

Structural Analysis of Sol-Gel Derived TiO₂ Nanoparticles: A Critical Impact of TiO₂ Nanoparticles on Thermo-Mechanical Mechanism of Glass Fiber Polymer Composites

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Structural analysis of sol-gel derived TiO₂ nanoparticles: A critical impact of TiO₂ nanoparticles on thermo-mechanical mechanism of glass fiber polymer composites

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Abstract

The incorporated of inorganic nanoparticles with thermosetting epoxy polymer are an emerging field of research over past few years. It is well analyzed that epoxy matrix are brittle in nature that shows poor crack initiation and propagation and results poor thermo-mechanical properties. Therefore, researchers are showing their interest towards nanoparticles embedded epoxy composites to improve their fracture resistance (brittleness and toughness). In this investigation, the dispersion of TiO₂ nanoparticles at different weight fraction (0-2%) with glass fiber reinforced epoxy composites is performed to enhance structural and thermo-mechanical properties. The TiO₂ nanoparticles are prepared by sol-gel method and structural analysis of TiO₂ nanoparticles shows greater interfacial bond with epoxy matrix and glass fibers due to fine dispersion of nanoparticles. From obtained results, a significant enhancement in their tensile strength (38.56%), flexural strength (30.52%), inter-laminar shear stress (25.22%), impact strength (327.10%), micro-hardness (48.53%) and fracture energy (40.19%) with minimum detrimental effect on toughness was revealed for GFRP-T1.0 compare to GFRP-T0.0 composite laminates. The stiffness and rigidity also improved up to 52.72% and 34.13% respectively for GFRP-T1.5 compare to GFRP-T0.0 composite laminates. The effects of nanoparticles contents and clustering size on thermal stability and glass transition temperature of developed composites are observed by thermo-gravimetric analysis. The surface morphology of TiO₂ nanoparticles are characterized by transmission electron microscope (TEM) while dispersion of nanoparticles and failure of developed composites were analyzed by scanning electron microscopy (SEM).

Keywords: TiO₂ nanoparticles; Sol-gel method; Physical properties; Mechanical properties; Microstructure analysis; Glass fiber

1. Introduction

The advance composite materials made from continues fiber polymer matrix composite laminates with different orientation and arrangements comprise a successful structural materials due to their less density, better specific strength and stiffness, lower thermal coefficient, high fatigue and damping resistance for various industrial application like construction, aerospace, marine, automobile and biomedical [1]. Fiber reinforced polymeric composites (FRPCs) such as glass fiber and carbon fiber have attracted much awareness as a substitute for conventional materials like wood, steel and plastics for various structural applications. The replacement of concrete structure with FRPCs gave more flexibility and suitability to the industrial society and has been also highlighted by American concrete institute (ACI) committee-503 where the adhesion and bonding between fibers and matrix acts a crucial role on the overall performance of the materials [2, 3]. Although, carbon fiber is not available plenty due to its more expensive and low reusability with high design and manufacturing cost, so glass fiber reinforced polymeric composite (GFRP) can be a better alternative materials for construction, automotive, marine, aerospace, wind turbine and power plants that also posses excellent chemical, physical and thermo-mechanical properties [4].

Thermoset epoxy resins are commonly used polymeric matrix for fabricating high performance and flexible composite due to their higher durability, better thermal stability and superior chemical, electrical, and mechanical properties [5, 6]. However, epoxy resins have highly brittle in nature that require high fracture and impact strength. Therefore, to overcome these disadvantages, incorporating or reinforcing fibers/fillers into polymer matrix enhanced the strength and stiffness of the composite without disclaiming its toughness [7]. To improve the performance and properties of FRPCs, nanoparticles were incorporated to identify as a prospective solution. A broad range of nanoparticles such as silica, clay, calcium carbonate, nickel oxide, layered silicate, titanium dioxide etc. have gained particular interest for inorganic/organic nanocomposites [8]. The impact of nanoparticles on the developed composite has been studied and investigated their performance including TiO₂ spherical nanoparticles [9], carbon oxide dots [10], carbon nanotubes [11], nanoclays [12], dichalcogenides transition metal [13], and graphene [14]. The several researches have been conducted on metallic or inorganic nanoparticles that explained about their performance and revealed an excellent candidate to reinforce with polymeric materials mainly thermosetting epoxy resin for better mechanical,

thermal and electrical properties [15]. Among various inorganic nanoparticles, TiO₂ has non-toxic, lower in cost, superior refractive index, and chemical inertness that attendant higher interest towards research and have been used for different commercial applications such as antibacterial activities, inhibition, photo electric and photo catalytic conversion in solar cells as well as utilized in food additives, self cleaning, corrosion resistant coating, water and air purification etc [16-18]. Therefore, TiO₂ nanoparticles have been recognized as promising reinforced materials to develop effective composite materials due to their light density, higher mechanical and thermal properties, reduce ultraviolet and thermal degradation [19-21]. Several investigations examined that, the performance and efficiency of the developed composites have been influenced with volume contents, shape and size that enhanced the thermal, mechanical, fatigue and toughness properties [22, 23] while greater surface area of nanoparticles makes chemically very active to bond with matrix materials [1]. However, these nanoparticles have been agglomerated and incompatibility with polymer matrix due to large surface area to volume ratio that results improper bonding and reduce their respective properties [8, 24]. Therefore, hybridization of these nanoparticles with different fiber reinforced materials was identified as potential solution that enhanced their compatibility and increases the durability of the materials where a low amount of homogeneous dispersion of nanoparticles with GFRP composites demonstrated higher mechanical, thermal, and fracture properties as well as improved their glass transition temperature (T_g) and maximum decomposition temperature [22, 25, 26].

Therefore, the current study deals with structure analysis of TiO₂ and the effects of nanoparticles at different weight fraction on thermo-mechanical characterization of GFRP composites have been examined. The TiO₂ nanoparticles are prepared by sol-gel methods for better dispersion with epoxy matrix and GFRP composites laminates are fabricated with vacuum assisted resin infusion molding techniques (VARIM).

2. Materials and methods

The bidirectional E-glass fiber was used for reinforcements that were purchased from Surfland Composites Pvt. Ltd. Bangalore, India. The E-glass fiber was having a density of 180g/m² with a 5-7micrometre diameter. Araldite LY556 (Diglycidyl ether of bisphenol, density~ 1.15-1.20g/cc and epoxy content~5.30-5.45eq/kg), HY 906 (methyl nadic anhydride, density~1.20-1.25g/cc) and DY 070 (density~0.95-1.05gm/cc) were used as resin, hardener and accelerator respectively as a

matrix material for fabrication of GFRP composites that were purchased from Huntsman International Pvt. Ltd. Mumbai, India.

For preparation and synthesis of titanium dioxide (TiO_2) by sol-gel method, Titanium Isopropoxide ($\text{C}_{12}\text{H}_{28}\text{O}_4\text{Ti}$) having density-0.96 gm/ml, molecular weight-284.22g/mol and Isopropanol as a liquid precursor were used that purchased from Sisco Research Laboratories Pvt. Ltd. Mumbai, India. TiO_2 nanoparticles were dispersed in resin solution then the solution was prepared with help of proper mixing from proper mechanical stirring then, ultrasonication was done for proper dispersion of nanoparticles in resin solution, then composites were fabricated with VARIM.

2.1 Preparation of TiO_2 nanoparticles

The TiO_2 nanoparticles was prepared through hydrolysis and condensation process with help of Sol-gel methods and shown in Fig. 1. Initially, 500ml distilled water having pH 7 was taken in a beaker then, put over a hot plate magnetic stirrer. The stirring process was done at 700 rpm and maintains 70°C temperature. Now, 10ml titanium Isopropoxide and 25ml Isopropanol was taken in two separate beakers. Instantly, the titanium Isopropoxide was added to Isopropanol with help of a dropper because titanium Isopropoxide was hazardous it gets react instantly when it comes in contact with moisture and finally mix it properly. The prepared solution was now added drop wise in distilled water beaker. Instantly the gel formation took place in the beaker and the hydrolysis process was started. The desired pH2 of the solution was attained by adding HNO_3 or NH_4OH . Hydrolysis of the solution produced turbid solution when it was heated at $60\text{-}70^\circ\text{C}$ for 14hrs. The process of producing a turbid solution was called peptization. After the peptization process, the decrement in the volume of the solution to 50cm^3 was recorded. Depending upon the preparatory conditions, the resultant solution was obtained in white colour with high viscosity. The prepared precipitates were washed with ethanol several times and then filtered with a bench top centrifuge. The prepared filtrated powders were drying in the oven for 3hrs, at 100°C then powders became yellow-white colour. The final obtained powder was coarse in structure, so, mortar and pestle were used to produce fine nanoparticles of TiO_2 .

2.2 Dispersion of Nanoparticles

The proper dispersion of nanoparticles in resin solutions was the essential requirements to minimize the scatter and agglomeration in the developed composites. The following techniques such as mechanical stirring, shear mixing, acoustic cavitations, evaporation with solvent,

ultrasonic waves, pulsed ultrasound vibration and direct in-situ incorporation with chemical and polymerization methods were used to improve the dispersion process of nanoparticles [25, 27]. In particular, it was observed that the sol-gel methods were effective to improve the bonding strength of TiO₂ nanoparticles and homogeneous dispersion of nanoparticles played a vital role in improving the thermo-mechanical properties of developed composites [28, 29]. The ultrasonic is a widely used technique to avoid scatter/ agglomeration during the mixing of nanoparticles with resin and provide fine quality of filled matrix material [30]. Thus, a fine structure of the matrix enhanced the interfacial bonding strength between fibers and matrix of developed composites.

