related to various factors such as particle size, band gap, defects and impurity centers.

The higher absorption intensity at Fe = 5% is due to the distortion caused by the Fe ions in the  $\text{SnO}_2$  lattice. In addition, the higher absorption in the lower wavelength region (300-350 nm) is accompanied by a higher absorption in the visible region. This higher UV absorption at Fe = 5% is the result of the improved transition of electrons from the valence band to the conduction band and a decrease in the size of the crystallites (size effect) is noted in this case.

The Optical Transmittance Spectra of Fe-doped SnO<sub>2</sub> nanoparticles from 300 nm to 700 nm are shown in Fig. 7. It can be seen that the sample doped with Fe = 5% has a lower transmittance value and the sample doped with Fe = 10% has a higher transmittance value. A higher optical absorption or lower transmittance may be due to the optical transitions from the 3d occupied band to the 4s-4p band of Fe point defects.

On the nanoscale, the band gap increases with the reduction of the size of the nanoparticles due to the quantum confinement effect, which generates discrete energy levels in the valence band and the conduction band. As the energy band gap increases, the restriction of electron movement increases. This shifts the absorption peak to lower wavelengths, i.e. to the blue region.

The direct optical band gap ( $E_g$ ) of Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> nanoparticles can be determined from the absorption coefficient ( $\alpha$ ) and photon energy ( $h\nu$ ) by the following relation [42]:

$$\alpha h\nu = A(h\nu - E_g)^n \quad (6)$$

Where A is a constant,  $E_g$  is optical band gap of the material and the exponent n depends upon the direct/indirect allowed transition. In the present case, n is taken as ½ because of allowed direct transition. The curves have been plotted as  $\alpha h\nu$  Vs  $h\nu$  for all the samples as exposed in Figure 8. The  $E_g$  value for undoped SnO<sub>2</sub> is 3.6 eV, which is in good accord with the previously reported values [43], the incorporation of Fe ions into the host SnO<sub>2</sub> decreased the band gap energy. The energy band gap reduced to 3.30 eV for 2% doping and then increased to 3.35 eV and 3.40 eV with an augmentation of the doping level by 5% and 10% respectively. These alterations in the band gap as dependent on the doping amount can be related to the increase in the defects of oxygen vacancies in the lattice. These defects result in impurity energy states in the band gap where the holes trap free electrons in the valence band and, as a consequence of instability, they will recombine with the holes in the conduction band, which results in a reduction of the band gap. Thus, the change in the energy band gap is a function of oxygen deficiency and lattice distortion generated as a response to the change in doping content in the host material.

The absorption spectra show a continuous red shift with increasing Fe content. The band gap energy decreases and the absorption edge increases with Fe content, is interpreted to

incorporation of Fe<sup>3+</sup> into SnO<sub>2</sub> NPs [44,45]. It was assigned to the charge transfer transitions between the d-electrons of Fe<sup>3+</sup> and the conduction or valence band of SnO<sub>2</sub> [44] and to the the reduction of the size of the nanoparticles

### 3.6 Photocatalytic activity

The evolution of the UV spectra of solutions irradiated from BM ( $8 \times 10^{-3} \text{M}$ ) at 250 nm in the presence of Sn<sub>0.95</sub>Fe<sub>0.05</sub>O<sub>2</sub> photocatalyst as a function of the irradiation time (Figure. 9) shows essentially a decrease of the optical density at the absorption maximum of MB at 664 nm. Images of MB samples irradiated in the presence of the Sn<sub>0.95</sub>Fe<sub>0.05</sub>O<sub>2</sub> photocatalyst after 3 hours clearly show the progressive discoloration of the pollutant compared to the initial non-irradiated sample. The rate in Fig.10 represents the ratio between the amount of reagent processed and the initial amount, using the following equation:

$$D_t(\%) = \frac{A_0 - A_t}{A_0} \times 100 \quad (7)$$

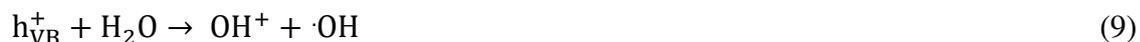
Where  $A_0$  is the initial absorbance of the dye,  $A_t$  the absorbance of the dye in solution at time  $t$

Figure .11 shows that irradiation leads to a progressive discoloration of the MB. A reduction of about 65% is observed after 3hours of irradiation, the kinetics of the degradation reaction of methylene blue in Solution, and he calculated kinetic parameters of MB degradation are shown in Table 3, according to the equation [46].

$$A = X * \exp(-k * t) + E \quad (8)$$

While the unit of (pseudo-) order rate constant  $k$  is the inverse of the unit of time used ( $\text{min}^{-1}$ ),  $X$  is the amplitude of the process,  $E$  is the endpoint, both of them have the same units as the measured quantity  $A$ . The photodegradation was linear ( $R^2 = 0.9513$ ), following an apparent kinetics of order 1 with an apparent speed constant of  $0.0358 \text{ min}^{-1}$ .

The reaction path for the degradation of methylene blue (MB) using Sn<sub>0.95</sub>Fe<sub>0.05</sub>O<sub>2</sub> fine nanopowders under UV irradiation is as follows:



Several factors influence the photocatalytic activity of Sn<sub>0.95</sub>Fe<sub>0.05</sub>O<sub>2</sub> fine nanopowder, namely, the morphology of the nanoparticles, as well as the porosity of the nanoparticles. In addition, it has been reported that increasing the specific surface area by reducing the size of SnO<sub>2</sub> crystallites also increases the efficiency of photocatalysis [47], which could provide more active sites for BM molecules and thus promote the efficiency of electron-hole separation.

The specific surface area of the samples was obtained using equation (14) [48]:

$$S = \frac{6}{\rho D} \quad (14)$$

S: the specific surface area of the nanoarticules.

D: the average size of crystals.

$\rho$  : the density of the material.

The increase in photocatalytic activity of fine nanopowder Sn<sub>0.95</sub>Fe<sub>0.05</sub>O<sub>2</sub> may be due to the increase in the number of crystal defects. In other terms, the network distortion ( $0,45 \times 10^{-3}$ ) and dislocation density ( $2,929 \times 10^{-3} \text{nm}^{-2}$ ) relative to the to undoped samples resulted in a reduction in particle size (18.47nm). This led to an increase in the specific surface area ( $46.74 \text{ m}^2 \cdot \text{g}^{-1}$ , Table. 1), this last parameter contributes very favourably in the degradation photocatalytic of pollutants in general and that of MB in our particular case.

It should also be noted that the shift of the optical gap of Sn<sub>0.90</sub>Fe<sub>0.05</sub>O<sub>2</sub> doped nanoparticles towards longer wavelengths (Red Shift) [49], leads to a decrease of the electron-hole recombination, thus producing a more efficient separation of the photogenerated pairs and consequently prolonging the life time of electrons and holes on the surface of SnO<sub>2</sub> doped nanowires.

#### **4. Conclusion**

In this work, nanoparticles of photosensitive SnO<sub>2</sub>, Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> [x=2, 5 and 10 at%] were synthesized, characterized and then used in photocatalytic applications. The nanomaterials, pure tin dioxide and iron-doped tin dioxide at different concentrations, were synthesized by a simple and classical "co-precipitation" method. The aim is to obtain materials with controlled properties, especially with regard to crystallite size, morphology and especially optical properties which play a very important role in photocatalytic applications.

All the X-ray diffraction spectra of our samples showed a polycrystalline growth and all peaks observed correspond to the rutile tetragonal structure of SnO<sub>2</sub>. The crystallite size of the layers calculated by the Debye-Scherrer method is 47.41 to 16.59 nm, the incorporation of Fe atoms in the SnO<sub>2</sub> matrix in substitution creates pressures in the crystal lattice and leads to a deterioration of the crystal quality of the doped SnO<sub>2</sub> nanoparticles.

The SEM characterization provided information on the morphology of surface of the processed nanoparticles. In the nanoparticles of SnO<sub>2</sub>, SnO<sub>2</sub>: Fe, the grains have quasi-spherical shapes with different sizes and uneven distribution on the surface. The growth mechanism of these crystallites involves the aggregation of atoms with each other to form nucleation centers. UV-vis results indicate that the band gap of SnO<sub>2</sub> NPs decrease with Fe doping.

The presence of dopant in the SnO<sub>2</sub> nanoparticles prevented the growth of these crystallites, so they appear smaller and denser compared to the pure SnO<sub>2</sub> film. The irregularity of the surface of the doped (5%) Fe SnO<sub>2</sub> nanoparticles also provided nucleation sites through pores and cracks.

This result makes the (5%) Fe doped SnO<sub>2</sub> sample potentially more suitable for photocatalytic applications. The nanoparticles of SnO<sub>2</sub> doped (5%) Fe exhibited higher photocatalytic activity with UV irradiation compared to those of pure SnO<sub>2</sub> with respect to degradation of the methylene blue dye.

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#### **Compliance with ethical standards**

**Conflict of interest:** No conflict of interest exists in the submission of this manuscript, and manuscript is approved by all authors for publication. The work described is original research that has not been published previously, and is not under consideration for publication elsewhere, in whole or in part.

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## **Table Captions**

**Table 1** X-ray diffraction data (grain size(D),  $c/a$  ratio, micro-strain ( $\epsilon$ ), dislocation ( $\delta$ ), specific surface area (S) and the lattice parameters “a” and “c”) of different  $\text{Sn}_{1-x}\text{Fe}_x\text{O}_2$  nanoparticles.

**Table 2** Dispersion parameters of the  $\text{Sn}_{1-x}\text{Fe}_x\text{O}_2$  nanoparticles extracted by fitting the experimental data.

**Table 3.** Pseudo-first-order kinetic parameters of MB degradation.

## Figure Captions

**Figure. 1** XRD patterns of SnO<sub>2</sub> and Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> (with x = 2, 5 and 10 at%) nanoparticles

**Figure. 2** SEM images of Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> [x=0, 2, 5 and 10 at%] nanoparticles.

**Figure. 3** EDX elemental composition analysis of a pure SnO<sub>2</sub> nanoparticles and b–d Fe–doped Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> [x= 2,5 and 10] nanoarticles.

**Figure. 4 a** The measured Fe/Sn atomic ratio and **b** the calculated Fe atomic content (from EDS analysis) plotted as function of the expected Fe content/doping

**Figure. 5** FTIR spectra of Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> [x=0, 2, 5 and 10 at%] nanoparticles at room temperature in the wave number from 400 to 4000 cm<sup>-1</sup>

**Fig. 6.** UV–Visible absorption spectra of Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> [x=0, 2, 5 and 10 at%] nanoparticles.

**Fig. 7.** Transmittance spectra of of Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> [x=0, 2, 5 and 10 at%] nanoparticles.

