

Effects of Different Interfacial Modifiers on the Properties of Digital Printing Waste Paper Fiber/Nano-Crystalline Cellulose/Poly (Lactic Acid) Composites*

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1 **Effects of different interfacial modifiers on the properties of digital**
2 **printing waste paper fiber/nano-crystalline cellulose/poly (lactic acid)**
3 **composites***

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8 **Abstract :** Digital printing waste paper fiber/nano-crystalline cellulose/poly (lactic acid)
9 (DPF/NCC/PLA) composites, modified through γ -methacryloxy propyl trimethoxy silane (KH570),
10 isopropyl tri (dioctylpyrophosphate) titanate (TMC201), sodium hydroxide (NaOH), polyethylene
11 glycol 6000 (PEG6000), and a composite silane coupling agent (KH570/PEG6000), were fabricated
12 by melt blending and injection molding and the effects of different modifiers on the properties of
13 composites were studied. Results showed that mechanical properties of the modified composites
14 generally improved, and the best mechanical properties, including flexural, tensile and impact strength,
15 were achieved PEG6000, KH570/PEG6000, and KH570 modification, respectively. Thermal
16 performance analysis showed improved thermal properties of composites treated by KH570, but the
17 crystallinity of the modified materials was increased. Both water absorption and degradation properties
18 showed a decreasing trend, and water absorption performance was obviously improved after
19 KH570/PEG6000 modification. Under the action of several modifiers, the diffusion coefficient,
20 thermodynamic solubility and permeability of composites were reduced to varying degrees.
21 Furthermore, scanning electron microscopy (SEM) demonstrated that interfacial adhesion and
22 composite compatibility were improved with significantly fewer and smaller pores, as well as a fuzzy
23 boundary among the three phases.

24 **Keywords:** digital printing waste paper fiber; nano-crystalline cellulose; poly (lactic acid); interfacial modifier;

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2 1 Introduction

3 With the continuous development of an industrial society, non-renewable resources such as
4 petroleum resources and plastic products have been consumed in enormous quantities, and their non-
5 degradability has brought severe environmental problems. In recent years, the public's awareness of
6 environmental protection has gradually strengthened, and research on composite materials technology
7 has made progress. Therefore, environmentally friendly composite materials have attracted the
8 attention of researchers worldwide.

9 Poly (lactic acid) (PLA), derived from renewable resources, such as starch, is an ideal
10 biodegradable polymer material^[1], which can be completely decomposed to obtain carbon dioxide and
11 water under specific conditions, such as composting and combustion, to realize an ecological carbon
12 cycle originated from nature and attributed to nature. PLA has preferable mechanical strength and
13 excellent biocompatibility, degradation, and sustainable utilization^[2], but at the same time, it is limited
14 by its chemical structure, resulting in poor toughness, poor hydrophilicity, slow crystallization rate and
15 great brittleness, thus limiting widespread applications. Nano-crystalline cellulose (NCC) is a new type
16 of biodegradable and renewable nanomaterial, which has good thermal stability, high strength, high
17 crystallinity and other properties, and can improve wide variety of composite materials. Existing
18 studies have shown that NCC can strengthen the characteristic defects of PLA materials^[3-6]. Similarly,
19 strengthening resin with plant fiber can also effectively improve its performance defects^[7-10].

20 In the printing industry, digitally printed papers are becoming a larger proportion of the incoming
21 waste paper stream to the recycling industry and increasing amounts of digital prints from offices
22 accounts for a certain proportion of the recovered paper^[11]. For traditional recycling methods, waste
23 paper is mostly used as raw material for recycled paper. However, deinking of digital prints involves
24 a deeper understanding of the ink and its interactions with various types of substrates^[12]. Consequently,
25 even though digital printing waste paper has excellent fiber quality, it is rarely used in the production
26 of recycled paper and provides little added value. Therefore, the efficient utilization of digital printing
27 waste paper attracts much attention. Blending filled digital printing waste paper fiber (DPF) can reduce
28 the production cost of composite materials, and the utilization of DPF is economical and environment-
29 friendly, has ample sources, and easily obtained, which can realize the idea of “turning waste into

1 treasure”^[13]. Therefore, in this paper, we will study the performance of a ternary DPF/NCC/PLA
2 degradable composite material with DPF as the blend filler material, PLA as the matrix and NCC as
3 the reinforcement, which is not only of academic significance, but also of great practical importance.

4 In this study, DPF was used as the blending filler material to prepare NCC/PLA composites, and
5 the modifier was used to react with PLA and cellulose to improve the interfacial conditions and the
6 composite properties^[14-17]. Modifiers, such as γ -methacryloxy propyl trimethoxy silane (KH570),
7 isopropyl tri (dioctylpyrophosphate) titanate (TMC201), sodium hydroxide (NaOH), polyethylene
8 glycol 6000 (PEG6000) and composite silane coupling agent (KH570/PEG6000), were added to the
9 composite material. DPF/NCC/PLA composites were prepared by melt blending and injection molding
10 process, and the influence of different modifiers on DPF/NCC/PLA composite material was studied.
11 The structural properties of the prepared degradable composites were analyzed and characterized by
12 mechanical properties, Fourier transform infrared spectroscopy (FTIR), thermogravimetry (TG),
13 differential scanning calorimetry (DSC), and scanning electron microscopy (SEM).

14 **2 Materials and methods**

15 2.1 Materials

16 DPF was obtained from the local digital printing waste paper of Xi’an, Shaanxi, China. NCC was
17 prepared from DPF using a sodium hydroxide (NaOH) treatment and sulfuric acid (H₂SO₄) solution
18 with a mass fraction of 66% in our laboratory. PLA (3052D) was purchased from Nature Works
19 (America). KH570 (Nanjing Chuangshi Chemical Co. Ltd., Nanjing, China), TMC201 (Dongguan
20 Dinghai Plastic Chemical Co. Ltd., Dongguan, China), NaOH (Zhengzhou Paini Chemical Reagent
21 Factory, Zhengzhou, China), PEG6000 (Tianjin Damao Chemical Reagent Factory, Tianjin, China)
22 and KH570/PEG6000 were used as interfacial modifiers for WP/NCC/PLA composite materials
23 respectively. Ethanol and phosphate buffered saline (PBS) were provided by Tianjin Fuyu Fine
24 Chemical Co. Ltd (Tianjin, China) and Xiamen Haibiao Technology Co. Ltd (Xiamen, China),
25 respectively.

26 2.2 Composite fabrication

27 According to the mass ratio of 90:10, anhydrous ethanol aqueous solution was prepared and
28 utilized as the solvent, and 3%KH570, 3%TMC201, 5%NaOH, 4%PEG6000 and KH570/PEG6000,
29 respectively, were used as the solute and then sprayed onto 3% NCC and 15% DPF. The solvent was

1 evaporated by drying in an oven at 80 °C for 12 h after reacting in a water bath at 60 °C for 5 h. DPF,
2 NCC, and PLA were blended and melted by the twin rotary mixer XH-401C (Dongguan, China) at
3 170 °C, and five composite samples were prepared by the injection molding machine TA-150 (Ningbo,
4 China) at 170 °C and 9 MPa injection pressure. Besides that, the overall processing techniques of NCC
5 and DPF/NCC/PLA modified composite preparations are schematically shown in Figure 1.

6 **3 Testing and characterization**

7 3.1 Mechanical property testing

8 The mechanical properties of standard samples of modified DPF/NCC/PLA composites were tested
9 by universal testing machine XWW-20A (Shanghai, China) and impact testing machine YF-8109
10 (Yangzhou, China). The flexural and tensile performances of the samples were tested according to
11 GB/T 1449-2005 and GB/T 1447-2005 standards with loading speeds of 2 mm/min and 10 mm/min,
12 respectively. Then, the impact performance test was carried out in accordance with the GB/T 1043-
13 1993 standard at a loading gear of 25 J and an impact angle of -150°.

14 3.2 Fourier transform infrared spectroscopy testing

15 Using a tablet press, the NCC and DPF modified samples were separately cold pressed with
16 potassium bromide (KBr) powder to prepare test samples, which were then scanned and analyzed by
17 the 8400S Fourier transform infrared spectrometer (FTIR) (SHIMADZU, Japan), including the test
18 band range of 500~4000 cm⁻¹ with 32 scanning times and a resolution of 4.0.

19 3.3 Thermogravimetric testing

20 The thermogravimetric (TG) curves of the modified composites were drawn using a STA 449F3
21 thermogravimetric analyzer (NETZSCH, Germany) over a temperature range of 30 to 600°C in an
22 argon (Ar) atmosphere with a 40 mL/min flow rate, and at a constant heating rate of 20°C/min.

23 3.4 Differential scanning calorimetry testing

24 Differential scanning calorimetry (DSC) was performed on a DSC 200F3 thermal analysis
25 instrument (NETZSCH, Germany) with two cycles (with a 5 min interval between them) at 200°C to
26 eliminate trace of thermal history^[18]. The first cycle was carried out from 20°C to 200°C under a
27 nitrogen flow of 60 mL/min followed by cooling from 200°C to 20°C. Then, a second heating was
28 performed from 20°C to 200°C. All heating scans were performed at a rate of 10°C/min. The composite
29 crystallinity (χ) was calculated using following equation (1)^[19]:

$$\chi = \frac{\Delta H_m}{\Phi \Delta H_m^0} \times 100 \quad (1)$$

where ΔH_m , Φ and ΔH_m^0 are the melting enthalpy of the composite, mass fraction of PLA in the composite, and the theoretical enthalpy of fully crystallized PLA, which is 93.7 J/g^[17,20], respectively

3.5 Water absorption testing

The water absorption of the modified DPF/NCC/PLA composites were investigated in accordance with GB/T 1034-2008. At room temperature, the samples were immersed in water for seven consecutive weeks. The water absorption percentage (MA) of the composite was calculated using following the equation (2):

$$MA = \frac{M_{TA} - M_{0A}}{M_{0A}} \times 100 \quad (2)$$

where M_{TA} and M_{0A} are the weight after and before water absorption, respectively.

The diffusion coefficient (D) was calculated based on the initial slope of the water absorption curve:

$$D = \pi \left(\frac{h}{4M_b} \right)^2 \left(\frac{M_2 - M_1}{\sqrt{t_2} - \sqrt{t_1}} \right)^2 \quad (3)$$

where, h is the specimen thickness, M_b is the balanced water absorption, M_2 and M_1 are the water absorption at t_2 and t_1 , respectively.

Furthermore, the thermodynamic solubility (S) can be calculated by the following formula:

$$S = \frac{m_b}{m_c} \quad (4)$$

where m_b and m_c are the mass of absorbed water at equilibrium and composite materials mass, respectively.

Permeability was obtained by the product of diffusion coefficient and solubility:

$$P = D \times S \quad (5)$$

3.6 Degradation performance testing

To simulate the soil degradation environment, the samples were placed in phosphate buffer saline (PBS) (pH=7.8) at room temperature for continuous soaking over 90 days. The degradation rate (MD) of the composite was calculated using the following equation (3):

$$MD = \frac{M_{0D} - M_{TD}}{M_{0D}} \times 100 \quad (6)$$

where M_{0D} is the quality before degradation, and M_{TD} is the quality after degradation.

3.7 Scanning electron microscopy testing

Scanning electron microscopy (SEM) of the tensile fracture surfaces of the composites were

1 performed with a SU-8000 scanning electron microscope (HITACHI, Japan). Prior to testing, the
2 samples were coated with a thin layer of conductive material (gold), and then observed under a
3 scanning voltage of 10 kV and a magnification of 1000 times.

4 **4 Results and discussion**

5 3.1 Mechanical property analysis

6 Figure 2 presents the flexural properties of modified DPF/NCC/PLA composites. It can be seen
7 that the flexural strength and modulus of the composites were improved to some extent with the
8 addition of interfacial modifier. After modification with KH570, TMC201, NaOH, PEG6000, and
9 KH570/PEG6000, the flexural strengths reached 98.3, 95.8, 100.4, 101.2 and 99.1 MPa, respectively,
10 among which PEG6000 showed the highest improvement with a 6% increase in flexural strength. The
11 flexural modulus increased by 7%, 13%, 2%, 12%, 1%, respectively, which means that the rigidity of
12 the PLA matrix has increased with the addition of modifier^[21].

13 The tensile strength and elongation at break of the modified DPF/NCC/PLA composites given in
14 Figure 3 show that the tensile strengths of the composites modified with KH570, NaOH, PEG6000
15 and KH570/PEG6000 improved to 65.6, 65.1, 66.8 and 68.6 MPa, respectively, but a decrease is
16 observed with the addition of TMC201. For elongation at break, the addition of NaOH shows no
17 changes before and after modification, and KH570, TMC201, PEG 6000, KH570/PEG6000 increased
18 by 12%, 7%, 5%, 12% respectively. Therefore, with the addition of modifiers, the toughness of the
19 composites has been improved.