For the present study, mechanical stirring and ultrasonication techniques were adopted for superior dispersion of nanoparticles. Initially, the mechanical stirring was done for 15min to prepare a mixture of TiO₂ nanoparticles and resin then finally, the sonication process was used for 30min to do proper mixing and dispersion to avoid agglomerations.

2.3 Preparation of GFRP Composite

Glass fiber reinforced TiO₂ nanoparticles with epoxy matrix composite laminates were prepared with VARIM methods and details methodology is showing in Fig. 2. For composite plate fabrication LY556, HY906 and DY090 were used as resin, hardener and accelerator respectively. The weight ratio of resin and hardener is 1:0.95 and the accelerator is taken as 2 % of resin weight. An initially adequate amount of resin was taken in hard plastic, then TiO₂ nanoparticles was added then, the whole resin solution (Part A) was taken for mechanical mixing for 15 min. Then slurry of (resin + TiO₂ nanoparticles) was then processed for ultrasonic dual mixing process (UDM) with simultaneous stirring by the impeller for proper dispersion of nanoparticles in a resin solution. After the dispersion of nanoparticles, hardener and accelerator were added into the solution (Part A) in a stoichiometric ratio. Then the mixture was again put under mechanical stirring action for 10 minutes for proper mixing. After that, the whole mixture is put inside desiccators to remove the gas or void entrapped into the solution during mixing. The prepared mixture was used for the fabrication of composite plates by the VARIM process.

Vacuum-assisted resin infusion techniques have become popular in the manufacturing of these composites to produce high-quality large-scale products where the process was well controlled, may produce composite plates with less than 1% voids. For this study, the single-side mould surface was taken and then dry release film (or peel ply) were placed on the mould

surface. Now, eight layers of glass fiber mats were arranged and cover on both side with peel ply then mesh was placed over the top surfaces of this arraignment for better flow of resin solution with high permeability inside the composite's laminates. The vacuum was created over the mould and the resin solution is infused with help of a vacuum pump where the resin solution was spread over the entire surface of the mould. The peel ply was used for superior smoothness and easy removal of composite plates from the mould surface. Finally, the prepared composite plate was taken from the mould and put inside an oven at 120°C for 2 hours called pre-curing and then post-curing of composite plates at 160°C for 6 hours. The prepared composite plates were cut as per ASTM standard for various testing.

3. Characterization and testing

3.1 Characterization of TiO₂ nanoparticles

3.1.1 Structural characterization

The physical structure of the nanoparticles was observed by Burcker D8 focus X-Ray diffraction machine (XRD), operated at a voltage of 30kv/15mA with cu radiation at an angle of 2θ from 5 to 75. The weight loss of nanoparticles at increasing temperature was analyzed by EXSTAR TGA/DTA 6300 Thermo-gravimetric analyzer, where nanoparticles was heated with a controlled heat flow rate of 10°C/min at nitrogen environment. To understand the functional groups (chemically) present in nanoparticles, the powder was going through Fourier transform infra-red (FTIR) spectrometer of PerkinElmer Spectrum machine in the range of 500-4000 cm^{-1} at a resolution of 4 cm^{-1} .

3.1.2 Surface morphological analysis

The surface morphology of TiO₂ nanoparticles was examined by transmission electron microscope (TEM) Philips, CM 200 operating voltages: 20-200kv with resolution:2.4 at room temperature. TEM is a microscopy technique in which specimen passed through electron beam and transmitted in the form of an image. TEM analysis is used for high resolution imaging compare to light microscopy. The microstructure and dispersion of TiO₂ nanoparticles were characterized with help of field emission scanning electron microscopy (FESEM). FESEM was conducted on Quanta 200 with 15kVvolatege at room temperature. The FESEM analysis was mainly used to characterize the dispersion and scattering of TiO₂ nanoparticles under high magnification of 1000x to 5000x.

3.2 Characterization of composite laminates

3.2.1 Mechanical characterization

The tensile testing was performed by using a 50 kN Instron, 8801 universal testing (UTM) machine. The test sample was prepared as per ASTM (D-638) [31] standard with a dimension of 160×15×3mm and testing has been done at 1mm/min of cross speed with a gauge length of 60 mm. The five specimens of each weight fraction were tested and the average value with an error bar has been reported in the graphical representation. The testing was carried under quasi-static tensile loading condition at room temperature (23°C), and the modulus of elasticity, yield strength, ultimate tensile strength and strain at failure of specimens at maximum load were measured.

The three-point bending test (also known as flexural test) was carried out by using a similar universal testing machine, INSTRON, 8801. The five samples of each weight fractions following ASTM D 790 [32] standards with a dimension of 130×14×3mm were tested and the average value was reported. The three-point bending test was conducted at 1mm/min of cross speed and a span length of 60 mm under quasi-static loading condition at room temperature and the flexural modulus, ultimate flexural strength, maximum deflection before the break of specimens at maximum load were measured.

The micro-hardness testing of the developed composite was done with the help of Vickers micro-hardness tester and following standard ASTM D384-99 [33]. The five different surface positions of each type of sample have been tested and the average value has been presented graphically.

The Charpy impact testing of the developed composite was carried out with the help of a Charpy impact tester made by FIE (Fuel Instruments and Engineers Pvt. Ltd.). The five notched samples of each weight fraction were tested and the average test result was adopted. The ASTM D256-10 [34] standard was followed for preparing the test samples. The dimension of the impact test specimen was 63.5 mm× 12.7 mm×3.2 mm and the depth of the notch was 2.54 mm.

The degree of fiber-matrix interfacial adhesiveness was evaluated by interlaminar shear test (also known as short beam shear test). To perform the short beam test, the previously mentioned UTM machine was used and samples were cut as ASTM D2344 [35] standard at room temperature (23°C). The dimension of the test sample was 37 mm× 9mm× 4.5 mm and the strain rate was maintained at 1mm/min. The five samples of each type of composites were tested and the average result has been reported.

The plane strain fracture toughness of the developed composites was investigated by using the previously mentioned UTM machine and following ASTM standard ASTM D5045-99b. The five specimens of each weight fraction composites with single edge notched were tested with a crosshead speed of 10mm/min and the average data has been reported. The fracture energy of each type of composite was found by using equation (1) mentioned below [36].

$$G_{IC} = \frac{U}{B W_{\phi}} \quad (1)$$

Where G_{IC} = fracture energy, U = corrected energy, B = thickness of the specimen, W = width of the specimen and ϕ = energy calibration factor.

3.2.2 Thermal stability

To understand the thermal stability and degradation behavior with glass transition temperature of the developed composites, Thermo gravimetric analysis (TGA) and Differential thermal analysis (DTA) were performed. The weight loss of the composites under the influence of temperature and maximum degradation temperature were characterized with help of EXSTAR TGA/DTA 6300 Thermo-gravimetric analyzer. The test was carried at a heating rate of 10°C/min from 30°C (room temperature) to 900°C at nitrogen gas environment.

3.2.3 Morphology analysis of fracture surface

The morphology behavior of developed composites laminates at fracture surface is characterized with scanning electron microscope (SEM). SEM is conducted on Sigma 300 with voltage of 5 kV at room temperature and used to magnify the microstructure and morphology behavior of composites in which high electron beam is incident on the surfaces of the specimens. To pass the electron smoothly, developed composite specimens are well coated with gold/silver coating for better electrical conductivity.

4. Result and discussion

4.1 Characterization of TiO₂ nanoparticles

The structural analysis of prepared TiO₂ nanoparticles is characterized by performing XRD, FTIR, TEM, FESEM and TGA. Fig. 3(a) showed the XRD results of prepared TiO₂ nanoparticles. The diffraction peak observed at 25.43°, 37.88°, 47.94°, 53.94°, 54.94°, 62.58° and 69.87° with corresponding to the crystalline plane of (101), (004), (200), (105), (211), (204), (116) and (220) of tetragonal anatase TiO₂ nanoparticles. The result confirmed that, the prepared TiO₂ nanoparticles with sol-gel methods are more adequate and fine to enhance the dispersion

qualities in the matrix materials [37]. From this diffraction pattern, Debye-Scherrer equation is used to calculate the average crystallites (grain) size of TiO₂ nanoparticles [38].

$$d = K \lambda / \beta \cos \theta \quad (1)$$

Where K= 0.89 (Scherrer constant), $\lambda = 0.15406$ nm (the wavelength of the x-ray source), $\beta =$ full width of the diffraction peak (radian) [39]. The average grain size of TiO₂ nanoparticles is found as 8.12 ± 0.21 nm.

The existence of chemical structure and functional group of TiO₂ nanoparticles has been examined with FTIR and measure the intensity of the TiO₂ nanoparticles as shown in Fig. 3(b). Broad intensity was observed at 3420 and 1630 cm⁻¹ due to the stretching vibration of O-H bonding due to presence of moisture [40]. The peak at 515 and 660 cm⁻¹ has been observed due to vibration of Ti-O-Ti bond and minor peaks at 775 and 565 cm⁻¹ has been observed due to Ti-O bonding of Titanium dioxide [41]. Based on the FTIR spectra, the analysis has been made that the TiO₂ nanoparticles can be the better inclusion for preparing composites laminates with better interfacial bonding and strength.

To identify the water content (H₂O) and weight loss in prepared TiO₂ nanoparticles, thermo-gravimetric analysis (TGA) and derivative thermo-gravimetric (DTG) have been performed. Two types of H₂O desorption has been observed. First, H₂O which is absorbed physically made a weak chemical bond with TiO₂ and evaporate at 120°C. Second, chemically absorbed H₂O (also called crystallographic water). These types of H₂O made a comparatively strong bond with TiO₂ and evaporate at 300°C [42]. The weight loss and derivative weight loss of TiO₂ nanoparticles with influence of temperature are shown in Fig. 3(c). From the DTG plot, one can find the points that showed the maximum weight loss rate at 118°C to 120°C and 298°C to 301°C. These two peaks are due to the evaporation of physically and chemically absorbed H₂O. The percentages of physically and chemically absorbed H₂O are 13.8% and 9.88% respectively in prepared TiO₂ nanoparticles.