**Fig. 8.** The  $(\alpha h\nu)^2$  versus  $h\nu$  curves of of Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> [x=0, 2, 5 and 10 at%] nanoparticles, for the optical energy gap calculation.

**Figure. 9** The effect of Sn<sub>0.95</sub>Fe<sub>0.05</sub>O<sub>2</sub> nanoparticles on the absorption spectra of MB solution for different reaction time under UV irradiation.

**Figure. 10** Degradation rate of  $\text{Sn}_{0.95}\text{Fe}_{0.05}\text{O}_2$  nanoparticles in MB degradation

**Figure. 11** Decolorization kinetics of MB aqueous solutions by  $\text{Sn}_{0.95}\text{Fe}_{0.05}\text{O}_2$  nanoparticles without UV irradiation. The nanoparticles were prepared by using co-precipitation method. The photocatalytic process at UV-irradiation follows first-order kinetics according to the equation  $A = X \cdot \exp(-k \cdot t) + E$  (circles) Experimental data point and (solid lines) fitted curve (solid lines). Here  $A_0$  is the initial absorbance of the dye solution,  $A(t)$  is the absorbance at time  $t$ , and  $k$  is the rate constant of photocatalysis.

**Table 1**

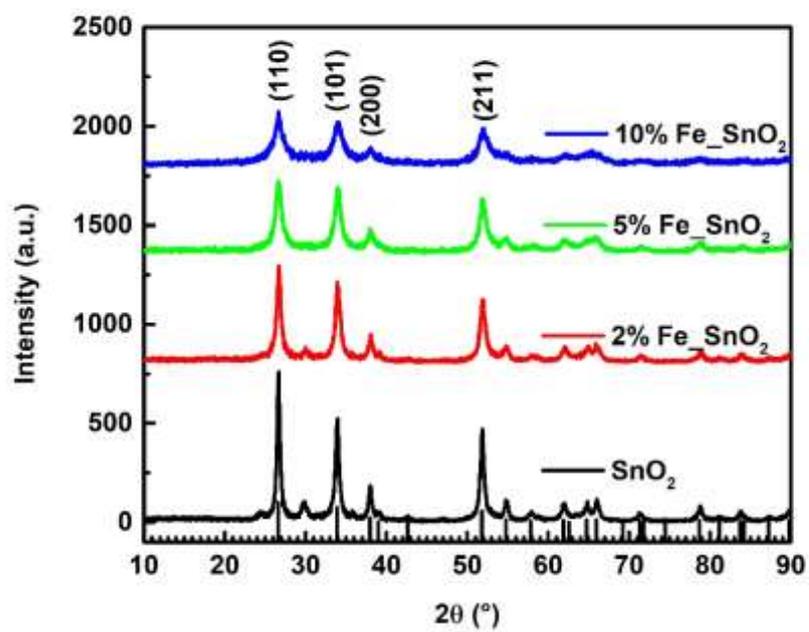
Sample	Crystallite Size (nm)	Lattice Parameters (Å)	c/a ration	Unit cell volume (Å <sup>3</sup> )	$\delta \times 10^{-3}$ (nm <sup>-2</sup> )	$\epsilon \times 10^{-3}$	S (m <sup>2</sup> /g)
SnO <sub>2</sub>	47.41	$a = 4.7227$ $c = 3.1878$	0.6749	71.100	0.444	0.17	18.20
Sn <sub>0.88</sub> Fe <sub>0.02</sub> O <sub>2</sub>	27.66	$a = 4.7233$ $c = 3.1817$	0.6736	70.982	1.306	0.30	31.21
Sn <sub>0.85</sub> Fe <sub>0.05</sub> O <sub>2</sub>	18.47	$a = 4.7026$ $c = 3.1848$	0.6772	70.430	2.929	0.45	46.74
Sn <sub>0.90</sub> Fe <sub>0.10</sub> O <sub>2</sub>	16.59	$a = 4.7269$ $c = 3.1806$	0.6728	71.066	3.631	0.50	52.03

**Table 2**

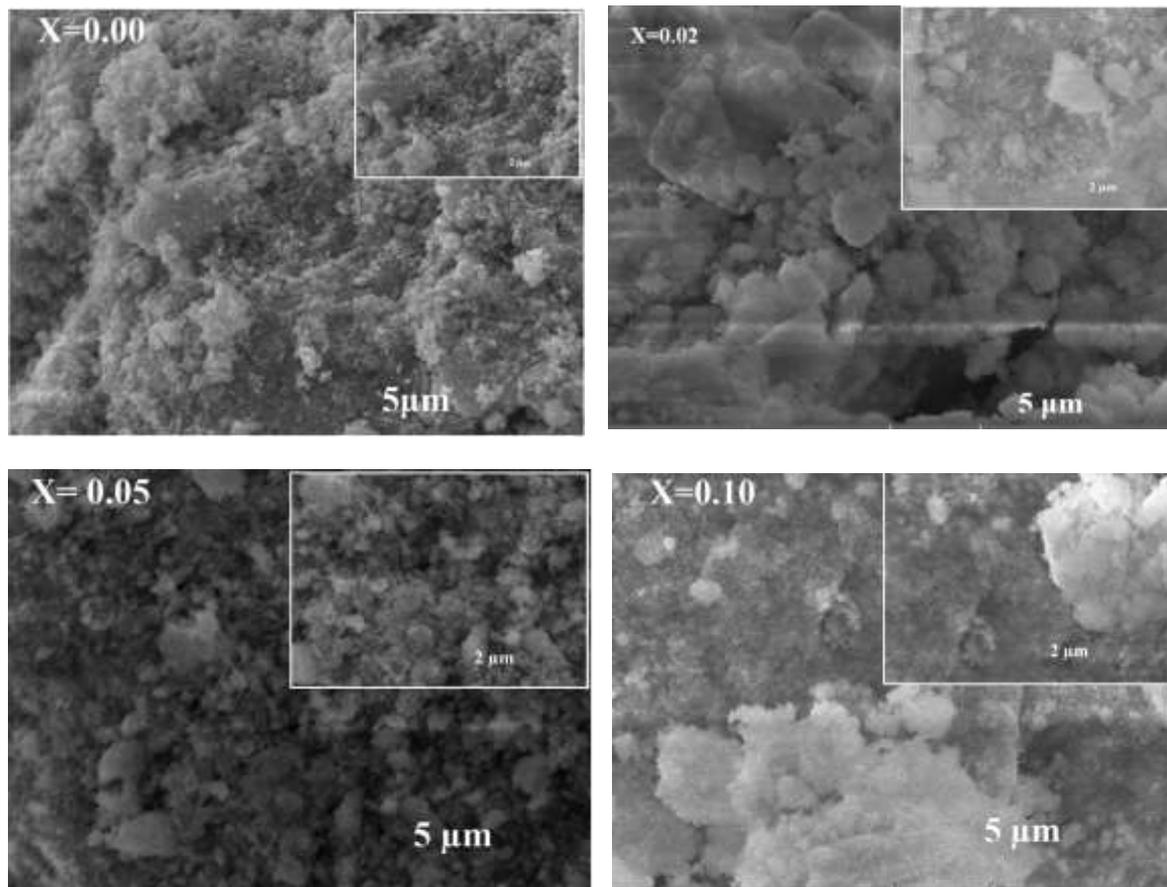
$\text{Zn}_{1-x}\text{Fe}_x\text{O}$ “Nominal” Fe content (at)	Sn (at.%)	Fe (at.%)	O (at.%)	$x$ (Fe) “EDS” Cu content (at)
$\text{SnO}_2$	34.47	/	65.53	0
$\text{Sn}_{0.98}\text{Fe}_{0.02}\text{O}_2$	37.54	1.01	61.46	0.0262
$\text{Sn}_{0.95}\text{Fe}_{0.05}\text{O}_2$	36.83	2.39	60.78	0.0609
$\text{Sn}_{0.90}\text{Fe}_{0.10}\text{O}_2$	37.40	5.35	57.26	0.1251

**Table 3**

Samples	Value			Standard deviation			R <sup>2</sup>
	K(min <sup>-1</sup> )	X	E	K(min <sup>-1</sup> )	X	E	
Sn <sub>0.95</sub> Fe <sub>0.05</sub> O <sub>2</sub>	0.0358	0.6023	0.3876	0.0108	0.0689	0.0404	0.9513