20 Figure 4 illustrates the impact properties of modified DPF/NCC/PLA composites. After adding
21 KH570, TMC201, NaOH, PEG6000, and KH570/PEG6000, the impact strengths increased by 19%,
22 3%, 5%, 12%, and 15%, respectively. This shows that when KH570 is added, the impact properties are
23 most improved with impact strengths reaching 12 KJ/m², which is then followed by the addition of
24 KH570/PEG6000.

25 In general, the mechanical properties of the modified composites have been improved to varying
26 degrees. The flexural strength, tensile strength, and impact strength show the most significant
27 improvements with the additions of PEG6000, KH570/PEG6000, and KH570, and show
28 improvements of 6%, 8%, and 19% respectively, when compared with the values prior to modification.
29 The reason is that the -Si-OH produced after the hydrolysis of KH570 is dehydrated to bond with the

1 polar hydroxyl (-OH) on the surface of DPF and NCC, which can then be grafted with PLA through
2 hydrogen bonding to make the interfacial bonding of the composites more compact^[22], as shown in
3 Figure 5 (a). At the same time, the surface of the reinforced materials becomes more wrinkled after
4 modification, and a strong mechanical interlocking with the molecular chain of PLA is generated, so
5 that DPF, NCC and PLA matrix have better compatibility and stronger adhesion. PEG6000 with
6 multiple hydroxyl-terminated groups not only increases the contact between the enhancer and matrix
7 through chemical bonding with the hydroxyl groups on the surface of DPF and NCC^[23], but also
8 conducts an esterification reaction, which is compatible with PLA and able to increase toughness^[24-25].
9 The composite silane coupling agent, KH570/PEG6000, provides a synergistic effect of the above two.

10 3.2 Fourier transform infrared spectroscopy analysis

11 The molecular structure of cellulose provides a basis for its surface modification due to the
12 multitude of hydroxyl groups^[26]. The FTIR spectra of NCC and DPF after interfacial modification are
13 shown in Figures 6 and 7, respectively. Near 3370 cm^{-1} is the stretching vibration peak of -OH, which
14 decreases after modification, and is caused by the reduction of -OH groups after reacting with the
15 modifier. The FTIR spectra also shows peaks at 1720 and 1634 cm^{-1} , which correspond to the
16 characteristic absorptions of C=O and C=C after modification with KH570^[27], further indicating that
17 KH570 was been successfully grafted. However, TMC201 is chemically and directly coupled by its
18 alkoxy group to the hydroxyl group on the cellulose surface, but no corresponding characteristic groups
19 are generated in the coupling reaction, and the connection is still the -O-Ti bond, as shown in Figure 5
20 (b). Therefore, the grafting effect can only be judged by the -OH and -CH₃ groups on the cellulose
21 surface. Thus, in addition to the weakened stretching vibration peak of -OH at 3370 cm^{-1} , the slight
22 stretching vibration peak of -CH₃ and -CH₂ at 2961 cm^{-1} also indicate that TMC201 has been
23 successfully grafted to the cellulose surface^[28].

24 3.3 Thermogravimetric analysis

25 Thermogravimetric analysis of modified DPF/NCC/PLA composites are presented in Figures 8
26 and 9, and all of the corresponding data are given in Table 1. The TG curves of the modified composites
27 show that the onset decomposition temperatures of the composites modified by TMC201, NaOH,
28 PEG6000, and KH570/PEG6000 are reduced to different degrees compared with that before
29 modification, as well as the temperature of maximum decomposition rate, indicating that these

1 modifiers have adverse effects on the thermal stability of the composites, especially the NaOH
2 treatment. The maximum decomposition temperatures of the composites modified by KH570 are
3 increased because the long chain alkanes of KH570 were successfully introduced onto the surface of
4 DPF and NCC, and a coating layer was formed. The thermal stabilities of the composites were
5 improved by absorbing some energy to destroy the interaction force and then pyrolyzing itself. The
6 onset temperature of the composites modified by KH570 remains unchanged, while the maximum
7 decomposition temperature increases, which is due to the fact that the KH570 long-chain alkanes were
8 introduced onto the surface of DPF and NCC successfully, forming a coating layer and providing better
9 cross-linking. In the thermal decomposition process, it is necessary to absorb a certain amount of
10 energy to destroy the force between the two, and then pyrolysis takes place, so the thermal stability of
11 the modified composites is improved^[29]. Meanwhile, Table 1 shows that the residual amount of the
12 modified composites increases, which is likely related to the residual organic modifier added to the
13 composite material.

14 3.4 Differential scanning calorimetry analysis

15 Figure 10 presents heat flow endo up vs. temperature of the modified DPF/NCC/PLA composites.
16 Table 2 shows the thermal parameters of DSC. The secondary heating curves show that the glass
17 transition temperature (T_g) of the modified composites slightly decreased, while the T_g of the KH570
18 modified composites have a slight increase. After modification, the crystallization temperature (T_c) of
19 composites is offset to the left, i.e. the T_c decreased, especially with the addition PEG6000, which
20 shows a decreased from 122.9 to 109.3°C after modification, indicating that the modifiers can promote
21 the movement of the PLA chain segment^[30], and provide the ability to block the crystallization process
22 of the composites. This may be attributed to more restricted molecular movements of PLA chains due
23 to the formation of hydrogen bonds^[31]. As the DSC heating curves demonstrate, bimodal endothermic
24 melting peaks could be seen for all samples except the material modified with KH570. Among which,
25 the lower one corresponds to crystalline melting with low perfection and thinner lamella and the higher
26 peak is related to the melting of more perfect recrystallized crystals^[32]. It can be seen from Table 2 that
27 the crystallinity of the modified composites is improved, because the addition of modifier promotes
28 the movement of the PLA molecular chain segment, which enables the molecular chain in the crystal
29 region of the composite to regularly arrange^[17].

1 3.5 Water absorption analysis

2 The cellulose molecules contain hydroxyl groups that attract water molecules through hydrogen
3 bonding, causing the fiber to expand and absorb water^[33-35]. Interfacial bonding between fiber and
4 resin is also an important factor affecting the water absorption properties of composites. Figure 11
5 illustrates the effect of modifier on the water absorption behavior of DPF/NCC/PLA composites. It
6 shows that water absorption increases with increasing immersion time for all modified samples. The
7 water uptake rate is linear and very rapid in the beginning of the exposure, but then slows and reaches
8 a saturation level. In the initial stage, the unmodified DPF/NCC/PLA showed a faster rate of water
9 absorption, which was due to the hydrophilic properties of cellulose fibers, and the abundant hydroxyl
10 groups that attract water molecules through hydrogen bonds, which in turn lead to fiber swelling and
11 water absorption. At the same time, the fiber has a hollow cavity in the center, allowing a large amount
12 of water to be absorbed by the capillary effect^[36]. In addition, the small amount of water absorption of
13 the PLA matrix and the gap water absorption existing in the combination of PLA, NCC and DPF are
14 also important reasons for the high water absorption rate. After modification, the water absorption of
15 the composites decreased to some extent. Among them, KH570/PEG6000 decreased the most,
16 followed by PEG6000, NaOH, KH570, and TMC201, and the reduction reached 23%, 21%, 20%, 19%
17 and 18% after seven weeks, respectively. This can be attributed to the reaction of modifiers with -OH
18 on the cellulose surface, which reduced the amount of -OH, resulting in decreased of polarity and
19 hydrophilicity. Furthermore, the interfacial affinity between reinforcement and matrix was improved
20 with a more compact internal structure which reduced the water absorption rate of the gap after
21 modification, so that the water absorption rate of the composites was improved. Due to the synergistic
22 effect of the two modifiers, the water absorption properties of the composite modified by
23 KH570/PEG6000 significantly improved.

24 The parameters related to water absorption of composites, including diffusion coefficient,
25 thermodynamic solubility and permeability, are shown in Table 3. The water absorption behavior of
26 composite materials (water diffusion behavior in composite materials) is usually represented by
27 diffusion mechanism, which basically conforms to Fick water absorption model^[37]. Compared with
28 the unmodified DPF/NCC/PLA, the diffusion coefficients of the composites with the addition of the
29 modifiers decreased, that is, the diffusion velocity of water molecules in the micro-gaps between the

1 polymer chains decreased^[38]. It may be that the interface binding between cellulose molecular chains
2 and matrix was improved under the action of modifiers. Through the reaction between the modifiers
3 and the hydroxyl group on the surface of the cellulose, the hydrophilicity of the cellulose molecular
4 chains was weakened, resulting in the decrease of the gap between the cellulose molecular chains and
5 the hydrophobic PLA molecular chains, and therefore the diffusion coefficient reduced. In addition,
6 the parameters of thermodynamic solubility and permeability can also help to further clarify the kinetic
7 behavior of water absorption. The solubility is related to the absorption of the penetrant to a certain
8 extent^[37]. With the addition of the modifiers, the diffusion channel of water was inhibited, and the
9 penetrant was prevented from penetrating the DPF/NCC/PLA composites, resulting in the decrease of
10 solubility and permeability values.

11 3.6 Degradation performance analysis

12 Due to acid degradation, unstable chemical bonds (such as ester bonds) in PLA molecules freely
13 hydrolyze and lead to chain scission, resulting in reduced relative molecular weight of PLA^[7]. After
14 degradation in PBS solution for 90 days, the DPF/NCC/PLA composite suffered from corrosion with
15 the surface roughened and cracked, which resulted in destruction of the basic structure and composite
16 morphology. The degradation performance of the modified DPF/NCC/PLA composites immersed in
17 PBS for 90 days is shown in Figure 12. With increased water absorption, the degradation rate of the
18 DPF/NCC/PLA composites reduced, especially with the addition KH570/PEG6000, which was then
19 followed by PEG6000, NaOH, KH570, and TMC201, showing decreases from 5% to 3%, 3%, 4%, 4%
20 and 4%, respectively. The lower degradation rate obtained for modified DPF/NCC/PLA composites is
21 attributed to the modifier introduction which increases interfacial compactness, affects the contact area
22 with phosphoric acid and, consequently, lowers the acid degradation rate^[39].

23 3.7 Scanning electron microscopy analysis

24 SEM images of the samples, with different modifiers added during the preparation process of
25 composites, are reported in Figure 13. After modification, a change in the section morphology of the
26 composites is observed. A clear surface defect with fibers exposed and several tiny holes on the surface
27 of the DPF/NCC/PLA composites are observed in Figure 12a, which shows that the bonding with the
28 matrix is not tight enough, and the adhesion and interfacial compatibility between the two are poor.
29 Figures 12b, c, d, e, and f demonstrate that DPF in the modified composite is firmly embedded in the

1 PLA matrix with significantly fewer and smaller pores as well as a fuzzy boundary between the fibers
2 and the matrix, which leads to improved interfacial adhesion and composite compatibility. Therefore,
3 the mechanical properties of DPF/NCC/PLA composites after modification have comprehensively
4 improved.

5 **5 Conclusion**

6 In this paper, DPF/NCC/PLA composites modified with different modifiers were prepared by melt
7 blending and injection molding. The mechanical, thermal, water absorption, and degradation properties
8 of the composites, as well as micro-morphology, were investigated. FTIR analysis showed that the
9 modifiers had been successfully applied to the composites. With the addition of modifiers, the
10 interfacial bonding force between reinforced phase fiber and PLA was improved, so that the
11 mechanical properties of the composites were further enhanced. Thermal property analysis showed
12 that the thermal properties of KH570 were slightly improved after modification, while all others had
13 adverse effects on the composites, but their crystallinity was improved. After modification, both water
14 absorption and degradation properties were reduced, and the water absorption performance was
15 improved most obviously after KH570/PEG6000 treatment, which was attributed to the improved
16 interfacial bonding of the modified composites. Furthermore, SEM demonstrated that interfacial
17 adhesion and composite compatibility were improved with significantly fewer and smaller pores as
18 well as fuzzy boundary among the three phases.

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27 **Table captions**

28 **Table 1.** Thermogravimetric analysis parameters of modified DPF/NCC/PLA composites.

29 **Table 2.** The thermal parameters of DSC of modified DPF/NCC/PLA composites.