The shape of the prepared TiO₂ nanoparticles has been observed with the help of a field emission scanning electron microscope (FESEM). Fig, 3(d) showed the microstructure behavior of TiO₂ nanoparticles and analyzed that, the microscopic view of nanoparticles tends to spherical shape. The spherical shapes of nanoparticles have a positive impact on the thermo-mechanical properties of developed nanocomposites and also help to enhance the interfacial bond between reinforcement and matrix materials [43, 44].

The high resolution transmission electron microscope (TEM) was performed to analyze the micrographic view of TiO₂ nanoparticles shown in Fig. 4(a). It was observed that, the nano ceramic particle forms dense aggregates made by crystallites whose sizes were varied from 10 to 20 nano meter. The Fig. 4(b) showed the selected area electron diffraction (SAED) ring pattern with the mark of Miller indices of TiO₂ nanoparticles. The SAED is the crystallographic experimental methods to examine crystal structure pattern of TiO₂ nanoparticles [25]. The ring pattern of TiO₂ nanoparticles confirmed that the granular appearances of ring are well balanced with polycrystalline in nature where sizes of the ingredient crystallites ranged between 10 to 20nm.

4.2 Characterization of composite laminates

4.2.1 Tensile properties

For the design and development of new composite materials, it is very essential to identify the tensile properties. To find out the tensile strength, tensile modulus and elongation at break of the developed composites, tensile testing has been performed. The maximum tensile strength identifies the ability of a material to resist failure, whereas, the ability of the material to resist deformation or plastic deformation characterized by tensile modulus also called as stiffness of the materials and elongation at break identifies the ability of elongation of the material before it goes through fracture and failure [45]. Fig. 5(a) showed the typical engineering stress-strain graph and Fig. 5(b) showed the comparative analysis of tensile properties with influence of different weight fraction of TiO₂ nanoparticles for developed composite laminates.

The addition of TiO₂ nanoparticles up to certain percentages along with glass fiber showed an increment in tensile strength and tensile modulus for developed composites. This may be due to the addition of fine and well dispersion of nanoparticles along with glass fibers that improve the interfacial bonding strength between matrix and fiber. The better interfacial bonding helps to transmit higher load and stress from matrix to reinforced materials [30]. From the results, it observed that, the tensile strength of the developed composites was recorded maximum for GFRP-T1.0 (341.53 MPa) which was 38.56% higher than GFRP-T0.0. Furthermore, the tensile modulus was recorded highest for GFRP-T1.5 (12.89 GPa) which was 52.72% greater than the GFRP-T0.0 composite laminates due to rigidity and stiffness provided by the nanoparticles. Further increments of nanoparticles loading (More than 1.5%) showed, decrement in tensile properties due to agglomeration/clustering effects of TiO₂ nanoparticles with epoxy

matrix while, presence of excessive nanoparticles in developed composites may also obstacles the matrix continuity, which may result the decrements in tensile properties [8, 30].

4.2.2 Flexural properties

To find the flexural properties of developed composite laminates such as flexural strength, flexural modulus and maximum deflection, the three-point bending test has been performed. Fig. 6(a) showed the flexural stress-strain plot and Fig. 6(b) showed the comparative analysis of flexural properties with influence of different weight fraction of TiO₂ nanoparticles for developed composite laminates. As similar to tensile properties, the addition of TiO₂ nanoparticles up to 1% by weight along with glass fiber showed an increment in flexural strength while maximum flexural modulus was recorded for GFRP-T1.5 composite laminates. The maximum flexural strength was observed for the GFRP-T1.0 (421.26 MPa) which was 30.52% greater than GFRP-T0.0 composite laminates. This increment may be due to dispersion of TiO₂ surface area and the adhesive bond between matrix and reinforcement at lower weight of TiO₂ nanoparticles. The maximum flexural modulus was observed for GFRP-T1.5 (16.19 GPa) which was 34.13% greater than GFRP-T0.0 composite laminates. This increment happened due to greater adhesion bond between fiber and matrix with fine dispersion of TiO₂ nanoparticles while orientation of fibers also enhanced and affects the properties of developed composite laminates [46]. But at higher weight fraction of nanoparticles loading created agglomeration/clustering effects which results in a sudden decrement in the mechanical properties [8, 47].

4.2.3 Micro-hardness

The ability to resist localized plastic deformation of any material is known as hardness of the material [48]. Fig. 7 showed the micro hardness of developed composite laminates and recorded highest micro hardness of 36.42 Hv for GFRP-T1.0 composites which was 48.53% higher than GFRP-T0.0 composite laminates. From result, it was notified that, the addition of TiO₂ nanoparticles from 0.5% to 1% increased the micro hardness of developed composites. This increment at lower weight fraction may be attributed to greater interfacial bond of matrix-fiber with dispersion of TiO₂ nanoparticles. But further increment of nanoparticles weight fraction (1.5% to 2%), the continuously decrements in micro hardness values were recorded. This may be due to the agglomeration/clusterization of nanoparticles at higher weight fraction [49].

4.2.4 Impact strength

To identify the shock and energy absorption ability of developed composite laminates, Charpy impact testing has been performed. Fig. 7 represented the impact strength of different developed composite laminates. From the test result, it was observed that, the addition of TiO₂ nanoparticles (up to 1%) along with glass fiber reinforcements showed increment in impact strength while, further increment of TiO₂ nanoparticles weight fraction (more than 1 wt%) observed decrement in impact strength. The better dispersion of nanoparticles in GFRP composite laminates helps to make better interfacial bonding between reinforcing material and matrix materials that showed increment in impact strength at a lower weight fraction of nanoparticles. But excessive addition of nanoparticles (more than 1 wt%) in developed composite laminates make it more brittle and that evidence of brittleness revealed reduction in impact strength values as well as reduction in elongation before break value for tensile and flexural specimen composite laminates [50]. The maximum impact strength was recorded for the GFRP-T1.0 (27.43 KJ/m²) which was 327.10% more than GFRP-T0.0 composite laminates.

4.2.5 Inter-laminar shear strength (ILSS)

One of the major failure criteria of composite laminates are investigated when load is subjected to transverse direction and these failures are analyzed due to inter-laminar shear stress. So, for the design and development process of laminated composites, it is very imported to investigate the ILSS. To examine the ILSS properties of developed composites, short beam shear testing has been performed. Fig. 7 represented the result of the inter-laminar shear test and maximum inter-laminar strength was observed for GFRP-T1.0 (39.44 MPa) which was 25.22% greater than GFRP-T0.0 composite laminates. The addition of TiO₂ nanoparticles with glass fiber reinforcement results better interfacial bond with better dispersion of nanoparticles that improved the ILSS of developed composites. This might be happened due to strong shear strength between laminates up to 1% loading of TiO₂ nanoparticles while, further increments of nanoparticles loading, the decrements was recorded in inter-laminar bonding strength due to high brittleness and agglomeration of the nanoparticles [51].

4.2.6 Fracture properties

The fracture toughness and fracture energy of developed composite laminates have been investigated and represented in Fig. 8. Due to the presence of a high cross-link structure in epoxy polymer, the polymer showed brittle in nature and low fracture toughness value (0.58 MPa-m^{1/2}). The addition of glass fiber with epoxy matrix increased the toughness value up to 3.35 MPa-m^{1/2}.

This increment suggests that the addition of fiber made the material ductile in nature. Now the addition of TiO₂ nanoparticles with glass fiber composite showed further increment in toughness value. The maximum toughness and fracture energy values of 3.92 MPa-m^{1/2} and 5.86 KJ/m² respectively were recorded for GFRP-T1.0 composite which was 17.01% and 40.19% higher than GFRP-T-0.0 composite laminates respectively. But further addition of nanoparticles (more than 1%) showed decrement in toughness and fracture energy values. The increments in fracture properties are analyzed due to crack path deflection, crack pinning and plastic void growth where dispersion of nanoparticles creates Vander wall bond between molecular chains of epoxy and TiO₂ nanoparticles that leads to enhance constrained between nanoparticles/polymer chain and polymer chain itself [52].

At low weight fraction, the nanoparticles interrupted the crack front propagation inside the composite laminates. These nanoparticles create obstacles at the crack front and result in crack front bowing in the middle of nanoparticles. This may lead to creating secondary crack which results in increments of fracture toughness and fracture energy [44]. But for higher weight fraction of TiO₂ nanoparticles, the bonding strength decreased rapidly due to agglomeration of nanoparticles that reduce the homogeneity and effective weight fraction of nanoparticles. Therefore, debonding of the nanoparticles with epoxy polymer chain reduces that creates the cavity and hence, lower fracture toughness and fracture energy are recorded at higher weight fraction of TiO₂ nanoparticles [53].

4.3 Thermal stability of composite laminates

To identify the thermal stability of the developed composites, thermo gravimetric analysis has been performed. Fig. 9 showed the thermo grams of the composite laminates. From obtained results, it was observed that the mass loss of the composites has occurred in three stages. The first stage was recorded at 450° C, second was at 550° C and the third stage was up to 650° C. The degradation of mass at each stage has been tabulated in Table 1. From results, it was analyzed that, the addition of TiO₂ nanoparticles increased the thermal stability of the developed composite laminates. This may be happened due to the catalytic effect of TiO₂ nanoparticles with cross-link nature epoxy polymer where the cross link density results to improve the decomposition temperature of the developed composites [54].