**Fig. 1**



**Fig. 2**

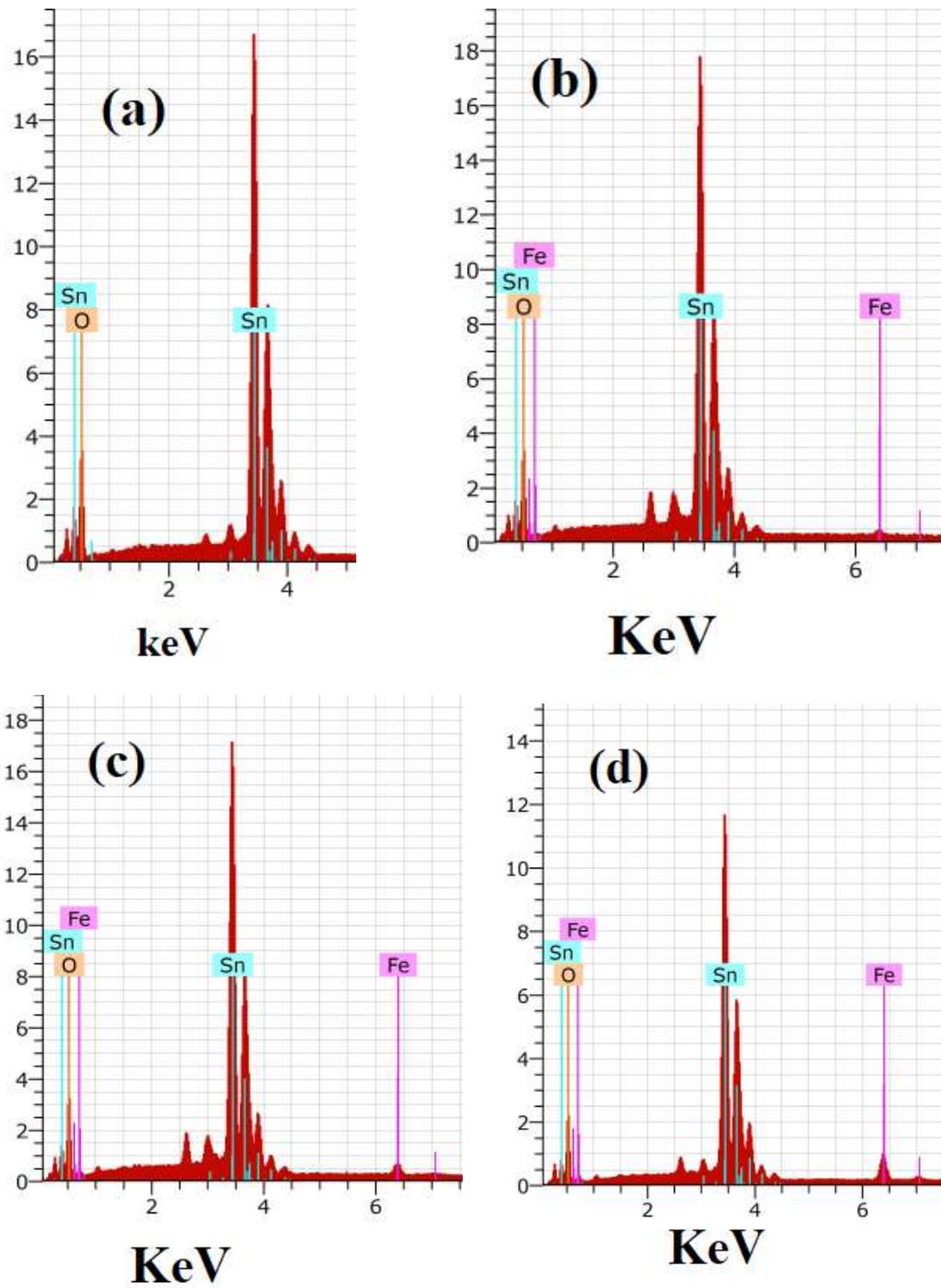
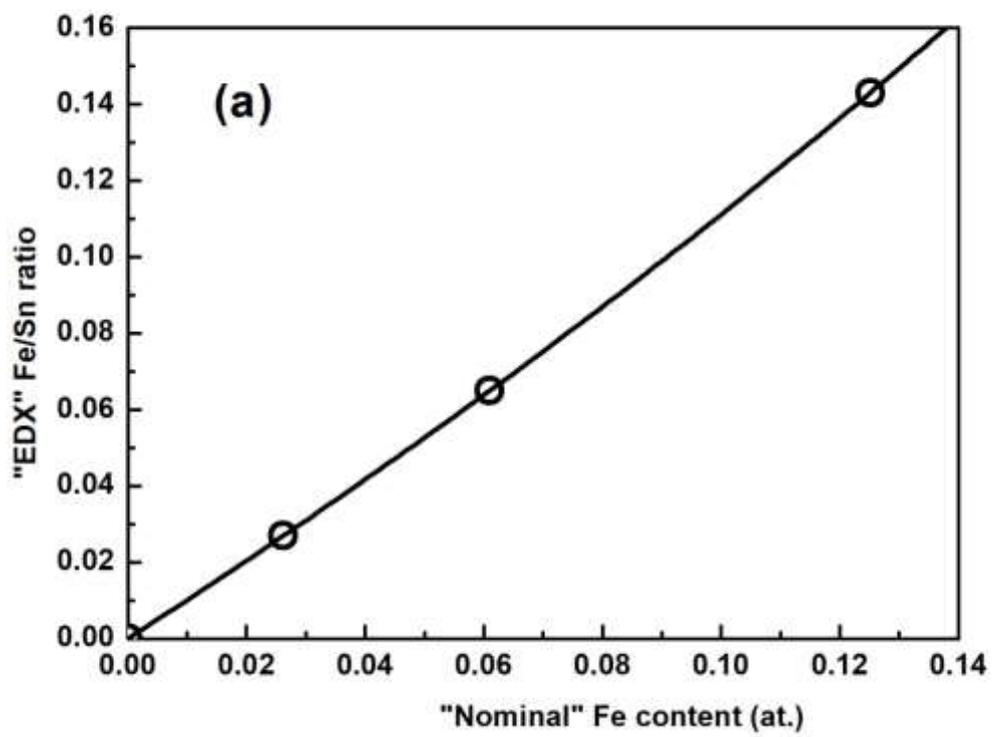


