

3.4 % solar-to-ammonia efficiency from nitrate using Fe single atomic catalyst supported on MoS₂ nanosheets

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2 **supported on MoS₂ nanosheets**

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26

27

28 **ABSTRACT**

29 **Electrochemical synthesis of NH₃ is a carbon-free alternative to the traditional**
30 **Haber-Bosch process. The challenge with nitrogen reduction reaction (NRR) to NH₃ is**
31 **cleavage of the inert N≡N triple bond of nitrogen gas. Obtaining NH₃ from**
32 **environmental pollutants, such as nitrates or nitrites, is a more practical route than NRR.**
33 **However, reduction of nitrates or nitrites to ammonia is currently hampered by modest**
34 **Faradaic efficiencies, typically below 10 %. Here, we report a novel heterogeneous**
35 **catalyst based on iron (Fe) single-atoms supported on two-dimensional MoS₂ (Fe-MoS₂)**
36 **for the nitrate reduction reaction (NO₃RR). We have found that Fe-MoS₂ exhibits**
37 **remarkable performance with a maximum Faradaic efficiency of 98 % for NO₃RR to**
38 **NH₃ at an overpotential of -0.48 V vs. the reversible hydrogen electrode (RHE) as**
39 **confirmed by our isotopic nuclear magnetic resonance (NMR) analyses. Density function**
40 **theory (DFT) calculations reveal that the enhanced selectivity for the production of NH₃**
41 **from single Fe atoms supported on MoS₂ is attributed to a reduced energy barrier of 0.38**
42 **eV associated with de-oxidation of *NO to *N – the usual potential limiting step in**
43 **NO₃RR. We assembled our catalyst in a two-electrode electrolyzer coupled to an**
44 **InGaP/GaAs/Ge triple-junction solar cell to demonstrate a solar-to-ammonia (STA)**
45 **conversion efficiency of 3.4 % and a yield rate of 0.03 mmol h⁻¹ cm⁻² equivalent to 510 μg**
46 **h⁻¹ cm⁻². Our results open new avenues for design of single-atom catalysts (SAC) for the**
47 **realization of solar-driven ammonia production.**

48

50 **MAIN TEXT**

51 Ammonia (NH_3) is an important industrial chemical that is widely applied as a nitrogen-
52 rich fertilizer for agriculture, textiles, plastics, and the pharmaceutical industry.^{1,2} The N cycle
53 perturbation is among the 3 identified planetary boundaries that have already been transgressed
54 by humanity.³ For renewable energy and net-zero carbon emission, ammonia has been
55 identified as a promising energy carrier because of its high hydrogen content of 17.7 wt % and
56 its high gravimetric energy density at 3 kWh kg⁻¹.⁴⁻⁷ The synthesis of NH_3 via the Haber-Bosch
57 reaction is done under harsh conditions with temperatures and pressures exceeding 400 °C and
58 200 bar, respectively. With an annual production of 250×10^6 tons, an estimated amount of 1%
59 of the world's energy is utilized in the Haber-Bosch, which is responsible for 1.4 % of the global
60 carbon dioxide emissions.⁸⁻¹⁰ As an alternative, the use of nitrogen and water for the electro-
61 catalytic synthesis of ammonia has recently attracted widespread attention.^{11,12} However, the
62 direct electrochemical reduction of N_2 for the production of ammonia under mild conditions is
63 severely limited by several bottlenecks such as: *i*) the high energy barrier required for cleavage
64 of inert $\text{N} \equiv \text{N}$ triple bond;^{13,14} *ii*) nonpolar nature of nitrogen molecules that results in a weak
65 interaction between N_2 and active sites of catalysts;¹⁵ *iii*) the very low solubility of N_2 in water
66 leading to slow reaction rates.^{16,17}

67 Alternatively, nitrates (NO_3^-) possess unique advantages as nitrogen sources for
68 electrosynthesis of NH_3 . The bond energy of polar $\text{N}=\text{O}$ (204 kJ/mol) is four times weaker than
69 the inert non-polar $\text{N} \equiv \text{N}$ triple bond, and therefore the $\text{N}=\text{O}$ bond can be easily activated at
70 lower energies.¹⁸ Nitrate is widely present in the environment and accumulates over time due
71 to agriculture and industrial production activity.^{19,20} NO_3^- is a common pollutant of water
72 resources that is responsible for birth defects such as infant methemoglobinemia and blue-baby
73 syndrome as well as thyroid and bladder cancers. NO_3^- RR for NH_3 production is therefore not

74 only in line with energy sustainability but is also a pollution mitigation strategy. The
75 electrocatalytic reduction of NO_3^- -to- NH_3 has been reported using Fe,²¹ Al,²¹ Cu bulk and
76 nanorods,^{22,23} Cu_2O/Cu wires,²⁴ Cu-Ni alloys,²⁵ Cu molecular catalyst,¹⁷ Co_3O_4 ,²⁶⁻²⁹ and
77 carbon-based materials.^{30,31} However, most NO_3R reactions typically suffer from modest
78 Faradaic efficiencies and/or stability.

79 We have recently identified chemically exfoliated MoS_2 nanosheets as two-dimensional
80 electrocatalysts for efficient oxidation of organic sulfides to sulfoxides with near-unity
81 selectivity.³² The nitrate reductase enzyme possesses a Mo (IV) active site coordinated with
82 sulfur coordinating ligands similar to the dimethyl sulfoxide (DMSO) reductase, whereas the
83 nitrogenase is a multinuclear enzyme with $MoFe_7$ clusters as the active sites. Fe-based catalysts
84 have been reported to be promising for NO_3RR but the Faradaic efficiency is low because of
85 the competing hydrogen evolution reaction and the formation of nitrogen *via* the five-electron
86 transfer pathway.³³⁻³⁵ To improve the selectivity to NO_3^- -to- NH_3 , Fe single-atom catalysts
87 (SACs) hold promise because the individual Fe atoms possess coordination environment that
88 results in efficient catalytic activity compared to bulk and nanostructured iron. To date, Fe-
89 SACs have been used in various heterogeneous catalytic reactions, such as CO oxidation,³⁶
90 oxygen reduction reaction (ORR),³⁷⁻³⁹ CO_2 reduction reaction (CO_2RR),^{40,41} and nitrogen
91 reduction reaction (NRR).^{42,43}

92 Inspired by the structure of active sites of enzymes, we report SACs based on individual
93 Fe atoms supported on MoS_2 nanosheets (Fe- MoS_2) for the electrocatalytic NO_3RR , which
94 exhibits excellent performance with a Faradaic efficiency of 98 % toward NH_3 at a low
95 overpotential of -0.48 V versus the reversible hydrogen electrode (*vs.* RHE) and a cathodic
96 energy efficiency of 31% at *vs.* RHE and -0.28 V *vs.* RHE, respectively. The optimized Fe-
97 MoS_2 SACs were implemented in a 2-electrode electrolyzer coupled to an external photovoltaic
98 (PV) device to allow solar-driven conversion of NO_3^- to NH_3 to demonstrate a maximum yield

99 rate of $0.03 \mu\text{mol h}^{-1} \text{cm}^{-2}$ equivalent to $510 \mu\text{g h}^{-1}$ with a near-unity FE for NH_3 . The system
100 delivers stable ammonia production and we estimated the solar-to-ammonia (STA) conversion
101 efficiency to be ca. 3.4 % – setting a new benchmark for the production of NH_3 from a PV-
102 powered electrolyzer.