At low TiO₂ nanoparticles loading the thermal degradation were high and low residual mass at 900° C was analyzed for developed composite laminates. The residual of 44.64% for

GFRP-T0.0 composite was recorded that followed by GFRP-T0.5 < GFRP-T1.0 < GFRP-T1.5 < GFRP-T2.0 composite laminates where the residual mass of 48.31%, 53.41%, 55.36% and 57.23% are observed. Therefore, it revealed that, as the weight fraction of TiO₂ nanoparticles increases, the percentage of residual mass also increases due to the ceramic nature of TiO₂ nanoparticles. These nanoparticles created a barrier to oxygen and heat inside the epoxy matrix that results the better thermal stability for developed composite laminates [55].

4.4 Glass transition temperature (T_g)

To investigate the glass transition temperature of developed composite, differential thermal analysis (DTA) has been performed. The DTA measured the temperature difference between the specimens and the reference materials. When the heating began, change in temperature (ΔT) was measured with differential thermocouple until the static state was reached [56]. The measured ΔT is also known as glass transition temperature (T_g) of the materials. Fig. 10(a) showed the DTA curve with T_g value and observed that, as the weight fraction of TiO₂ nanoparticles increased, the T_g of the developed composite was also increased (up to 1.5% of TiO₂ loading), moreover, there is a sudden drop was observed for 2% nanoparticles loading. The comparative T_g value at different weight fraction of TiO₂ nanoparticles are represent in Fig. 10(b). The maximum T_g was observed for GFRP-T1.5 (116°C) which was 20.83% greater than GFRP-T0.0 composite laminates. For thermosetting polymer (epoxy resin) based composites, factors like-molecular weight, cross-link density, fiber/filler- matrix interfacial bonding and polymer tacticity played a vital role to identify the T_g value of the composites [57]. In the present study, addition of nanoparticles (up to lower weight fraction, 1.5%) along with glass fiber increased the interfacial interaction between matrix and reinforcing material where the dispersion of TiO₂ nanoparticles played vital role for increments in thermal stability and glass transition temperature. However, at higher weight fraction of nanoparticles, clusterization and agglomeration effect has been observed, which reduced the cross link density of epoxy matrix and interfacial interaction between matrix and reinforcing material [58]. Therefore, the glass transition temperature for GFRP-T2.0 decreased rapidly.

4.5 Surface morphology and failure mechanism

The surface microstructure and fracture behavior of developed composite laminates are characterized with help of scanning electron microscopy (SEM). The morphology analysis was examined with high magnification of 100x to 5000x and voltage of 5kV at atmospheric

temperature. The dispersion of TiO₂ nanoparticles with epoxy matrix and stacking sequence of glass fiber were analyzed and showed in Fig. 11. The different types of failure such as brittle fracture of epoxy matrix, fiber breakage, debonding, pullout, crushing and delamination of developed composites under tensile and flexural testing were also analyzed on the fracture surfaces. The results examined that, dispersion of TiO₂ nanoparticles in epoxy matrix are properly mixed with less than one % of voids that, results superior interfacial bond with glass fiber reinforcements. The similar observation is also reported with previous studies [59, 60]. The thermo-mechanical performances of developed composites indicated that, incorporation of TiO₂ nanoparticles up to 1.5% TiO₂ nanoparticles are well dispersed with epoxy matrix and create strong interfacial bond with glass fibers. The glass fiber can be transferred their high reinforcing potential to TiO₂ nanoparticles and matrix materials that revealed high mechanical and thermal properties with less fiber breakage and pullout [25]. Whereas, the presence of excess amount of nanoparticles showed visible clusterization and agglomerate with epoxy matrix. The effects of clustering/agglomeration act as the stress concentration sites on the developed composite due to generation of cavitations, void, debonding, and cracks with epoxy matrix and finally, affect the performances of composite materials [25, 38]. Thus, further increments of TiO₂ nanoparticles loading (more than 1.5%) showed improper bonding with epoxy matrix and reduction in their mechanical and thermal performances were recorded for developed composite laminates. The current developed composite laminates are mostly failed due to brittle fracture of epoxy resin with linear behavior of stress-strain curve up to failure and there in no plastic deformation occurred.

5. Conclusion

Nanoparticles are gaining too much attention to embedded with glass fiber reinforced epoxy composites due to their advanced physical and structure properties. In this study, the effective role of TiO₂ nanoparticles has been demonstrated towards fabrication of superior physical, mechanical and thermal properties based on epoxy glass fiber composites. The TiO₂ nanoparticles are prepared with sol-gel methods and mix in epoxy matrix with help of mechanical stirring and sonication to ensure better dispersion. Then, vacuum assisted resin infusion molding techniques (VARIM) are used to fabricate the composite laminates with reinforcements of glass fibers. The effects of different weight fraction (0.5%, 1.0%, 1.5%, and 2.0%) of TiO₂ nanoparticles on glass fiber reinforced epoxy composites are studied and compare

the thermo-mechanical properties with glass fiber reinforced composites (GFRP-T0.0). Based on the characterization, the following conclusions are drawn.

- The physical and chemical characterization of derived TiO₂ nanoparticles with XRD and FTIR confirmed better structure and functional properties to enhance the dispersion qualities with matrix and proved that, the dispersion of nanoparticles to be very efficient to make better interfacial bond with epoxy matrix that reduce the voids and defaults in developed composites.
- The physical and chemical absorption of water contents were analyzed with TGA/DTG and percentage weight loss and weight loss rate were recorded. The physical absorbed water evaporated at 120°C with 13.8% of water contents that made weak bond with TiO₂ while, chemical absorption of water made strong bond with TiO₂ at 300°C with 9.88% of water contents.
- The surface morphology and microstructure of TiO₂ nanoparticles were characterized by TEM and FESEM and showed better crystallographic structure with spherical in shape to improve the thermo-mechanical properties of developed composite laminates.
- The developed composite laminates were used to perform tensile, flexural, impact, inter-laminar shear, micro-hardness, and fracture tests and observed that, the inclusion of TiO₂ nanoparticles up to 1.0% with glass fiber epoxy composites (GFRP-T1.0) enhanced their tensile, flexural, ILSS and impact strength while tensile and flexural modulus were recorded maximum for GFRP-T1.5 due to high stiffness and rigidity provided by TiO₂ nanoparticles.
- The highest micro-hardness (36.42Hv), fracture toughness (3.92MPa-m^{0.5}) and fracture energy (5.86KJ/m²) exhibited for GFRP-T1.0 composite laminates while, further increments of nanoparticles loading agglomerate with epoxy matrix that increased the brittleness of materials and reduction in thermo-mechanical properties were recorded.
- The maximum thermal stability and decomposition of developed composites are characterized where the degradation temperature starts to 450°C and degraded at 650°C. The maximum glass transition temperature (T_g) of 116°C for GFRP-T1.5 composites were recorded due to mobility loss of cross link density of epoxy with interaction of TiO₂ nanoparticles.
- The morphology analysis at fracture surfaces was characterized by SEM and failure of developed composites were analyzed with fiber breakage, fiber pullout, fiber/nanoparticles

debonding, cracking of epoxy matrix with brittle fracture. The excess presence of nanoparticles showed agglomeration and clusterization due to cavitations and debris formed with epoxy matrix.

The effective results and investigations conclude that, the improvement of thermo-mechanical properties due to principal of toughening mechanism may have interesting for various industrial applications.

Declaration of conflicting interest

The authors declare that there are no potential conflicts towards financial interest for investigation, authorship, and/or publication of this study.