Fig. 3



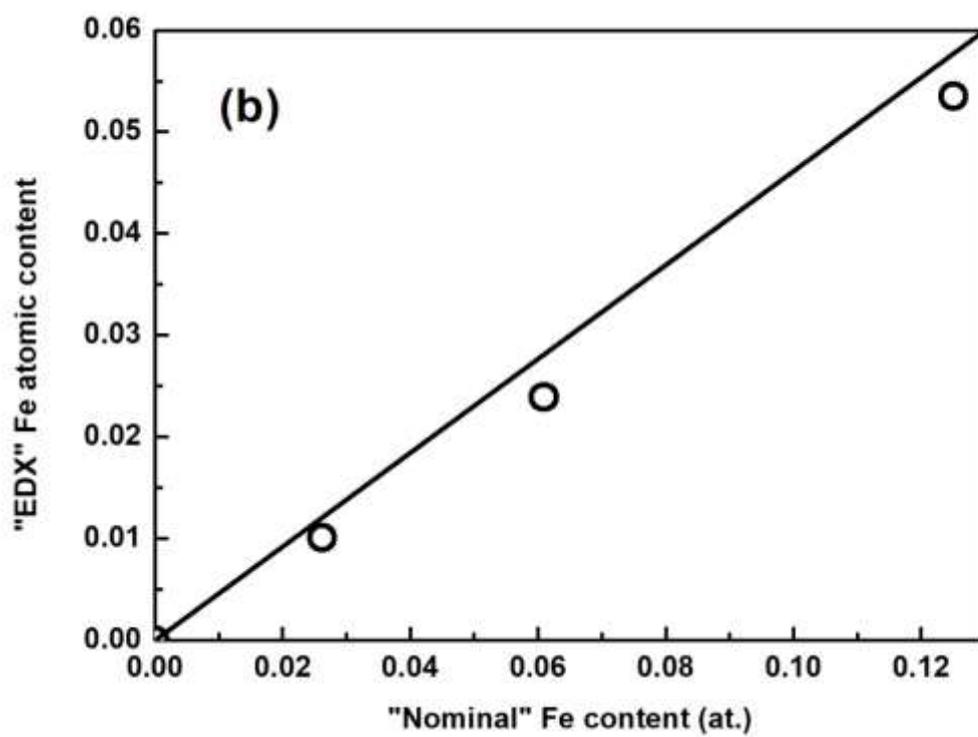


Fig. 4

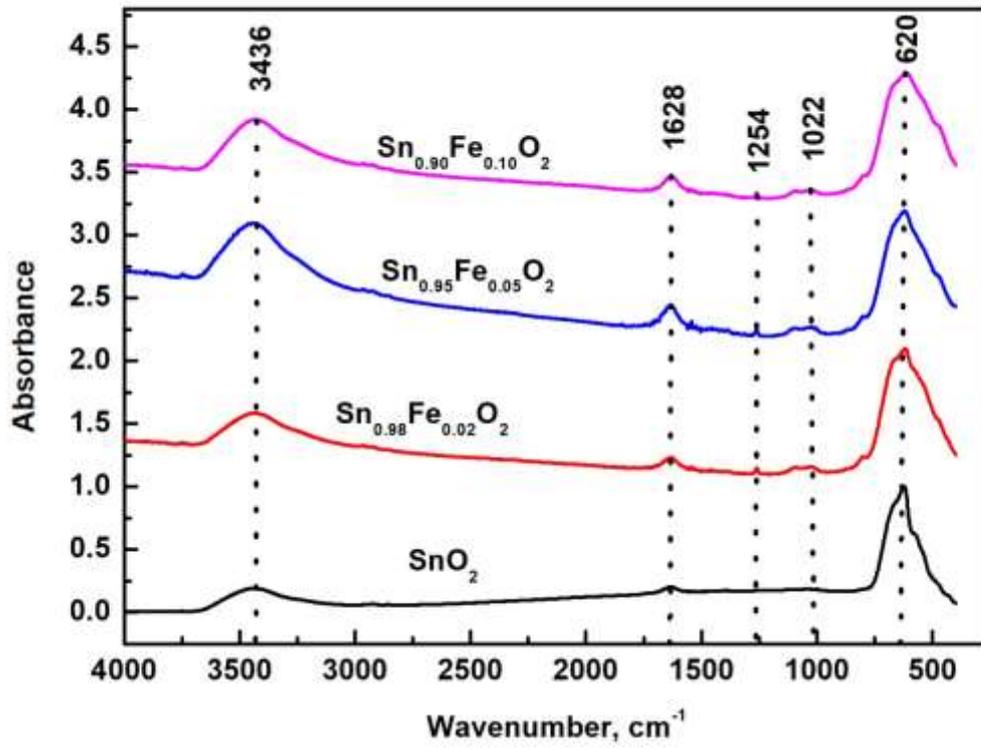


Fig. 5

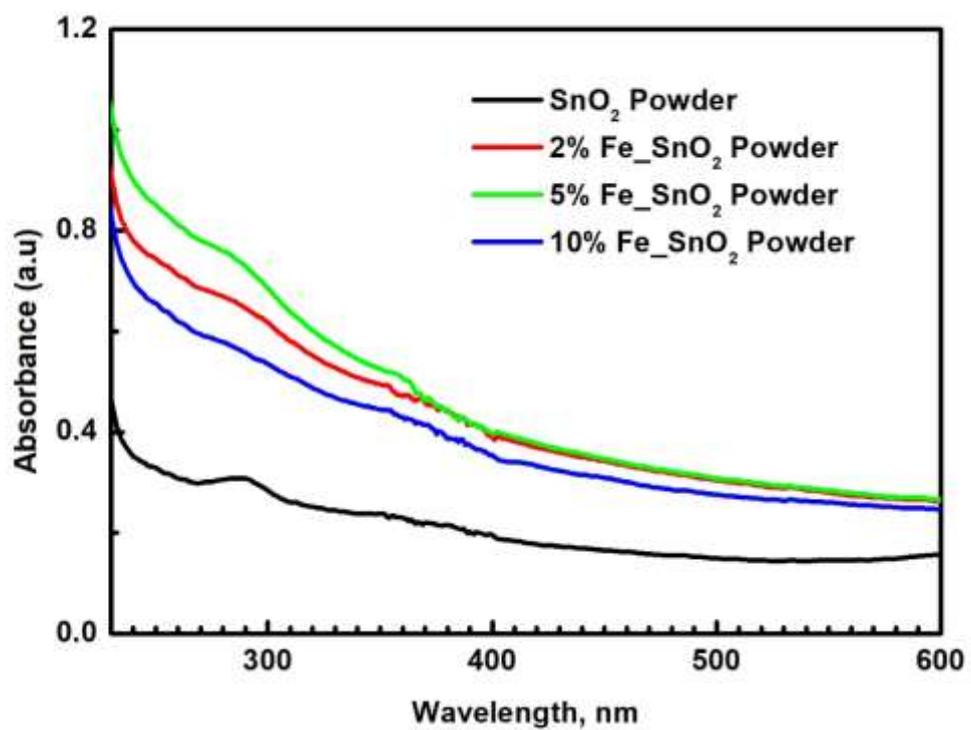


Fig.6

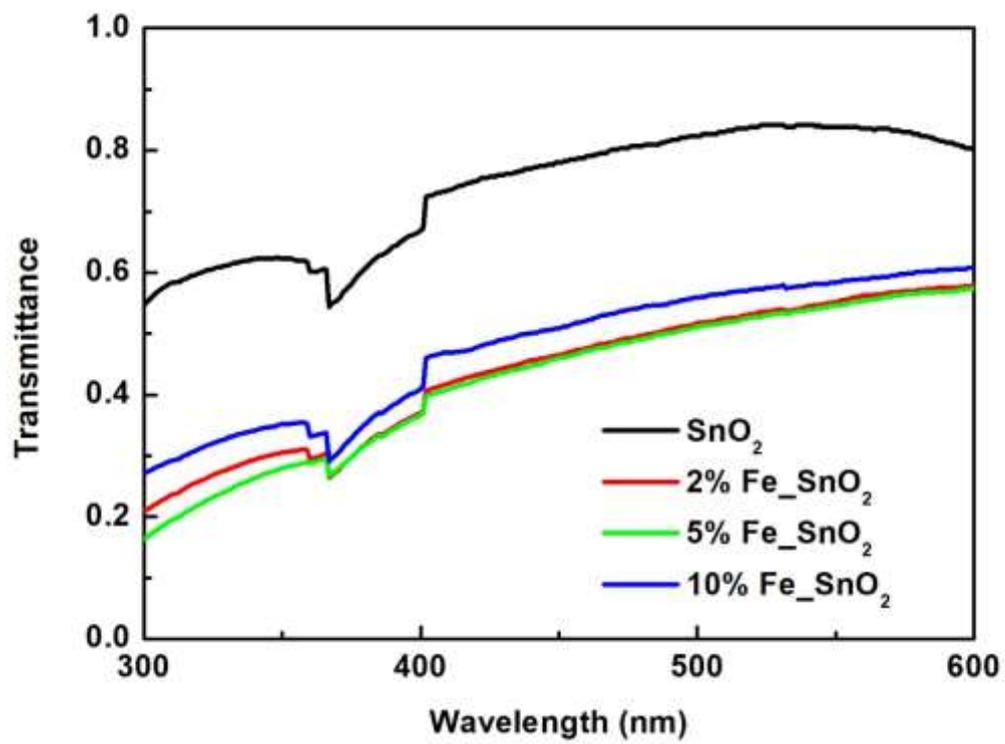


Fig.7

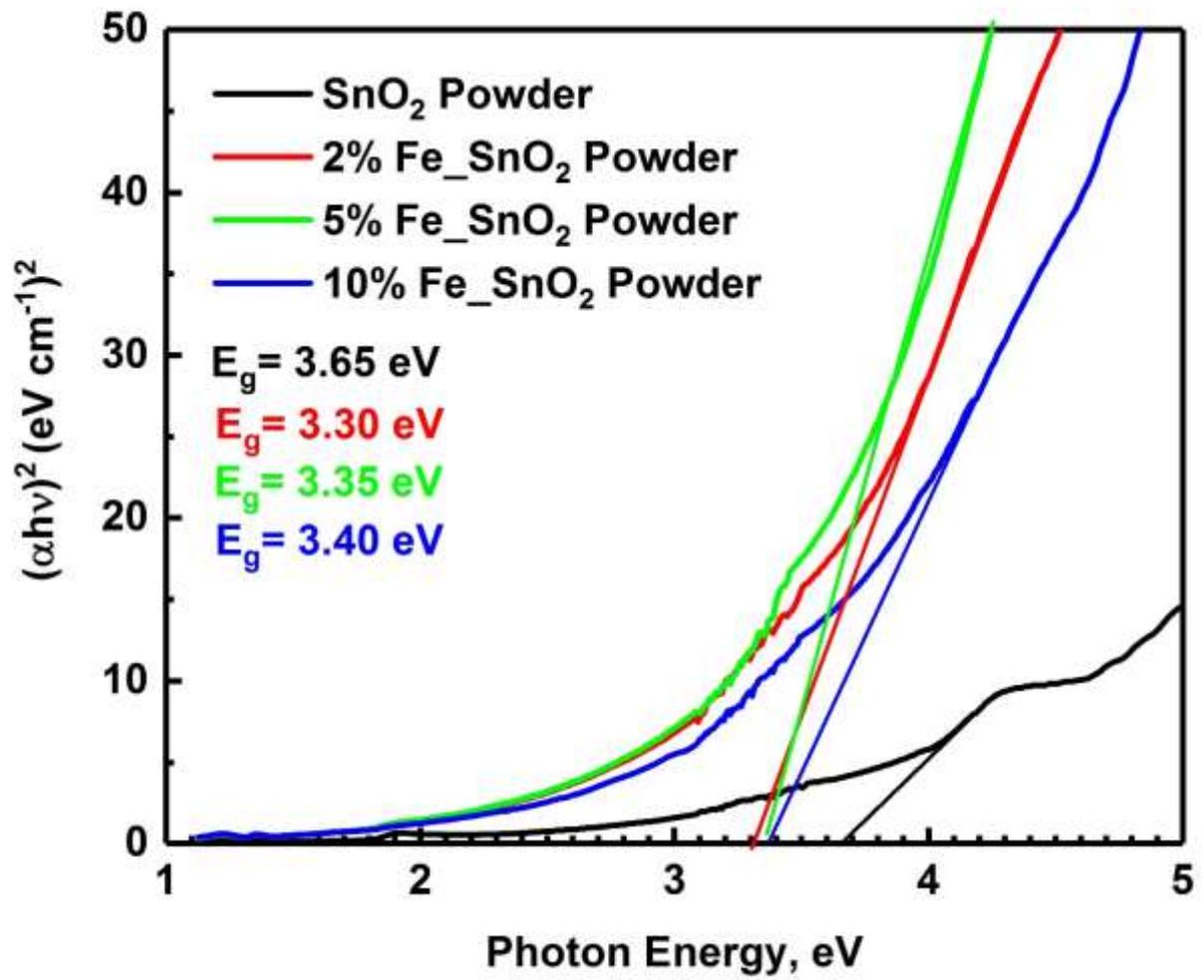