103

104 **Synthesis and characterization of the Fe-MoS₂ SAC**

105 MoS₂ and Fe-MoS₂ nanosheets were synthesized *via* a hydrothermal reaction using
106 ammonium tetrathiomolybdate: $(\text{NH}_4)_2\text{MoS}_4$ and iron(III) nitrate nonahydrate as precursors of
107 MoS₂ and Fe, respectively.⁴⁴ The morphology of as-synthesized Fe-MoS₂ was characterized by
108 scanning electron microscopy (SEM) and transmission electron microscopy (TEM). SEM
109 revealed the 2D of the MoS₂ nanosheets (**Figure 1a**), cross-sectional TEM revealed the uniform
110 coverage of the MoS₂ nanosheets film (thickness ~ 150 nm) on the conducting carbon cloth
111 (**Figure 1b**). The HAADF-STEM analysis of single-layer catalyst regions shows that the
112 nanosheets are highly crystalline as evidenced by the fast Fourier transform (FFT) patterns in
113 **Figure 1c** and **Supplementary Figure 1**. The observation along [110] direction of the MoS₂
114 slabs reveal the presence of the 1T polytype in Fe-MoS₂ in agreement with the presence of *J*
115 peaks in the Raman spectra (**Supplementary Figure 2**). Energy-dispersive X-ray spectroscopy
116 (EDS) elemental mapping was used to confirm the presence of Fe on MoS₂. **Figure 1d,e** show
117 uniform distribution of Fe atoms on the nanosheets and no sign of aggregation was observed
118 even at the highest magnification (**Supplementary Figure 3**). The presence of Fe is also
119 highlighted by spatially-resolved electron energy loss spectroscopy (SR-EELS, **Figure 1f** and
120 **Supplementary Figure 4**). In particular, the EELS chemical maps evidence that the domain
121 size of Fe is equal or below the EELS voxel size (0.16 nm). It should be noted that the presence
122 of oxygen is also highlighted by EELS and will be discussed later. The high-resolution TEM
123 (HR-TEM) images pointed out that Fe-MoS₂ retains the layered structure of MoS₂ with an

124 interlayer distance of $\approx 9.1 \text{ \AA}$, ascribed to the (002) crystalline plane (**Figure 1g**). Compared
125 with pristine MoS₂, the interlayer distance is $\sim 2.8 \text{ \AA}$ larger in the present samples, which is
126 attributed to the presence of Fe atoms on the surface of the nanosheets (**Figure 1g** and
127 **Supplementary Figure 5**). This observation is further corroborated by our X-ray diffraction
128 (XRD) analyses in which the d-spacing of Fe-MoS₂ was found to be 9.8 \AA , which is larger than
129 that of MoS₂ at 6.5 \AA (**Figure 1h**). We studied the influence of Fe atoms on interlayer spacing
130 by preparing Fe-MoS₂ catalysts with increasing loadings from 1.36 % to 2.14 % (see
131 Supplementary Information for details). The position of the (002) did not change significantly
132 with the Fe: Mo ratio suggesting a similar average *d*-space for all Fe-MoS₂ samples.
133 Interestingly the half maximum (FWHM) decreases with the Fe content whereas the intensity
134 of the (002) peak increases; indicating improved stacking order and crystallinity of the MoS₂
135 nanosheets at higher Fe loadings.⁴⁵

136 Next, we used X-ray photoelectron spectroscopy (XPS) to elucidate the chemical
137 composition of the MoS₂ and Fe-MoS₂ catalysts. The Fe2p spectra of Fe-MoS₂ unambiguously
138 revealed the presence of Fe on the samples. The spectra can be decomposed into two doublets
139 at 707.9/720.8 eV and 709.6/723.2 eV, which are ascribed to contributions from Fe bonded to
140 S and the satellite peaks, respectively (**Figure 1i**).^{46,47} The splitting of the S2p and Mo3d spectra
141 (**Supplementary Figure 6a, b**) for both MoS₂ and Fe-MoS₂ suggests the coexistence of the 1T
142 and the 2H phase MoS₂ in agreement with our Raman and STEM analyses. From the
143 deconvolution of the Mo3d and S2p signals, the amount of 1T phase in MoS₂ and Fe-MoS₂ is
144 estimated to be 42.5 % and 47.0 % respectively, whereas minimal amount of Mo⁶⁺ is detected
145 at 16.9 % and 10.2 %. The presence of an additional doublet at 162.6 eV and 163.9 eV was also
146 identified in the S2p region and attributed to S2p_{1/2} and S2p_{2/3} signals from S-Fe bond.⁴⁸⁻⁵⁰
147 X-ray absorption near-edge spectroscopy (XANES) and extended X-ray absorption fine
148 structure (EXAFS) were acquired to elucidate the electronic and coordination structure of Fe-

149 MoS₂. **Figure 1j** shows the Fe K-edge XANES profiles for Fe-MoS₂, compared to that of Fe₂O₃,
150 FeS, and Fe metal used as references for Fe³⁺, Fe²⁺, and Fe⁰, respectively. The near-edge
151 absorption energy position of Fe-MoS₂ was found to be between that of FeS and Fe₂O₃,
152 suggesting that the oxidation state of Fe in Fe-MoS₂ is comprised of between +2 and +3. Only
153 one main peak was visible at 1.78 Å in the Fourier transformed EXAFS (FT-EXAFS) spectrum
154 of Fe-MoS₂ (**Figure 1k**), which is close to that of Fe-S in FeS at 1.87 Å and is attributed to the
155 first coordination shell of Fe-S. It should be noted that no contribution for the Fe-Fe bond
156 expected at ~2.20 Å and ~2.58 Å for Fe⁰ and Fe³⁺ in Fe metal and Fe₂O₃ were observed from
157 the FT-EXAFS spectrum. Our X-ray absorption data suggest the fact that Fe is dispersed on the
158 MoS₂ nanosheets at the atomic level. To gain more information on the structural parameters,
159 we performed least-square EXAFS fittings on Fe-MoS₂ (see **Supplementary Figure 7** and
160 **Supplementary Table 1**). The corresponding coordination number of Fe atom in the Fe-MoS₂
161 was estimated to be ≈ 3.5 with the bond lengths of 2.25 Å and 1.97 Å for Fe-S and Fe-O
162 respectively. The identification of the Fe-O bond may be attributed to the presence of axial
163 oxygen atoms bonded to Fe, in good agreement with the observed spatial overlapping of the Fe
164 and O EELS chemical maps (**Supplementary Figure 4**). All the above results corroborate the
165 single atomic nature of Fe in Fe-MoS₂ catalysts as illustrated in **Figure 1l**.