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Figures

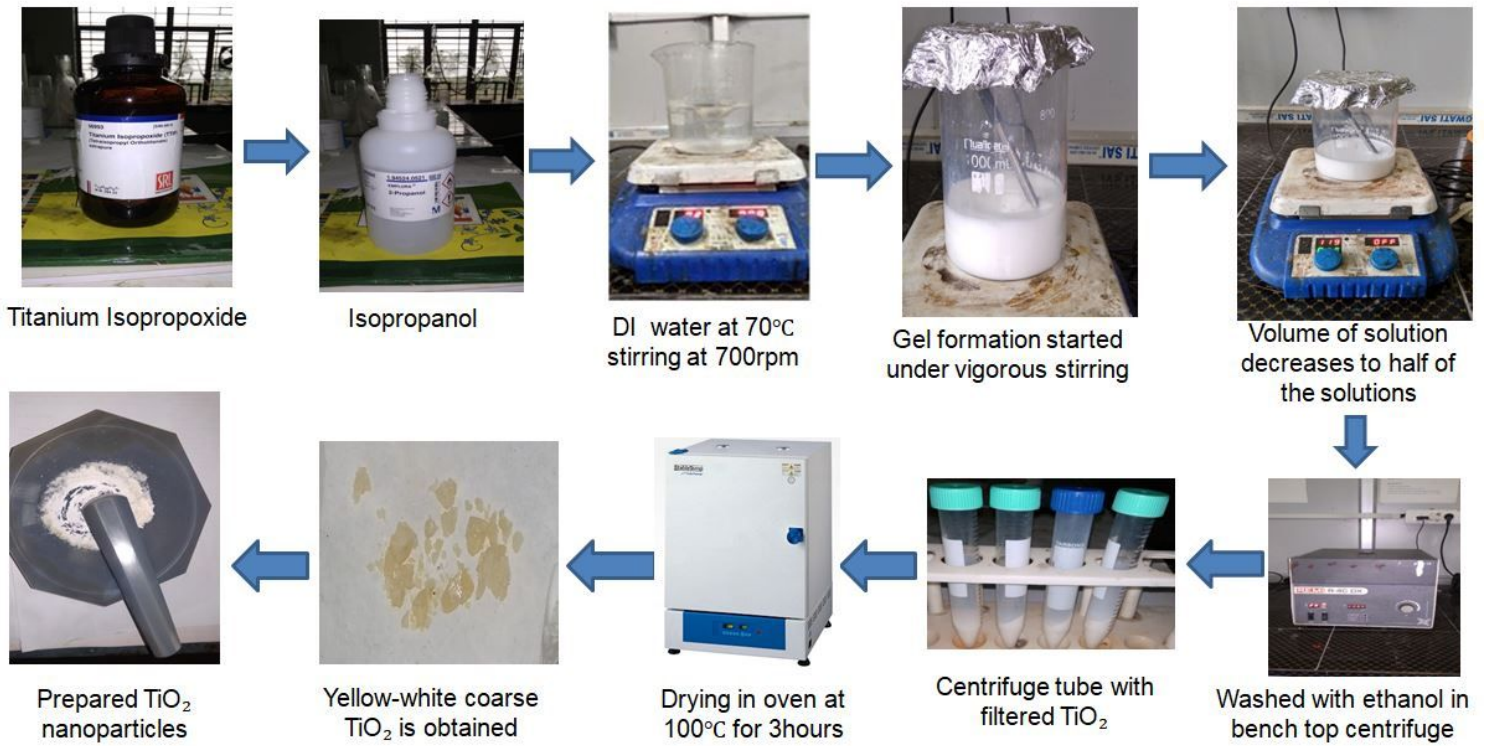


Figure 1

Preparation of TiO₂ nanoparticles with sol-gel method

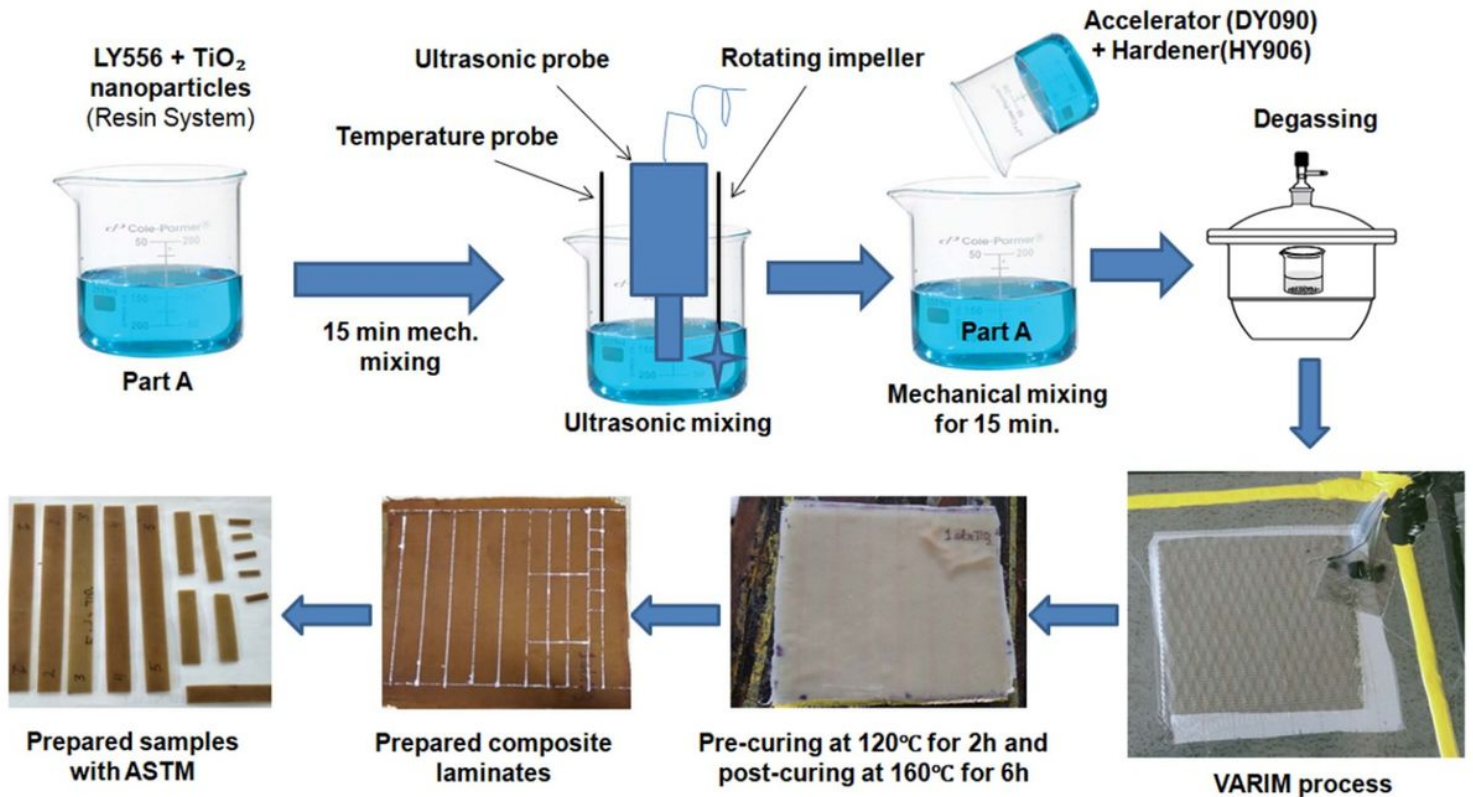
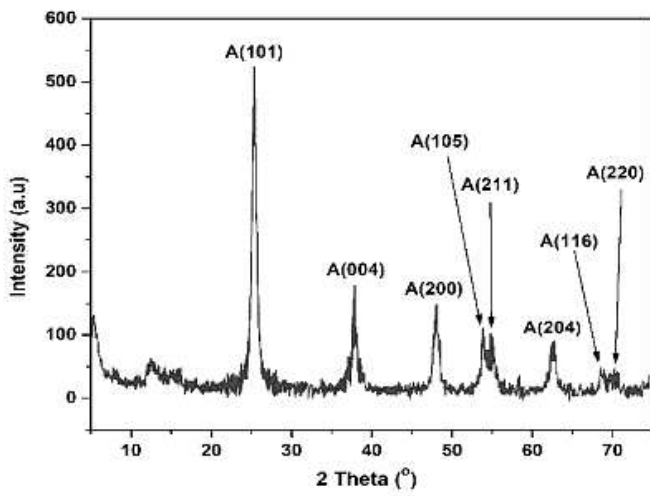
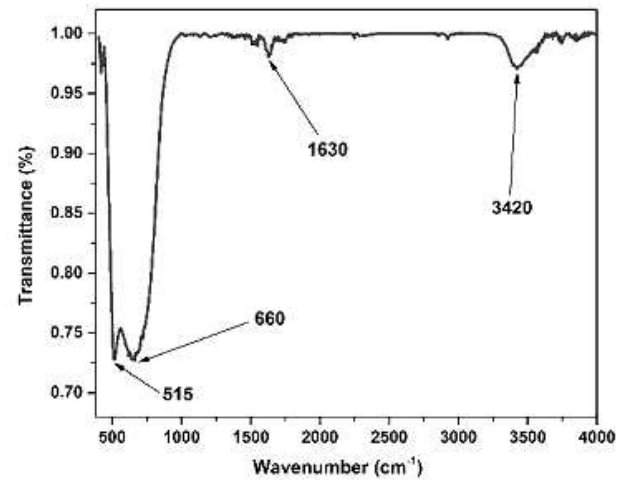


Figure 2

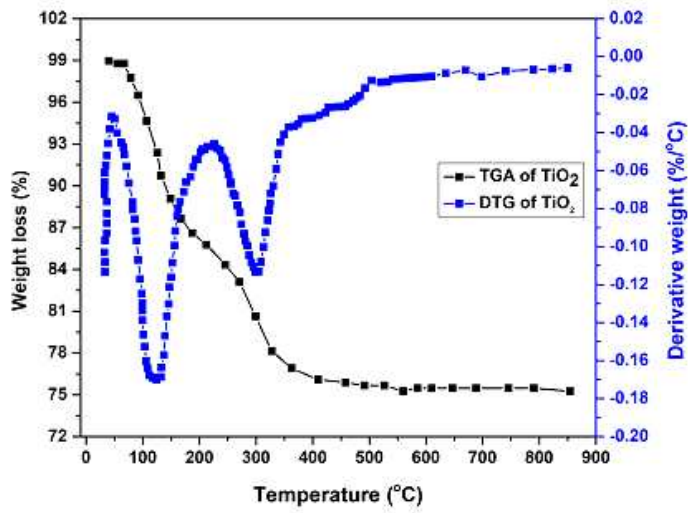
Fabrication processes of GFRP composite laminates



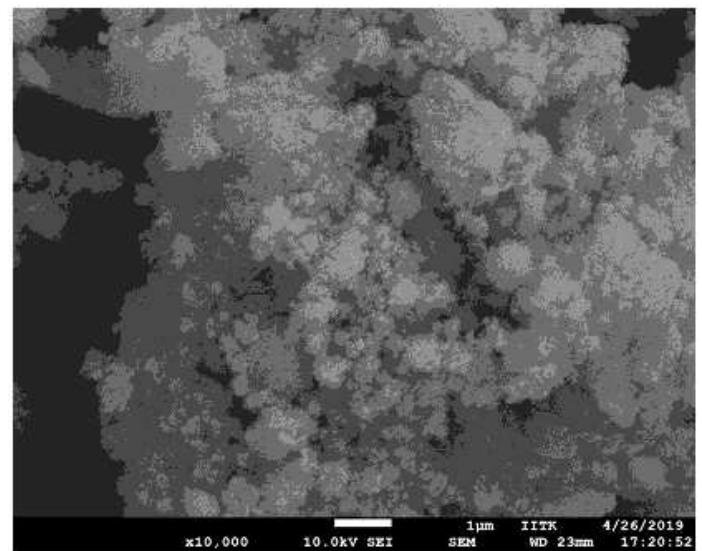
(a)