Fig.8

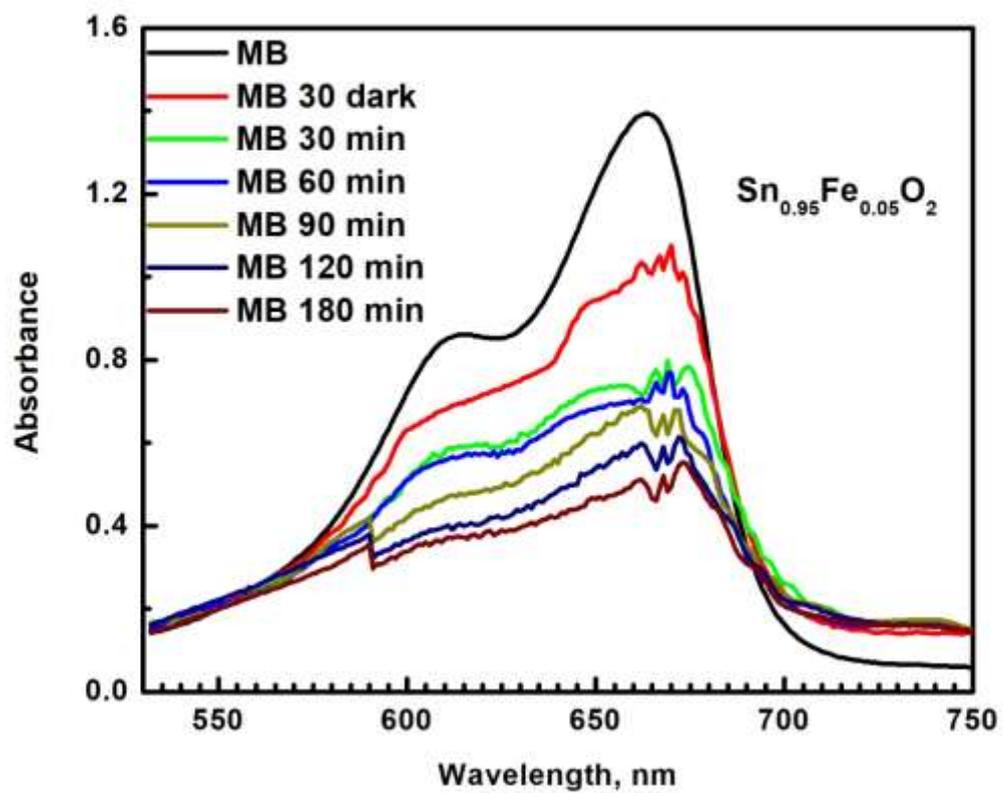


Fig. 9

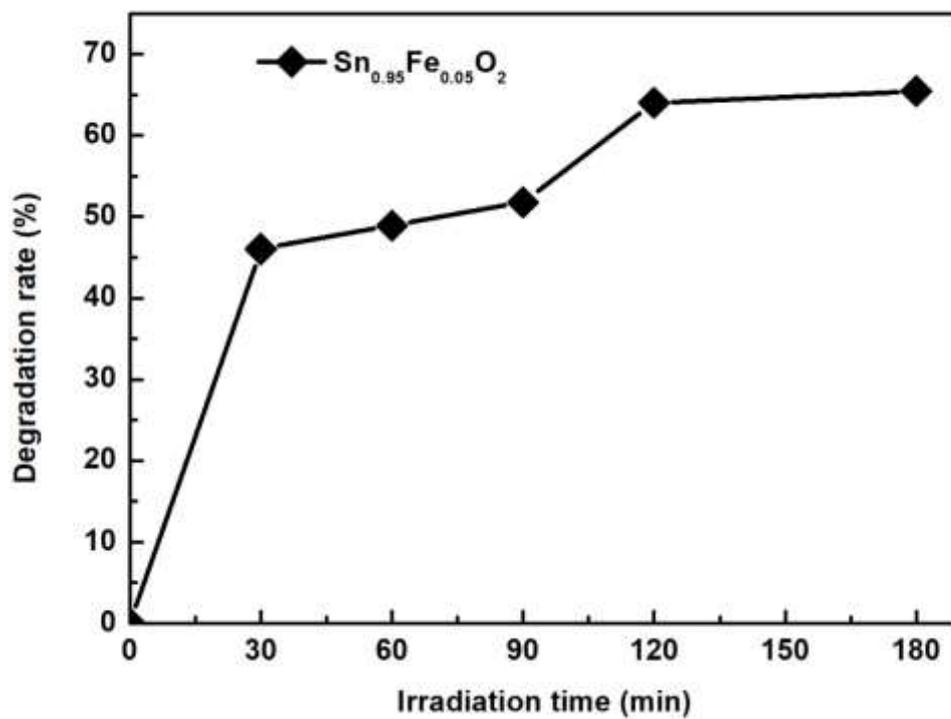


Fig. 10

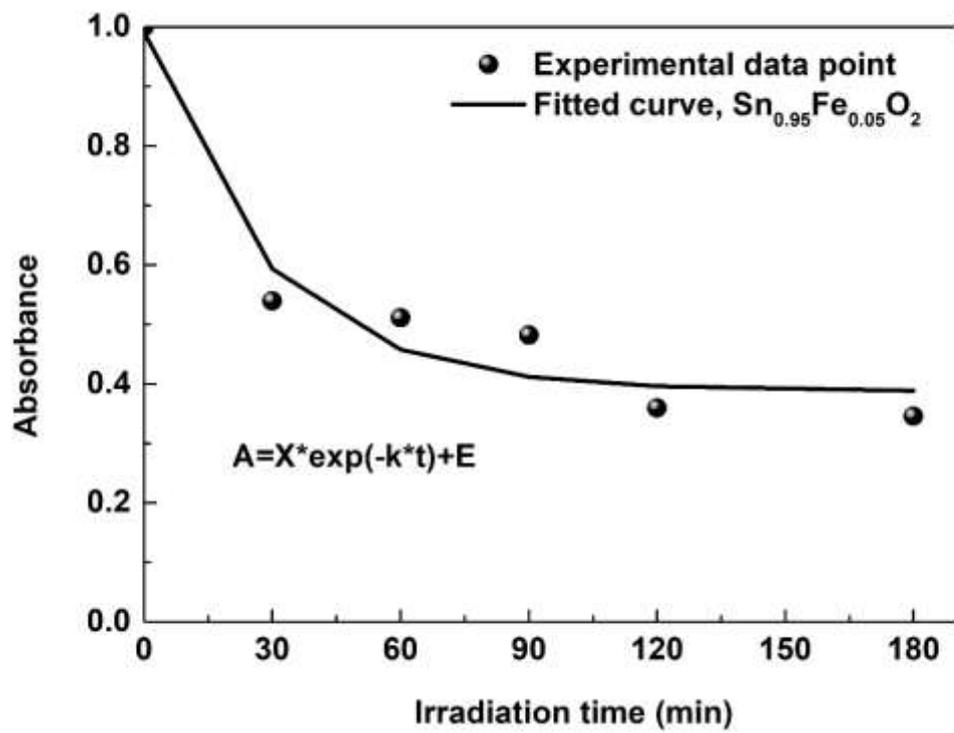


Fig. 11



# Figures

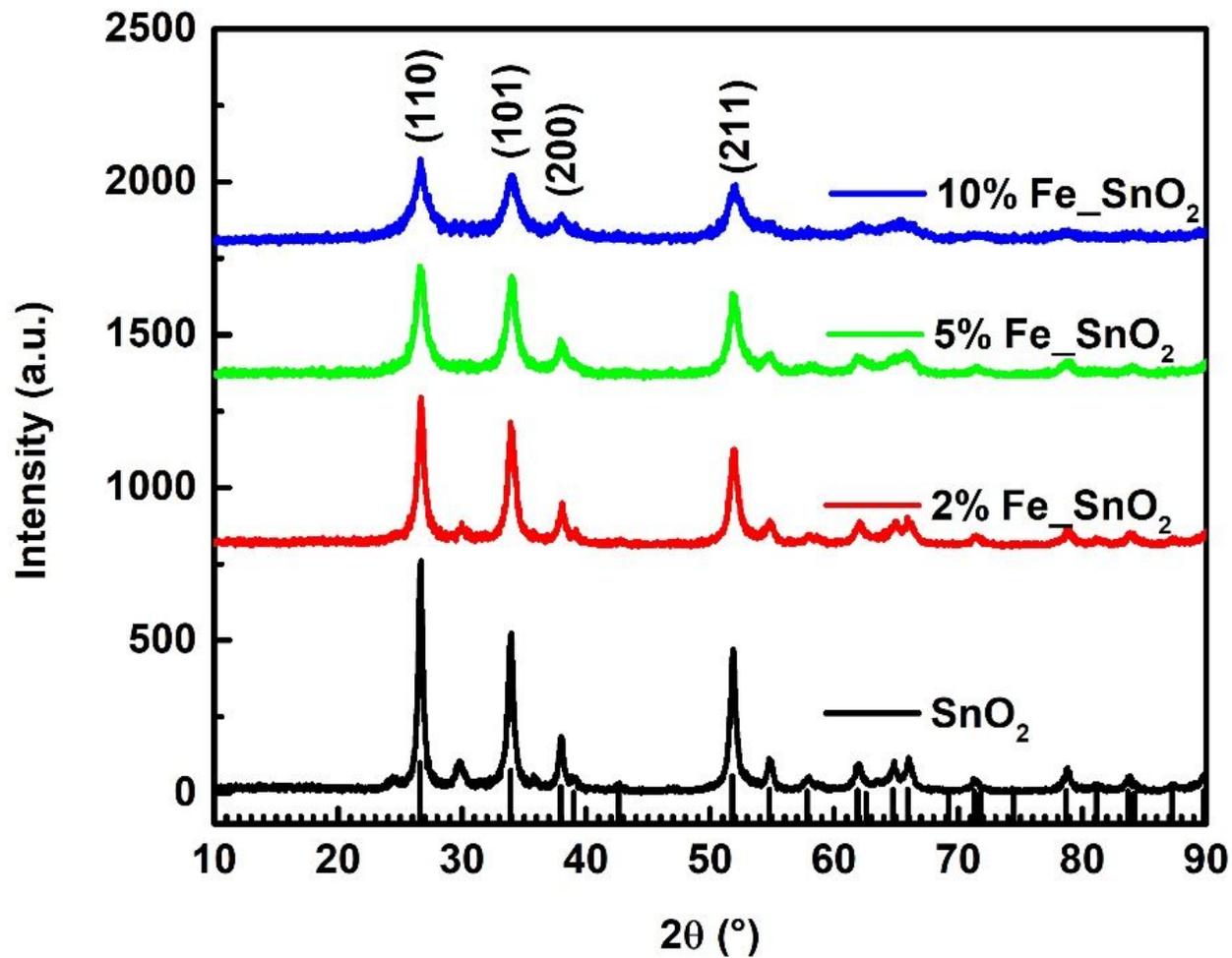
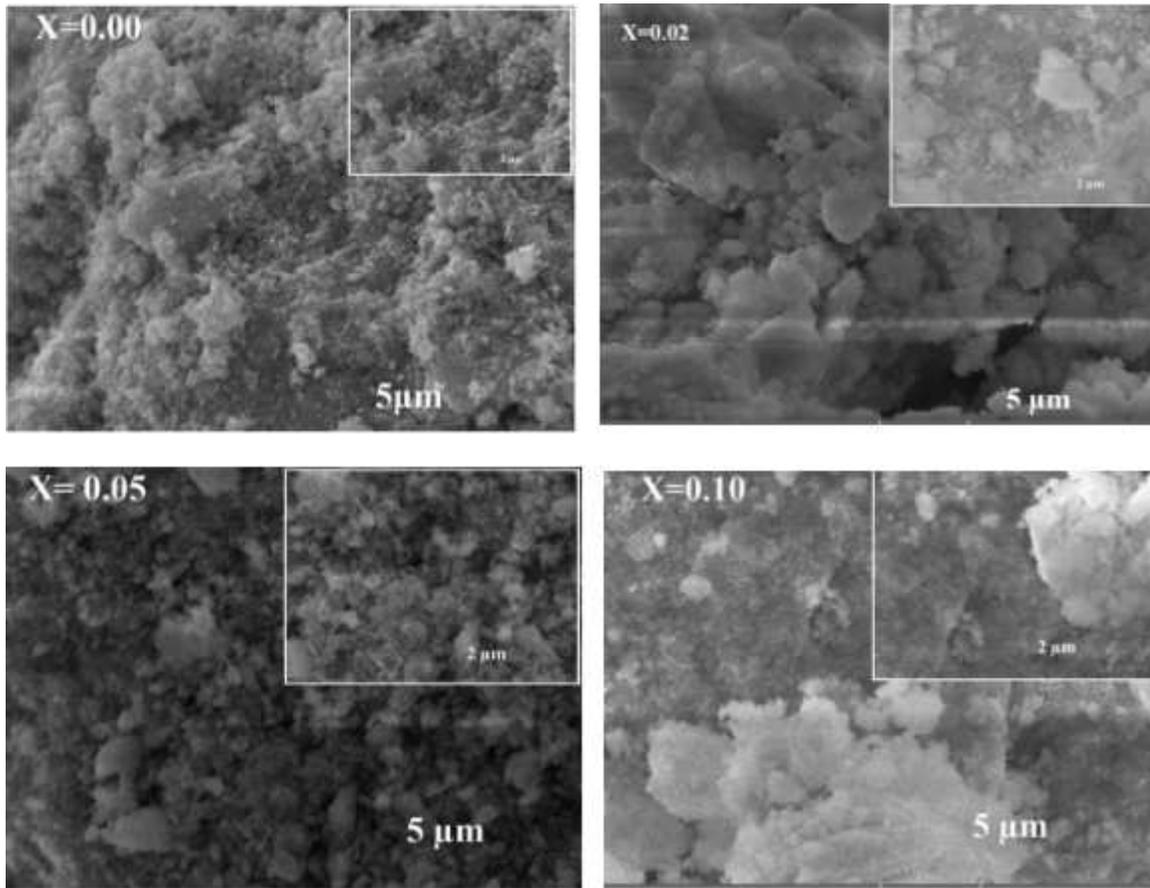