- 1 **Table 3.** Diffusion coefficient, solubility, and permeability of modified DPF/NCC/PLA composites.
- 2 **Figure captions**
- 3 **Figure 1.** Schematic diagram of NCC and DPF/NCC/PLA modified composites preparation.
- 4 **Figure 2.** Flexural properties of modified DPF/NCC/PLA composites.
- 5 **Figure 3.** Tensile properties of modified DPF/NCC/PLA composites.
- 6 **Figure 4.** Impact properties of modified DPF/NCC/PLA composites.
- 7 **Figure 5.** Schematic diagram of KH570 modification mechanism.
- 8 **Figure 6.** FTIR spectra of interface modified NCC.
- 9 **Figure 7.** FTIR spectra of interface modified DPF.
- 10 **Figure 8.** TG curves of modified DPF/NCC/PLA composites.
- 11 **Figure 9.** DTG curves of modified DPF/NCC/PLA composites.
- 12 **Figure 10.** DSC curves of modified DPF/NCC/PLA composites.
- 13 **Figure 11.** Water absorption property of modified DPF/NCC/PLA composites: (a) within seven days; (b) within seven weeks.
- 14 **Figure 12.** Degradation properties of modified DPF/NCC/PLA composites.
- 15 **Figure 13.** SEM micrographs of modified DPF/NCC/PLA composites ($\times 1000$): (a) DPF/NCC/PLA, (b) KH570 DPF/NCC/PLA, (c)
- 16 TMC201 DPF/NCC/PLA, (d) NaOH DPF/NCC/PLA, (e) PEG6000 DPF/NCC/PLA, (f) KH570/PEG6000
- 17 DPF/NCC/PLA

Table 1. Thermogravimetric analysis parameters of modified DPF/NCC/PLA composites.

Samples	$T_{\text{onset}} / ^\circ\text{C}$	$T_{50} / ^\circ\text{C}$	$T_{\text{max}} / ^\circ\text{C}$	$R_{600} / \%$
DPF/NCC/PLA	291	345	346	4.6

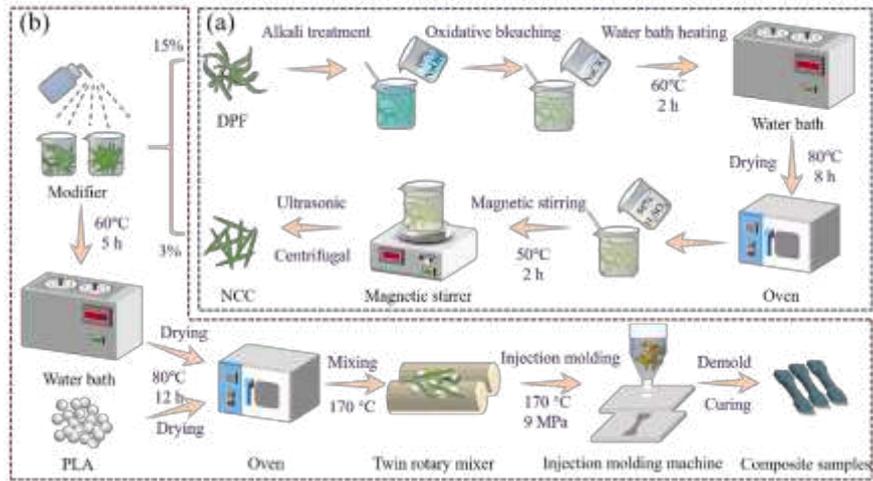
KH570	291	344	351	5.4
TMC201	281	369	364	7.8
NaOH	256	322	328	4.6
PEG6000	276	334	337	7.8
KH570/PEG6000	279	338	340	8.8

Table 2. The thermal parameters of DSC of modified DPF/NCC/PLA composites.

Samples	$T_g / ^\circ\text{C}$	$T_c / ^\circ\text{C}$	$\Delta H_c / (\text{J/g})$	$\Delta H_m / (\text{J/g})$	$T_{m1} / ^\circ\text{C}$	$T_{m2} / ^\circ\text{C}$	$\chi_c / \%$
DPF/NCC/PLA	62.4	122.9	-19.78	17.18	151.5		22.4
KH570	62.6	112.0	-21.81	18.25	149		23.8
TMC201	62.0	110.6	-21.97	19.57	148.4	154.6	25.5
NaOH	61.4	113.3	-23.28	25.42	147.2	153.9	33.1
PEG6000	61.5	109.3	-22.01	19.11	148.1	154.6	24.9
KH570/PEG6000	61.9	109.7	-18.46	20.15	148.9	154.1	26.2

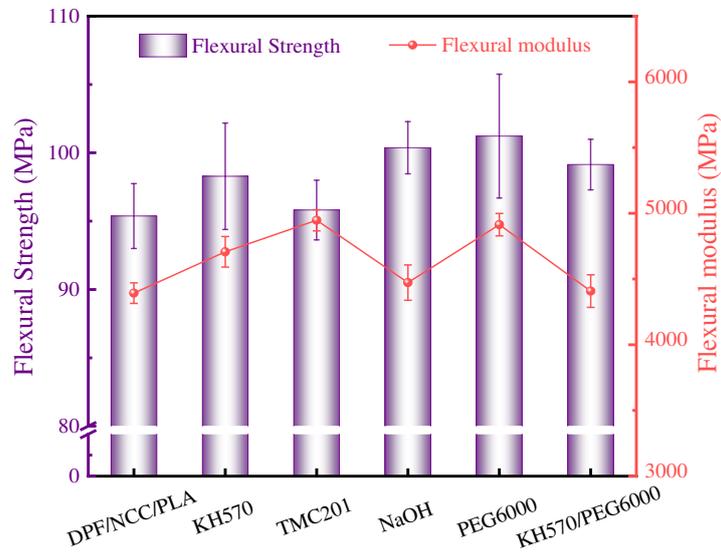
Table 3. Diffusion coefficient, solubility, and permeability of modified DPF/NCC/PLA composites.

Samples	$D \times 10^{-13} (\text{m}^2/\text{s})$	$S (\text{g/g})$	$P \times 10^{-14} (\text{m}^2/\text{s})$
DPF/NCC/PLA	7.7164	0.0280	2.1606
KH570	7.1979	0.0228	1.6411
TMC201	6.6111	0.0230	1.5206
NaOH	6.1470	0.0224	1.3769
PEG6000	6.0231	0.0220	1.3251
KH570/PEG6000	5.9514	0.0215	1.2796



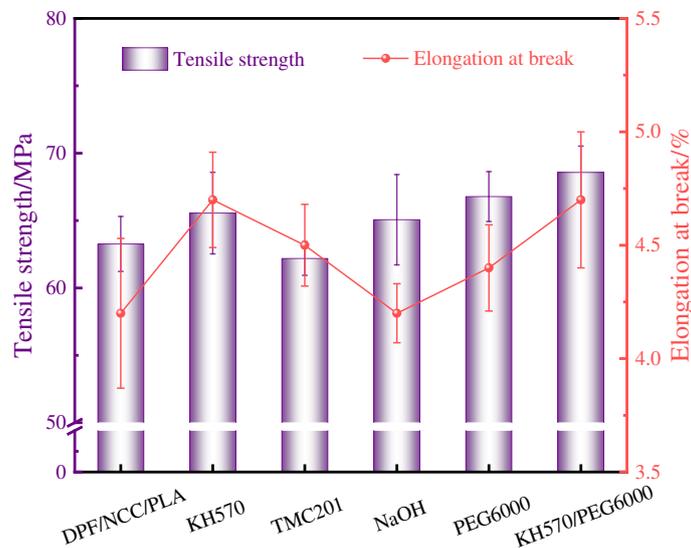
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Figure 1. Schematic diagram of NCC and DPF/NCC/PLA modified composites preparation: (a) process of preparation of NCC; (b) process of preparation of DPF/NCC/PLA modified composites.



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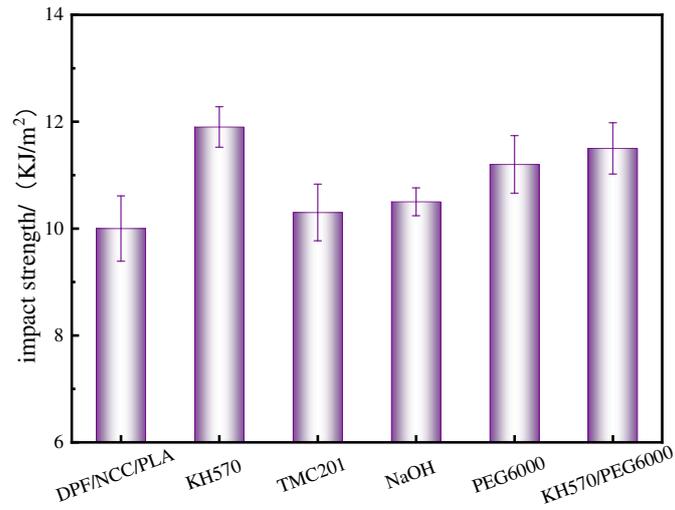
Figure 2. Flexural properties of modified DPF/NCC/PLA composites.



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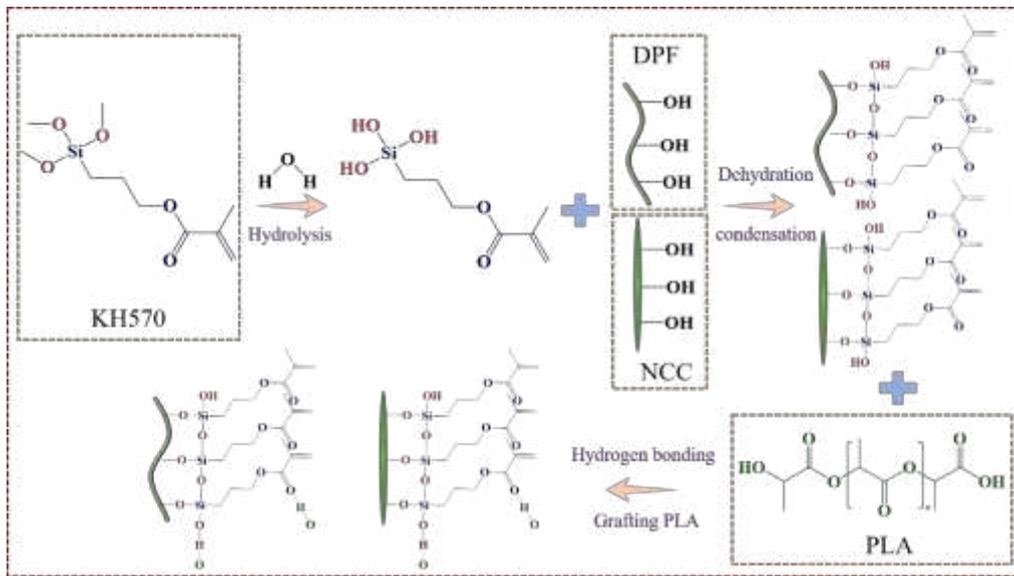
Figure 3. Tensile properties of modified DPF/NCC/PLA composites.



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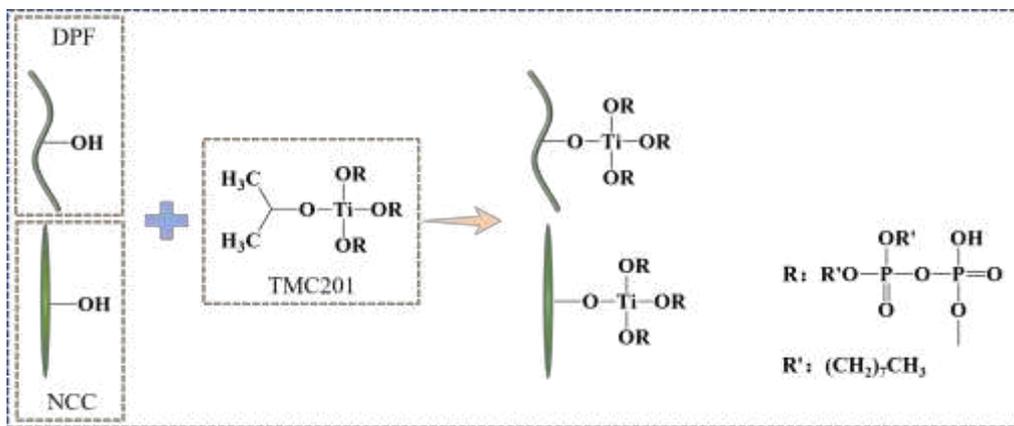
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Figure 4. Impact properties of modified DPF/NCC/PLA composites.



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Figure 5. Schematic diagram of modification mechanism: (a) KH570; (b) TMC201.

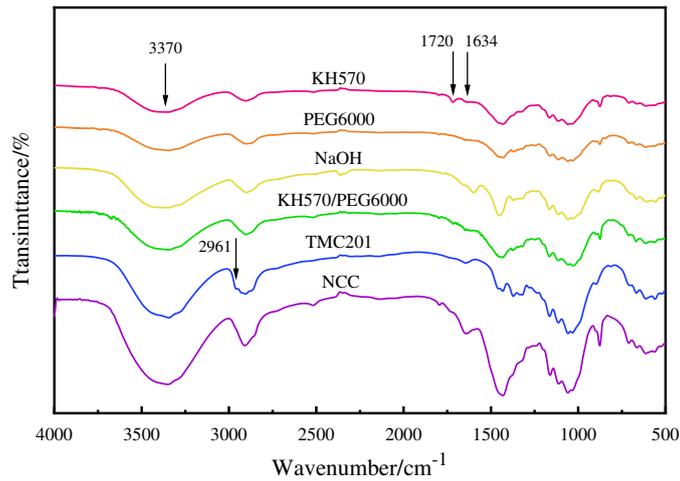


Figure 6. FTIR spectra of interface modified NCC.