166 **Investigations of the catalytic activity on Fe-MoS₂**

167 The electrocatalytic properties of Fe-MoS₂ catalyst for the reduction of NO₃⁻ were
168 investigated in an H-cell reactor using a 0.1 M of K₂SO₄ + 0.1 M NaOH electrolyte solution
169 (See the Methods section for details about the electrochemical measurements). **Figure 2a** shows
170 linear sweep voltammetry (LSV) curves measured with and without 0.1 M NaNO₃ for MoS₂,
171 and Fe-MoS₂ nanosheets compared to results from carbon support used as reference. Higher
172 current density and lower onset potential were detected in the presence of NO₃⁻. To prove that
173 the change in current density originates from the reduction of NO₃⁻ to NH₃ rather than a change

174 in the electrolyte concentration, we estimated the concentration of ammonia after the reaction
175 *via* a colorimetric method using the indophenol blue method (See details in the Supplementary
176 Information file and **Supplementary Figure 8**). Ammonia concentrations were detected in the
177 range of 0.6~2.0 mM after 1 hour of electrolysis under a constant applied potential of -0.48 V
178 versus the reversible hydrogen electrode (*vs.* RHE), which indicates that the change in the LSV
179 data originates from electrocatalytic conversion of NO_3^- . To exclude contamination that could
180 lead to overestimation of ammonia production, isotopic experiments were conducted using 0.1
181 M $Na^{14}NO_3$ and 0.1 M $Na^{15}NO_3$. The 1H nuclear magnetic resonance (NMR) spectra of the
182 electrolyte after electrolysis are shown in **Figure 2b**. In the case of $Na^{15}NO_3$ electrolyte solution,
183 the 1H NMR spectrum exhibits two clear symmetric signals at $\delta = 7.02$ and 7.14 ppm with a
184 spacing of 73.1 Hz assigned to $^{15}NH_3$, due to scalar interaction between 1H and ^{15}N . Conversely,
185 in presence of $Na^{14}NO_3$ three symmetric signals located at 6.99, 7.08, and 7.17 ppm were
186 detected with a spacing of 52.2 Hz ascribed to $^{14}NH_3$.¹² We performed a blank experiment in
187 the absence of NO_3^- and no signals from $^{15}NH_3$ nor $^{14}NH_3$ were detected. Overall, our results
188 confirmed the successful reduction of NO_3^- to NH_3 and rule out contamination. The content of
189 ammonia in the electrolyte was further quantified *via* 1H NMR using an external standard for
190 calibration (**Supplementary Figure 9**). The ammonia quantification using 1H NMR and
191 colorimetry are comparable and validates our colorimetric strategy for the quantification of the
192 Faradaic efficiency of the reaction on Fe-MoS₂ (**Supplementary Table 3**). **Figure 2c** shows
193 the Faradaic efficiency for the formation of NH_3 on Fe-MoS₂ compared to pristine MoS₂, Cu,
194 and Cu₅₀Ni₅₀ alloy used as the benchmarked catalyst for the NO₃RR (Ref. ²²). The selectivity
195 of Fe-MoS₂ nanosheets rapidly increases with the onset potential and the Faradaic efficiency
196 for NH_3 reaches a maximum value of 95.8% at -0.48 V *vs.* RHE, which is higher than that for
197 MoS₂ nanosheets and Cu at 40.8 % and 22.5 % respectively. The enhanced FE on Fe-MoS₂
198 translates to a specific current density j_{NH_3} of -8.4 mA cm⁻², which represents 7.1- and 2.8-fold

199 increase compared to MoS₂ and Cu, respectively. The onset potential for NO₃RR – measured
200 at -1 mA cm⁻² – was found to be ≈ 100 mV vs. RHE for Fe-MoS₂ compared to 200 mV vs. RHE
201 for Cu (**Figure 2d**). The j_{NH_3} and onset potential on Fe-MoS₂ are comparable with those from
202 Cu₅₀Ni₅₀ alloy. To further quantify the catalytic properties of Fe-MoS₂, we estimated the
203 cathodic energy efficiency ($EE_{\text{NO}_3\text{RR}}$) of the different catalysts. The $EE_{\text{NO}_3\text{RR}}$ for Fe-MoS₂ was
204 found to be the highest at 31 % for a cathodic potential of -0.28 V vs. RHE, which is at least 4
205 times higher than that of MoS₂, Cu, and Cu₅₀Ni₅₀ alloy (**Figure 3b**). These results also
206 compared favorably with previous reports from the literature based on CuNi alloy,⁵¹ Cu
207 nanosheets,⁵² Cu/CuO nanowire arrays,⁵³ titanium electrode,⁵⁴ and copper-molecular solid
208 catalyst (**Supplementary Table 5**),⁵⁵ and other MoS₂-based catalysts those we synthesized
209 (**Supplementary Figure 10**). Our investigations indicate that the presence of atomically
210 dispersed Fe on the two-dimensional MoS₂ matrix enhances both the intrinsic catalytic activity
211 and the selectivity with respect to H₂.

212 We then sought to better understand the behavior of Fe-MoS₂ by exploring the influence
213 of Fe loading and the nitrate concentration in the electrolyte. At -0.48 V vs. RHE, the Faradaic
214 efficiency continuously increases from 30.1 % up to 95.8 % as the Fe content is increased from
215 0 % (i.e. pristine MoS₂) to 2.13% (**Supplementary Figure 11a,b**). The effect of nitrate
216 concentration on catalytic properties was explored by varying the NO₃⁻ concentration from 10
217 mM up to 100 mM. Remarkably we observed that the FE for NH₃ on Fe-MoS₂ is largely
218 maintained in presence of diluted NO₃⁻ and the FE remains as high as 70% for a nitrate
219 concentration as low as 10 mM. On the contrary, the concentration profoundly affects the
220 NO₃RR performance of MoS₂ and the FE decreases to ~3.8 % for a nitrate concentration of 10
221 mM (**Figure 2e**). This apparent 18-fold increase of FE_{NH_3} in dilute medium highlights the high
222 selectivity of Fe-MoS₂ towards the NO₃RR versus the competing hydrogen evolution reaction
223 (HER).

224 We assessed the charge transfer resistance (R_{CT}) at the interface between Fe-MoS₂ and
225 the electrolyte using electrochemical impedance spectroscopy (EIS) (**Supplementary Figure**
226 **12a**). The Nyquist plots of different catalysts exhibit the typical semicircle shape, which reflects
227 the interface resistance on the electrode surface. By modeling the EIS responses with the
228 Randles equivalent circuit, the values of R_{CT} are found to be low in the case of Fe-MoS₂ at 13.1
229 Ω compared to 24.8 Ω for MoS₂ and 9.8 Ω for Cu. This points to faster kinetics for electron
230 transfer at the surface of the catalyst in agreement with the reduction of the Tafel slope at 260
231 mV dec⁻¹ compared to >500 mV dec⁻¹ for pristine MoS₂ and other Cu-based catalysts
232 (**Supplementary Figure 12b**). We conclude that the reduced R_{CT} and Tafel slope result from
233 improved conductivity of the Fe-MoS₂ nanosheets and the presence of the metallic 1T polytype
234 within the MoS₂ lattice. To evaluate the stability toward the conversion of NO_3^- to NH₃, we
235 applied a steady-state potential on the Fe-MoS₂ electrode, while recording the current density
236 and measuring the FE using colorimetry. **Figure 2f** shows that the FE at -0.48 V vs. RHE
237 displays minimal changes over 7 hours with an average value of 98 %. The robustness of the
238 Fe-MoS₂ performance was further corroborated by the high retention of the current density,
239 showing that the 7-hour average of yield rate was 431.8±38.6 $\mu\text{g h}^{-1} \text{cm}^{-2}$.