(b)



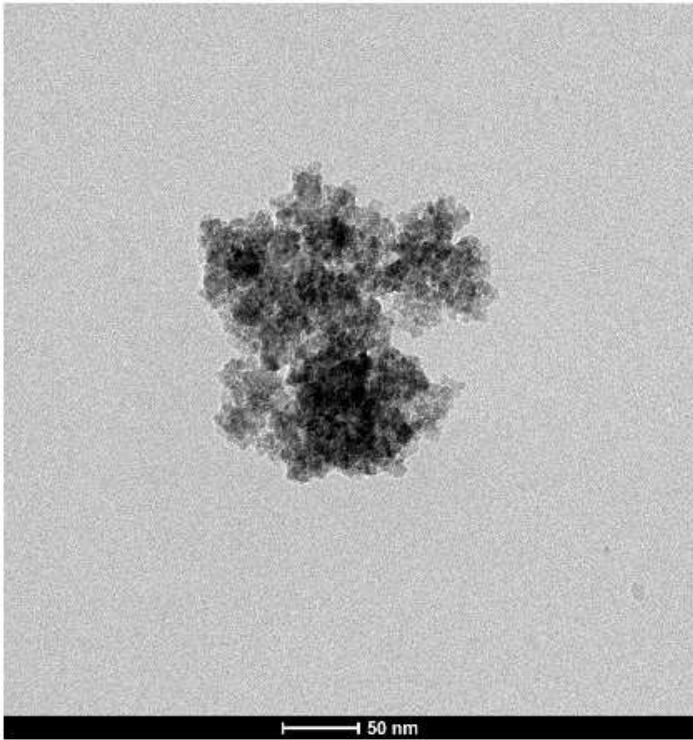
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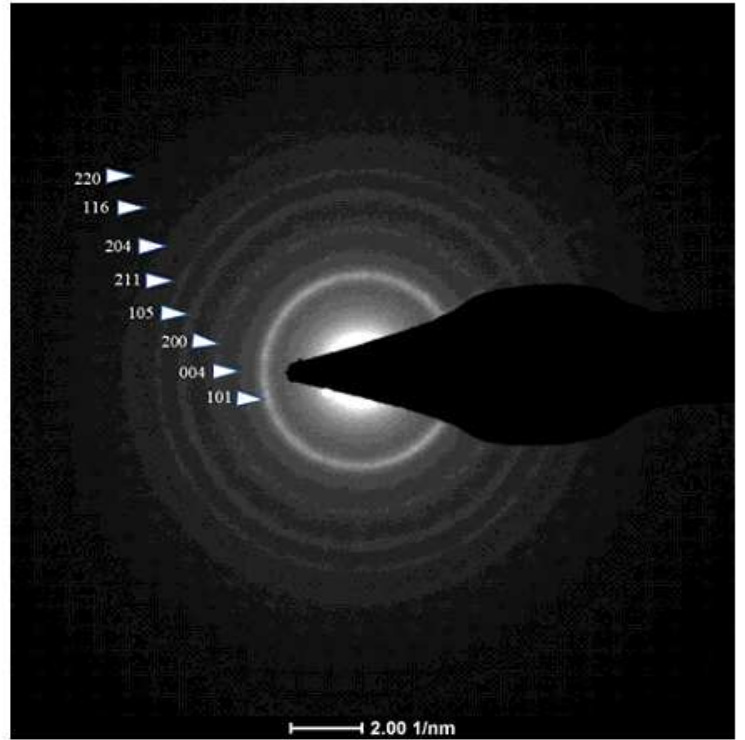
(d)

Figure 3

(a) XRD pattern, (b) FTIR spectra, (c) TGA and DTG, (d) FESEM morphology of TiO₂ nanoparticles



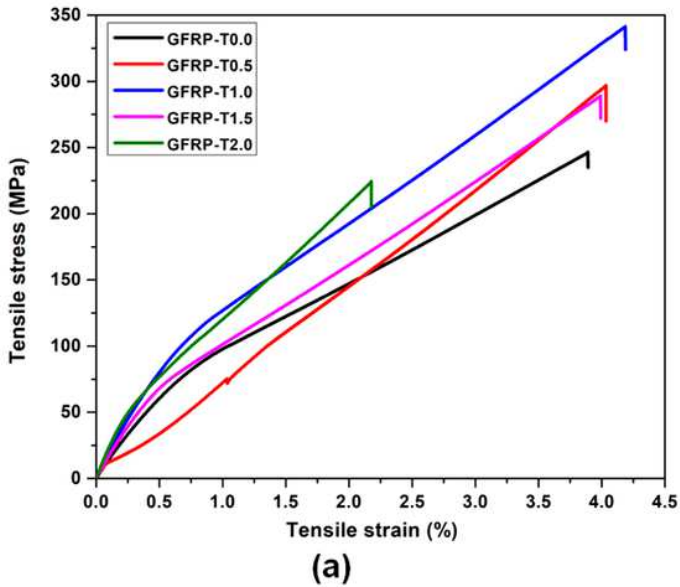
(a)



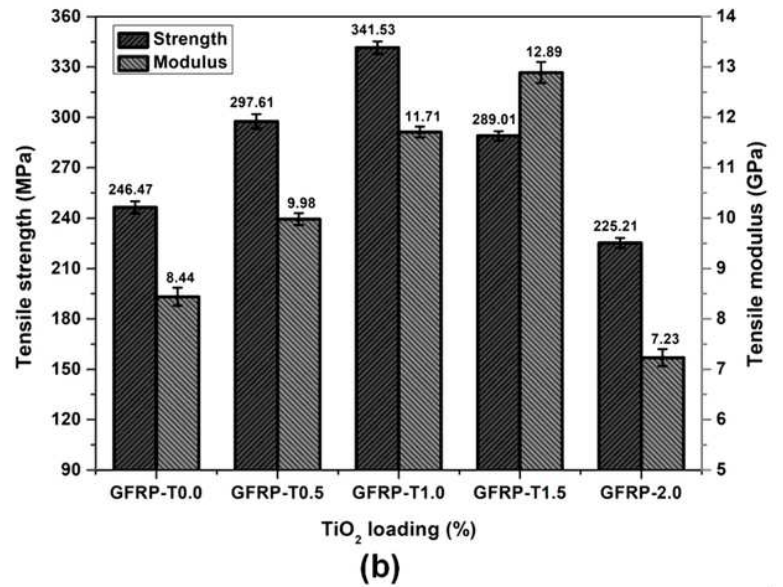
(b)

Figure 4

(a) TEM, and (b) SAED of TiO₂ nanoparticles



(a)



(b)

Figure 5

(a) Tensile stress-strain and (b) Comparative representation of tensile properties for developed composites

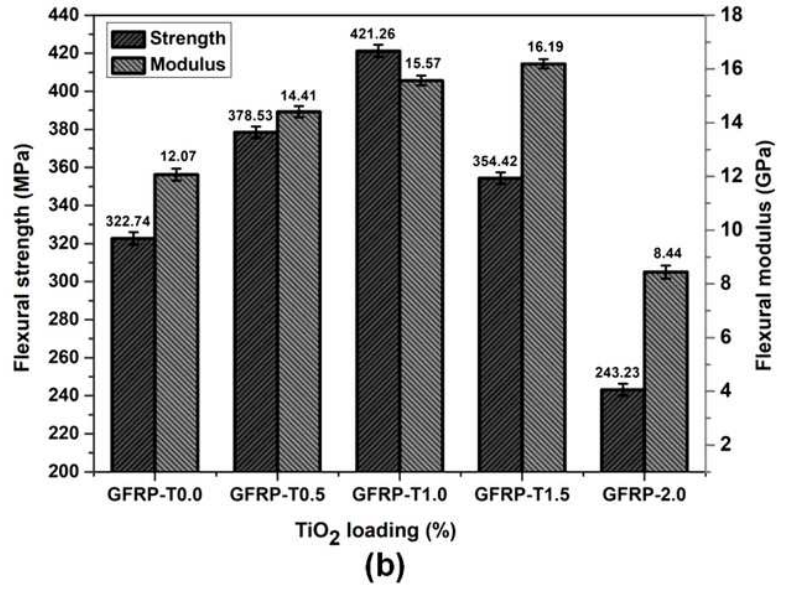
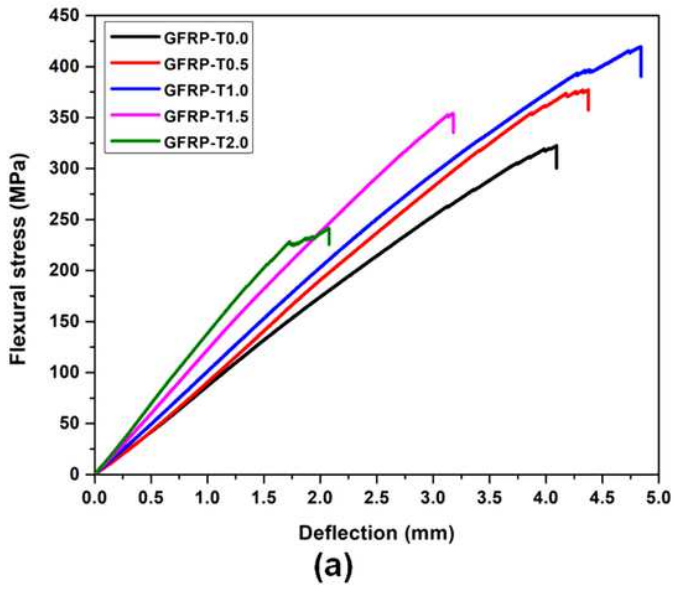