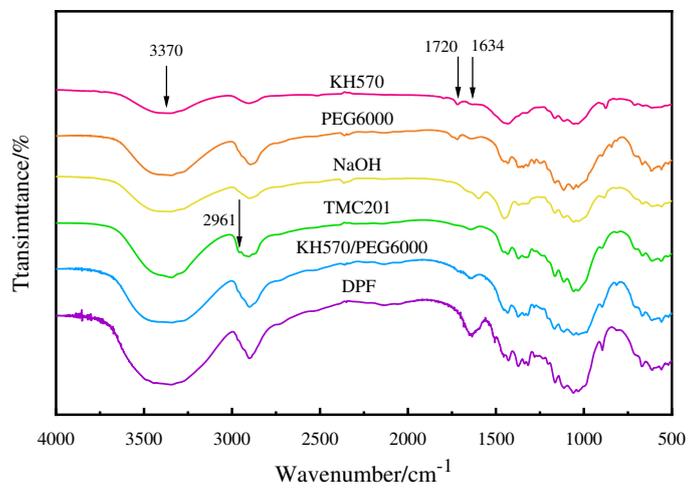


Figure 7. FTIR spectra of interface modified DPF.

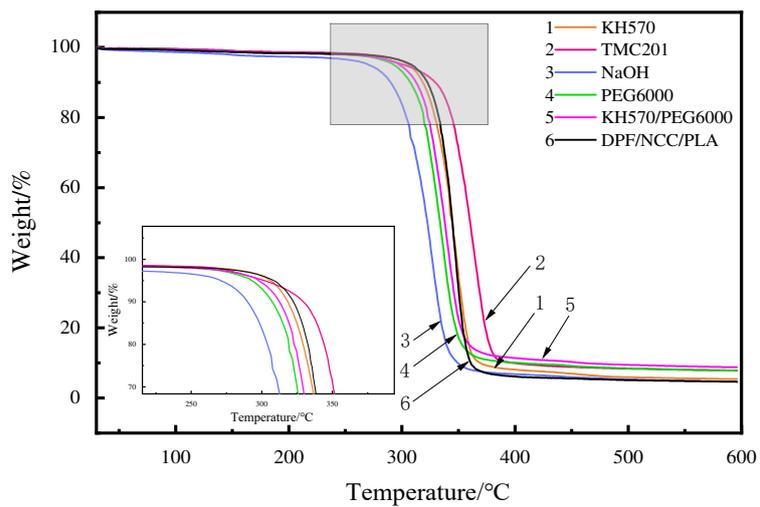
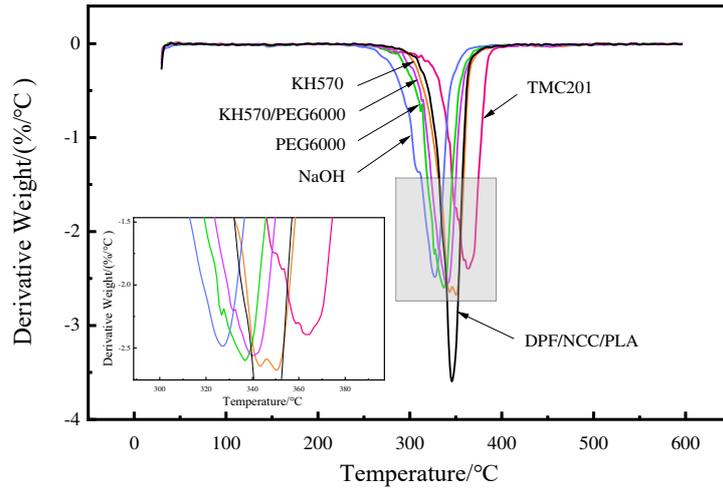


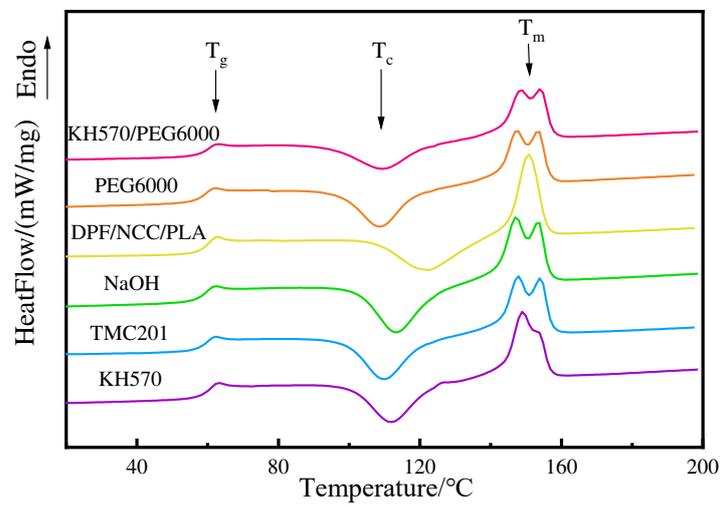
Figure 8. TG curves of modified DPF/NCC/PLA composites.



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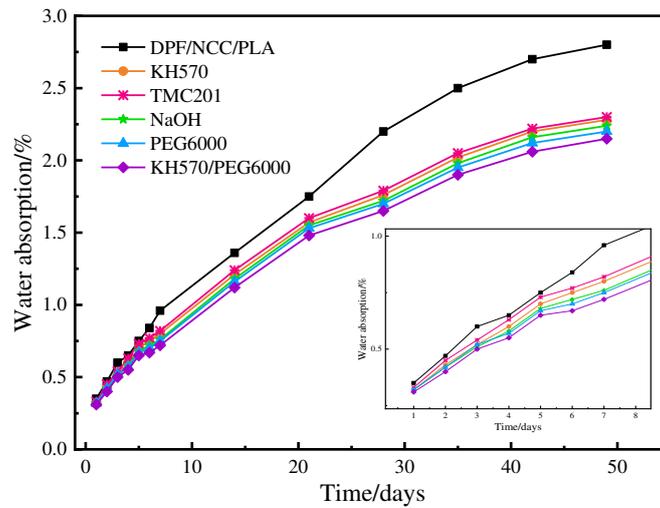
Figure 9. DTG curves of modified DPF/NCC/PLA composites.



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Figure 10. DSC curves of modified DPF/NCC/PLA composites.



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Figure 11. Water absorption property of modified DPF/NCC/PLA composites.

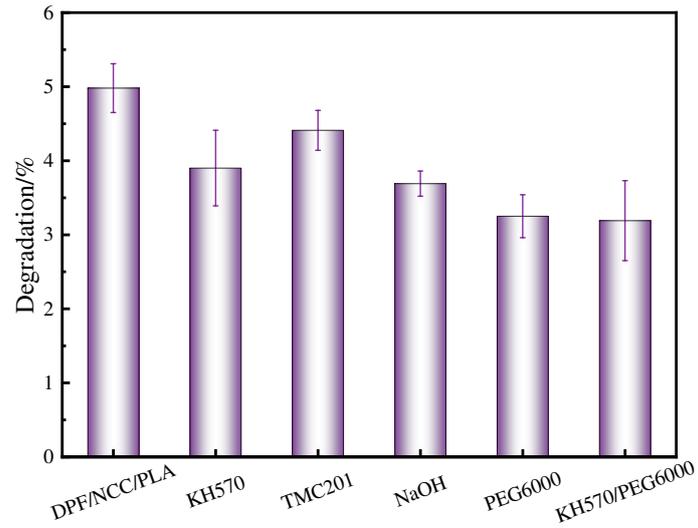


Figure 12. Degradation properties of modified DPF/NCC/PLA composites.

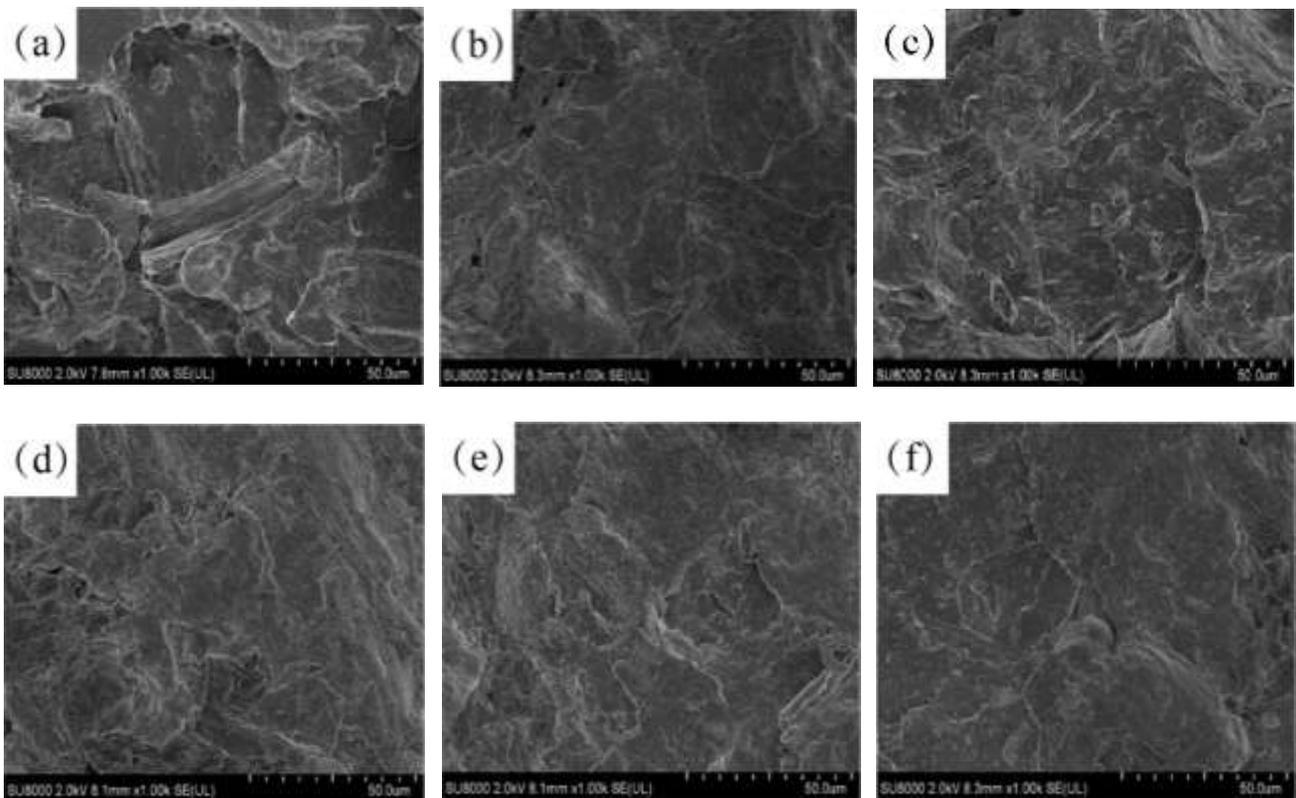


Figure 13. SEM micrographs of modified DPF/NCC/PLA composites ($\times 1000$): (a) DPF/NCC/PLA, (b) KH570 DPF/NCC/PLA, (c) TMC201 DPF/NCC/PLA, (d) NaOH DPF/NCC/PLA, (e) PEG6000 DPF/NCC/PLA, (f) KH570/PEG6000 DPF/NCC/PLA