240 We explored the isotopic effect on the NO₃RR by investigating the electrochemical
241 responses of Fe-MoS₂ in presence of Na¹⁴NO₃ and Na¹⁵NO₃ respectively.⁵⁶ Using Na¹⁵NO₃, we
242 observe a decrease in the current density together with an increase of the overpotential of ~40
243 mV (**Figure 3a**). The LSV and chronoamperometry responses were recorded for each nitrate
244 isotope with increasing temperatures from 25 °C to 65 °C and the apparent activation energies
245 were obtained by fitting the *Arrhenius* plot of the specific current density j_{NH_3} as a function of
246 T^{-1} (**Figure 3a**). The slope of the *Arrhenius* plot was estimated to be 0.18 and 0.21 for Na¹⁴NO₃
247 and Na¹⁵NO₃ respectively. The apparent difference between the two isotopes is in agreement

248 with the change in the polarization curves and is attributed to the slow diffusion of $\text{Na}^{15}\text{NO}_3$ in
249 the Helmholtz layer as well as in the interlayer of Fe-MoS₂ nanosheets.

250 To obtain more insight into the remarkable selectivity of the Fe-MoS₂ for the
251 electrosynthesis of NH₃, we investigated NO₃RR on different MoS₂-based SACs with different
252 transition metals. **Supplementary Figure 13** shows the polarization curves on Co-MoS₂, Ni-
253 MoS₂, Cu-MoS₂, and Fe-MoS₂ nanosheets in the presence of NO_3^- . Fe-MoS₂ outperformed the
254 other MoS₂ SACs as evidenced by the larger current density and the lower onset potential
255 compared to Co-MoS₂, Ni-MoS₂, Cu-MoS₂. The FE for NH₃ was found to be 59.6 % at -0.3 V
256 vs. RHE on Co-MoS₂, which is lower than that of Fe-MoS₂ at 86.4 % (**Figure 3c**). Remarkably
257 the values of the onset potential for NO₃RR and j_{NH_3} on Fe-MoS₂ are also 4~10 folds higher
258 and at least 400 mV vs. RHE lower than other MoS₂ catalysts (**Figure 3d**) – strongly suggesting
259 that the dispersed Fe atoms on the MoS₂ nanosheets are key for enhancing the intrinsic catalytic
260 activity and selectivity towards the NO₃RR.

261

262 **Computational investigations of the NO₃RR on MoS₂-based SACs**

263 To rationalize our experimental results, we conducted density functional theory
264 calculations to investigate the NO₃RR on the different single-atom catalysts supported on MoS₂.
265 The structural models of SACs consisted of four different transition metals: Fe, Co, Ni, and Cu
266 atoms coordinated with 3 sulfur atoms as presented in **Supplementary Figure 14**.⁵⁷ **Figure 4a**
267 shows the successive steps associated with nitrate reduction on the MoS₂ SACs, which can be
268 decomposed into two main electrochemical processes: $^*\text{NO}_3^- \rightarrow ^*\text{NO}_2 \rightarrow ^*\text{NO} \rightarrow ^*\text{N}$ and $^*\text{NH}$
269 $\rightarrow ^*\text{NH}_2 \rightarrow ^*\text{NH}_3$ corresponding to the deoxidation and hydrogenation mechanisms,
270 respectively.^{23,58} There is a general agreement that the catalytic active sites of MoS₂ are located
271 at the edges of the nanosheets.⁵⁹ We first computed the Gibbs free energy of the full NO₃RR
272 pathway on different MoS₂ SACs. On pristine MoS₂, the first deoxidation step: $^*\text{NO}_3^- \rightarrow ^*\text{NO}_2$

273 is considered as the potential-dependent step (PDS) with a very high reaction free energy of
274 1.47 eV. Such a high value is attributed to the formation of strong covalent bonds between two
275 adjacent Mo atoms and two oxygen atoms of NO_3^- (with $\Delta E_{NO_3^-} = -4.04$ eV for the largest charge
276 transfer of $0.85 e^-$) (**Supplementary Figure 15**). Conversely, all SACs display a different
277 energy profile illustrating that the inclusion of metallic single atoms profoundly modifies the
278 thermodynamics landscape of the NO_3RR on MoS_2 . The PDS step is associated with the $*NO$
279 deoxidation step. Among the different metal single atoms, the lowest reaction free energy is
280 obtained for Fe- MoS_2 at 0.38 eV – in qualitative agreement with our experimental observations.
281 When plotting the energy barrier for the reaction as function of the experimental onset potential,
282 we observed a linear relationship, which clearly suggests that the catalytic activity of MoS_2
283 SACs is controlled by the barrier of the deoxidation step (**Figure 4d**).

284 To elucidate the origin of the reduced energy barrier for the PDS on the active sites, we
285 examined the projected densities of states (PDOS) of NO adsorption on Fe- MoS_2
286 (**Supplementary Figure 16**). We found that there is a strong overlap between energy levels of
287 the α -spin and β -spin d orbitals of Fe- MoS_2 and the π^* orbitals of NO , which leads to an orbital
288 splitting and rearrangement to form new d - π^* bonding and antibonding orbitals (**Figure 4b**).
289 According to our calculations of the adsorption energy of $*NO$ on the MoS_2 SACs ($E_{ads, NO}$),
290 we identified the following trend: $E_{ads, NO}(Fe-MoS_2) > E_{ads, NO}(Co-MoS_2) > E_{ads, NO}(Ni-MoS_2) >$
291 $E_{ads, NO}(Cu-MoS_2)$ (**Supplementary Figure 17** and **Supplementary Table 6**). Our results
292 suggest that the stabilization of $*NO$ on Fe- MoS_2 enhances the catalytic activity – in agreement
293 with the Brønsted–Evans–Polanyi (BEP) relation that has been proposed to describe multi-step
294 processes (see Supplementary Information). To visualize the bonding situations between the
295 active site and the N atom of NO , we employed the Crystal Orbital Hamiltonian Populations
296 (COHP) analysis,^{60,61} which is a theoretical method for partitioning the band-structure energy
297 into orbital-pair interactions. The integrated projected COHP (IpCOHP) below the Fermi level

308 provides qualitative estimation of the bond strength. Usually, the bonding state is characterized
309 by a positive overlap population that leads to negative Hamiltonian off-site elements that
300 contribute to increasing the interactions between adsorbates and the catalyst surface. A
301 comparison of the COHP curves for the four different catalysts (as seen in **Figure 4e**) reveals
302 that a significant number of antibonding states of M-N interactions (with M and N being the
303 transition metal atom and the N atom in NO) are below the Fermi level of the Ni-MoS₂ and Cu-
304 MoS₂. Conversely, fewer antibonding states of M-N are present below the Fermi levels of the
305 Fe-MoS₂ and Co-MoS₂, which indicates that there is a larger orbital overlap between the *d* band
306 of the catalyst and the 2π* of *NO and a lower electron density in the antibonding orbitals
307 below the Fermi level in both Fe-MoS₂ and the Co-MoS₂. The IpCOHP between Fe and N for
308 Fe-MoS₂ was found to be -2.74 eV more negative than the other three MoS₂ SACs suggesting
309 that stabilization of *NO intermediate is responsible for the reduction of the barrier for the
310 formation of the *N intermediate associated with the PDS step of the reaction.