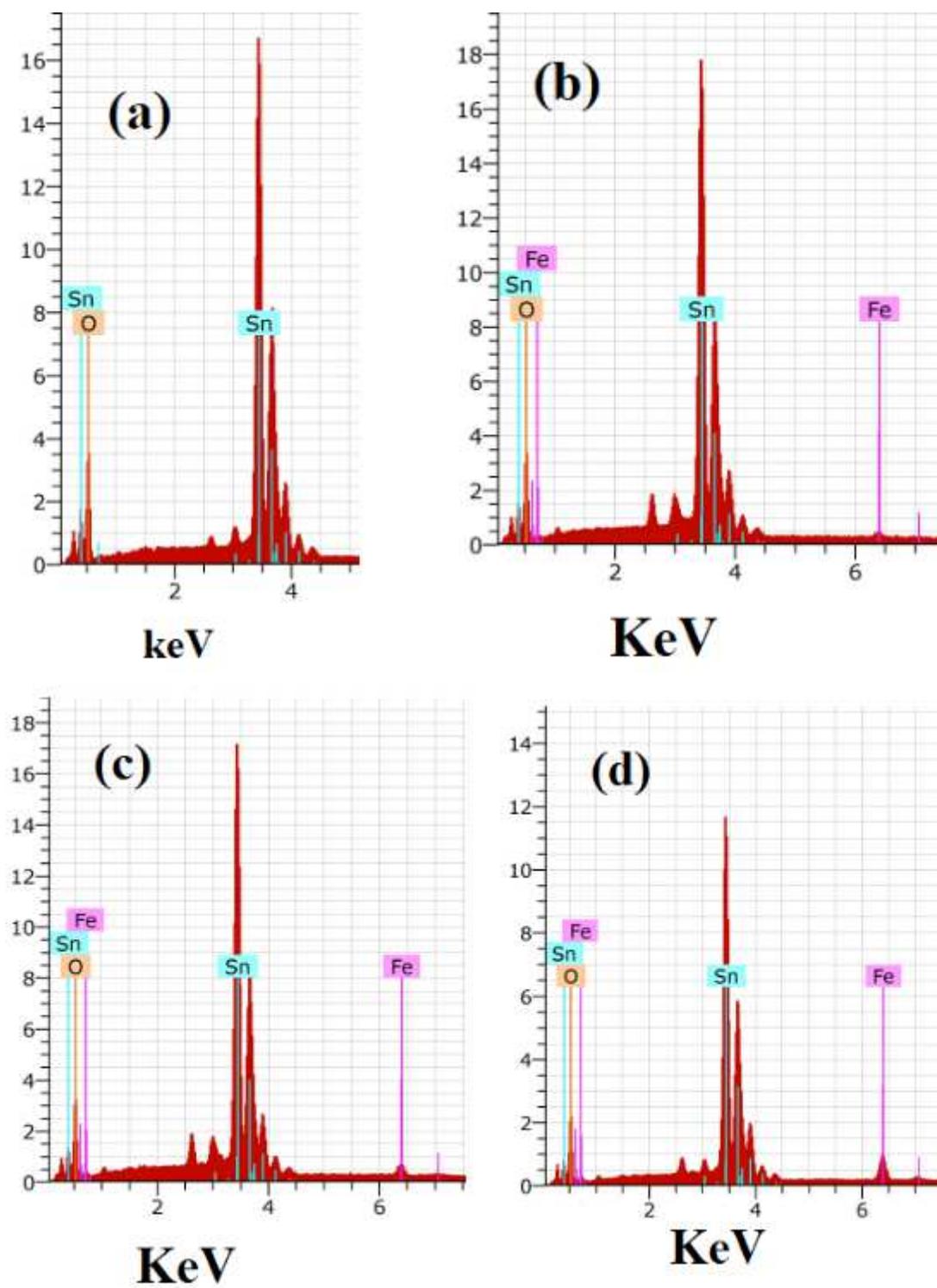
Figure 1

XRD patterns of SnO<sub>2</sub> and Sn 1-xFexO<sub>2</sub> (with x = 2, 5 and 10 at%) nanoparticles



**Figure 2**

SEM images of Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> [ $x=0, 2, 5$  and  $10$  at%] nanoparticles.



**Figure 3**

EDX elemental composition analysis of a pure SnO<sub>2</sub> nanoparticles and b–d Fe–doped Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> [x= 2, 5 and 10] nanoparticles.

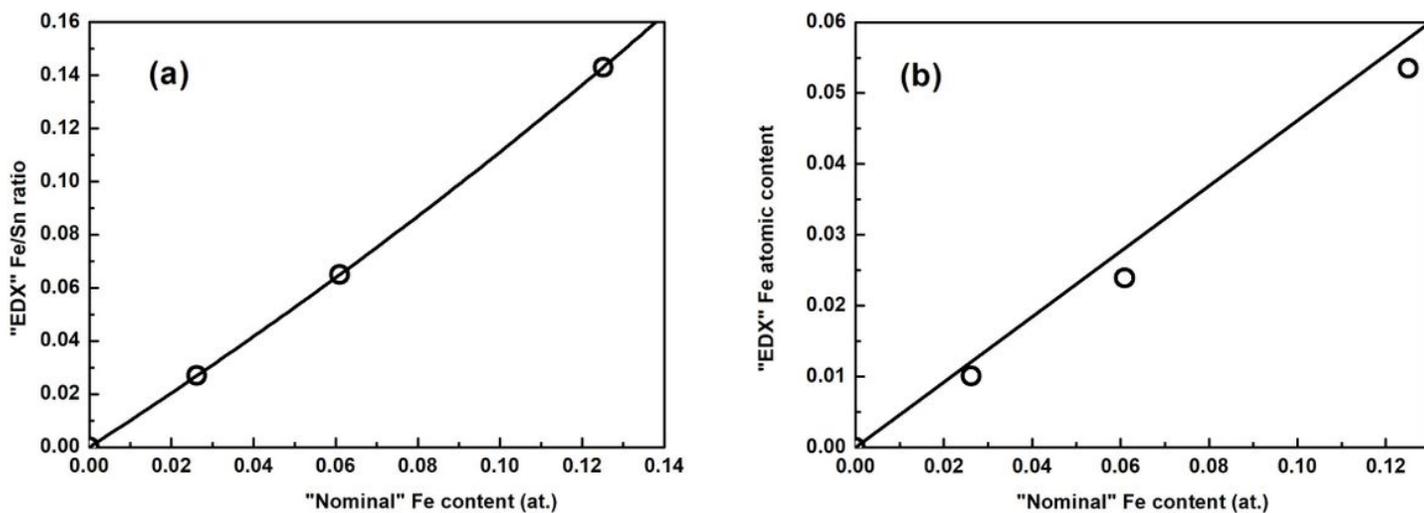


Figure 4

a The measured Fe/Sn atomic ratio and b the calculated Fe atomic content (from EDS analysis) plotted as function of the expected Fe content/doping

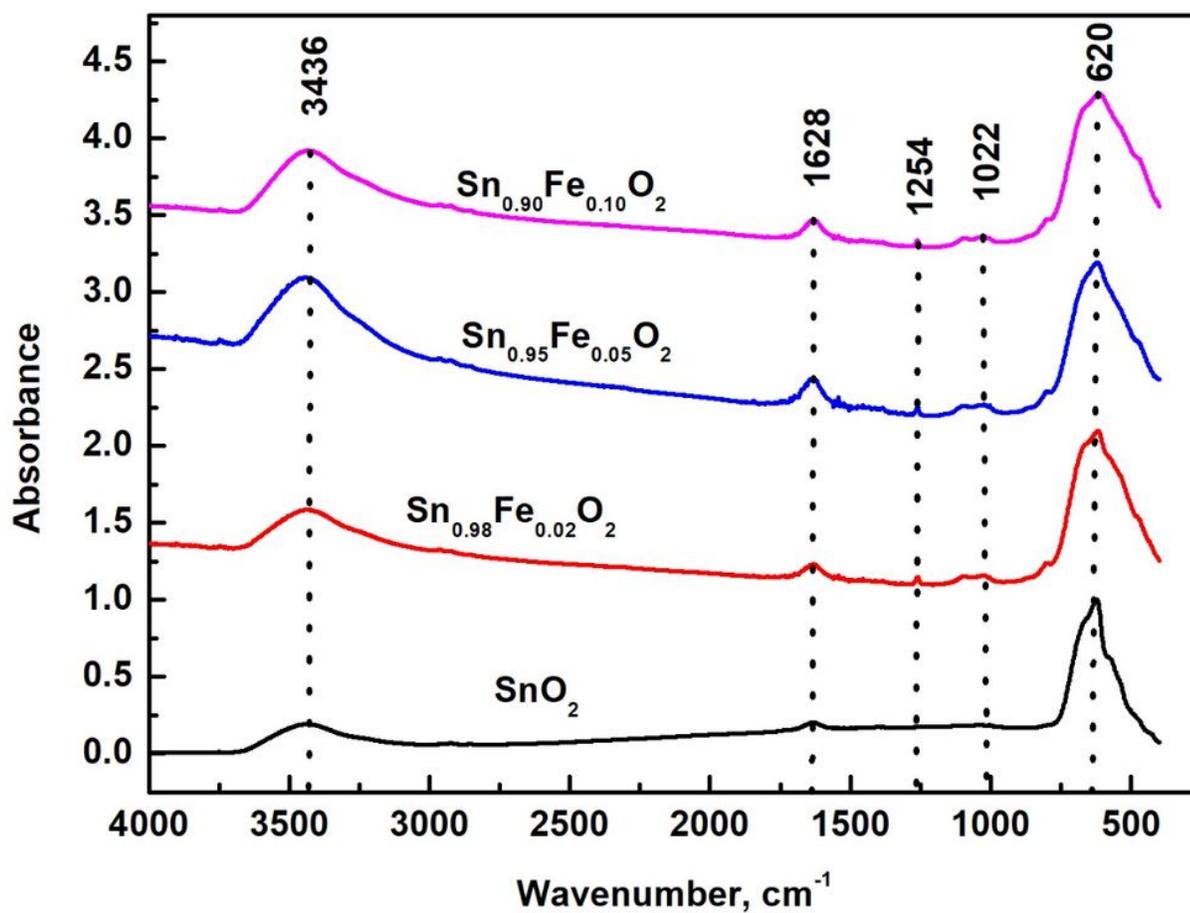


Figure 5

FTIR spectra of Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> [x=0, 2, 5 and 10 at%] nanoparticles at room temperature in the wave number from 400 to 4000 cm<sup>-1</sup>