Figures

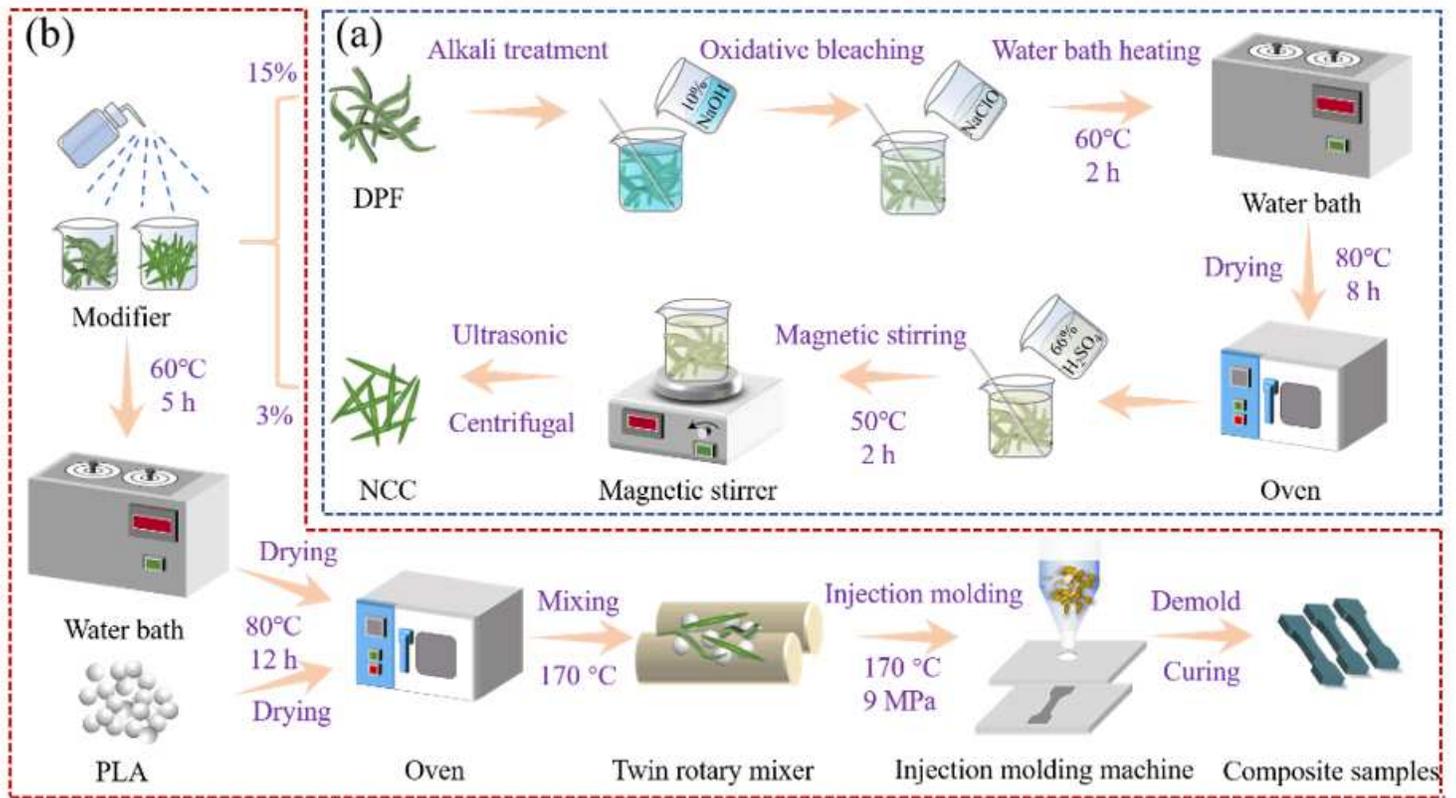


Figure 1

Schematic diagram of NCC and DPF/NCC/PLA modified composites preparation: (a) process of preparation of NCC, (b) process of preparation of DPF/NCC/PLA modified composites.

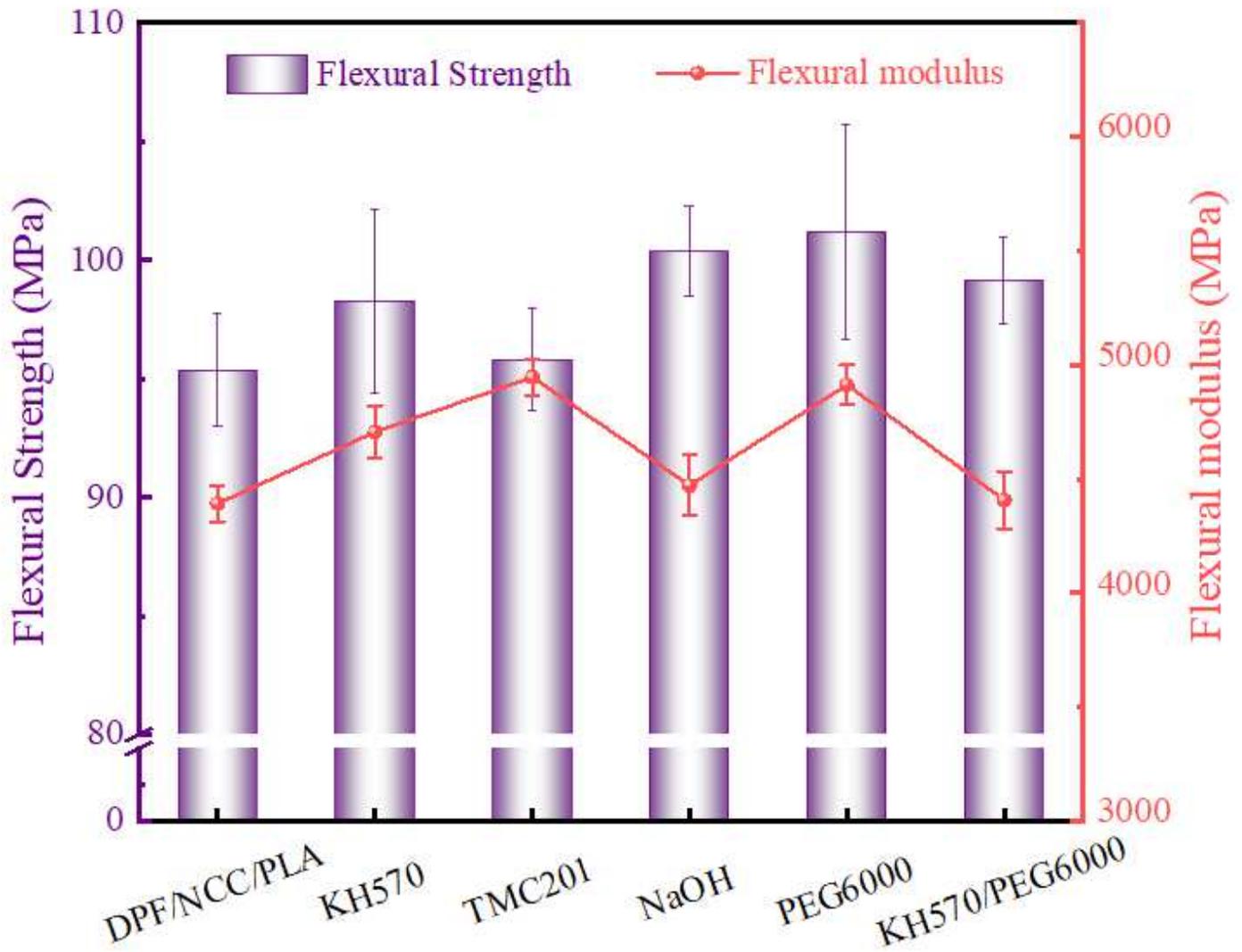


Figure 2

Flexural properties of modified DPF/NCC/PLA composites.

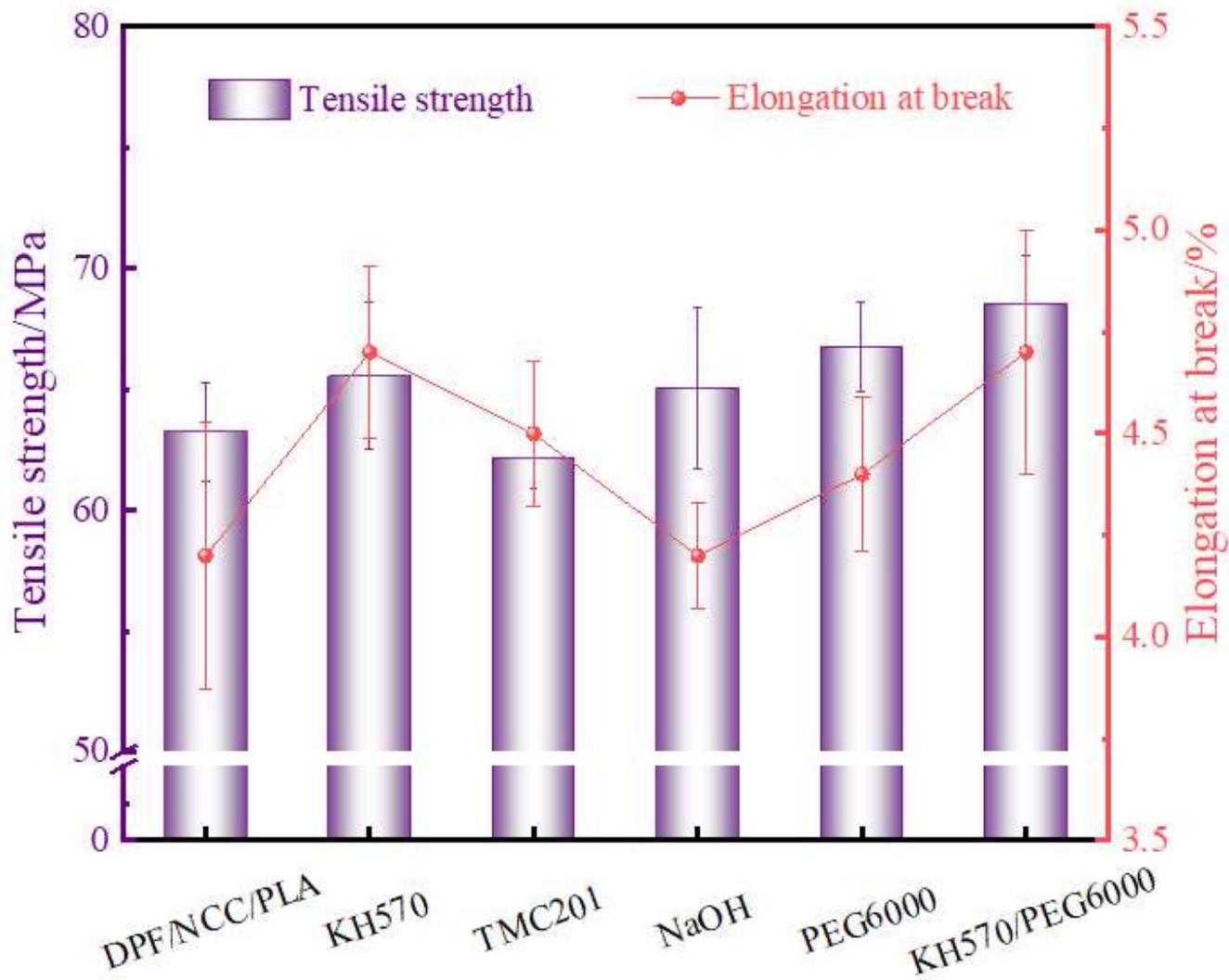


Figure 3

Tensile properties of modified DPF/NCC/PLA composites.