311 **Solar-driven NO₃RR**

312 To evaluate the potential of Fe-MoS₂ for practical ammonia production devices, we
313 integrated the catalysts in a two-electrode H-cell reactor powered by an external photovoltaic
314 (PV) cell. **Figure 5a** shows the respective polarization curves of the cathode and the anode
315 measured in a 3-electrode configuration. The onset potentials for the NO₃RR and OER –
316 measured by definition at 1 mA cm² were estimated to be -0.13 V and 2.06 V vs. RHE,
317 respectively, giving an onset potential of 2.2 V for the full cell. **Figure 5b** presents the
318 electrocatalytic response of the electrolyzer for applied potentials between 0 to 5.0 without
319 compensating for internal resistance. The 2-electrode electrolyzer exhibited an onset potential
320 of 2.26 V consistent with the value predicted from our 3-electrode experiments. We then
321 evaluated the electrolysis property of the full cell, while the Faradaic efficiency was
322 systematically measured at increasing cell voltage. The Faradaic efficiency for NH₃

323 continuously increased up to 4 V to approach a near-unity value. **Figure 5c,d** summarize the
324 full cell energy efficiency (EE), the yield rate, EPC, and the electric power consumption (EPC,
325 in kWh kg⁻¹) obtained by varying the voltage. The EE_{full-cell} was estimated to be 24.2 %, which
326 indicates that our Fe-MoS₂ nanosheets possess a high conversion capability from electric to
327 chemical energy. In addition, the EPC was found to be 46.6 kWh kg⁻¹ at 4.0 V while the yield
328 rate reaches a value of 0.024 mmol h⁻¹ cm⁻² equivalent to 412.6 μg h⁻¹ cm⁻². We coupled the 2-
329 electrode H-cell reactor with a PV cell to achieve indirect photocatalytic conversion of nitrate
330 to ammonia. The relatively large potential associated with the electrochemical synthesis of
331 ammonia typically makes the combination of PV and electrolysis processes difficult to realize
332 experimentally at reasonable efficiencies. A GaInAs/Ga(In)As/Ge triple-junction solar cell was
333 used to generate sufficient photovoltage to drive the catalytic reaction. As presented in **Figure**
334 **5d** and **Supplementary Figure 18**, the negative and positive poles were connected with wires
335 to the Fe-MoS₂ cathode and Pt anodes, respectively, and the solar cell was illuminated by
336 standard AM 1.5G spectrum (100 mW cm⁻²) provided by a Xe solar simulator. **Figure 5e**
337 presents the J–V characteristic curve of the tandem cell under 1 sun, yielding a V_{OC} of ≈ 5 V
338 while exhibiting a solar-to-electric power conversion efficiency (PCE) of 29.89 %. To
339 accurately calculate the solar-to-ammonia (STA) conversion efficiency, the LSV curve of the
340 full cell was measured. The size of the electrodes was adjusted in order to minimize the energy
341 loss of the PV-driven electrolysis. We estimated the energy loss between the PV and the PV-
342 electrolysis systems to be only 4.25 % (**Supplementary Figure 19**). The operating point was
343 determined from the intersection of the J–V curves of both the solar cell and the H-cell reactor.⁶²
344 The results show that the current density and the cell tension at the operating point reached \sim
345 7.19 mA cm⁻² and 4.02 V respectively as shown by the red point in **Figure 5e**. Using the
346 thermodynamic potential of the reaction, the electrolysis current, and the Faradaic efficiency
347 (See Supplementary information for details about the calculations), we estimated the STA

348 efficiency to be $\sim 3.9\%$. To confirm the accuracy of the predicted operating point, we also
349 measured the photocurrents from the NO_3RR process in an unbiased light-driven configuration
350 (**Figure 5f**). The operating point of the NO_3RR was close to the maximum power point (MPP)
351 of the solar panel tandem cell (7.06 mA cm^{-2} at $V_{\text{MP}}=4.23\text{ V}$) (the orange point in **Figure 5e**).
352 We performed solar-driven electrolysis of nitrate for 1 hour and the amount of NH_3 was
353 estimated to reach $60.7\text{ }\mu\text{mol}$; equivalent to a concentration of 1.02 mM or 17.3 ppm – larger
354 the contamination threshold of $\approx 1\text{ ppm}$.¹² The photocatalytic activity corresponds to a yield
355 rate of $\sim 0.03\text{ mmol h}^{-1}\text{ cm}^{-2}$, equivalent to $510\text{ }\mu\text{g h}^{-1}\text{ cm}^{-2}$, which outperforms the previous
356 report on the photocatalysis of the NO_3RR (see **Supplementary Table 7**). The estimated STA
357 efficiency was found to be $\sim 3.4\%$. To the best of our knowledge, such a value for the STA has
358 not been previously reported and sets a new benchmark for solar-driven ammonia production.

359

360 CONCLUSION

361 In summary, we have designed a Fe single-atomic catalyst on two-dimensional MoS_2
362 nanosheets for electrocatalytic conversion of nitrate to ammonia. Fe- MoS_2 catalysts exhibit
363 excellent NO_3RR properties with Faradaic efficiency as high as 98% for production of NH_3 at
364 a low overpotential of $< -0.5\text{ V vs. RHE}$ and a cathodic energy efficiency of 31% at -0.28 V vs.
365 RHE . The catalytic performance of Fe- MoS_2 towards NO_3RR is among the highest reported.
366 The excellent activity and selectivity of Fe- MoS_2 are supported by DFT analysis, which
367 confirmed the superior ability of individual Fe atoms on MoS_2 to activate NO_3^- due to the strong
368 interaction between $2\pi^*$ orbital of NO species and d band orbitals of Fe atoms that leads to low
369 energy barrier for the limiting $^*\text{NO}$ to $^*\text{N}$ reaction. We integrated Fe- MoS_2 in a two-electrode
370 H cell reactor coupled to a PV cell and achieved a solar-to-ammonia conversion efficiency of
371 ca. 3.4% with a yield rate of $510\text{ }\mu\text{g h}^{-1}\text{ cm}^{-2}$. Our investigation sheds light on a practical

372 strategy for the realization of PV-electrolysis systems for the production of ammonia and opens
373 up future applications for solar-driven NH₃ production.

374

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378 **Notes**

379 The authors declare no competing financial interest

380

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400

401

402 **AUTHOR CONTRIBUTIONS**

403 D.V. conceived the idea, designed the experiments and wrote the manuscript. J.L. designed the
404 experiments with D.V., synthesize the materials, performed the electrochemical measurements
405 and the DFT calculations. J.L. and D.V. analyzed the data and wrote the manuscript. C.L. and
406 L.Z. synthesized the materials and performed the physical characterizations. E.P. carried out
407 the NRM measurements. K.Q, Z.Y., H.W. assisted J.L. with the electrochemical measurements.
408 W.W. performed the Raman spectroscopy on the samples with J.L.. X.W. and M.C. discussed
409 the data with J.L. and D.V. L.L. performed high-resolution TEM and EELS on the Fe-MoS₂
410 catalysts. R.Y. performed the STEM measurements. All of the authors edited the manuscript
411 before submission.

412

413 **DATA AVAILABILITY**

414 The data that support the plots within this paper and other findings of this study are available
415 from the corresponding authors upon reasonable request.