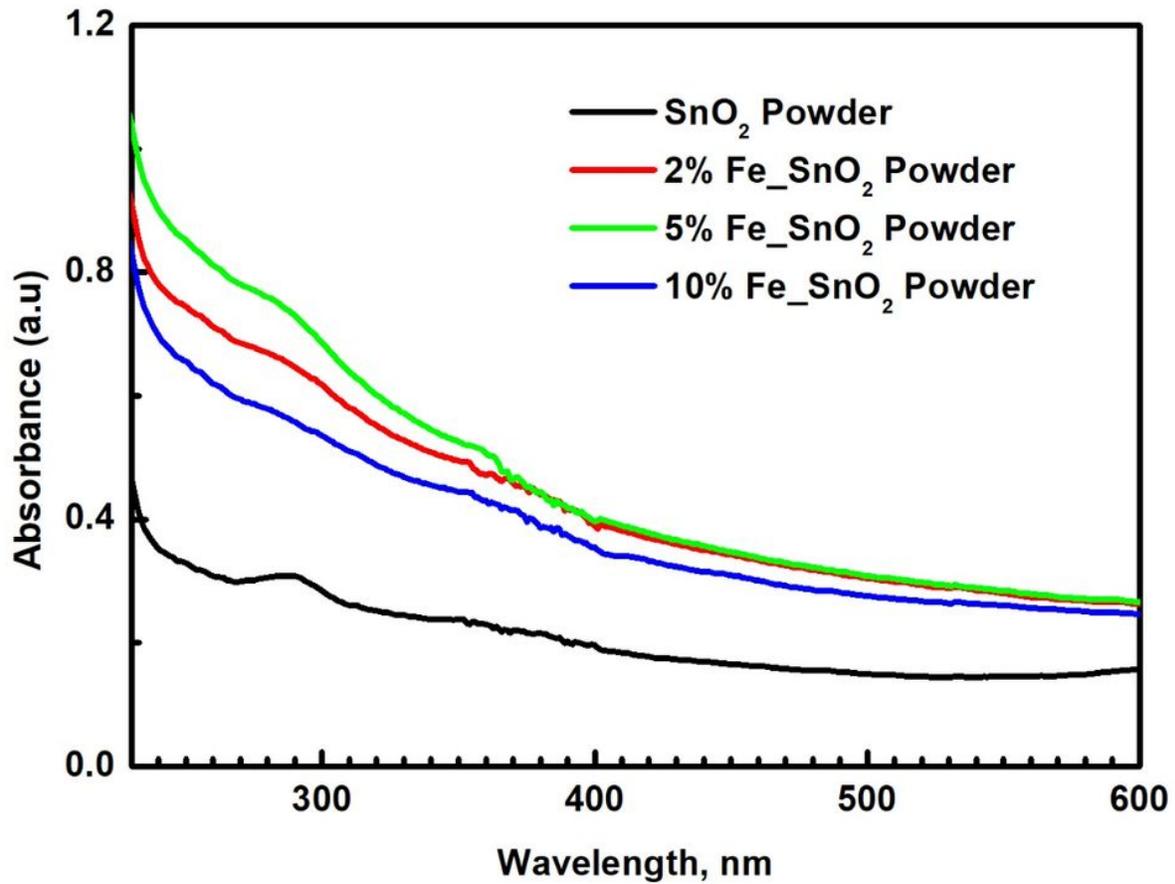


Figure 6

UV-Visible absorption spectra of Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> [x=0, 2, 5 and 10 at%] nanoparticles.

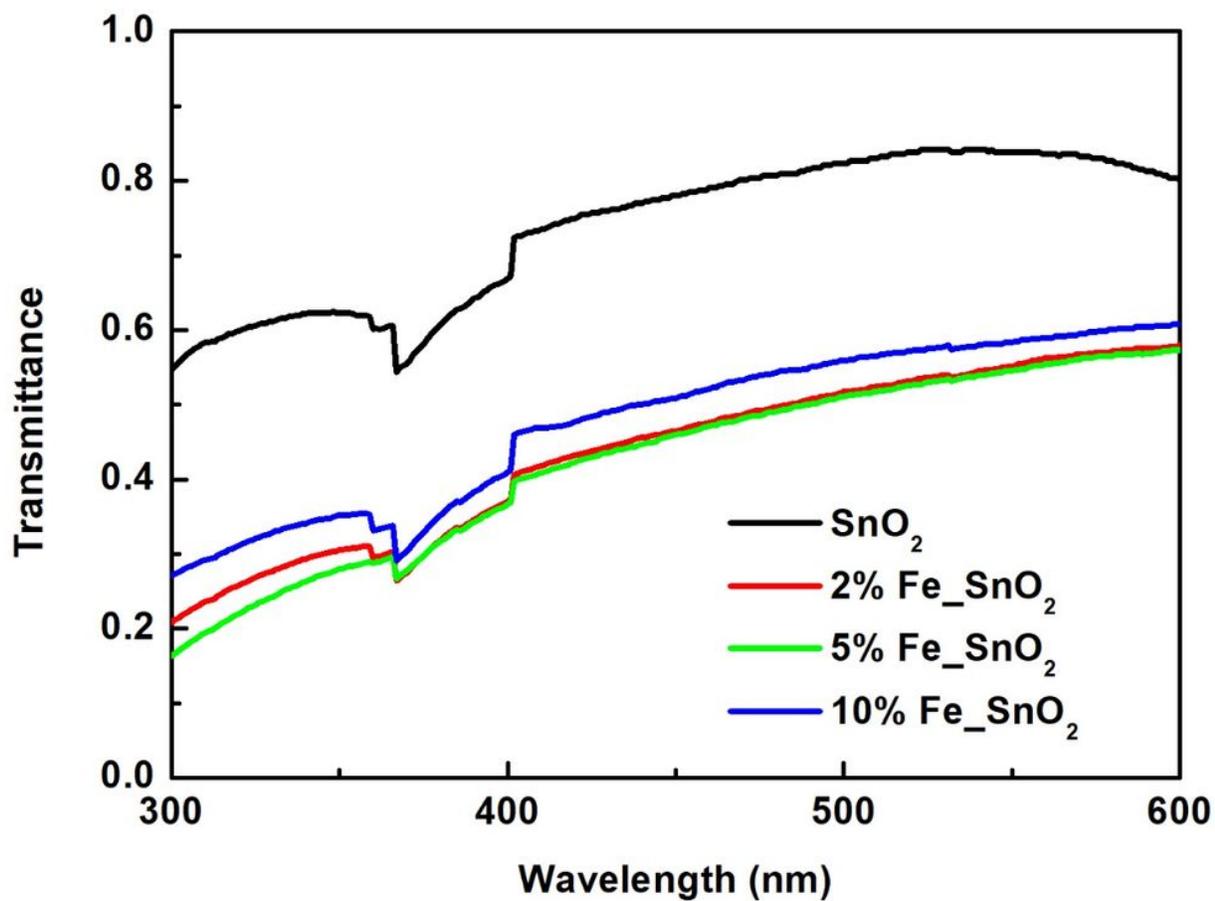


Figure 7

Transmittance spectra of of Sn<sub>1-x</sub>Fe<sub>x</sub>O<sub>2</sub> [x=0, 2, 5 and 10 at%] nanoparticles.

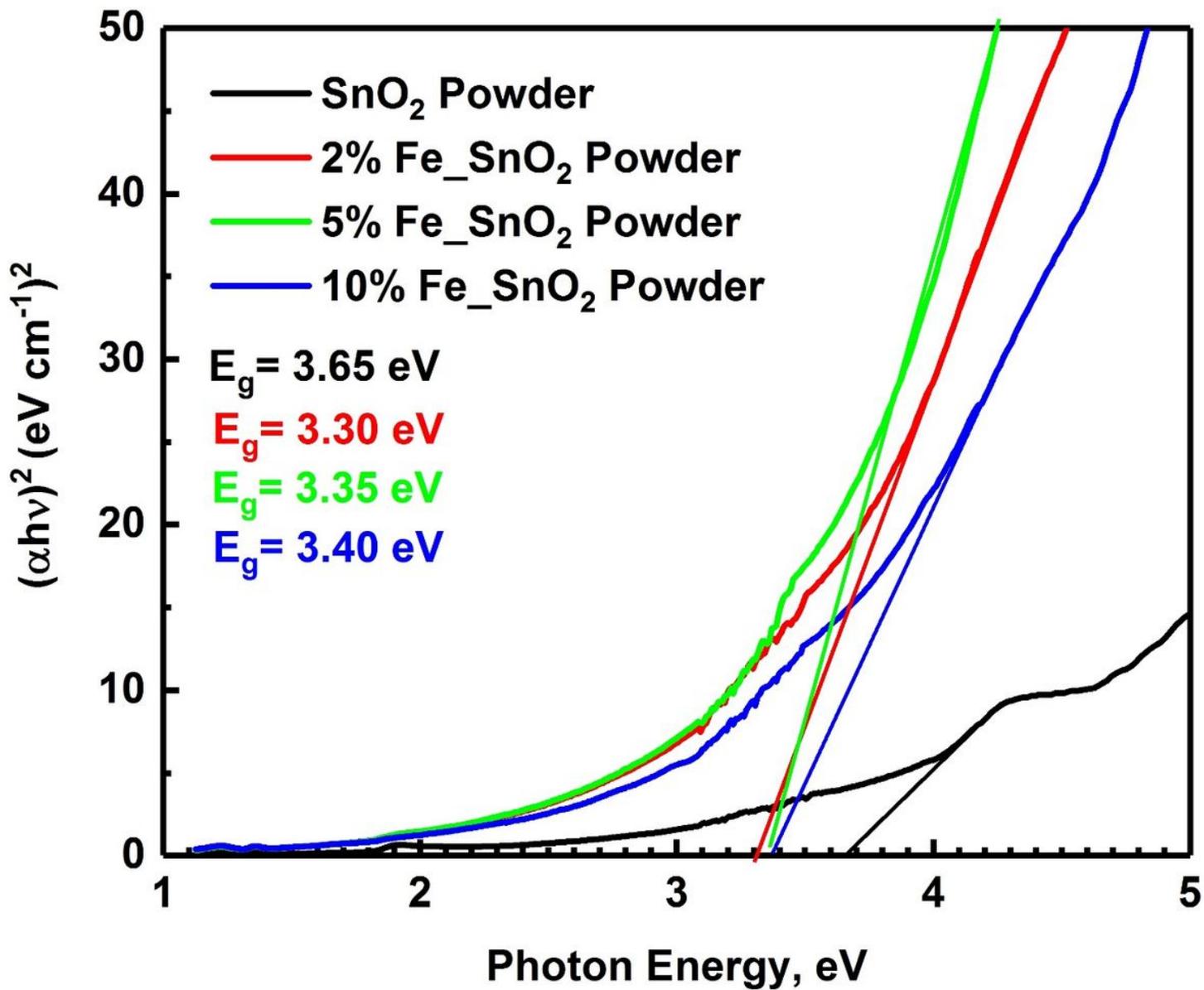


Figure 8

The  $(\alpha h\nu)^2$  versus  $h\nu$  curves of  $\text{Sn}_{1-x}\text{Fe}_x\text{O}_2$  [ $x=0, 2, 5$  and  $10$  at%] nanoparticles, for the optical energy gap calculation.