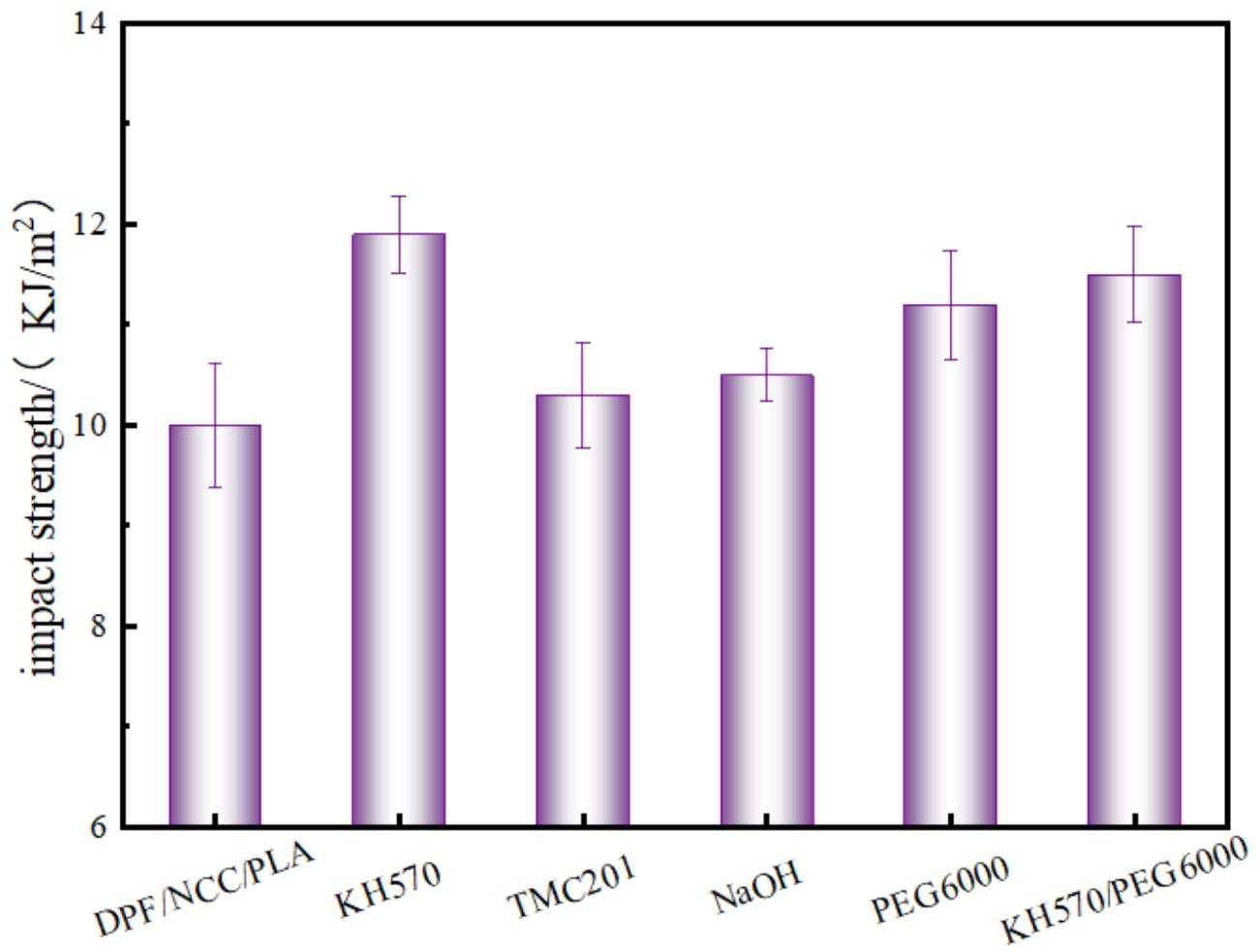
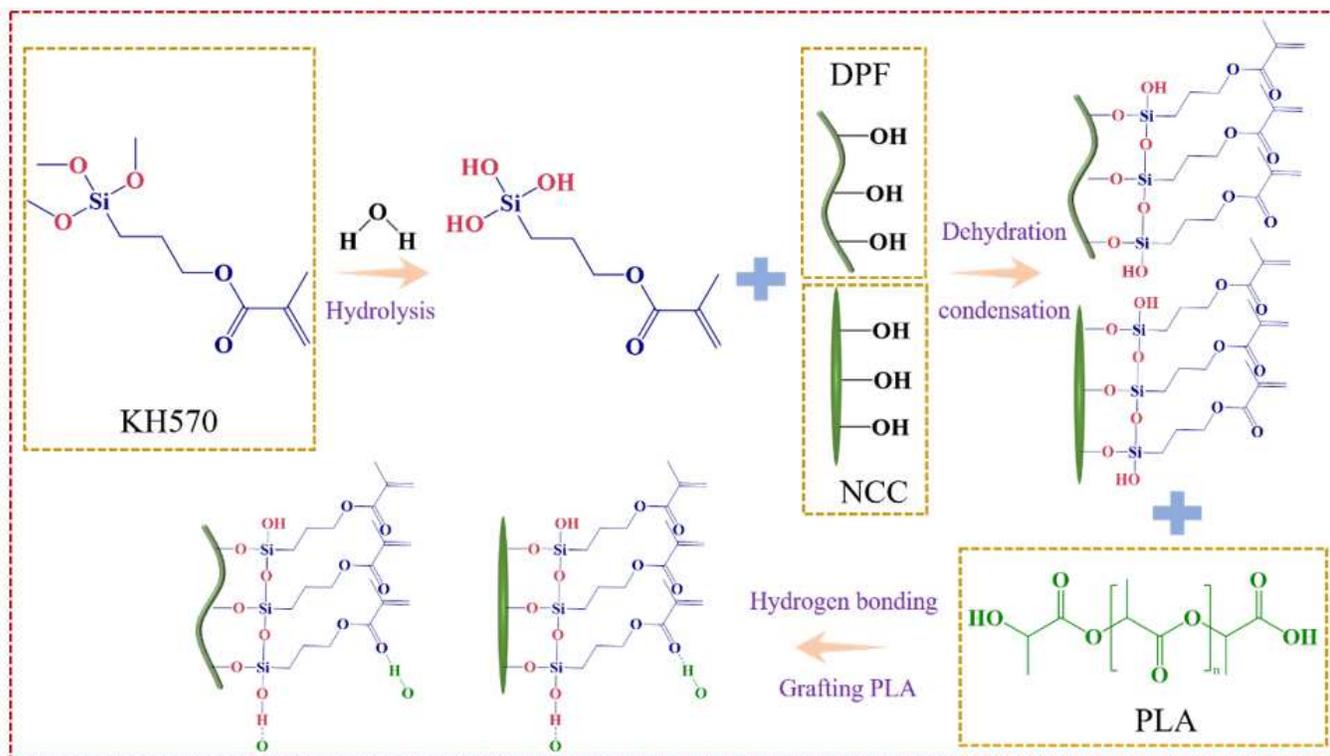
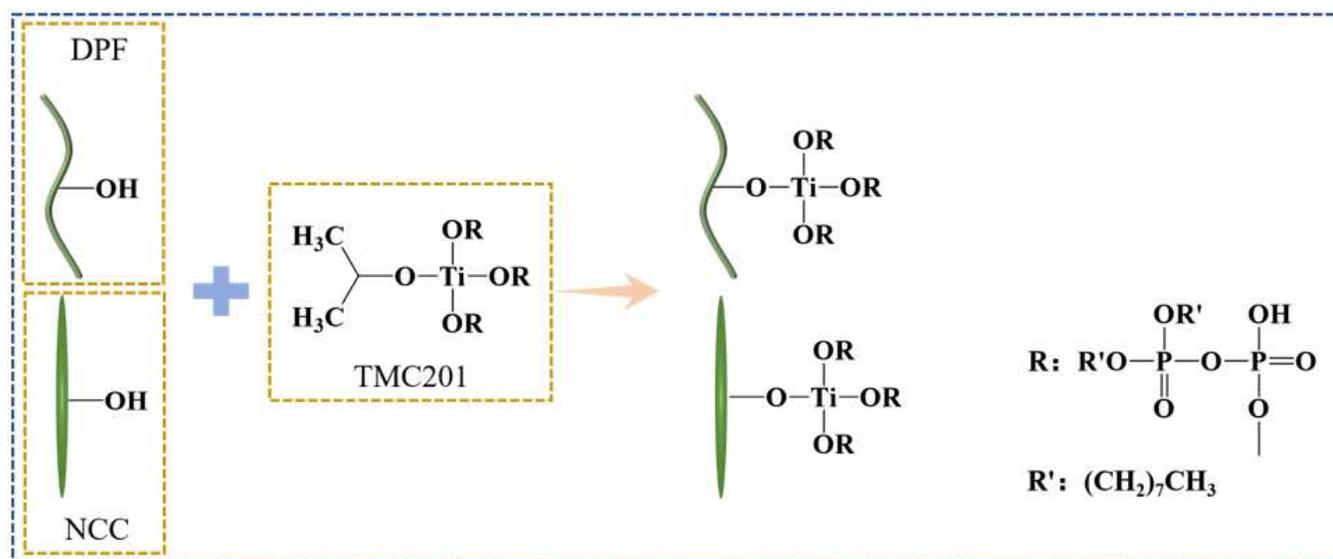


Figure 4

Impact properties of modified DPF/NCC/PLA composites.



(a)



(b)

Figure 5

Schematic diagram of modification mechanism: (a) KH570, (b) TMC201.

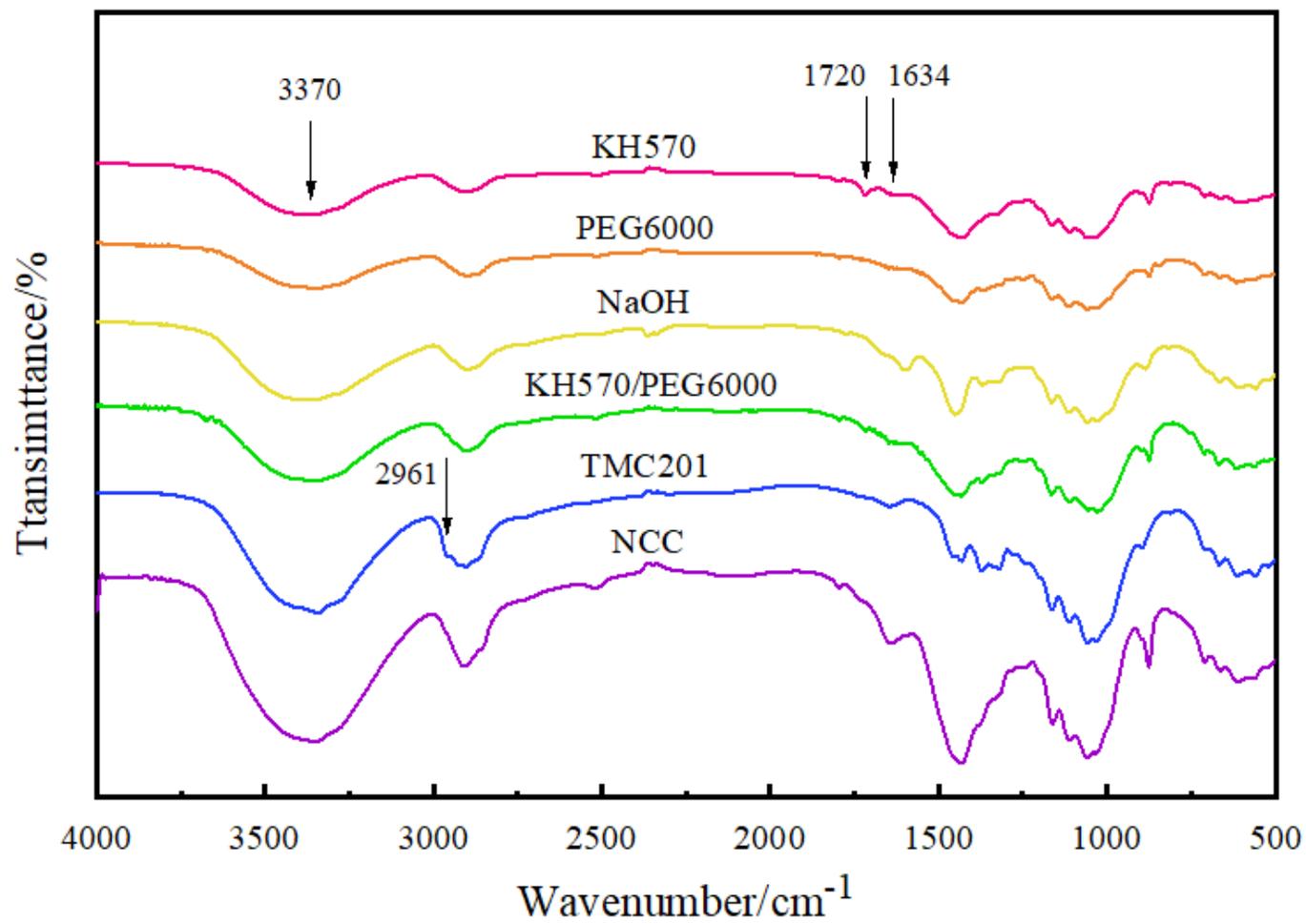


Figure 6

FTIR spectra of interface modified NCC.