416

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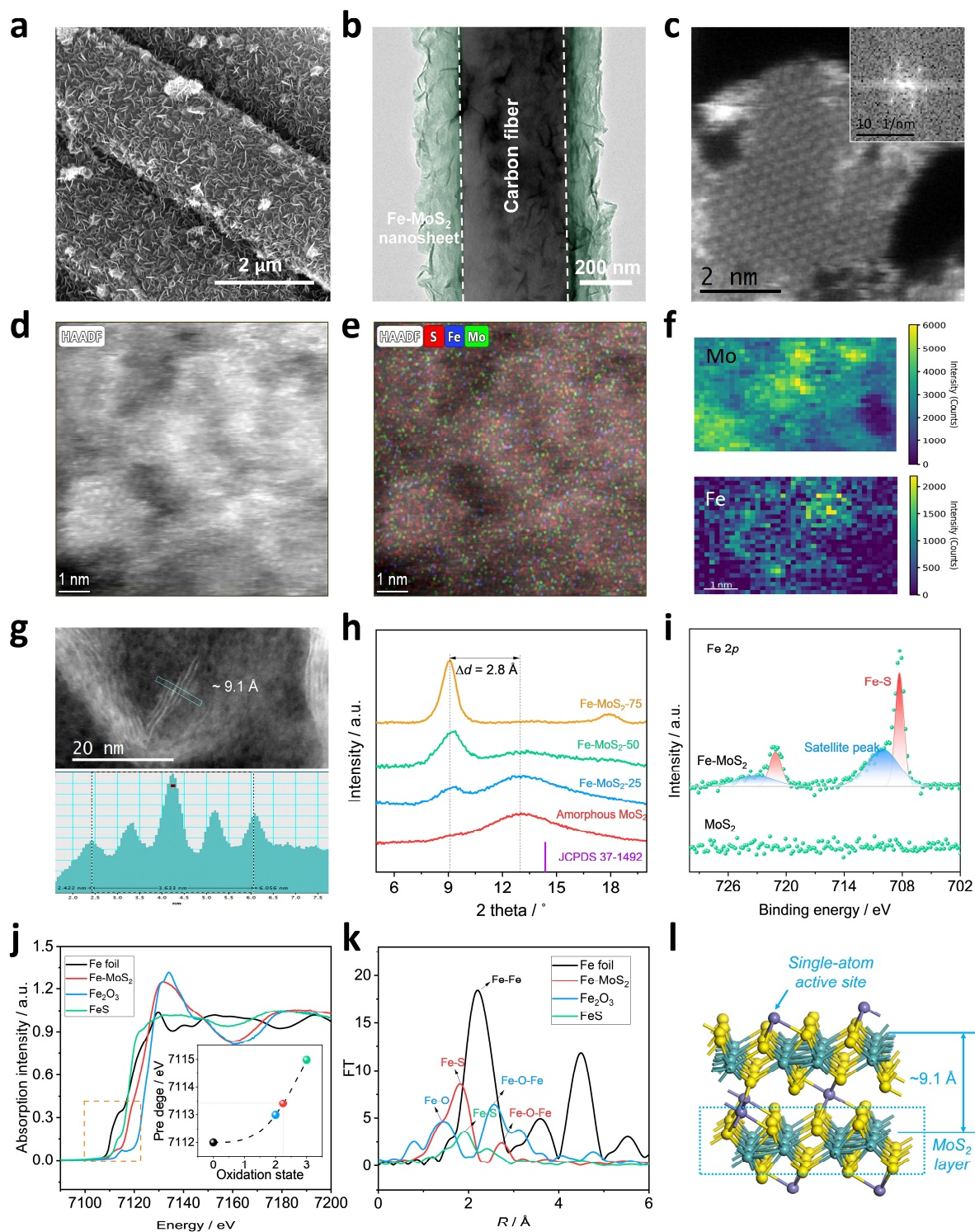
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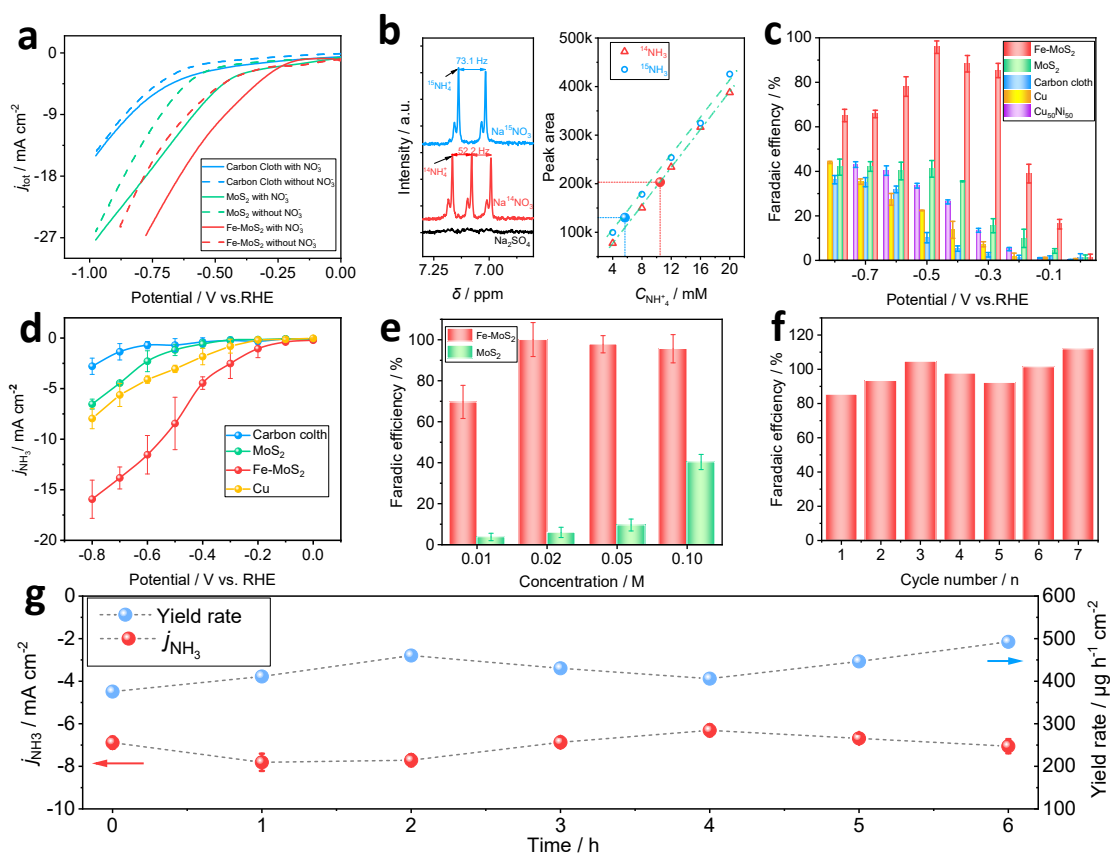
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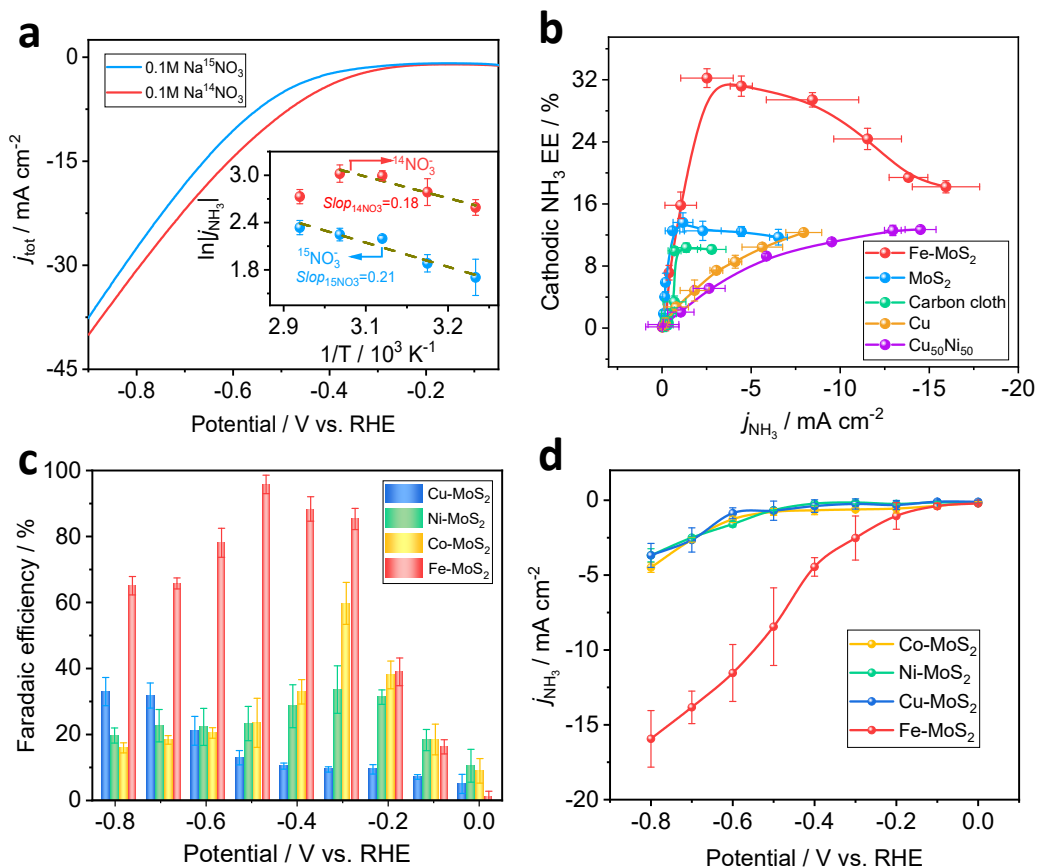
566 **Figure 1 | The physical characterizations of Fe-MoS₂.** a,b, Scanning electron microscopy
 567 (SEM) and transmission electron microscopy (TEM) images of the Fe-MoS₂ nanosheets grown
 568 on the carbon support. c,d, High-angle annular dark-field imaging scanning transmission
 569 electron microscopy (HAADF-STEM) image of Fe-MoS₂ nanosheets. Inset: corresponding