Figure 6

(a) Flexural stress-strain and (b) Comparative representation of flexural properties for developed composites

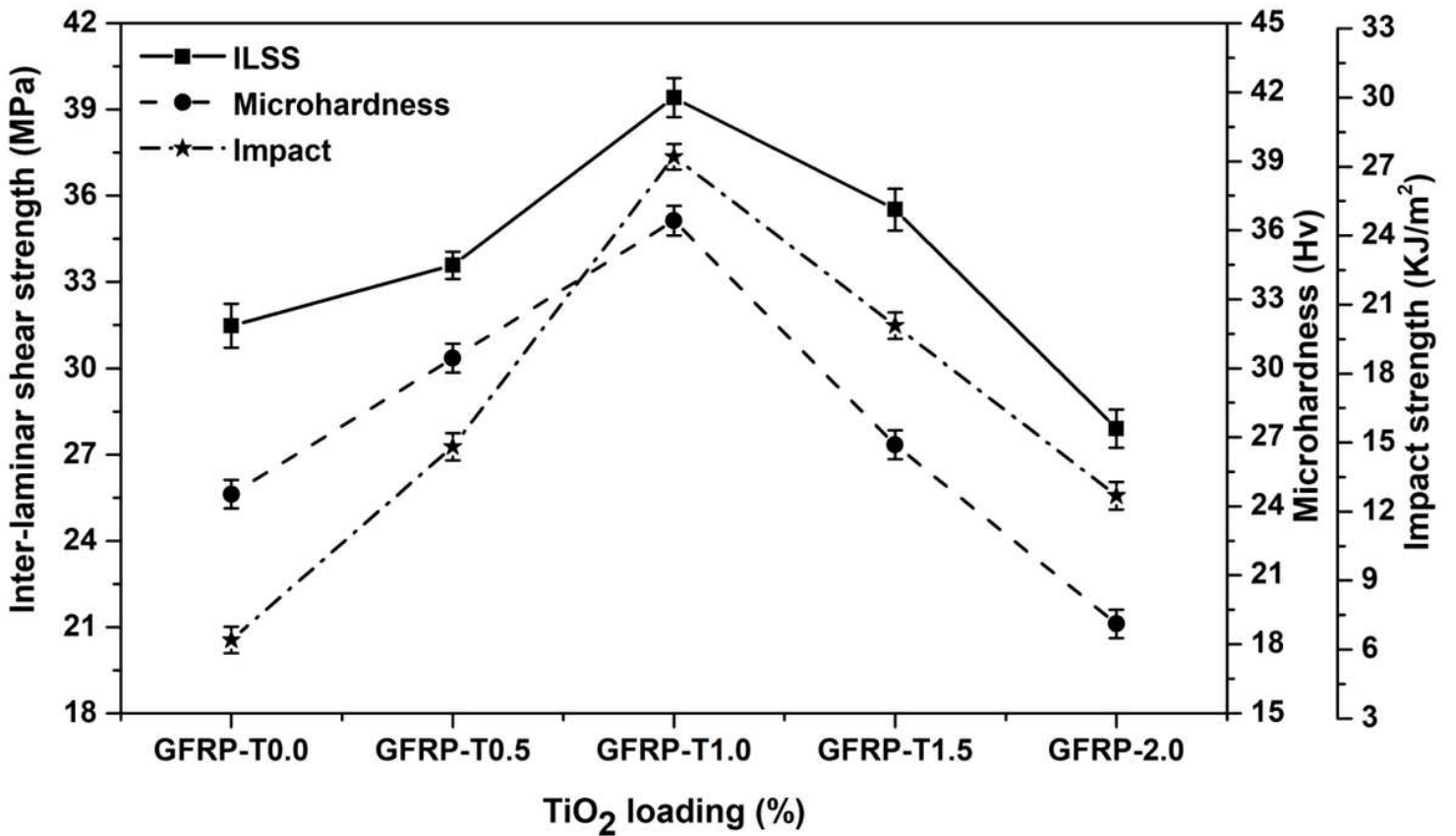


Figure 7

ILSS, micro-hardness and impact strength of developed composite laminates

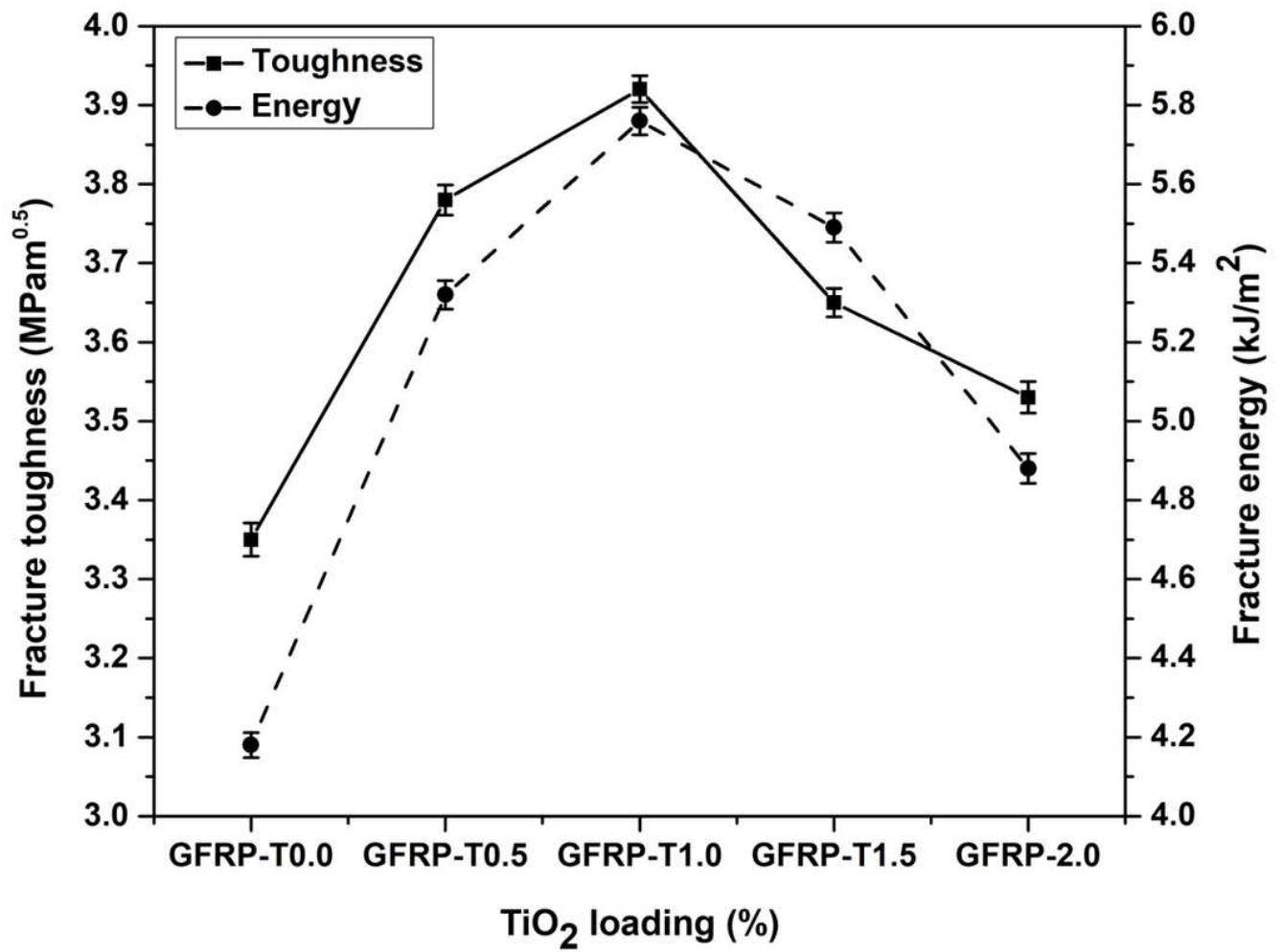


Figure 8

Fracture properties of developed composite laminates