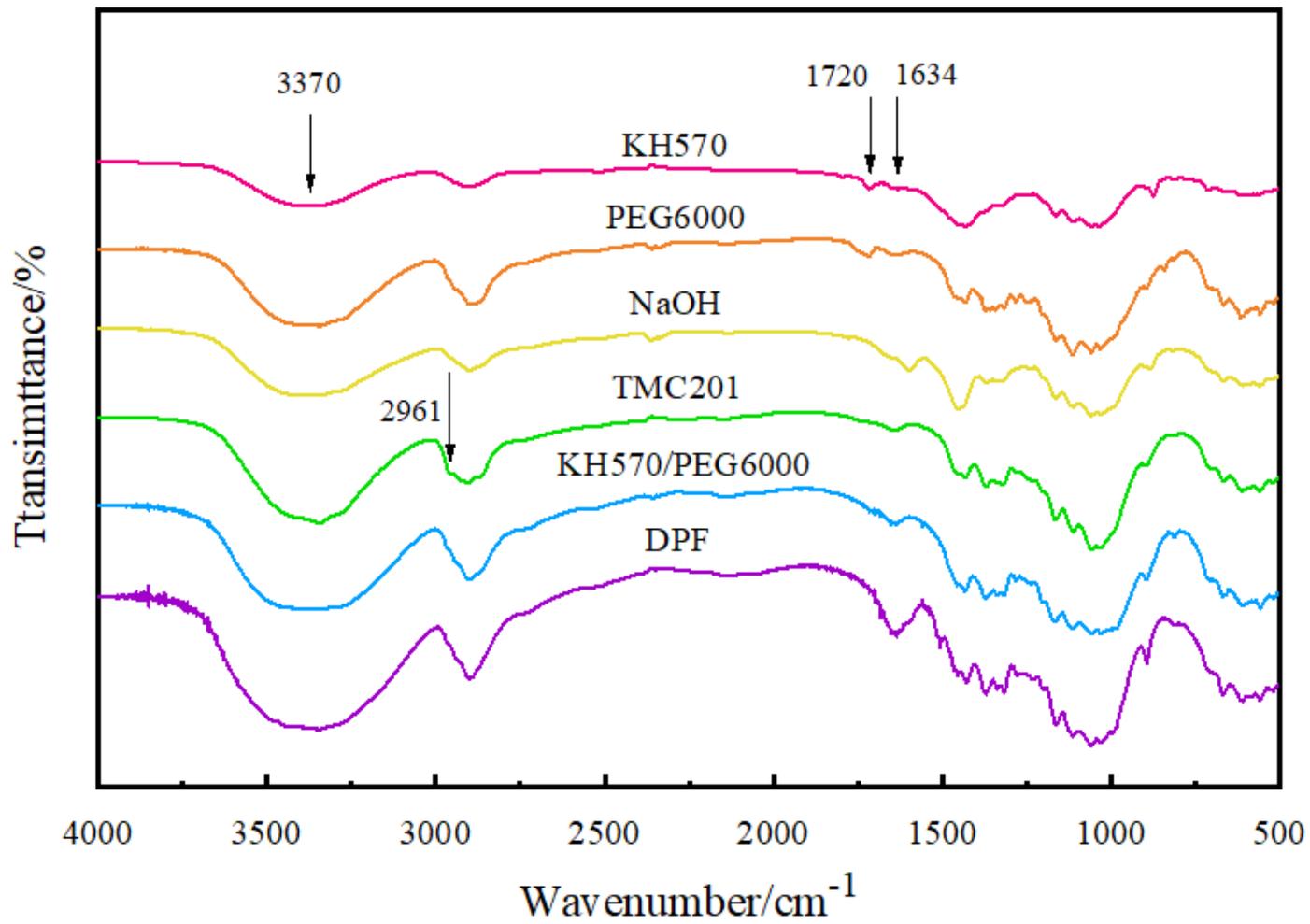


Figure 7

FTIR spectra of interface modified DPF.

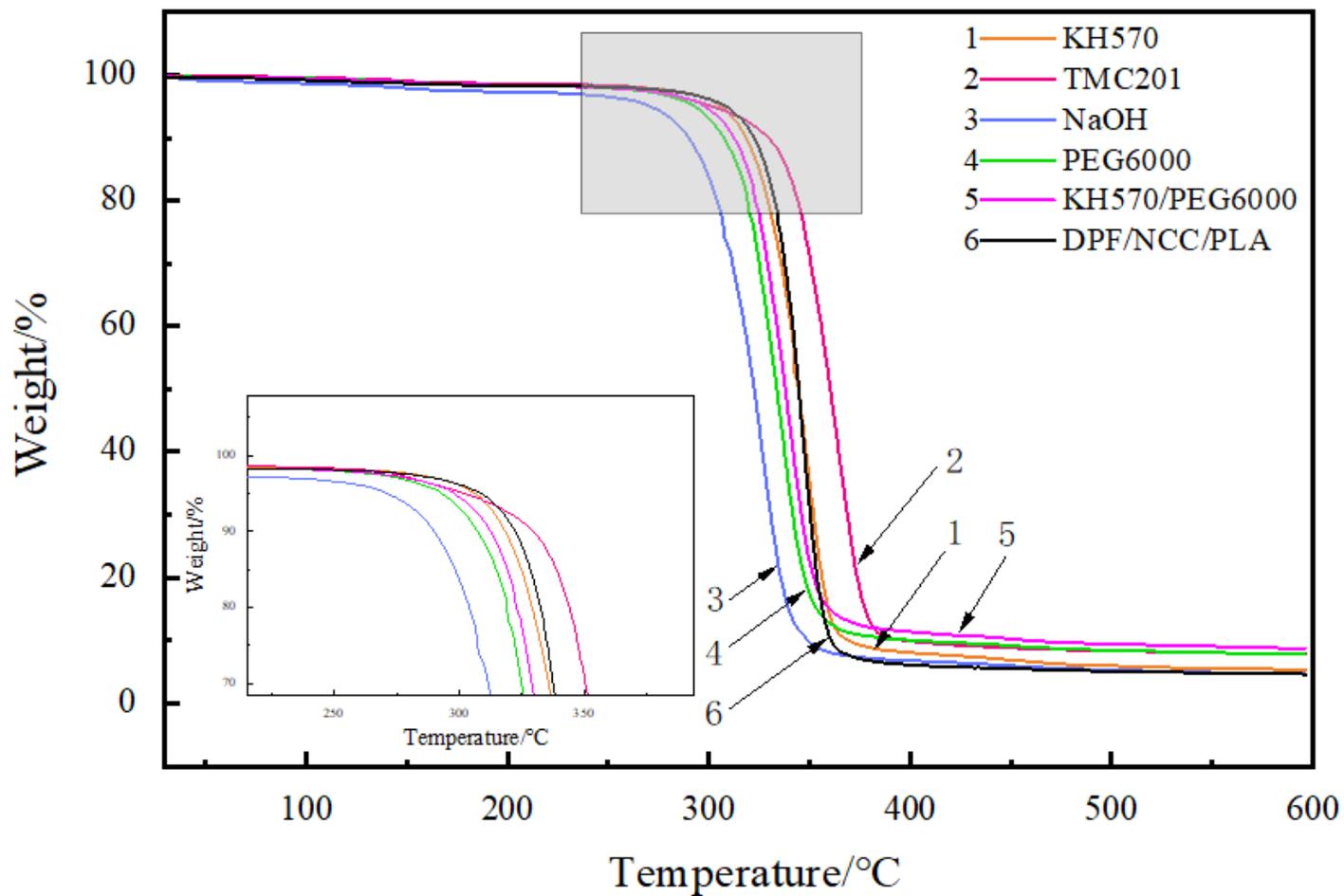


Figure 8

TG curves of modified DPF/NCC/PLA composites.

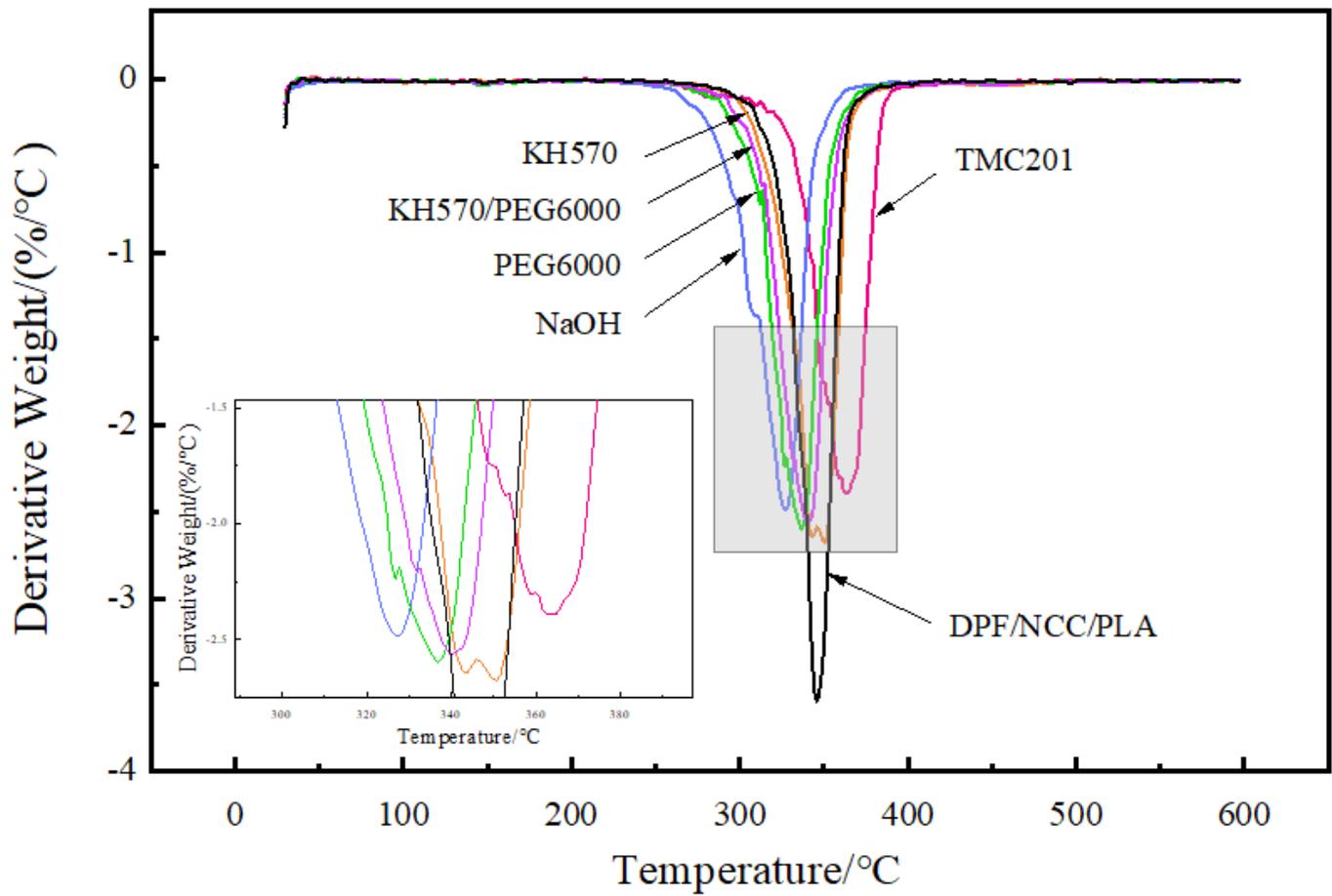


Figure 9

DTG curves of modified DPF/NCC/PLA composites.

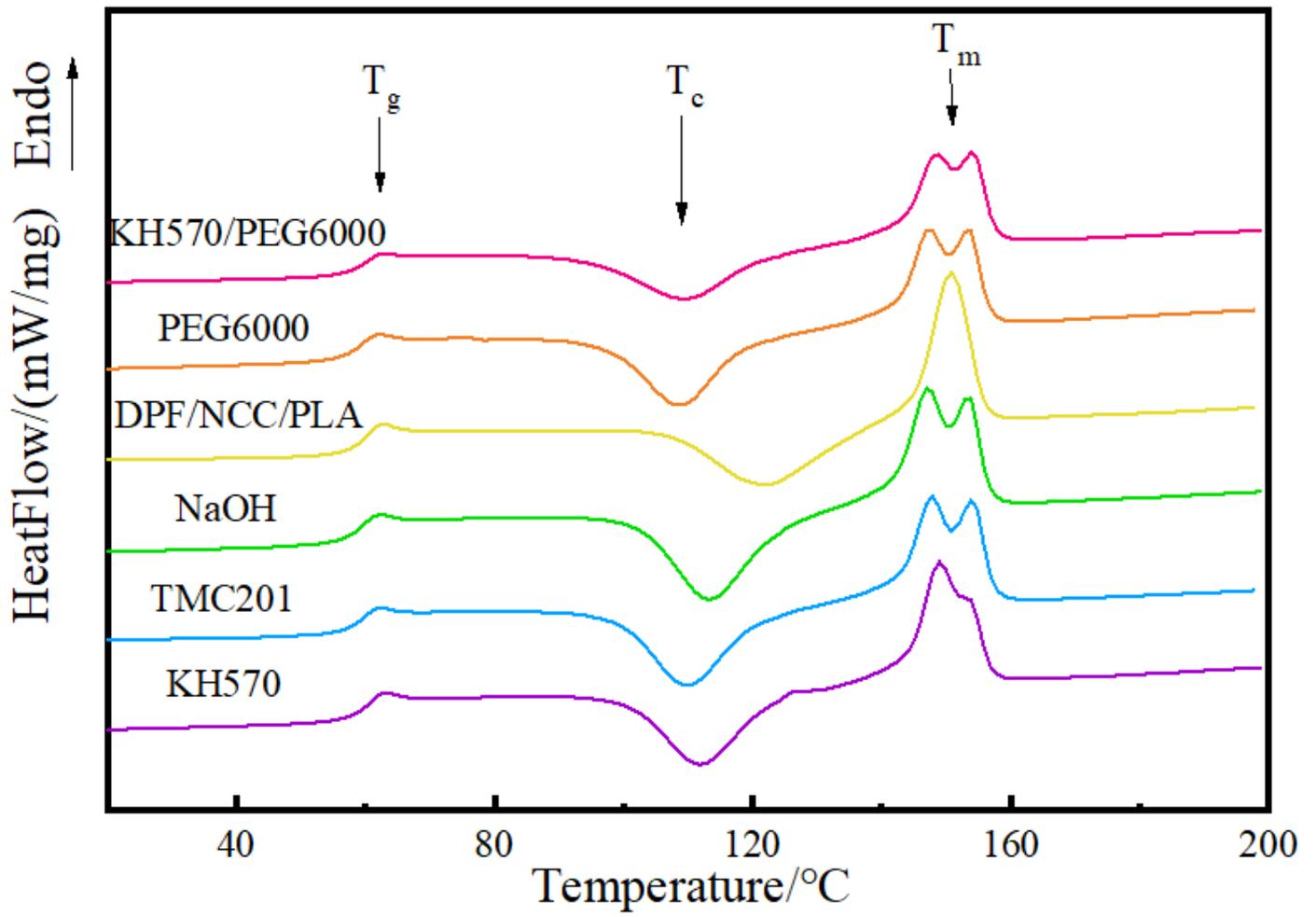


Figure 10

DSC curves of modified DPF/NCC/PLA composites.