570 diffraction pattern confirming the single-crystalline nature of the nanosheet. **e,f**, High resolution
571 TEM (HRTEM) - energy dispersive X-ray analyses (EDX) and electron energy loss (EELS)
572 elemental mapping images of Fe-MoS₂ nanosheets. **g**, HR-STEM image of the interlayer
573 spacing of Fe-MoS₂ and the corresponding line profiles showing an average d-spacing of ≈ 9.1
574 Å. **h**, X-ray diffraction (XRD) patterns of the Fe-MoS₂ nanosheets with different iron content
575 and compared to pristine MoS₂. **i**, High resolution Fe2p XPS spectra for Fe-MoS₂ and MoS₂. **j**,
576 Normalized XANES spectra and **k**, Fourier transform magnitudes in *R* space of the EXAFS at
577 the Fe K edge of Fe-MoS₂ nanosheets, Fe₂O₃, FeS, and Fe foil, **i**, Proposed structure of Fe-
578 MoS₂.
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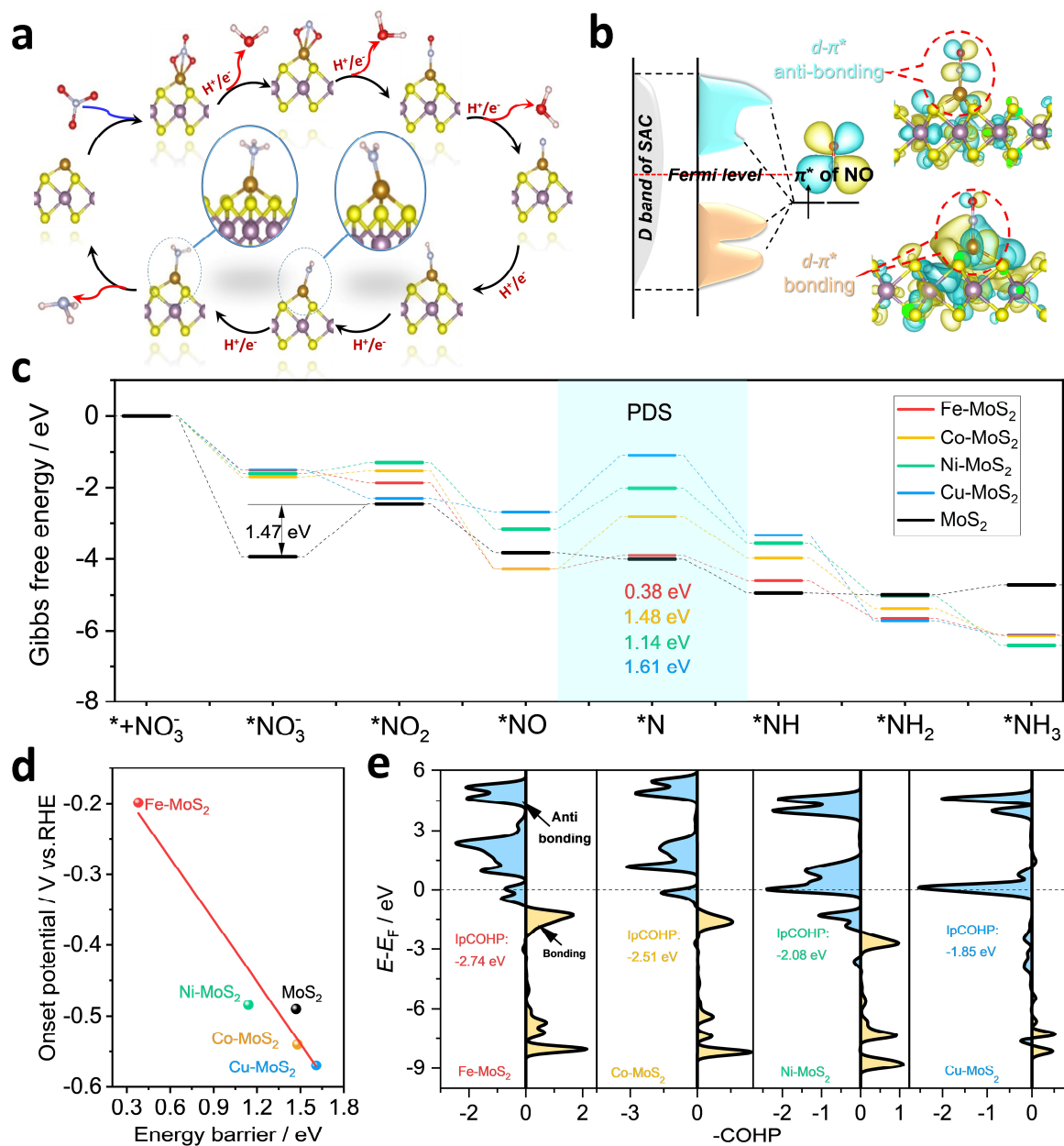
580