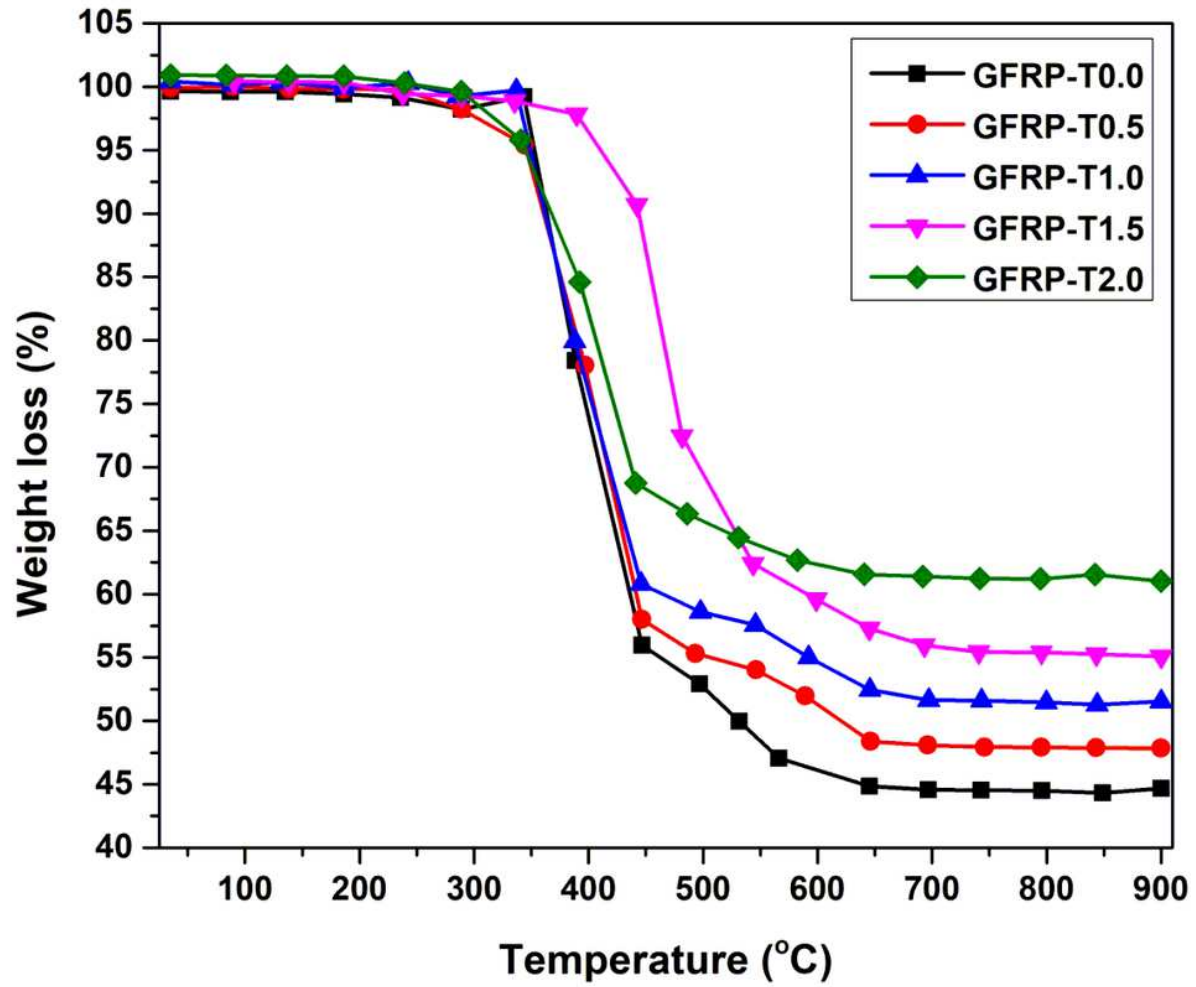


Figure 9

TGA of developed composite laminates

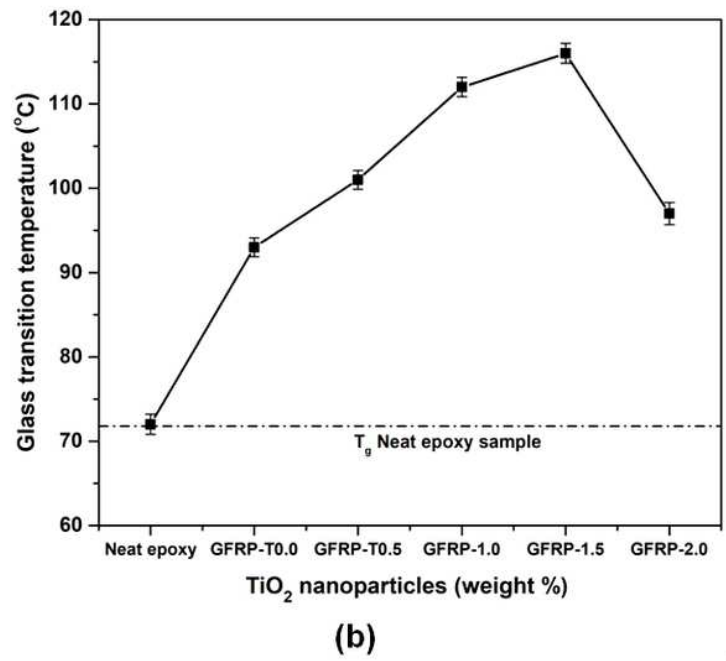
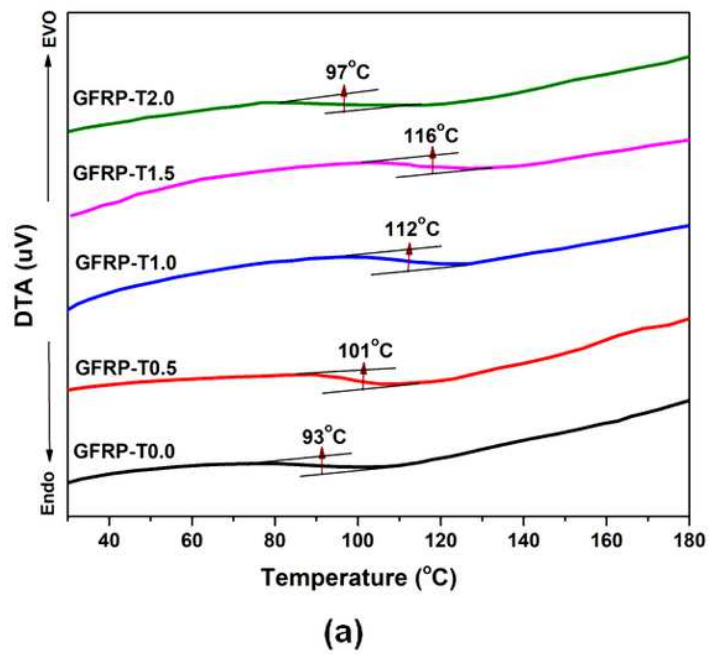


Figure 10

(a) DTA and (b) T_g at different weight fraction of developed composite laminates

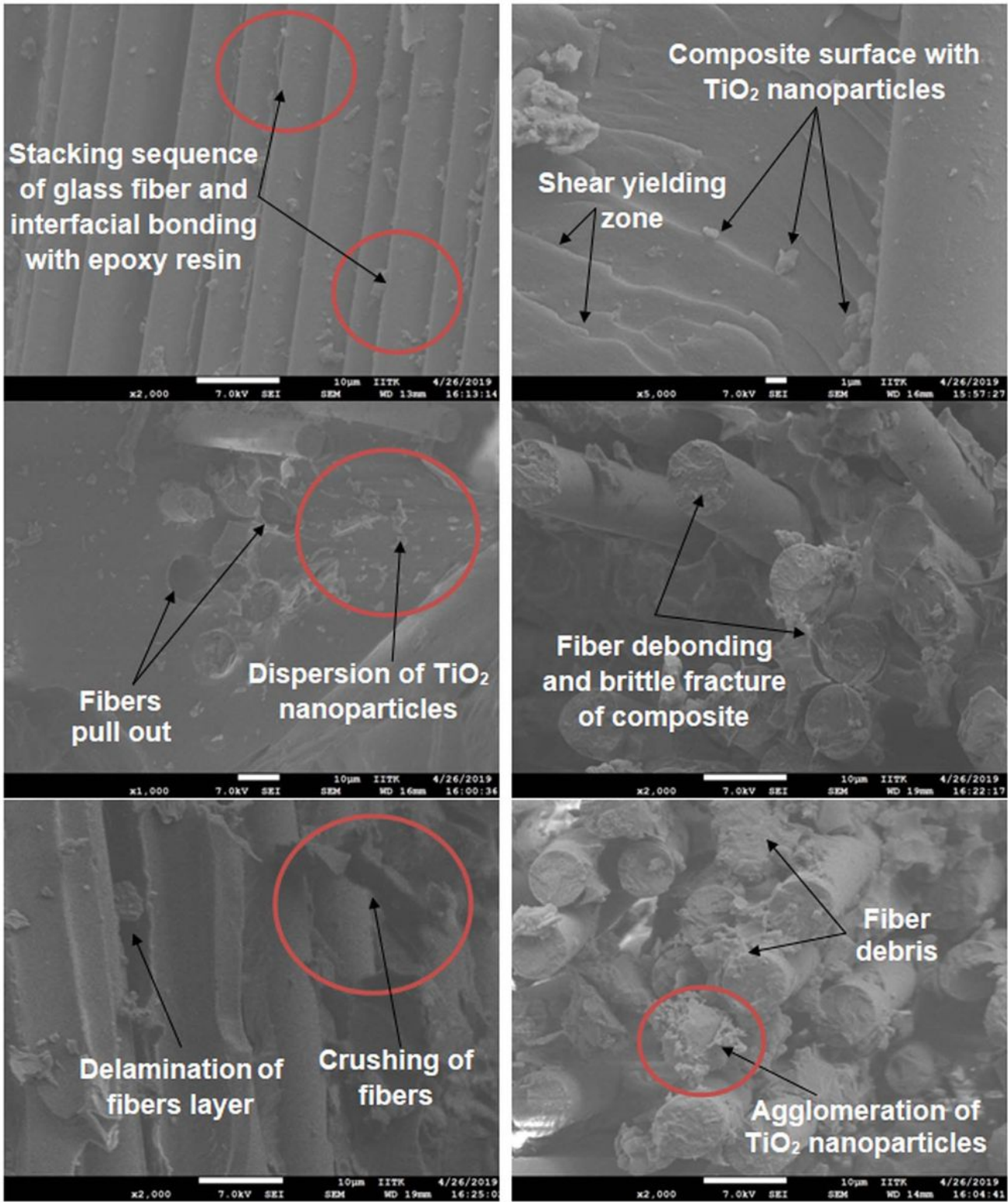


Figure 11

Morphology analyses at fracture surface of developed composite laminates.