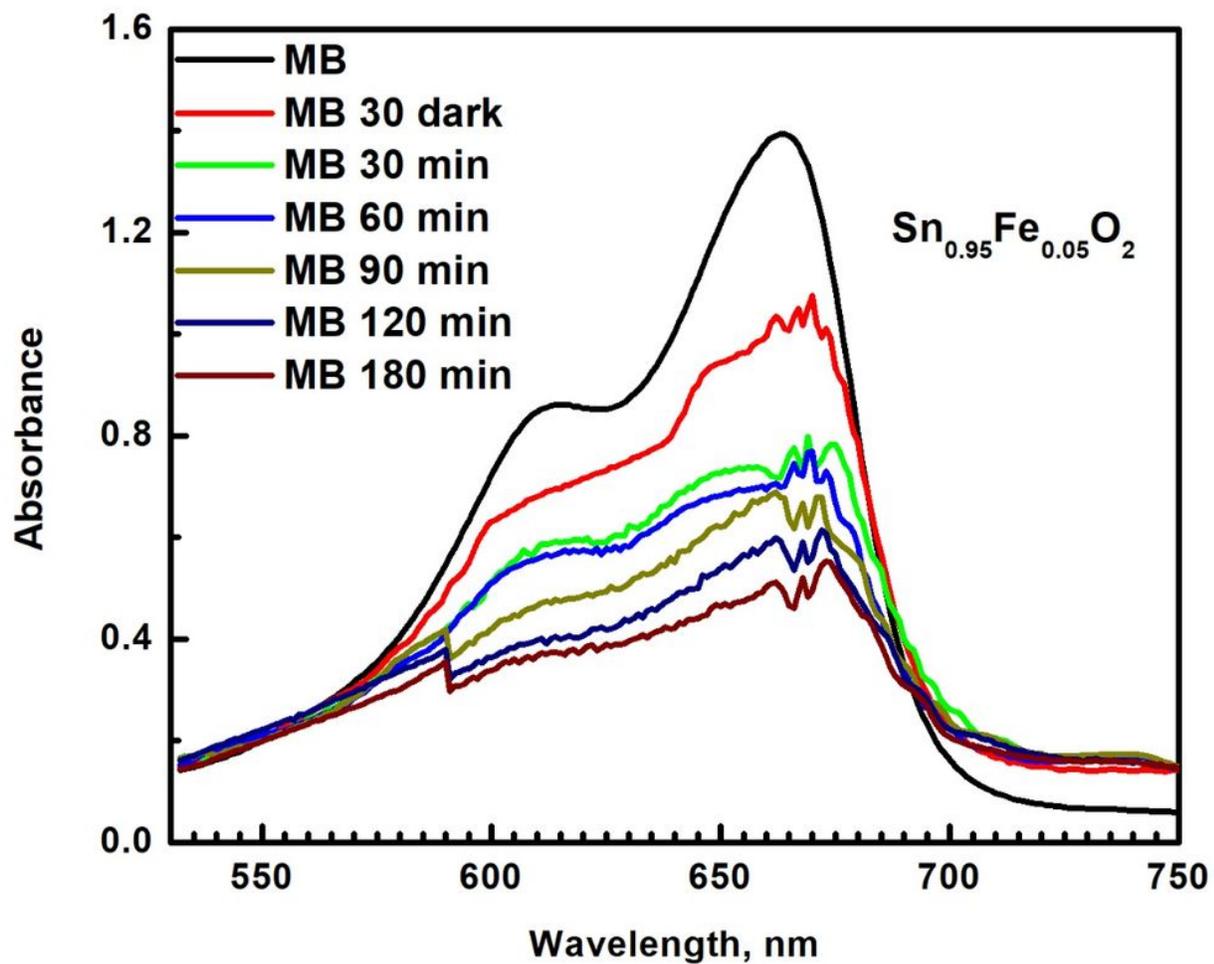


Figure 9

The effect of  $\text{Sn}_{0.95}\text{Fe}_{0.05}\text{O}_2$  nanoparticles on the absorption spectra of MB solution for different reaction time under UV irradiation.

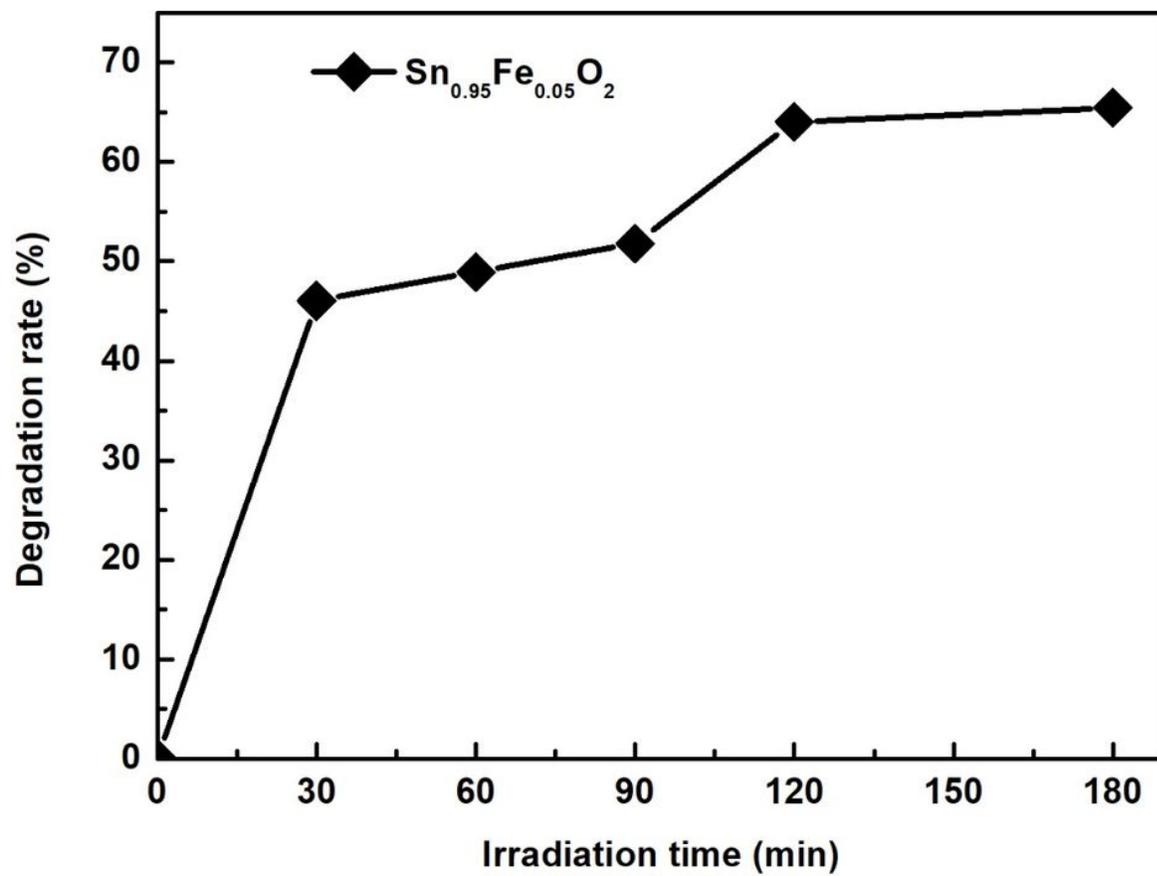


Figure 10

Degradation rate of Sn<sub>0.95</sub>Fe<sub>0.05</sub>O<sub>2</sub> nanoparticles in MB degradation

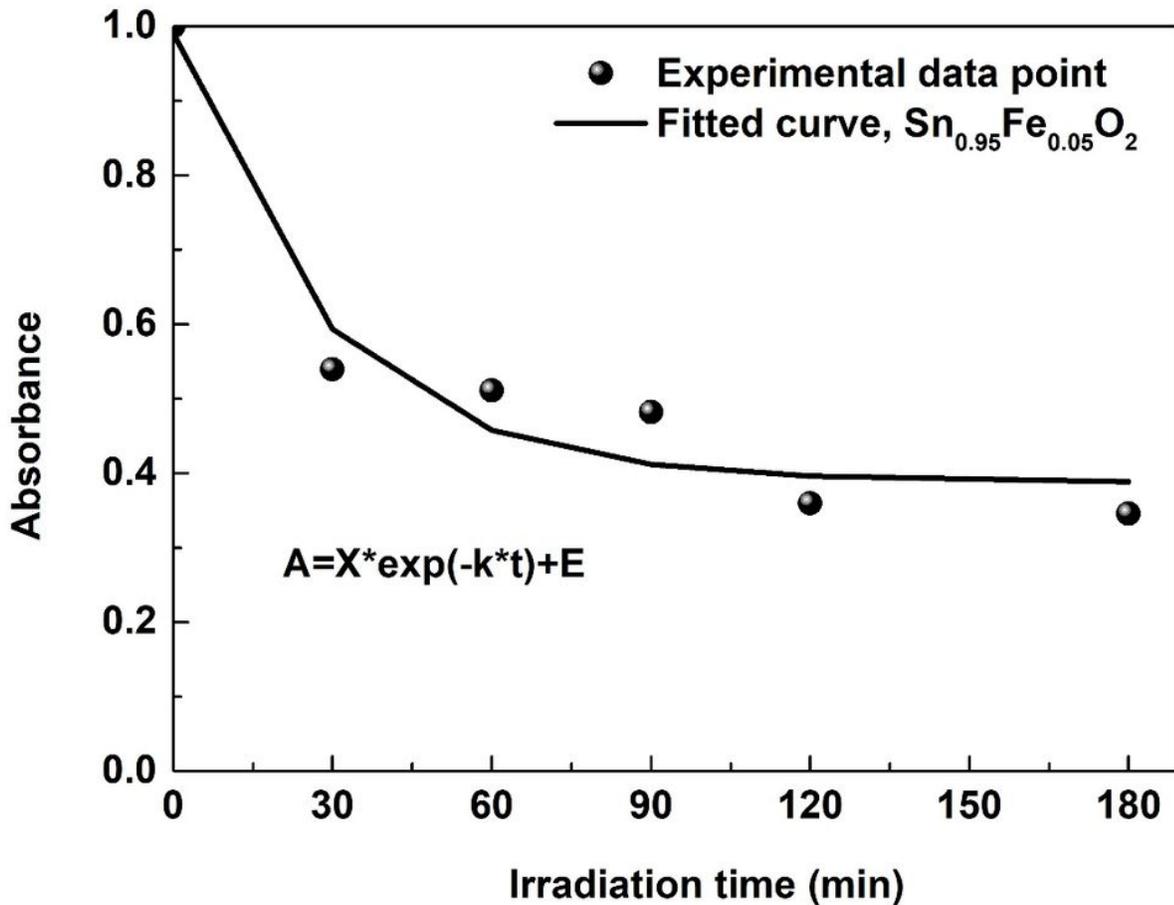


Figure 11

Decolorization kinetics of MB aqueous solutions by Sn<sub>0.95</sub>Fe<sub>0.05</sub>O<sub>2</sub> nanoparticles without UV irradiation. The nanoparticles were prepared by using co-precipitation method. The photocatalytic process at UV-irradiation follows first-order kinetics according to the equation  $A=X*\exp(-k*t)+E$  (circles) Experimental data point and (solid lines) fitted curve (solid lines). Here  $A_0$  is the initial absorbance of the dye solution,  $A(t)$  is the absorbance at time  $t$ , and  $k$  is the rate constant of photocatalysis.