581 **Figure 2 | The electrochemical performance investigation for NO₃RR on Fe-MoS₂.** a, LSV
 582 curves of Fe-MoS₂ nanosheets, MoS₂ nanosheets, and the carbon support measured with and
 583 without 0.1 M NaNO₃ electrolyte. b, *Left*: ¹H nuclear magnetic resonance (NMR) spectra (600
 584 MHz) of electrolyte produced from NO₃RR under -0.58 V vs. RHE using 0.1 M Na¹⁴NO₃ and
 585 Na¹⁵NO₃ as N source. *Right*: Calibration curve of ¹⁴NH₄Cl and ¹⁵NH₄Cl measured by ¹H NMR
 586 (right). The experimental NO₃RR results are show as solid symbols. c, Potential-dependent
 587 Faradaic efficiency of ammonia on Fe-MoS₂, MoS₂, Cu, Cu₅₀Ni₅₀ alloy, compared with the
 588 carbon support. d, Evolution of the specific current density: j_{NH_3} as a function of the potential
 589 (vs. RHE). e, Comparison of the Faradic efficiency for ammonia on the Fe-MoS₂ and MoS₂
 590 nanosheets at different nitrate concentrations measured at an applied potential of -0.48 V vs.
 591 RHE. f, Evolution of the Faradaic efficiency at -0.48 V over 7 cycles of 1 hour. The electrolyte
 592 was refreshed for every cycle. h, Evolution of j_{NH_3} and the yield rate of Fe-MoS₂ nanosheets
 593 over time.



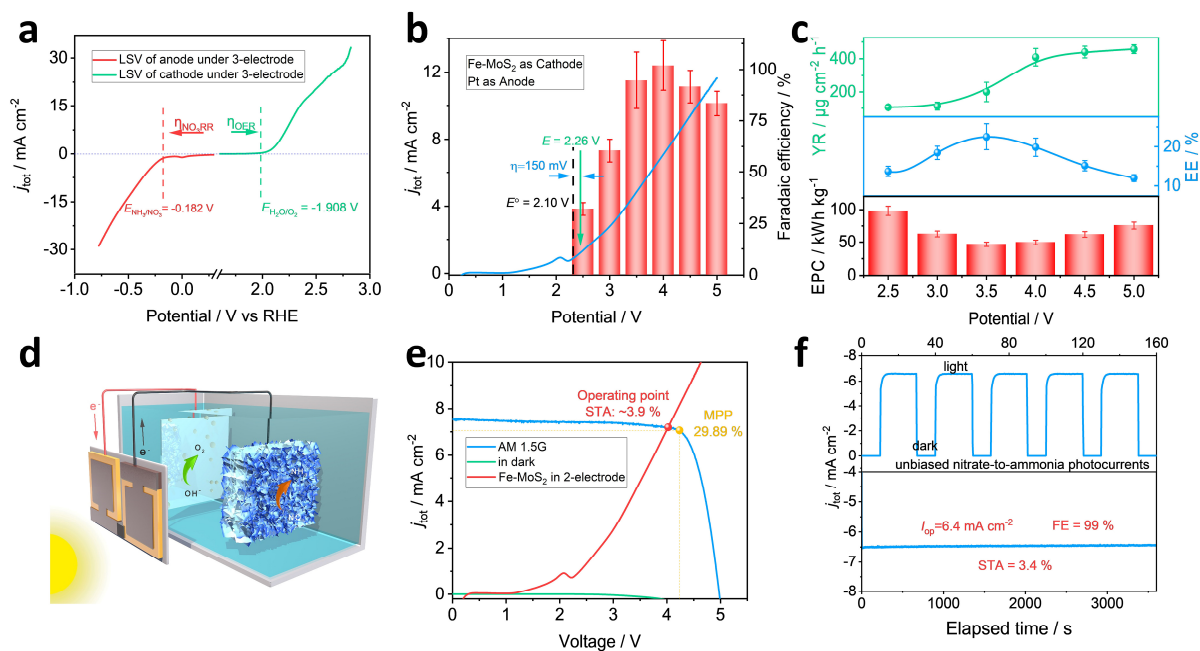
594

595 **Figure 3 | The electrochemical performance towards NO_3RR on MoS_2 -based SACs. a,**
 596 **Linear scanning voltammetry (LSV) and Arrhenius plots (inset) of Fe-MoS₂ measured in**
 597 **$\text{Na}^{14}\text{NO}_3$ and $\text{Na}^{15}\text{NO}_3$ at different temperatures. b, The cathodic energy efficiency (EE) for**
 598 **NO_3^- -to- NH_3 conversion on Fe-MoS₂ and MoS₂ nanosheets compared with Cu, Cu₅₀Ni₅₀, and**
 599 **the carbon support. c, Potential-dependent Faradaic efficiency for ammonia on Cu-MoS₂, Ni-**
 600 **MoS₂, Co-MoS₂ and Fe-MoS₂ nanosheets. d, Evolution of j_{NH_3} on Co-MoS₂, Co-MoS₂, and Co-**
 601 **MoS₂ nanosheets as a function of the applied potential.**



602

603 **Figure 4 | DFT calculations of the NO₃RR on MoS₂ SACs.** **a**, Reaction pathway for the
 604 NO₃RR on M-MoS₂ nanosheets (M: Fe, Co, Ni, and Cu, respectively). **b**, Schematic diagram
 605 of the interaction between NO and the M-MoS₂ nanosheets. **c**, Reaction Gibbs free energies for
 606 different intermediates on M-MoS₂ nanosheets. **d**, Scaling relationship between energy barrier
 607 and onset potential of NO₃RR for pristine MoS₂ and M-MoS₂. **e**, Projected crystal orbital
 608 Hamilton population (pCOHP) of NO adsorbed on M-MoS₂ nanosheets.



609

610 **Figure 5 | The performance of Fe-MoS₂ for NO₃RR using a 2-electrode electrolyzer. a,**
 611 **LSV curves of Fe-MoS₂ nanosheets and Pt used as a cathode and anode. The LSV responses**
 612 **were measured in a 3-electrode configuration. b, Polarization curve of the full cell electrolyzer**
 613 **with a total geometric area of 2 cm². c, Potential-dependent electric power consumption of**
 614 **ammonia, yield rate, and energetic conversion efficiency of the full-cell device. d, Schematic**
 615 **of the photovoltaic-electrolysis system for the conversion of nitrate to ammonia. e, J - V**
 616 **characteristics of the triple junction solar under dark and simulated AM 1.5G 100 mW cm^{-2}**
 617 **illumination. Polarization curves of the full cell device based on Fe-MoS₂ nanosheets as NO₃RR**
 618 **catalyst at the cathode. f, *Top*: Current density-time curve of the PV-EC system without**
 619 **external bias under chopped simulated AM 1.5G 100 mW cm^{-2} illumination. *Bottom*: Stability**
 620 **of the photocatalytic current over 1 hour.**

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Supplementary Files

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