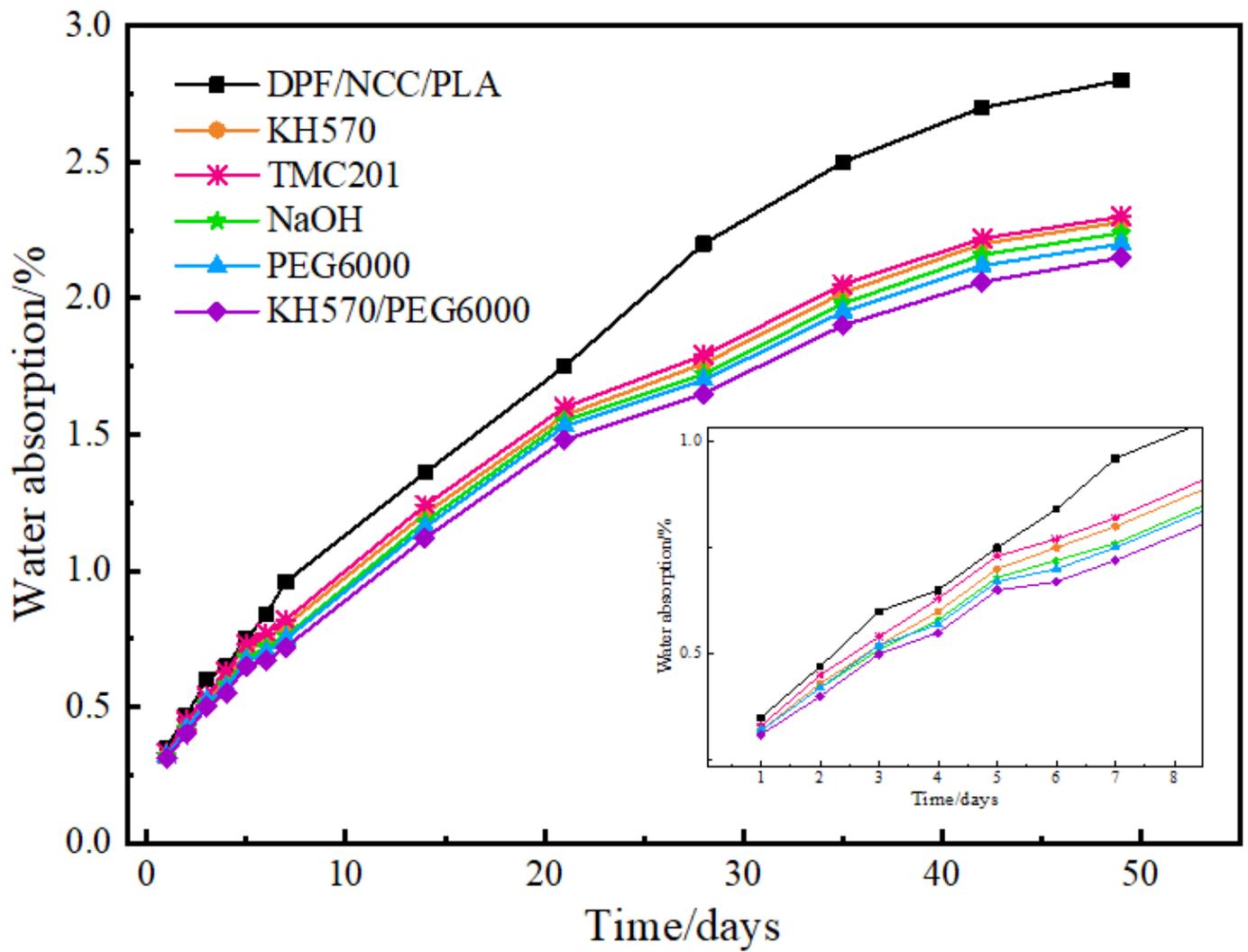


Figure 11

Water absorption property of modified DPF/NCC/PLA composites.

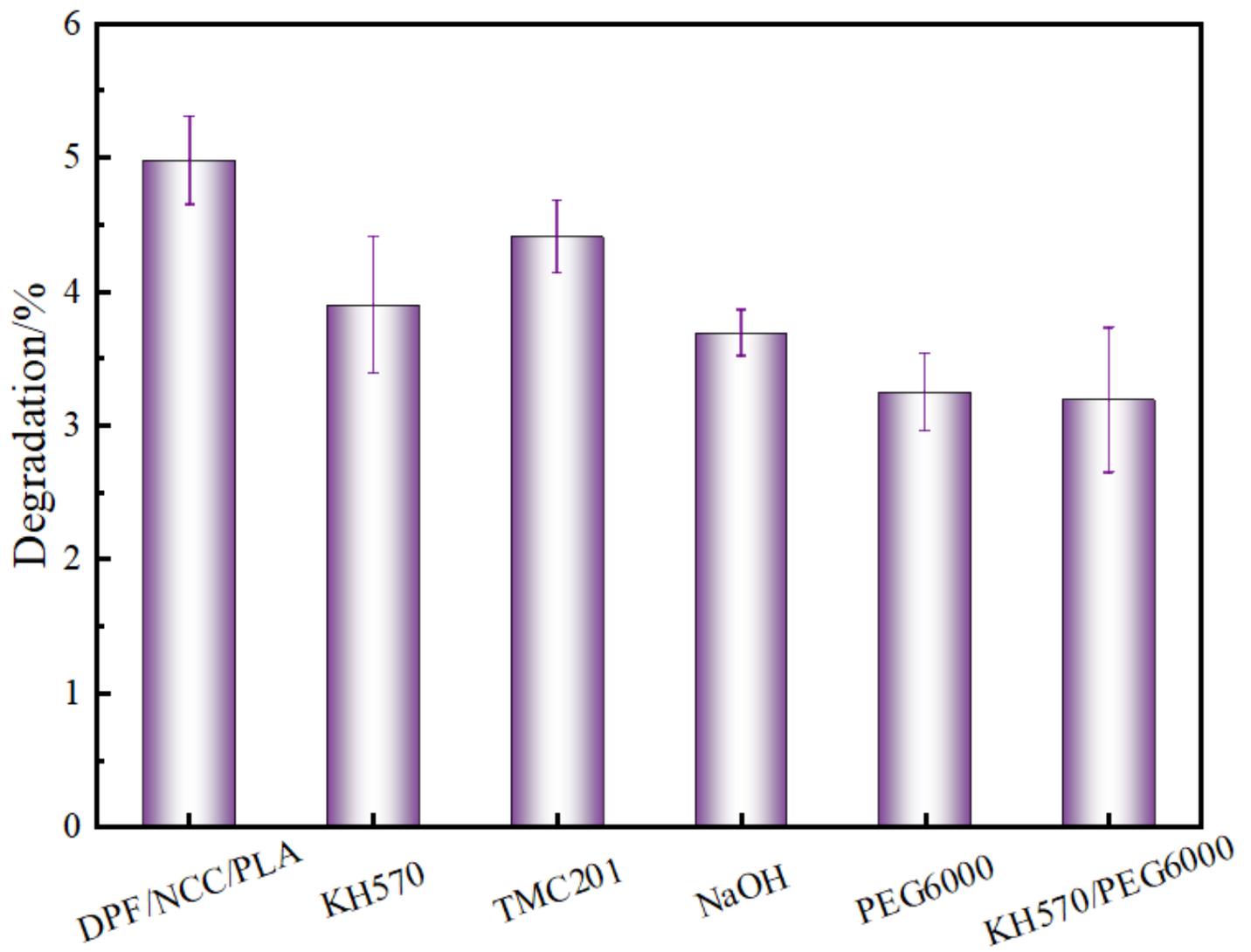


Figure 12

Degradation properties of modified DPF/NCC/PLA composites.

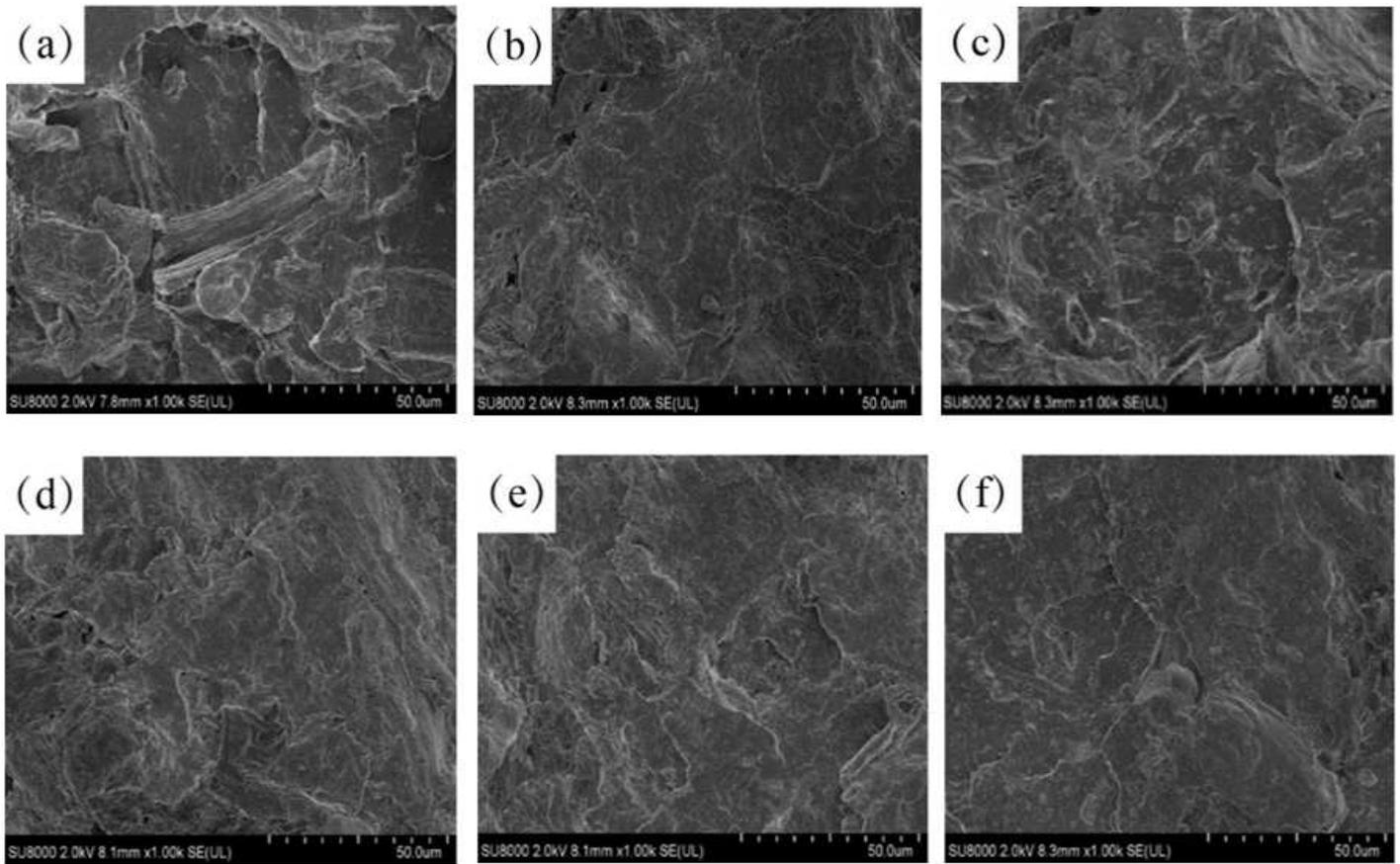


Figure 13

SEM micrographs of modified DPF/NCC/PLA composites ($\times 1000$): (a) DPF/NCC/PLA, (b) KH570 DPF/NCC/PLA, (c) TMC201 DPF/NCC/PLA, (d) NaOH DPF/NCC/PLA, (e) PEG6000 DPF/NCC/PLA, (f) KH570/PEG6000 DPF/NCC/PLA